BEDT-TTF SALTS WITH SQUARE PLATINATES(II) AS COUNTERIONS: [BEDT-TTF]₄[Pt(C₂O₄)₂], A NEW ORGANIC METAL

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Abstract

The title compound, [BEDT-TTF]₄[Pt(C₂O₄)₂], is obtained by electrocrystallization, which yields black platelets of stoichiometry $C_{44}H_{32}O_8PtS_{32}$, $M_r = 1909.87$. Crystal structure determination at room temperature gives a triclinic cell P1, with a = 8.678(2) Å, b = 11.878(5) Å, $c = 15.757(7) \text{ Å}, \alpha = 105.49(3)^{\circ}, \beta = 91.05(3)^{\circ}, \gamma = 91.96(3)^{\circ}, V = 1563.64 \text{ Å}^3$ $d_c = 2.03 \text{ g/cm}^3$, Z = 1. The solid contains sheets of BEDT-TTF cations separated by sheets of $[Pt(C_2O_4)_2]^{2-}$ counterions. The BEDT-TTF sheets are made up of columns that interact strongly through S-S contacts. The columns are built up by pairs of BEDT-TTF molecules. There are strongly varying bond distances in the two different BEDT-TTF moieties that make up a pair. This fact is especially clearly expressed in the surprisingly short C-C distances of the central C=C double bond of the two BEDT-TTF ions: 1.273 Å and 1.327 Å respectively. These are by far the shortest distances for the central C=C bonds observed so far in conductive BEDT-TTF salts. The compound behaves like a metal down to about 60 K. Temperaturedependent d.c. conductivity, e.s.r. and thermopower measurements show metallic behaviour at room temperature ($\sigma_{300} = 20 \text{ S/cm}$) with a metal-tometal phase transition around 200 K. At about 60 K a second broad phase transition occurs and the crystals become semiconducting.

Introduction

The combination of BEDT-TTF⁺ radical cations with anions of transition metal complexes has proven to be exceptionally successful in the search for new molecular metals [1 - 7] and superconductors [1, 7 - 9]. The organic conductor with the highest transition temperature to superconductivity so far observed, 10.4 K in [BEDT-TTF]₂[Cu(SCN)₂] [8, 9], belongs to this group of solids.

For several reasons we have decided to combine two successful principles in one solid: the properties of the 'inorganic', one-dimensional, molecular metals, which were of central interest a decade ago because of their potential to form one-dimensionally interacting transition metal ion chains (typically in mixed valence platinates [10 - 12]) on one hand and the special often pronounced two-dimensional properties of metallic, organic solids containing radical cations like BEDT-TTF+ on the other hand. Recently we obtained two phases of [BEDT-TTF]₄[Pt(CN)₄] as the first solids in our efforts to combine lattice elements of typical 'inorganic' with those of well-known 'organic' molecular metals [13]. One of them turned out to be semi-conducting at room temperature and below while the second is metallic at room temperature [13]. In this communication we report the preparation, structure and some physical properties of a metallic phase of another compound from this class of materials, namely [BEDT-TTF]₄[Pt(C₂O₄)₂].

Experimental

Preparation

The crystals of 1 were obtained by electrocrystallization. In a typical experiment 60 mg of BEDT-TTF were dissolved in 100 ml warm 1,1,2-trichloroethane. To this solution was added 225 mg of $K_2[Pt(C_2O_4)_2]$ finely ground in a mortar and the mixture was stirred for several hours. Finally, 1000 mg of 18-crown-6-ether were added. The mixture was put into a three-compartment cell without filtering and electrolysed for about five weeks using a voltage of 1.8 V, resulting in a current of 0.3 to 0.5 μ A. This dark crystals of the title compound are obtained.

X-ray structure

A dark brownish black prismatic crystal, $0.05 \times 0.2 \times 0.2$ mm³, was used for the structure determination. Lattice parameters were obtained from setting angles of 25 reflections ($3 < 2\theta < 35^{\circ}$) centred on a diffractometer (Syntex R3, Mo K α radiation). Data collection at room temperature by Ω scan ($3 < 2\theta < 60^{\circ}$) yielded 4577 observed independent reflections with I > 2.5 σ (I). An empirical absorption correction using psi-scans of five reflections with $6 < 2\theta < 27^{\circ}$ was applied ($\mu = 33.5$ cm⁻¹). The structure was solved by direct methods and Fourier syntheses. Refinement of 385 variables (fixed hydrogen positions) converged with R = 0.055 and $R_{\rm w} = 0.047$. The largest features in a final Fourier map were +0.94 and -1.96 eA⁻³.

Calculations were performed with the SHELXTL program package [14] on an Eclipse computer (Data General) using scattering factors from the International Tables of X-ray Crystallography [15].

Physical measurements

Temperature-dependent resitivity measurements were carried out with the typical four-probe technique in the temperature range 4.2 - 300 K.

Thermopower measurements were done with an apparatus described earlier [16] in the same temperature range. The e.s.r. investigations were performed with an X-band spectrometer (Bruker ESP 300) from 4.2 to 300 K. A single e.s.r. line was observed, as is typical for conduction electrons of a metal. The spin susceptibility was obtained in the usual way from the e.s.r. linewidth and the intensity of the signal. The unit cell data of the crystals used for the measurements of the physical properties were controlled by X-ray (after the measurements).

Results

Atomic coordinates are listed in Table 1, bond distances and angles in Table 2. Figure 1 shows the two crystallographically different BEDT-TTF moieties of the unit cell and Fig. 2 the $[Pt(C_2O_4)_2]^{2-}$ counterion. The numbering scheme is indicated in these Figures. The two different BEDT-TTF molecules occur in pairs, which make up columns. Projections of the unit cell along (Fig. 3) and perpendicular (Fig. 4) to these columns give an impression of the 'pairing' and they show that there are strong S—S contacts

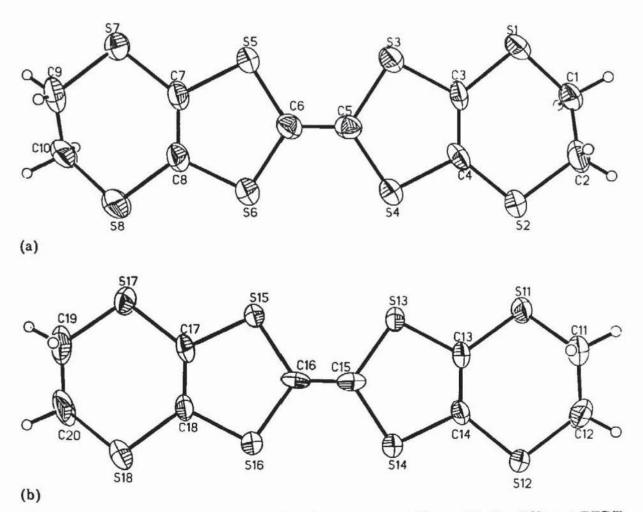


Fig. 1. Numbering schemes (a) and (b) of the two crystallographically different BEDT-TTF mojeties.

TABLE 1 Atomic coordinates

Atom	x/a	y/b	z/c	Atom	x/a	y/b	-1-
Atom		910	2/0	Atom		9/0	z/c
T1	.00000	.20000	.50000	112A	.47302	.52139 .00074	1.42470
01	.20511 .00058	.42076 .00044	.49165 .00035	1128	.57220 .00099	-44107 -00074	1.34330
32	.07034	.34444	.50120 .00037	C3	.32771	.28277	1.21187
33	3:305	.25430 .00047	.50603 .00037	C4	.31834	.37727	1.21272
04	.02025	.16443	.49771 .00045	C5	.17772 .00094	.28013 .00064	1.06061
021	.20210	.31320	.47775 .00050	CG	.11700	.24678	.78458
C22	.03722	.26447	.49837 .00055	C7	01507 .00098	.13011	.03963
3 1	.37517 .00034	.22564 .00020	1.27363	CB	02133	.24484	.83870 .00053
52	.36196	.52049 .00020	1.27843	C9	14712 .00134	.06334	.67393
23	.26615 .00031	-17875 -00019	1.11550	Н9А	.22606	.00846	.64033 .00058
34	.24501 .00029	.42437	1.11645	1198	.06258	.06701	.63771
S5	.07137	.10177	.73141	C10	21075	.17878 .00075	.69613
56	.05533	.34697	.93127 .00015	HIOA	24207	.17637	.6424
57	08336	.00662	.75945 .00018	1110B	30027	.17276	.72972
SB	.10245	.27730	.75803 .00019	311	.30710 .00033	.26715	.00015
C1	-40470 -00108	.35221 .00068	1.30737	312	.34703 .00036	02460	.6476
H1A	-44304 -00108	.33407 .00068	1.43605 .00052	313	-44612 -00027	.31306	.0301
HIB	.30136	.36941	1.40795	314	.46706 .00029	.06562	.00014
C2	-47538 -00099	-46193 -00074	1.37047	315	.63224	.30211	1.0267 .0001 continued

TABLE 1 (continued)
Atomic coordinates

Atomic	coordinates						
Atom	x/a	y/b	z/c	Atom	x/a	y/b	z/c
516	.65162 .00029	.13367	1.01660	C15	.51211 .00091	-21045 -00060	.87223
517	.77564 .00032	.46560 .00017	1.19797	C16	.58320 .00090	.23774	.96584 .00056
518	.02542 .00035	.16739 .00022	1.10240	C17	-72661 -00089	.34674 .00070	1.11337
C11	.34637 .00102	.15343 .00068	.56356 .00052	C10	.73572 .00094	.23197	1.10312
H11A	.45614 .00102	.14634 .00060	.56050 .00051	C19	.71174 .00114	.37818 .00090	1.26353 .00061
H11B	.30819 .00102	.17311	.51177 .00051	H19A	.72616 .00114	.45252 .00070	1.32063
C12	.27833 .00108	.03680 .00070	.56591 .00055	H19B	1.00702	.38475 .00090	1.23594
H12A	.13974 .00108	-04681 -00070	.57380 .00055	C20	.05687 .00138	.20554 .00082	1.27525 .00057
H12B	.27458 .00108	01817 .00070	.51020 .00055	H20A	.72702 .00138	.26214 .00082	1.31357
C13	.37678 .00091	.21135 .00068	.74185 .00049	H20B	.75766 .00138	.27764	1.30350
C14	.38755 .00094	.09322	.73330 .00050				
03'	C21'	01'				C2?	04
	THE PROPERTY OF THE PROPERTY O						
	C22'					C21	

Fig. 2. Numbering scheme of the $[Pt(C_2O_4)_2]^{2-}$ counterions.

TABLE 2 Bond lengths (Å) and bond angles (°)

```
PT1 -01
          1.970( .005)
                         82.6( .2)
130.0( .0)
97.4( .2)
PT1 -02 1.771( .006)
PT1 -01 A 1.770( .005)
                                        97.4( .2)
                                                     02.6( .2)
FT1 -02 A 1.971( .006)
                                       180.0( .0)
                              01
                                           02
                                                        01 A
    -PT1
01
          1.970( .005)
                         113.9( .5)
01
   -C21
          1.320( .010)
                              PT1
    -FT1
          1.971( .006)
02
                         115.6( .5)
    -C22
          1.313( .010)
02
                              PT1
03
   -C21
          1.200( .011)
04 -022
          1.202( .011)
C21 -01
          1.320( .010)
C21 -03
          1.200( .011)
                         124.8( .8)
C21 -C22
          1.544( .012)
                          114.5( .7)
                                       120.6( .8)
                             01
                                           03
C22 -02
          1.313( .010)
C22 -04
          1.202( .011)
                         125.8( .8)
C22 -C21
          1.544( .012)
                          112.5( .7)
                                       121.6( .8)
                              02
                                           04
   -C1
          1.803( .007)
31
S1
   -C3
                          102.3( .4)
                              C1
          1.780( .010)
52
    -C2
                         101.6( .4)
S2
   -C4
          1.729( .007)
                              C2
33
   -C3
          1.744( .007)
S3
   -C5
                           75.9( .4)
          1.769( .009)
                              C3
          1.747( .009)
1.736( .007)
    -C4
S4
                           97.1( .4)
34
    -05
                              C4
          1.733( .007)
55
    -C6
55
                           95.6( .4)
    -C7
                              C6
    -C6
56
           1.727( .009)
          1.732( .008)
56
    -08
                           96.1( .4)
                              C6
    -C7
           1.738( .008)
57
          1.755( .011)
                          103.4( .4)
37
    -C9
                              C7
   -C8
58
           1.725( .010)
38 -C10
          1.735( .009)
                          101.8( .5)
                              CB
    -51
           1.803( .007)
C1
C1
    -HIA
           .960( .000)
                          108.7( .3)
C1
    -H1B
            .960( .000)
                          108.1( .3)
                                      109.5( .0)
C1
    -C2
           1.524( .013)
                          114.1( .6)
                                                    107.0( .5)
                                      107.4( .5)
                              51
                                                        H1B
                                           H1A
            .960( .000)
H1A -C1
H1B -C1
            .960( .000)
C2
    -S2
           1.780( .010)
C2 -C1
           1.524( .013)
                          114.6( .6)
C2 -H2A
           .960( .000) 103.4( .3) 109.4( .4)
                                                                   (continued)
```

TABLE 2 (continued)
Bond lengths (Å) and bond angles (°)

C2 -H2B	.9600	.000)	108.0(.3)	107.0(.5)	109.5(.0)	
H2A -C2	.9600	.000)	52	C1	H2A	
H2B -C2	.9600	.000)				
C3 -S1	1.739(.009)				
C3 -S3	1.7440	.007)	114.8(.5)			
C3 -C4	1.3590	.012)	128.0(.6) S1	117.2(.6) S3		
C4 -S2	1.7290	.007)				
C4 -S4	1.7460	.009)	115.2(.5)			
C4 -C3	1.3590	.012)	120.3(.6) 52	116.0(.5) S4		
C5 -S3	1.7690					
C5 -S4	1.7360	.007)	113.0(.4)			
C5 -C6	1.3260	.011)	122.4(.6) S3	124.6(.7) S4		
C6 -S5	1.733(.007)				
C6 -S6	1.7270	.009)	115.1(.5)			
C6 -C5	1.3260	.011)	123.2(.7) S5	121.8(.4) S6		
.C7 -S5	1.7340	.009)				
C7 -S7	1.7380	.008)	114.8(.5)			
C7 -C8	1.369(.012)	116.9(.6) S5	128.3(.7) S7		
C8 -S6	1.7320	.008)				
C8 -S8	1.7250	.010)	116.4(.5)			
C8 -C7	1.3690	.012)	116.2(.7) S6	127.4(.6) S8		
C9 -S7	1.7550	.011)				
C9 -H9A	.9600	.000)	107.2(.3)			
C9 -H9B	.7600	.000)	107.1(.4)	109.5(.0)		
C9 -C10	1.448(.013)	118.7(.7) S7	108.7(.6) H9A	105.4(.6) H7B	
H9A -C7	-9600	.000)				
119B -C9	.7600	.000)				
C10 -S8	1.735(.009)				
C10 -C9	1.448(.013)	121.0(.8)			
C10 -H10A		.000)	107.1(.4)	108.2(.5)	100 51 01	
C10 -H10B		.000)	105.9(.3) S8	104.9(.6) C9	109.5(.0) H10A	
H10A-C10	.7600					
H10B-C10		.000)				
S11 -C11	1.781(.007)	0.2 2 22 2 2 v			
S11 -C13	1.753(.007)	100.3(.4) C11			
512 -C12	1.7720	.010)				
S12 -C14	1.753(.007)	103.3(.4) C12			
S13 -C13	1.7490	.007)				
S13 -C15	1.7680	.009)	96.7(.4) C13			
S14 -C14	1.740(.009)				(continued)

TABLE 2 (continued)
Bond lengths (Å) and bond angles (°)

VIEW CO Program					
S14 -C15	1.7590	-007)	97.1(.4) C14		
S15 -C16	1.7550	.007)	614		
S15 -C17			96.3(.4)		
S16 -C16	1 7441	0081	C16		
S16 -C18			96.3(.4)		
			C16		
S17 -C17 S17 -C19			100 7/ 43		
317 -017	1./05(.0117	102.7(.4) C17		
S18 -C18			DECEMBER WHILE THE		
S18 -C20	1.7460	.008)	(
C11 -S11	1.781(.007)	C18		
C11 -H11A	-9600	.000)			
			108.6(.3)		
C11 -C12	1.4970	.012)	311	107.1(.5) H11A	109.5(.5) H11B
H11A-C11	.9600	.000)	011	112211	HILL
H11B-C11	.960(.000)			
C12 -S12	1.7720	.010)			
C12 -C11			117.6(.6)		
C12 -H12A			107.3(.3)	106.2(.5)	
C12 -H12B		.000)		108.4(.4)	109.5(.0)
		525	S12	C11	H12A
H12A-C12	.960(.000)			
H12B-C12	.9600	.000)			
C13 -511	1 757/	0001			
C13 -S13			116.1(.5)		
C13 -C14				116.9(.6)	
			511	513	
C14 -S12					
C14 -S14	1.740(.009)	114.1(.5)		
C14 -C13	1.3380	.012)	128.7(.7)	117.1(.6)	
045 045			312	S14	
C15 -S13					
C15 -S14 C15 -C16					
612 -610	1.2/3(.011)	124.2(.6) S13	123.8(.7) S14	
C16 -S15	1.7550	-007)	313	214	
C16 -S16			113.2(.5)		
C16 -C15	1.273(124.0(.7)	122.8(.6)	
			315	516	
C17 -S15	1.731(.009)			
C17 -S17			115.4(.5)		
C17 -C18	1.351(.012)	116.7(.5)		
C18 -S16	1 7244	0071	315	317	
C18 -S18			114 7/		
C18 -C17				100 04 41	
240 017	1.001(.0127	117.4(.6) S16	126.2(.6) S18	
C19 -S17	1.7850	.011)	510	210	
C19 -H19A	.9600		107.4(.3)		
C19 -H19B		.000)		109.5(.0)	

(continued)

```
C19 -C20 1.464( .015)
                       118.0( .7) 108.4( .5) 106.1( .6)
                           S17
                                       H17A
                                                   H19D
H19A-C17
          .960( .000)
H19B-C19
          .760( .000)
C20 -S13
         1.746( .008)
C20 -C19
         1.464( .015)
                       118.9( .7)
C20 -H20A .760( .000)
                      107.3( .4) 108.3( .6)
                       106.8( .4) 105.8( .4)
                                               109.5( .0)
                                       C17
                           S18
                                                   H20A
H20A-C20
          .960( .000)
H20B-C20
        .760( .000)
```

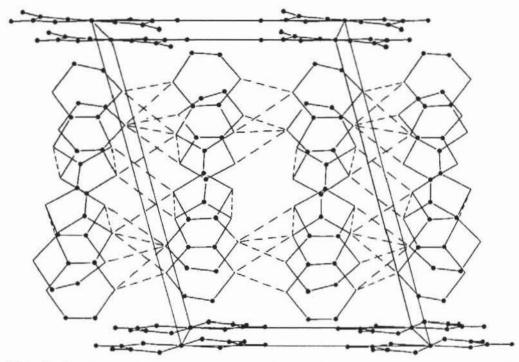


Fig. 3. Section of the structure projected along the columns. Dotted lines represent intermolecular S....S contacts < 3.8 Å.

between molecules of the different columns. Figure 5 gives a different view of the intermolecular S-S contacts between BEDT-TTF moieties of different columns. The dotted lines in Figs. 3, 4 and 5 connect S atoms that are less than 3.8 Å apart. 3.6 Å is the sum of the covalent S radii [17].

The bond distances in the two different BEDT-TTF molecules of a pair vary appreciably. Especially pronounced is this difference in the bond distances of the central C=C bond. It is 1.273 Å in one and 1.327 Å in the second molecule of the pair. These values are very unusual. There have been several earlier efforts to find a relation between the amount of positive

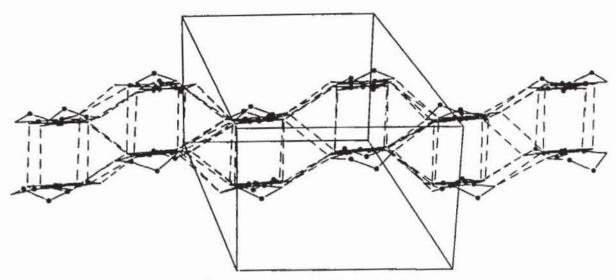


Fig. 4. S....S contacts < 3.8 Å between pairs of BEDT-TTF moieties.

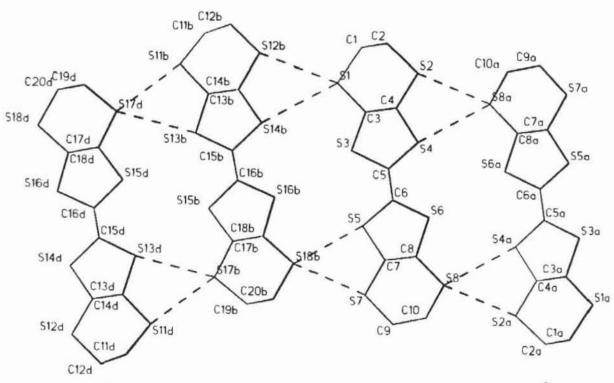


Fig. 5. The shortest S....S contacts between the columns $(S_2-S_{8a}=3.432 \text{ Å}, S_5-S_{18b}=3.382 \text{ Å} \text{ and } S_7-S_{18b}=3.373 \text{ Å}).$

charge on the (BEDT-TTF)ⁿ⁺ ions and the bond distances in these moieties [18, 19]. Compared to these earlier findings, the C=C distances reported here would suggest a negatively-polarized BEDT-TTF molecule. Since this is not reasonable for the DEDT-TTF moieties in the title compound, the determination of charge densities on the basis of bond distances in the BEDT-TTF salts has to be treated with caution. Either packing effects or the charge on the counterion (in our exceptional case two), which may lead to molecular distortions or polarizations, seem to play an important part in the bond distances too.

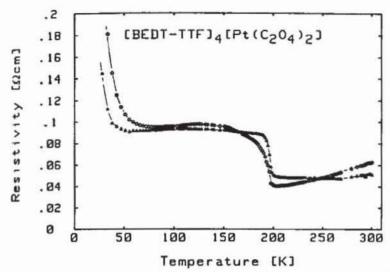


Fig. 6. Temperature dependence of the resistivity of two typical crystals of [BEDT-TTF]₄[Pt(C_2O_4)₂].

Figure 6 shows the temperature dependence of the resistivity of two typical crystals of our samples. The room-temperature conductivity σ_{300} is about 20 S/cm. For both crystals a weak decrease of the resistivity is observed by lowering the temperature down to 200 K. Such a behaviour is typical for a metal.

At around 200 K both crystals show a relatively sharp resistivity increase, indicating a structural phase transition at this temperature. As pointed out later, e.s.r. and thermopower investigations support this conclusion. Since the resistivity decrease continues below 120 K, as can clearly be seen in the blow-up in Fig. 7, it seems that the phase transition again leads into a metallic state. This conclusion is again supported by the results of the thermopower measurements (Fig. 8).

It is evident from Fig. 7 that the resistivity increases by lowering the temperature below 80 K in one crystal, and below 60 K in the other. Since this resistivity increase occurs in different crystals at different temperatures,

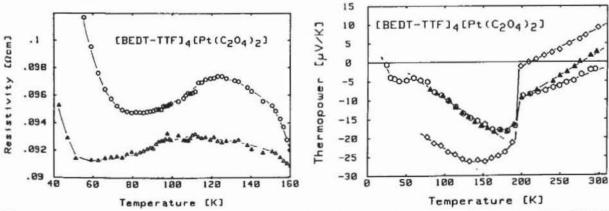


Fig. 7. Resistivity of the two crystals shown in Fig. 6 in the temperature range 40 - 160 K (expanded scale).

Fig. 8. Temperature dependence of the thermopower of three different crystals of $[BEDT-TTF]_4[Pt(C_2O_4)_2]$.

it seems that a very broad phase transition into a semiconducting state exists (activation energy = 0.006 eV). The broadness of the transition might result from residual stresses due to the phase transition at 200 K, as well as from variations in crystal quality. A further indication for the existence of residual stresses in the crystals at lower temperatures is the fact that we were only in one case able to measure thermopower data below 70 K. With five other crystals only values down to 70 K could be obtained.

Figure 8 shows the temperature dependence of the thermopower of three crystals in the temperature range 20-300 K. Above 200 K the thermopower is directly proportional to the temperature, as expected for a metal. The sharp decrease at 200 K indicates the above-mentioned phase transition. In the lower temperature regime (70-140 K) the thermopower is again linearly proportional to the temperature as in a metal. Above 200 K the conductivity seems to be dominated by holes (positive thermopower at room temperature), while below 200 K the conductivity might be dominated by electrons. This change in charge carriers at the phase transition could be due to an opening of a gap in the conduction band, resulting in a different conduction band filling. In the crystal measured down to 20 K, the phase transition into the semiconducting state at about 60 K can be clearly seen.

The e.s.r. results are presented in Figs. 9 - 11. At room temperature the e.s.r. linewidth is weakly angle dependent and varies between 43 and 46 G. Figure 9 shows the temperature dependence of the e.s.r. linewidth starting with a room-temperature value of 43 G. By lowering the temperature the e.s.r. linewidth decreases to a value of about 3.5 G at 4.2 K. This is a typical behaviour expected for a quasi two-dimensional metal. Around 200 K a sharper decrease of the linewidth seems to occur, which in our opinion is due to the above-mentioned phase transition. The otherwise monotonic decrease of the linewidth with decreasing temperature can be attributed to temperature-dependent scattering processes. Therefore, the e.s.r. linewidth might be described by the Elliot formula for the spin relaxation in metals:

$$\Delta H = (\Delta g)^2 \tau^{-1}/\gamma$$

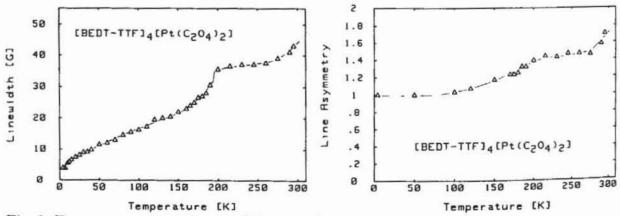


Fig. 9. Temperature dependence of the e.s.r. linewidth.

Fig. 10. Temperature dependence of the asymmetry of the e.s.r. line (Dyson-type e.s.r. line).

where $\Delta g = g - 2.0023$ measures the spin-orbit coupling, γ is the electron gyromagnetic ratio and τ^{-1} the scattering rate of the conduction electrons. At room temperature the e.s.r. line shows the typical Dyson-like asymmetry usually observed in metals. The temperature dependence of the asymmetry of the e.s.r. line is shown in Fig. 10. As can be seem from this diagram, the line becomes symmetric with a Lorentzian shape below 100 K in the region where the crystals become semiconducting.

Figure 11 shows the temperature dependence of the spin susceptibility as obtained from the e.s.r. measurements. Above and below 200 K the susceptibility is more or less temperature independent, while at 200 K a weak decrease appears. This result supports the assumption of a metal-to-metal phase transition at 200 K. The susceptibility is even constant below 60 K but at temperatures below 20 K an increase (probably due to paramagnetic impurities) is observed. These paramagnetic impurities might be responsible for the nearly temperature-independent behaviour of the susceptibility even in the semiconducting region.

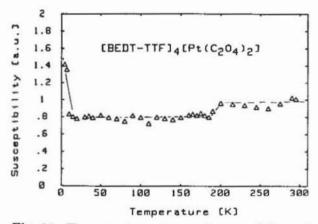


Fig. 11. Temperature dependence of the spin susceptibility (arbitrary units).

It should be mentioned that during the e.s.r. investigations we found a second crystallographic phase, which shows an e.s.r. linewidth of about 20 G at room temperature. Further investigations of this phase are in progress.

Conclusion

So far there have been only a very few attempts to react the doubly-charged anionic metal complexes, which are well known in the realm of one-dimensional 'inorganic' molecular metals, with organic donors [5, 20, 21]. Doubly-charged negative counterions seem to have a strong influence on the charge distribution in the 'organic part' of BEDT-TTF salts. Though the general features of the structure are as 'usual', the structural details are very different. To the best of our knowledge this is the first example of a metallic BEDT-TTF salt with a doubly negatively-charged counterion whose molecular structure has been unambiguously characterized. The solids with

'polymeric' anions [2, 3, 22, 23] with charges distributed over several metal centres are considered as a separate group, because the charge density on a special site in the counterion is difficult to determine.

There is one other example of this type of compound: [BEDT-TTF]4-[Pt(CN)₄] [5, 13]. One phase of this solid turned out to be a semiconductor. In one of the phases of the latter material, we again found two crystallographically different BEDT-TTF moieties with very different and comparatively short (1.306 and 1.350 Å) C=C central double bonds [13]. It is very surprising to note that despite the strongly alternating bonds in adjacent BEDT-TTF moieties and the clear pairing of these non-equivalent units along the stacks in the [Pt(C2O4)2]2- derivative, a metallic behaviour results. On the basis of rules developed so far to predict high conductivity in molecular metals (regular, mixed valence stacks), one has to assume that the conduction path in the new compound follows solely along the intermolecular S-S contacts perpendicular to the stacks. The results obtained here and earlier [13] show that the molecular structure and physical behaviour of the BEDT-TTF salts with doubly-charged negative counterions are quite different from those of their mono-anionic 'relatives'. We will try to find out more about these special materials by using other planar platinum complexes as counterions.

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