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Naval Construction Battalion Center
Port Hueneme, California 93043


CORROSION OF METALS AND ALLOYS
IN THE DEEP OCEAN
by F. M. Reinhart

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Between 1960 and 1970 , about 20,000 specimens of 475 alloys were exposed in the seawater in the Pacific Ocean in order to conduct a program on the effects of deep-ocean environments on materials. The test specimens included steels, cast irons, stainless steels, copper, nickel, aluminum, titanium, miscellaneous alloys, and wire ropes. They were exposed at the surface and at nominal depths of 2,500 and 6,000 feet for periods of time varying
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## SECTION 1

## INTRODUCTION

Between 1962 and 1970 the Civil Engineering Laboratory, Naval Construction Battalion Center, Port Hueneme, California, exposed approximately 20,000 specimens of about 475 different alloys in the Pacific Ocean. These specimens were exposed at the surface and at nominal depths of 2,500 and 6,000 feet for periods of time varying from 123 to 1,064 days.

The purpose of these exposures was to provide the Naval Facilities Engineering Command (NAVFAC) with information on the deterioration of materials in deep-ocean environments. Such information was needed to improve techniques, to develop new techniques pertaining to naval material, and to support the increasing interest in the deep ocean as an operating environment.

The Naval Facilities Engineering Command is charged with the responsibility for the construction and maintenance of all fixed Naval facilities; hence, the construction and maintenance of Naval structures at depths in the oceans are but one facet of its overall responsibility. Fundamental to the design, construction, maintenance, and operation of structures and their related facilities is information on the deterioration of materials in a particular environment. Since there was very little published information on the behavior of construction materials in deep-ocean environments, this program was initiated in 1960 to obtain such information.

In-situ testing was chosen because it is not possible to duplicate all the variables and the changes in these variables that prevail in any one environment or location. A test site was considered suitable if the circulation (currents), sedimentation, and bottom conditions were representative of open ocean conditions: (1) the bottom should be reasonably flat, (2) the site should be open and not located in an area of restricted circulation, such as a silled basin, (3) the site should be reasonably close to Port Hueneme for ship operations, and (4) the site should be within the operating range of the more precise navigating and locating techniques.

A Pacific Ocean site meeting these requirements was selected at a nominal depth of 6,000 feet. The ocean bottom at this site is relatively flat in a broad submarine valley southwest of San Miguel Island, California; it is readily accessible to the Civil Engineering Laboratory; and it is subject to the effects of ocean currents. This site, designated Test Site I, is approximately 81 nautical miles southwest of Port Hueneme, latitude $33^{\circ} 44^{\prime} \mathrm{N}$, longitude $120^{\circ} 45^{\prime} \mathrm{W}$.

Oceanographic data collected between 1961 and 1963 [1,2] show the presence of an oxygen minimum zone at depths between 2,000 and 3,000 feet. This minimum oxygen zone was present at all sites investigated when the ocean floor was at depths varying between 2,000 and 13,000 feet.

It is well known that the corrosion rates of many materials (e.g., steels) are affected by the concentration of oxygen in the environment. Because of this, it was decided to establish a second test site, Test Site II, in this minimum oxygen concentration zone where, it was thought, much pertinent information could be obtained. Test Site II (nominal depth of 2,500 feet) is 75 nautical miles west of Port Hueneme, latitude $34^{\circ} 06^{\prime} \mathrm{N}$, longitude $120^{\circ} 42^{\prime} \mathrm{W}$.

The oceanographic investigations by the Civil Engineering Laboratory also disclosed that the ocean floor at these sites is rather firm and was characterized as sandy, green cohesive mud (partially glauconite) with some rocks. Biological cultures of these bottom sediments showed the presence of sulfate-reducing bacteria in at least the first 6 inches of sediment.

In order to determine the differences between the corrosiveness of seawater at depths and at the surface in the Pacific Ocean, it is desirable to compare deep-ocean corrosion data with surface immersion data. Since surface data from the Pacific Ocean in the vicinity of Port Hueneme were not available in the literature for most of the alloys exposed at depths in the Pacific Ocean, it was decided to establish a surface exposure site to obtain this information. Therefore, a third site, Test Site V, was established at the

Naval Pacific Missile Range, Point Mugu, California, latitude $34^{\circ} 06^{\prime} \mathrm{N}$, longitude $119^{\circ} 07^{\prime} \mathrm{W}$. Test Site V is about 10 miles east of Port Hueneme.

The specific geographical locations of the test sites and the average characteristics of the seawater 10 feet above the ocean floor at these sites are given in Table 1. Their positions relative to the California coast are shown in Figure 1. The variation of the temperature, pH , salinity and oxygen content of the seawater with depth at the STU sites is shown in Figure 2.

Other naval activities were invited to participate in this program and, if possible, to contribute to the funding. From 1962 to 1966 the Naval Air Systems Command supplied funds for partial support of the program. Navy contractors and other companies also participated in this program. The participants are listed in Table 2 as well as those who evaluated the materials and whether or not the evaluators, other than CEL, supplied CEL with the results of their evaluations.

This report presents the performance data obtained by CEL and other participants from the seawater exposures at the sites given in Table 1. The performance of the various materials as supported by this data is also discussed.

Figure 1. STU sites off the Pacific Coast; STU structure in inset.


Figure 2. Variation of environment with depth at STU sites.

Table 1. Exposure Site Locations and Seawater Characteristics

| Site <br> No. | Latitude <br> N | Longitude <br> W | Depth <br> $(\mathrm{ft})$ | Exposure <br> (day) | Temperature <br> $\left({ }^{\circ} \mathrm{C}\right)$ | Oxygen <br> $(\mathrm{ml} / \mathrm{l})$ | Salinity <br> $(\mathrm{ppt})$ | pH | Average <br> Current <br> (knot) |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{I}-1$ | $33^{\circ} 46^{\prime}$ | $120^{\circ} 37^{\prime}$ | 5,300 | 1,064 | 2.6 | 1.2 | 34.51 | 7.5 | 0.03 |
| $\mathrm{I}-2$ | $33^{\circ} 44^{\prime}$ | $120^{\circ} 45^{\prime}$ | 5,640 | 751 | 2.3 | 1.3 | 34.51 | 7.6 | 0.03 |
| $\mathrm{I}-3$ | $33^{\circ} 44^{\prime}$ | $120^{\circ} 45^{\prime}$ | 5,640 | 123 | 2.3 | 1.3 | 34.51 | 7.6 | 0.03 |
| $\mathrm{I}-4$ | $33^{\circ} 46^{\prime}$ | $120^{\circ} 46^{\prime}$ | 6,780 | 403 | 2.2 | 1.6 | 34.40 | 7.7 | 0.03 |
| $\mathrm{I}-5$ | $33^{\circ} 51^{\prime}$ | $120^{\circ} 35^{\prime}$ | 5,900 | 189 | 2.3 | 1.6 | 34.6 | 7.4 | 0.03 |
| $\mathrm{II}-1$ | $34^{\circ} 06^{\prime}$ | $120^{\circ} 42^{\prime}$ | 2,340 | 197 | 5.0 | 0.4 | 34.36 | 7.5 | 0.06 |
| $\mathrm{II}-2$ | $34^{\circ} 06^{\prime}$ | $120^{\circ} 42^{\prime}$ | 2,370 | 402 | 5.0 | 0.4 | 34.36 | 7.5 | 0.06 |
| V | $34^{\circ} 06^{\prime}$ | $119^{\circ} 07^{\prime}$ | 5 | $181-763$ | $12-19$ | $3.9-6.6$ | 33.51 | 8.1 | variable |

Table 2. Participants in Test Program

| Name | Materials <br> Evaluated <br> By | Report <br> Submitted <br> to CEL |
| :--- | :---: | :---: |
| Aerojet-General Corp. | AGC | no |
| Aluminum Company of America | CEL | - |
| Allegheny Ludlum Steel Corp. | CEL | - |
| American Chain and Cable Co. | CEL | - |
| American Steel and Wire Div., U.S.S. | CEL | - |
| Anaconda American Brass Co. | CEL | - |
| Anaconda Wire and Cable Co. | AWCC | CEL |
| Armco Steel Corp. | CEL | yes |
| Baldt Anchor Chain and Forge Div., Boston Metals Co. | BTL | - |
| Bell Telephone Laboratories | CEL | yes |
| Bethlehem Steel Co. | CEL | - |
| Boeing Co. | CEL | yes |
| Brush Beryllium Co. | CEL | -- |
| Carpenter Steel Co. | CEL | - |
| E. I. Dupont Co. | CEL | - |
| Elgiloy Co. | GAC | - |
| Fansteel Metallurgical Corp. | CEL | - |
| Goodyear Aerospace Corp. | CEL | yes |
| Haynes Stellite Div., Cabot Corp. | - |  |
| Hooker Chemical Corp. | - |  |

. Continued

## Table 2. Continued.

| Name | Materials <br> Evaluated <br> By | Report <br> Submitted <br> to CEL |
| :--- | :---: | :---: |
| International Nickel Co., Inc. | INCO/CEL | yes/- |
| Joseph T. Reyerson \& Son | CEL | - |
| Kaiser Aluminum and Chemical Corp. | CEL | - |
| Kawecki Berylco Industries | CEL | - |
| Lukens Steel Co. | CEL | - |
| Menasco Manufacturing Co. | MMC | no |
| Metco Inc. | CEL | - |
| Military Consultant Service | CEL | - |
| Minnesota Mining and Manufacturing Co. | 3M | - |
| Mobay Chemical Co. | CEL | - |
| Naval Air Development Center | NADC | yes |
| Naval Air Systems Command | NASC | yes |
| Naval Electronics Laboratory | NEL | no |
| NAVFAC, Code 042 | CEL | - |
| Naval Ordnance Test Station | NOTS | no |
| Naval Pacific Missile Range | CEL | - |
| Naval Ship Research and Development Center, Annapolis Div. | NSRDC(A) | yes |
| Naval Underwater Ordnance Station | NUOS | no |
| Owens Corning Fiberglass Corp. | CEL | - |
| Reactive Metals Inc. | CEL | - |
| Republic Steel Corp. | CEL | - |
| Reynolds Metals Co. | RMC | yes |
| Scripps Institution of Oceanography | CEL | - |
| Shell Development Co. | Shell | yes |
| Standard Pressed Steel Co. | CEL | - |
| Taylor Fibre Co. | CEL | - |
| Texas Instruments, Inc. | CEL | - |
| Titanium Metals Corporation of America | CEL | - |
| TRW Space Technology Laboratories | TRW | yes |
| Tube Turns Plastic Co. | CEL | - |
| U.S. Rubber Co. | CEL | - |
| U.S. Steel Corp. | USS/CEL | yes/- |
| Valley Bolt Co. | CEL | - |
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## SECTION 2

## STEEL AND CAST IRONS

The data discussed in this section were obtained from the reports given in References 3 through 19. The chemical compositions of the alloys are given in Table 3; their surface conditions and heat treatments, if any, are given in Table 4.

The corrosion rates and types of corrosion of all the alloys are given in Table 5. Inorganic coatings were applied to some steels to evaluate their protective qualities. These coatings and their conditions are given in Table 6. Steels that were exposed in a stressed condition to determine their susceptibility to stress corrosion cracking are given in Table 7.

The effects of corrosion on the mechanical properties of many of the alloys were determined after various periods of exposure; these results are given in Table 8.

Water near the surface in the open sea is quite uniform in its composition throughout the oceans [20]; therefore, the corrosion rates of steels exposed under similar conditions in clean seawater should be comparable. The results of many investigations on the corrosion of structural steels in surface seawater at many locations throughout the world show that after a short period of exposure the corrosion rates are constant and amount to between 3 and 5 mils per year $[21,22]$. Factors which can cause differences in corrosion rates outside these limits are variations in marine fouling, contamination of the seawater near the shorelines, variations in seawater velocity, and differences in the surface water temperature.

### 2.1. IRONS AND STEELS

The corrosion rates of the irons; mild steels; highstrength low-alloy steels; high-strength steels; other alloy steels; and nickel alloy steels are given in Table 5. Analysis of the corrosion rates of these alloys shows that for all practical purposes their corrosion rates were comparable for any one duration of exposure at any one depth or at the surface. Therefore, these data were treated statistically to obtain one median value for each time of exposure and each
depth. These average data values were used to plot curves to show the general corrosion behavior to be expected from these alloys with regard to duration of exposure, depth in the ocean, and concentration of oxygen in seawater.

### 2.1.1. Duration of Exposure

The effects of the duration of exposure on the corrosion of steels in seawater at the surface and at depth are shown in Figure 3. The corrosion rates of the steels exposed in seawater at nominal depths of 2,500 and 6,000 feet in the Pacific Ocean decreased with increasing duration of exposure and were consistently lower than the surface corrosion rates by a factor of approximately 3 . The corrosion rates at the 2,500 -foot depth also were lower than those at the 6,000 -foot depth. The corrosion rates decreased asymptotically with increasing duration of exposure both at the surface and at the 6,000-foot depth.

The performance of the steels when partially embedded in the bottom sediments at the 2,500- and 6,000-foot depths is shown in Figure 4. Here, also, the average corrosion rates of the steels at the 6,000 -foot depth decreased asymptotically with increasing duration of exposure. During the initial exposures the steels corroded at faster rates in seawater than in the bottom sediments at the 6,000 -foot depth, but after approximately 2 years of exposure, their average corrosion rates were approximately the same as shown by comparing the curves in Figures 3 and 4 . Here, also, the average corrosion rates at the 2,500 -foot depth were lower than at the 6,000 -foot depth, but they increased with increasing duration of exposure.

### 2.1.2. Depth

The effect of depth of exposure in seawater on the average corrosion rates of the steels is shown in Figure 5. The variation of the concentration of oxygen in seawater with depth is also shown in Figure 5 for comparison purposes. The shape of the curve
for steels shows that corrosion of steels is not affected by depth (pressure), at least to a depth of 6,000 feet ( $2,700 \mathrm{psi}$ ) for a period of 1 year of exposure. The shape of this curve is practically identical to that of the oxygen concentration curve. The identical shape of these curves indicate that the concentration of oxygen in seawater exerts a major influence on the corrosion of steels in this environment.

### 2.1.3. Concentration of Oxygen

The effect of the variation in the concentration of oxygen in seawater on the corrosion of steels after 1 year of exposure is shown in Figure 6. The curve for the average corrosion rates of the steels after 1 year of exposure versus the concentration of oxygen is a straight line. This indicates that the corrosion of steels in seawater is proportional to the concentration of oxygen.

### 2.1.4. Nickel

The effect of the variation of the concentration of nickel on the corrosion of steels is shown in Figure 7. Variations of from 1.5 to $9 \%$ in the nickel content were ineffectual with respect to the corrosion of steel both at the surface and at depth. However, the corrosion rates in surface exposures were higher than at depth by about a factor of 7 .

### 2.1.5. Type of Corrosion

All the steel, except AISI Type 502, in general, corroded uniformly except for some slight pitting in surface seawater which was caused by fouling. The corrosion rates of AISI Type 502 steel (5\% Cr-0.5\% Mo) were erratic and higher than those of the other steels. This behavior is attributed to the broad, shallow pitting and the severe crevice corrosion caused by the chromium content of the steel.

### 2.1.6. Metallic Coatings

Zinc, aluminum, sprayed aluminum, titaniumcadmium, cadmium, copper, and nickel-coated steel specimens were exposed at depth.

A $1 \mathrm{oz} / \mathrm{sq} \mathrm{ft}$ of zinc on galvanized steel sheet exposed at a depth of 2,500 feet protected the steel
for from 3 to 4 months in the seawater and for about 7 months when partially embedded in the bottom sediments.

A $1 \mathrm{oz} / \mathrm{sq} \mathrm{ft}$ of aluminum on aluminized steel sheet exposed at a depth of 2,500 feet protected the steel for at least 13 months in the seawater and when partially embedded in the bottom sediments.

A 6 -mil-thick hot-sprayed aluminum coating over steel, which had been subsequently primed and sprayed with two coats of clear vinyl sealer, protected the underlying steel from corroding for 1,064 days at the 6,000 -foot depth. After removal from exposure the aluminum coating was dark gray and speckled with pin-point size areas of white corrosion products. Since no red rust was present, it is evident that this coating would provide added protection to the steel for an additional period of time, possibly another 3 years.

A titamium-cadmium coating on AISI 4130 steel was completely sacrificed, and the underlying steel was covered with a layer of red rust after 402 days of exposure at a depth of 2,500 feet. Such a coating would not provide satisfactory protection for seawater applications.

An electrolytically applied cadmium coating on steels, both stressed and unstressed, did not provide adequate protection for 1 year of exposure at depths of 2,500 and 6,000 feet.

Electrolytically applied copper and nickel coatings on steels, both stressed and unstressed, failed within 6 months after exposure at the 2,500 -foot depth and caused galvanic corrosion of the underlying steels.

### 2.1.7. Inorganic Coatings

A few steels were coated with selected paint coatings to determine their performance at depths in the Pacific Ocean. Table 6 shows the results of this test.

The multicoat epoxy systems exhibited, in general, satisfactory performance, while the multicoat polyurethane system behaved erratically, varying from cracked and blistered paint to no paint failures. The single-coat, zinc-rich primer coating did not afford satisfactory protection for a period of 6 months at a depth of 6,000 feet.

### 2.1.8. Cathodic Protection

Sacrificial zinc anodes were attached to AISI Type 1015 steel to determine its effectiveness in providing cathodic protection to a more noble material at these depths.

The sacrificial zinc anodes were effective in reducing the corrosion of the AISI Type 1015 steel. They provided nearly complete protection for 123 days, $50 \%$ protection during 751 days of exposure, and $30 \%$ during 1,064 days of exposure.

### 2.1.9. Galvanic Corrosion

A few galvanic couples (dissimilar metals) of AISI Type 4130 and AISI Type $4140,1 \times 7$-inch steel strips with 1 -inch-square pieces of 6061 and 7075-T6 aluminum alloys, AZ31B magnesium alloy, aluminum bronze alloy, titanium metal, and AISI Type 308 stainless steel attached to them were exposed at depths of 2,500 and 6,000 feet for 400 days to determine their compatibilities.

After 400 days of exposure at a depth of 6,000 feet aluminum alloy 6061 attached to AISI Type 4130 steel was moderately corroded with practically no corrosion of the steel; the aluminum alloy 7075-T6 was severely corroded under the same conditions. Magnesium alloy AZ31B was nearly completely sacrificed when attached to AISI Type 4130 steel, but the steel was also corroded because of the insulating layer of magnesium alloy corrosion products which accumulated at the faying surfaces of the two alloys. AISI Type 4130 steel was extensively corroded when in contact with the aluminum bronze.

After 400 days of exposure at a depth of 2,500 feet, AISI Type 4340 steel was rusted considerably from being in contact with titanium metal or AISI Type 308 stainless steel.

### 2.1.10. Stress Corrosion

Some of the steels were exposed in a stressed condition at stresses equivalent to from 30 to $75 \%$ of their respective yield strengths. The steels, stresses, depths, days of exposure, and their susceptibility to stress corrosion cracking are given in Table 7.

One-half-inch AISI Type 4140 steel bolts, heattreated to about 175,000 psi tensile strength, failed
during 400 days of exposure - one in the bottom sediment and two in the seawater at the 2,500 -foot depth. Whether these failures were due to stress corrosion or hydrogen embrittlement is not certain. Bolts of such hardness should not be used in deep-sea applications.

One nickel-plated specimen of AISI Type 4130 steel, stressed at $127,000 \mathrm{psi}$, failed during 197 days of exposure at a depth of 2,500 feet. Since no unplated specimens failed, it is possible that the failure was caused by the nickel plating. Hydrogen absorbed into the metal during the plating process could have caused hydrogen enbrittlement, which in turn caused the failure.

Some 18 Ni maraging specimens failed by stress corrosion when stressed at various levels, under different conditions, for different periods of time at different depths. These results indicate that the stress corrosion behavior of this steel is unpredictable and unreliable when used at high stress levels (above about 150,000 psi yield strength) for seawater applications.

The other steels were not susceptible to stress corrosion.

### 2.1.11. Mechanical Properties

The percent changes in the mechanical properties of the steels resulting from corrosion are given in Table 8.

The percent elongation of HSLA No. 5 in thicknesses of $1 / 4$ inch and $1 / 8$ inch was decreased by 77 and $82 \%$, respectively, after 400 days of exposure at the 2,500 -foot depth.

The mechanical properties of AISI Type 4130 steel, bare, cadmium, copper, or nickel-plated were affected after 400 days of exposure at the 2,500 -and 6,000 -foot depths. Cadmium, copper, or nickel plating on AISI Type 4340 steel also caused decreases in the mechanical properties of the steel after exposure for 400 days at the 2,500 -foot depth.

Because of pitting corrosion the elongation of AISI Type $502(5 \% \mathrm{Cr})$ steel was decreased from 13 to $38 \%$ during all exposures at both depths, except for 197 days at the 2,500 -foot depth.

The mechanical properties of the 18 Ni maraging steels were, in general, adversely affected by exposure at depth in the Pacific Ocean.

### 2.1.12. Corrosion Products

The corrosion products from some of the steels were analyzed by X-ray diffraction, spectrographic analysis, quantitative chemical analysis, and infrared spectrophotometry. The constituents found were:

$$
\begin{aligned}
& \text { Alpha iron oxide }-\mathrm{Fe}_{2} \mathrm{O}_{3} \cdot \mathrm{H}_{2} \mathrm{O} \\
& \text { Iron hydroxide }-\mathrm{F} 3(\mathrm{OH})_{2} \\
& \text { Beta iron (III) oxide hydroxide }-\mathrm{FeOOH} \\
& \text { Iron oxide hydrate }-\mathrm{Fe}_{2} \mathrm{O}_{3} \cdot \mathrm{H}_{2} \mathrm{O} \\
& \text { Significant amounts of chloride, } \\
& \text { sulfate, and phosphate ions. }
\end{aligned}
$$

### 2.2. ANCHOR CHAINS

Two types of $3 / 4$-inch-diameter anchor chains, Dilok and welded stud link, were exposed as shown in Table 9. The chain links were covered with layers of loose, flaky rust which varied from thin to thick as the time of exposure increased. Exposure for as long as 751 days did not decrease the breaking loads of the chains as shown in Table 9. In most cases there was rust in the bottoms of the sockets of the Dilok chain, indicating that seawater had penetrated the sockets. This could be a source of additional corrosion and early failure of this type of chain.

### 2.3. CAST IRONS

The corrosion rates of the cast irons are given in Table 5. Analysis of this data shows that for all practical purposes the corrosion rates of the alloy cast irons (nickel, nickel-chromium No. 1 and 2, and ductile irons No. 1 and 2) are comparable. This is also true of the austenitic cast irons. These average data values were used to plot curves to show the general corrosion behavior to be expected from these alloys with regard to duration of exposure, depth in the ocean, and concentration of oxygen in seawater.

### 2.3.1. Duration of Exposure

The effects of duration of exposure on the corrosion of cast irons in seawater at the surface and at depth are shown in Figure 3.

There was no measurable corrosion of the high silicon and the high silicon-molybdenum cast irons in seawater, either at the surface or at depth.

In all three environments (surface, 2,500-, and 6,000 -foot depths), the corrosion rates decreased with increasing duration of exposure and were consistently lower at depth than at the surface. The corrosion rates at the 2,500 -foot depth were lower than those at the 6,000 -foot depth. At the surface and at the 6,000 -foot depth the corrosion rates decreased asymptotically with increasing duration of exposure. At the 6,000 -foot depth the corrosion rates of the austenitic cast irons, for the first 400 days of exposure, were lower than those of the gray and alloy cast irons, but they were comparable after longer periods of exposure, about 1 mpy . However, at the 2,500-foot depth, the corrosion rates of the austenitic cast irons were lower than those of the alloy and gray cast irons for exposures of up to 400 days.

The corrosion of the cast irons when partially embedded in the bottom sediments is shown in Figure 4. Here again, there was no measurable corrosion of the high silicon and high siliconmolybdenum cast irons in the bottom sediment at either depth.

The other cast irons behaved essentially the same as in the seawater except that the alloy cast irons initially corroded at slower rates than in the seawater at the 6,000 -foot depth. After 2 years of exposure at the 6,000 -foot depth in both the seawater and the bottom sediments, all the steels and cast irons corroded at essentially the same rate.

In the sediments at the 2,500 -foot depth the corrosion rates of the austenitic cast irons tended to increase very slightly with increasing duration of exposure, while those of the alloy cast irons increased considerably.

### 2.3.2. Depth

The effect of depth of exposure in seawater on the average corrosion rates of the alloy and austenitic cast irons as well as those of the gray and high silicon cast irons is shown in Figure 5. The variation of the concentration of oxygen in seawater with depth is also shown in Figure 5 for comparison purposes. The shapes of the curves for the cast irons show that the corrosion of the cast irons is not directly affected by depth (pressure), at least to a depth of 6,000 feet for a period of 1 year.

### 2.3.3. Concentration of Oxygen

The effect of the variation in the concentration of oxygen in seawater on the corrosion of cast irons after 1 year of exposure is shown in Figure 6. The curves for the average corrosion rates of the gray, alloy, and austenitic cast irons versus the concentration of oxygen are essentially straight lines. This indicates that the corrosion of the cast irons in seawater is proportional to the concentration of oxygen. However, the different slopes of the curves indicate different degrees of influence, the influence being greatest on the alloy cast irons and least on the gray cast irons. Oxygen exerted no influence on the corrosion of high silicon or high silicon-molybdenum cast irons.

### 2.3.4. Type of Corrosion

All the cast irons corroded uniformly both in the seawater and in the bottom sediments. The high silicon and high silicon-molybdenum cast irons were uncorroded in any of the environments.

### 2.3.5. Mechanical Properties

The percent changes in the mechanical properties of the cast irons due to exposure in seawater are given in Table 8. The mechanical properties of the Type 4 austenitic cast iron were not affected by exposure either at the surface or at the 2,500 -foot depth. However, the mechanical properties of the D-2C austenitic cast iron were significantly lowered. About $80 \%$ of the surfaces of fracture of the D-2C specimens were black in contrast to the gray surfaces of fracture of unexposed specimens. Metallographic examinations of polished cross sections of the D-2C alloy adjacent to the surfaces of fracture showed that the alloy had been attacked by selective interdendritic corrosion. This selective corrosion was the cause of the decrease in mechanical properties of the alloy.


Figure 3. Effect of duration of exposure on corrosion of steels and cast irons.


Figure 4. Effect of duration of exposure on corrosion of steels and cast irons in bottom sediments.


Figure 5. Effect of depth on the corrosion of steels and cast irons after 1 year of exposure in seawater.


Figure 6. Effect of concentration of oxygen in seawater on the corrosion of steels and cast irons after 1 year of exposure.


Figure 7. Effect of nickel on the corrosion of steel in seawater.
Table 3. Chemical Composition of Steels and Irons, Percent by Weight

| Alloy | C. | Mn | P | S | Si | Ni | Cr | Mo | Cu | Co | Other | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Wrought iron | 0.02 | 0.06 | 0.13 | 0.01 | 0.13 | - | - | - | - | -- | 2.5 slag | CEL (4) |
| ARMCO iron | - | 0.02 | - | - | - | - | - | - | - | - | - | INCO (3) |
| AISI 1010 | 0.12 | 0.50 | 0.004 | 0.023 | 0.060 | - | - | - | - | - | - | CEL (4) |
| AISI 1010 | 0.11 | 0.52 | 0.016 | 0.024 | 0.048 | - | - | - | - | - | - | CEL (4) |
| AISI 1010 | - | 0.34 | 0.01 | - | 0.02 | 0.04 | 0.02 | - | 0.03 | - | - | INCO (3) |
| AISI 1015 | - | - | - | - | - | - | - | - | - | - | - | MEL (5) |
| Copper steel | - | 0.40 | 0.01 | - | 0.02 | 0.01 | 0.03 | - | 0.28 | - | -- | INCO (3) |
| ASTM A36 | 0.24 | 0.70 | 0.011 | 0.027 | 0.055 | - | - | - | - | - | - | CEL (4) |
| ASTM A 36 | 0.20 | 0.55 | 0.010 | 0.020 | 0.064 | - | - | - | - | - | - | CEL (4) |
| ASTM A387-D | 0.06 | 0.49 | 0.013 | 0.021 | 0.24 | - | 2.20 | 1.02 | - | - | - | CEL (4) |
| Plow steel ${ }^{b}$ |  |  |  |  |  |  |  |  |  |  |  | CEL (4) |
| HSLA No. $1^{\text {c }}$ | 0.18 | 0.86 | 0.014 | 0.023 | 0.28 | 0.05 | 0.64 | 0.18 | - | - | $\begin{aligned} & 0.047 \mathrm{~V} \\ & 0.0028 \mathrm{~B} \\ & 0.020 \mathrm{Ti} \end{aligned}$ | CEL (4) |
| HSLA No. 2 | 0.12 | 0.30 | 0.015 | 0.025 | 0.27 | 2.34 | 1.25 | 0.20 | 0.17 | - | - | CEL (4) |
| HSLA No. 3 | 0.17 | 0.28 | 0.020 | 0.018 | 0.20 | 2.96 | 1.76 | 0.40 | - | - | - | CEL (4) |
| HSLA No. 3 | 0.10 | 0.28 | 0.014 | 0.010 | 0.25 | 2.91 | 1.59 | 0.52 | - | - | - | CEL (4) |
| HSLA No. 3 | - | - | - | - | - | - | - | - | - | -- | - | Boeing (6) |
| HSLA No. 4 | 0.07 | 0.38 | 0.11 | 0.025 | 0.54 | 0.31 | 0.88 | - | 0.28 | - | - | CEL (4) |
| HSLA No. 4 | - | 0.36 | 0.08 | - | 0.41 | 0.32 | 0.72 | - | 0.38 | - | - | INCO (3) |
| HSLA No. 5 | 0.14 | 0.78 | 0.020 | 0.025 | 0.23 | 0.74 | 0.56 | 0.42 | 0.22 | - | $\begin{aligned} & 0.36 \mathrm{~V} \\ & 0.0041 \mathrm{~B} \end{aligned}$ | CEL (4) |
| HSLA No. 5 HSLA No. $5^{d}$ | 0.20 | 0.83 | - | 0.03 | - | 0.38 | 0.54 | 0.21 | 0.45 | - | 0.35 V | NADC (7) INCO (3) |
| HSLA No. 6 | 0.26 | 0.13 | 0.007 | 0.008 | 0.01 | 3.07 | 1.43 | 0.97 | - | - | 0.07 Cb | CEL (4) |
| HSLA No. 6 | - | - | - | - | - | - | - | - | - | - | - | Bocing (6) |
| HSLA No. 7 | - | 0.43 | 0.12 | - | 0.13 | 0.54 | - | - | 1.0 | - | - | INCO (3) |
| HSLA No. 8 | - | 0.24 | 0.03 | - | 0.004 | 0.47 | 0.51 | - | 0.51 | - | - | INCO (3) |
| HSLA No. 9 | - | 0.75 | 0.12 | - | 0.55 | 1.00 | 0.70 | - | 0.50 | - | - | INCO (3) |
| HSLA No. 10 | - | 0.63 | 0.01 | - | - | 0.99 | - | - | 1.42 | - | - | INCO (3) |

Table 3. Continued.

| Alloy | C | Mn | 1 | S | Si | Ni | Cr | Mo | Cu | Co | Other | Source ${ }^{a}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| HSLA No. 11 | - | 0.69 | 0.08 | - | - | 0.50 | 0.26 | - | 0.30 | - | - | INCO (3) |
| HSLA No. 12 | 0.14 | 0.26 | 0.011 | 0.009 | 0.27 | 2.60 | 1.55 | 0.46 | - | - | 0.02 V | CEL (4) |
| HSLA No. 13 | 0.23 | 1.18 | 0.04 | 0.05 | 0.30 | - | - | - | - | - | $\begin{aligned} & 0.02 \mathrm{Cb}+\mathrm{V} \\ & 0.015 \mathrm{~N} \end{aligned}$ | CEL (4) |
| HS No. $1^{e}$ | 0.11 | 0.78 | 0.008 | 0.006 | 0.29 | 5.03 | 0.56 | 0.42 | - | - | 0.05 V | CEL (4) |
| HS No. 1A | 0.12 | 0.84 | 0.003 | 0.005 | 0.32 | 4.91 | 0.56 | 0.48 | - | - | $\begin{aligned} & 0.07 \mathrm{~V} \\ & 0.021 \mathrm{Al} \\ & 0.003 \mathrm{O} \\ & 0.010 \mathrm{~N} \end{aligned}$ | CEL (4) |
| HS No. 2 | 0.002 | 0.018 | 0.004 | 0.005 | 0.05 | 12.20 | 5.07 | 3.12 | - | - | $\begin{aligned} & 0.21 \mathrm{Ti} \\ & 0.25 \mathrm{Al} \end{aligned}$ | CEL (4) |
| HS No. 2A | 0.032 | 0.073 | 0.007 | 0.013 | 0.062 | 12.04 | 5.04 | 3.39 | - | - | $\begin{aligned} & 0.24 \mathrm{Ti} \\ & 0.41 \mathrm{Al} \\ & 0.008 \mathrm{~N} \end{aligned}$ | USS (8) |
| HS No. 3 | 0.28 | 0.29 | 0.005 | 0.005 | 0.10 | 8.26 | 0.53 | 0.47 | - | 3.82 | 0.15 V | CEL (4) |
| HS No. 3A | 0.24 | 0.19 | 0.004 | 0.010 | 0.01 | 8.36 | 0.47 | 0.47 | - | 3.90 | 0.06 V | CEL (4) |
| HS No. 4 | 0.11 | 0.38 | 0.006 | 0.013 | 0.27 | 2.76 | 1.23 | 0.30 | - | - | $\begin{aligned} & 0.10 \mathrm{~V} \\ & 0.035 \mathrm{Al} \end{aligned}$ | CEL (4) |
| HS No. 5 | 0.11 | 0.06 | 0.005 | 0.005 | 0.067 | 9.91 | 2.20 | 0.98 | - | 8.00 | $\begin{aligned} & 0.003 \mathrm{Al} \\ & 0.001 \mathrm{O} \\ & 0.002 \mathrm{~N} \end{aligned}$ | CEL (4) |
| HS No. 6 | 0.18 | 0.30 | 0.007 | 0.004 | 0.02 | 9.18 | 0.77 | 1.01 | 0.13 | 4.39 | 0.09 V | CEL (4) |
| 12 Ni , maraging | - | - | - | - | - | - | - | - | - | - | - | Boeing (6) |
| 18 Ni , maraging | 0.02 | 0.10 | 0.005 | 0.007 | 0.14 | 17.92 | - | 4.78 | - | 8.75 | $\begin{aligned} & 0.003 \mathrm{~B} \\ & 0.94 \mathrm{Ti} \\ & 0.17 \mathrm{Al} \end{aligned}$ | CEL (4) |
| 18 Ni , maraging | 0.02 | 0.05 | 0.005 | 0.010 | 0.06 | 18.17 | - | 4.85 | 0.10 | 8.13 | $\begin{aligned} & 0.36 \mathrm{Ti} \\ & 0.08 \mathrm{Al} \end{aligned}$ | CEL (4) |
| 18 Ni , maraging | - | - | - | - | - | 18.0 | - | 5.0 | - | 7.0 | - | INCO (3) |
| 18 Ni , maraging | - | - | - | - | - | 18.0 | - | 4.72 | - | 7.85 | - | NADC (7) |
| 18 Ni , maraging | - | - | - | - | - | - | - | - | - | - | - | Boeing (6) |
| 1.5\% Ni | - | - | - | - | - | - | - | - | - | - | - | INCO (3) |

Table 3. Continued.

| Alloy | (: | Mn | $1 '$ | S | Si | Ni | C ${ }^{\circ}$ | Mos | Cu | Co | ()ther | Source ${ }^{a}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 3\% Ni | - | - | - | - | - | - | - | -- | - | - | - | INCO (3) |
| 5\% Ni | - | - | - | - | - | - | - | - | - | - | - | INCO (3) |
| 9\% Ni | - | - | - | - | - | - | - | - | - | - | - | INCO (3) |
| AISI 4130 | 0.29 | 0.42 | - | - | - | - | 0.90 | 0.18 | - | - | - | NADC (7) |
| AISI 4140 | 0.42 | 0.87 | 0.016 | 0.032 | 0.24 | - | 0.87 | 0.22 | - | $\sim$ | - | Shell (9) |
| AISI 4340 | 0.40 | 0.73 | 0.013 | 0.014 | 0.27 | 1.77 | 0.82 | 0.24 | - | - | - | CEL (4) |
| AISI 4340 | 0.33 | 0.65 | - | - | - | 2.0 | 0.73 | 0.26 | 0.11 | - | - | NADC (7) |
| AISI 502 | 0.06 | 0.48 | 0.020 | 0.010 | 0.33 | - | 4.75 | 0.55 | - | - | - | CEL (4) |
| AISI 502 | 0.06 | 0.5 | - | - | - | 0.4 | 5.2 | 0.5 | - | - | - | INCO (3) |
| D6AC | 0.45 | - | - | - | - | 0.6 | 1.1 | 1.1 | - | - | 0.1 V | Boeing (6) |
| Gray cast iron | - | - | - | - | - | - | - | - | - | - | - | INCO (3) |
| Ni cast iron | - | 0.68 | - | - | 2.47 | 1.56 | -- | - | - | - | - | INCO (3) |
| Ni-Cr cast iron No. 1 | - | 0.73 | $\rightarrow$ | - | 1.64 | 1.66 | 0.60 | - | - | - | - | 1NCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 2 | - | 0.86 | - | - | 1.99 | 3.22 | 0.98 | - | - | - | - | INCO (3) |
| Ductile cast iron No. 1 | - | 0.35 | - | - | 2.50 | 0.91 | - | - | - | - | - | INCO (3) |
| Ductile cast iron No. 2 | - | 0.34 | - | - | 2.24 | - | $\cdots$ | - | $-$ | - | - | INCO (3) |
| Si cast iron | - | - | - | - | 14.5 | - | - | - | - | - | - | INCO (3) |
| Si-Mo cast iron | - | - | - | - | 14.0 | - | - | 3.0 | - | -- | - | INCO (3) |
| Si-Mo cast iron | 0.4-1.0 | 0.4-1.0 | - | - | 14-17 | - | -- | 3.5 | - | - | - | NADC (7) |
| Austenitic cast iron, Type 1 | - | 1.4 | - | - | 2.05 | 15.8 | 1.79 | - | 6.71 | - | - | INCO (3) |
| Austenitic cast iron, Type 2 | - | 1.01 | - | - | 2.29 | 18.2 | 2.04 | - | - | - | - | INCO (3) |
| Austenitic cast iron, Type 3 | - | 0.6 | - | - | 1.15 | 28.4 | 2.87 | - | - | - | $\sim$ | INCO (3) |
| Austenitic cast iron, Type 4 | - | 0.56 | - | - | 5.34 | 29.7 | 4.94 | $\cdots$ | - | - | - | INCO (3) |
| Austenitic cast iron, Type 4, | 2.13 | 0.79 | - | - | 5.60 | 29.98 | 5.02 | - | 0.16 | - | - | CEL (4) |
| Austenitic cast iron D-2 | - | 0.94 | - | - | 3.0 | 21.4 | 2.26 | - | - | -- | - | INCO (3) |
| Austenitic cast iron D-2b | - | 0.96 | - | - | 2.0 | 20.8 | 3.19 | - | - | - | - | INCO (3) |
| Austenitic cast iron D-2c | 2.45 | 2.12 | 0.017 | - | 2.38 | 22.34 | 0.08 | - | - | - | - | CEL (4) |
| Austenitic cast iron D-3 | - | 0.5 | - | - | 1.83 | 29.8 | 2.70 | - | - | - | $\cdots$ | INCO (3) |


| Alloy | C | Mn | P | S | S1 | Ni | Cr | Mo | Cu | Co | Other | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Austenitic cast iron, hardenable | - | - | - | - | - | - | - | - | - | - | - | INCO (3) |
| Galvanized steel, $1 \mathrm{oz} / \mathrm{ft}^{2} \mathrm{f}$ | $\begin{aligned} & 0.15 \\ & \max \end{aligned}$ | 0.25-0.60 | $\begin{gathered} 0.040 \\ \max \end{gathered}$ | $\begin{gathered} 0.050 \\ \max \end{gathered}$ | - | - | - | - | - | - | - | CEL (4) |
| Aluminized steel, Type 2, $1.03 \mathrm{oz} / \mathrm{ft}^{2}$ | - | - | - | - | - | - | - | - | - | - | - | CEL (4) |

[^1]Table 4. Condition of the Steels, As Received

| Alloy | Condition |
| :---: | :---: |
| Wrought iron | As fabricated pipe |
| Armeo iron | Mill finish, anodically cleaned |
| AISI C1010 | Hot rolled (mill) and pickled (laboratory) |
| AISI 1015 | Grit blasted |
| ASTM A36 | Hot rolled (mill) and pickled (laboratory) |
| ASTM A387-D | Hot rolled (mill) and pickled (laboratory) |
| HSLA No. 1 | Water quenched from $1,650^{\circ} \mathrm{F}$ to $1,750^{\circ} \mathrm{F}$ and tempered at $1,100^{\circ} \mathrm{F}$ to $1,275^{\circ} \mathrm{F}$ (mill), blast cleaned (laboratory) |
| HSLA No. 2 | Hot rolled and pickled |
| HSLA No. 3 | Water quenched from $1,650^{\circ} \mathrm{F}$ and tempered at $1,150^{\circ} \mathrm{F}$ to $1,200^{\circ} \mathrm{F}$ (mill), blast cleaned (laboratory) |
| HSLA No. 4 | Hot rolled (mill) and pickled (laboratory) |
| HSLA No. 5 | Water quenched from $1,650^{\circ} \mathrm{F}$ to $1,750^{\circ} \mathrm{F}$ and tempered at $1,150^{\circ} \mathrm{F}$ to $1,275^{\circ} \mathrm{F}$ (mill), blast cleaned (laboratory) |
| HSLA No. 6 | Consumable electrode vacuum melt, hot rolled, annealed, cleaned and oiled |
| HSLA No. 12 | Quenched and tempered |
| HS No. 1 | Quenched and tempered |
| HS No. 2 | Solution annealed and aged |
| HS No. 3 | Consumable electrode vacuum melt, hot rolled, annealed, cleaned and oiled |
| AISI 4130 | Quenched and tempered |
| AISI 4140 | Austenized at $1,550^{\circ} \mathrm{F}$ for 0.7 hr , oil quenched to room temperature, tempered at $900^{\circ} \mathrm{F}$ for 1 hr , air cooled to room temperature |
| AISI 4340 (200 ksi) | Oil quenched from $1,550^{\circ} \mathrm{F}$, tempered for 1 hr at $750^{\circ} \mathrm{F}$, blast cleaned (laboratory) |
| AISI 4340 (150 ksi) | Oil quenched from $1,550^{\circ} \mathrm{F}$, tempered for 1 hr at $1,050^{\circ} \mathrm{F}$, blast cleaned (laboratory) |
| AISI 502 | Annealed and pickled, No. 1 sheet finish (mill) |
| $18 \% \mathrm{Ni}$, maraging (0.202) | Electric furnace air melt, air cast, annealed, desealed and oiled |

Table 4. Continued.

| Alloy | Condition |
| :--- | :--- |
| $18 \%$ lNi, maraging (0.082) | Electric furnace air melt, air cast, annealed, desealed and <br> oiled (mill); CEL unwelded, aged at $900^{\circ} \mathrm{F}$ for 3 hr, air <br> cooled, then welded |
| $18 \%$ Ni, maraging | Electric furnace air melt, air cast, annealed, aged at <br> $950^{\circ} \mathrm{F}$ for 3 hr , air cooled, as rolled surfaces |
| $18 \%$ Ni, maraging | Electric furnace air melt, air cast, annealed, aged at <br> $950^{\circ} \mathrm{F}$ for 3 hr , air cooled, surfaces ground to RMS-125 |
| Austenitic cast iron, Type 4 | As cast |
| Nodular austenitic cast iron, Type D-2c | As cast |
| Galvanized steel, 18 gage | 1.0 oz/ft ${ }^{2}$ |
| Aluminized steel, Type 2 | Commercial quality, $1.03 \mathrm{oz} / \mathrm{ft}^{2}$ |

Table 5. Corrosion Rates of Irons and Steels

| Alloy | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) ${ }^{\text {a }}$ |  | Type of Corrosion ${ }^{c}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $w^{b}$ | $M^{\text {b }}$ |  |  |
| Armco iron | 123 | 5,640 | 3.1 | 2.4 | U | INCO (3) |
| Armco iron | 403 | 6,780 | 1.5 | $0.5{ }^{e}$ | U | INCO (3) |
| Armco iron | 751 | 5,640 | 0.8 | 0.7 | G | INCO (3) |
| Armco iron | 1,064 | 5,300 | 0.7 | $1.9{ }^{\text {e }}$ | U | INCO (3) |
| -Armco iron | 197 | 2,340 | 1.9 | 0.9 | G | INCO (3) |
| Armeo iron | 402 | 2,370 | 1.4 | 1.4 | G | INCO (3) |
| Armeo iron | 181 | 5 | 6.9 | - | $\mathrm{U}, \mathrm{CR} f$ | INCO (3) |
| Armco iron | 366 | 5 | 7.1 | - | G | INCO (3) |
| Wrought iron | 123 | 5,640 | 2.6 | - | U | CEL (4) |
| Wrought iron | 403 | 6,780 | 1.4 | 1.2 | U | CEL (4) |
| Wrought iron | 751 | 5,640 | 0.9 | - | U | CEL (4) |
| Wrought iron | 1,064 | 5,300 | 0.6 | - | U | CEL (4) |
| Wrought iron | 197 | 2,340 | 2.0 | 1.2 | U | CEL (4) |
| Wrought iron | 402 | 2,370 | 1.5 | 1.5 | G | CEL (4) |
| Wrought iron | 181 | 5 | 5.3 | - | U | CEL (4) |
| Wrought iron | 364 | 5 | 4.8 | - | G | CEL (4) |
| Wrought iron | 723 | 5 | 4.0 | - | G | CEL (4) |
| Wrought iron | 763 | 5 | 4.8 | - | G | CEL (4) |
| AISI 1010 | 123 | 5,640 | 3.0 | 2.2 | U | CEL (4) |
| AISI 1010 | 123 | 5,640 | 2.4 | 1.5 | U | INCO (3) |
| AISI 1010 | 403 | 6,780 | 1.5 | 1.7 | U | CEL (4) |
| AISI 1010 | 403 | 6,780 | 2.3 | 0.5 | G | INCO (3) |
| AISI 1010 | 751 | 5,640 | 0.9 | - | U | CEL (4) |
| AISI 1010 | 751 | 5,640 | 0.8 | 0.6 | G | INCO (3) |
| AISI 1010 | 1,064 | 5,300 | 0.8 | - | U | CEL (4) |
| AISI 1010 | 1,064 | 5,300 | 1.1 | 1.0 | U | CEL (4) |
| AISI 1010 | 1,064 | 5,300 | 0.9 | 0.5 | U | INCO (3) |
| AISI 1010 | 197 | 2,340 | 1.5 | 1.7 | U | CEL (4) |
| AISI 1010 | 197 | 2,340 | 1.7 | 0.6 | G | INCO (3) |
| AISI 1010 | 402 | 2,370 | 1.2 | 1.1 | U | CEL (4) |
| AISI 1010 | 402 | 2,370 | 1.1 | 1.1 | G | INCO (3) |
| AISI 1010 | 181 | 5 | 9.1 | - | U | CEL (4) |
| AISI 1010 | 181 | 5 | 9.0 | - | G | INCO (3) |
| AISI 1010 | 366 | 5 | 8.0 | - | G | INCO (3) |
| AISI 1010 | 398 | 5 | 8.2 | - | U | CEL (4) |
| AISI 1010 | 588 | 5 | 8.9 | - | G | CEL (4) |
| AISI 1015 | 123 | 5,640 | 3.0 | - | U | MEL (5) |
| AISI 1015 | 751 | 5,640 | 1.7 | - | U | MEL (5) |
| AISI 1015 | 1,064 | 5,300 | 0.6 | - | U | MEL (5) |
| AISI $1015^{g}$ | 386 | 5 | 5.3 | - | G | MEL (5) |
| Copper steel | 123 | 5,640 | 1.9 | 1.6 | U | INCO (3) |
| Copper steel | 403 | 6,780 | 2.1 | 0.7 | G | INCO (3) |
| Copper steel | 751 | 5,640 | 1.4 | 0.6 | G | INCO (3) |
| Copper steel | 1,064 | 5,300 | 0.5 | 0.4 | U | INCO (3) |
| Copper steel | 197 | 2,340 | 2.0 | 0.5 | G | INCO (3) |
| Copper steel | 402 | 2,370 | 1.1 | 1.2 | U | INCO (3) |
| Copper steel | 181 | 5 | 9.0 | - | G | INCO (3) |
| Copper steel | 366 | 5 | 6.0 | - | G | INCO (3) |

Table 5. Continued

| Alloy | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) ${ }^{\text {a }}$ |  | Type of Corrosion ${ }^{\text {c }}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $w^{b}$ | $\mathrm{M}^{\text {b }}$ |  |  |
| ASTM A36 | 123 | 5,640 | 3.1 | 2.4 | U | CEL (4) |
| ASTM A36 | 403 | 6,780 | 1.5 | 1.8 | U | CEL (4) |
| ASTM A36 | 751 | 5,640 | 0.9 | - | U | CEL (4) |
| ASTM A36 | 1,064 | 5,300 | 0.6 | - | U | CEL (4) |
| ASTM A36 | 197 | 2,340 | 1.7 | 1.7 | U | CEL (4) |
| ASTM A36 | 402 | 2,370 | 1.3 | 1.5 | U | CEL (4) |
| ASTM A36 | 181 | 5 | 10.7 | - | G, C (8) | CEL (4) |
| ASTM A36 | 398 | 5 | 6.2 | - | G | CEL (4) |
| ASTM A36 | 540 | 5 | 6.3 | - | G | CEL (4) |
| ASTM A36 | 588 | 5 | 5.8 | - | G | CEL (4) |
| ASTM A387-D | 123 | 5,640 | 3.0 | 2.3 | U | CEL (4) |
| ASTM A387-D | 403 | 6,780 | 2.0 | 1.9 | U | CEL (4) |
| ASTM A387-D | 751 | 5,640 | 0.9 | 0.9 | U | CEL (4) |
| ASTM A387-D | 197 | 2,340 | 1.8 | 2.0 | U | CEL (4) |
| ASTM A387-D | 402 | 2,370 | 1.3 | 1.3 | U | CEL (4) |
| HSLA No. 1 | 123 | 5,640 | 2.9 | 2.2 | U | CEL (4) |
| HSLA No. 1 | 403 | 6,780 | 2.0 | 1.2 | U | CEL (4) |
| HSLA No. 1 | 751 | 5,640 | 0.9 | - | U | CEL (4) |
| HSLA No. 1 | 1,064 | 5,300 | 0.6 | - | U | CEL (4) |
| HSLA No. 1 | 1,064 | 5,300 | 0.6 | 0.7 | U | CEL (4) |
| HSLA No. 1 | 197 | 2,340 | 1.4 | 1.4 | U | CEL (4) |
| HSLA No. 1 | 402 | 2,370 | 1.0 | 1.0 | U | CEL (4) |
| HSLA No. 1 | 181 | 5 | 9.7 | - | G, P | CEL (4) |
| HSLA No. 1 | 398 | 5 | 5.2 | - | G, P | CEL (4) |
| HSLA No. 1 | 588 | 5 | 4.7 | - | G, P | CEL (4) |
| HSLA No. 2 | 123 | 5,640 | 4.7 | 4.3 | U | CEL (4) |
| HSLA No. 2 | 403 | 6,780 | 2.1 | 2.2 | U | CEL (4) |
| HSLA No. 2 | 751 | 5,640 | 0.9 | - | U | CEL (4) |
| HSLA No. 2 | 1,064 | 5,300 | 0.5 | - | U | CEL (4) |
| HSLA No. 2 | 197 | 2,340 | 1.4 | 1.4 | U | CEL (4) |
| HSLA No. 2 | 197 | 2,340 | 1.1 | - | G | Boeing (6) |
| HSLA No. 2 | 402 | 2,370 | 1.3 | 1.1 | U | CEL (4) |
| HSLA No. 2 | 181 | 5 | 6.8 | - | G, P | CEL (4) |
| HSLA No. 2 | 398 | 5 | 4.5 | - | U, P | CEL (4) |
| HSLA No. 2 | 540 | 5 | 4.4 | - | G, P | CEL (4) |
| HSLA No. 3 | 1,064 | 5,300 | 0.7 | - | U | CEL (4) |
| HSLA No. 4 | 123 | 5,640 | 3.6 | - | U | CEL (4) |
| HSLA No. 4 | 123 | 5,640 | 4.3 | 1.8 | U, C (9) | INCO (3) |
| HSLA No. 4 | 403 | 6,780 | 3.3 | 2.3 | U | CEL (4) |
| HSLA No. 4 | 403 | 6,780 | 2.1 | 0.4 | G | INCO (3) |
| HSLA No. 4 | 751 | 5,640 | 1.2 | - | U | CEL (4) |
| HSLA No. 4 | 751 | 5,640 | 0.9 | 0.7 | G | INCO (3) |
| HSLA No. 4 | 1,064 | 5,300 | 0.3 | - | U | CEL (4) |
| HSLA No. 4 | 1,064 | 5,300 | 1.1 | - | U | CEL (4) |
| HSLA No. 4 | 1,064 | 5,300 | 0.6 | 0.6 | U | INCO (3) |
| HSLA No. 4 | 197 | 2,340 | 1.4 | 0.9 | U | CEL (4) |
| HSLA No. 4 | 197 | 2,340 | 2.2 | 0.7 | G | INCO (3) |
| HSLA No. 4 | 402 | 2,370 | 1.1 | 1.1 | G | CEL (4) |

Continued

Table 5. Continued

| Alloy | Exposure (day) | Depth <br> (ft) | Currosion Rate (mpy) ${ }^{\text {a }}$ |  | Type of Corrosion ${ }^{\text {c }}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $\mathrm{w}^{\text {b }}$ | $\mathrm{M}^{\text {b }}$ |  |  |
| HSLA No. 4 | 402 | 2,370 | 1.3 | 1.0 | G | $1 \mathrm{NCO}(3)$ |
| HSLA No. 4 | 181 | 5 | 11.0 | - | G | INCO (3) |
| HSLA No. 4 | 366 | 5 | 8.0 | - | G | INCO (3) |
| HSLA No. 5 | 123 | 5,640 | 3.1 | 1.6 | U | CEL (4) |
| HSLA No. 5 | 123 | 5,640 | 6.0 | 3.5 | E | INCO (3) |
| HSLA No. 5 | 403 | 6,780 | 2.7 | 1.8 | U | CEL (4) |
| HSLA No. 5 | 403 | 6,780 | 7.4 | 0.2 | S-E, P (3) | INCO (3) |
| HSLA No. $5^{\text {b }}$ | 403 | 6,780 | 3.7 | - | G | NADC (7) |
| HSLA No. 5 | 751 | 5,640 | 1.4 | 0.9 | U | CEL (4) |
| HSLA No. 5 | 751 | 5,640 | 3.1 | 3.2 | G, S-E | INCO (3) |
| HSLA No. 5 | 1,064 | 5,300 | 0.9 | - | U | CEL (4) |
| HSLA No. 5 | 1,064 | 5,300 | 0.7 | 1.0 | U | CEL (4) |
| HSLA No. 5 | 1,064 | 5,300 | 0.9 | 1.0 | U | INCO (3) |
| HSLA No. 5 | 197 | 2,340 | 1.4 | 1.5 | U | CEL (4) |
| HSLA No. 5 | 197 | 2,340 | 3.3 | 0.9 | E, I-P | INCO (3) |
| HSLA No. 5 | 197 | 2,340 | 2.5 | - | U | NADC (7) |
| HSLA No. 5 | 402 | 2,370 | 1.1 | 1.3 | U | CEL (4) |
| HSLA No. 5 | 402 | 2,370 | 1.4 | 1.3 | U, G | INCO (3) |
| HSLA No. $5^{\text {b }}$ | 402 | 2,370 | 1.1 | - | G | NADC (7) |
| HSLA No. 5 | 181 | 5 | 8.9 | - | U | CEL (4) |
| HSLA No. 5 | 181 | 5 | 11.0 | - | G | INCO (3) |
| HSLA No. 5 | 366 | 5 | 8.0 | - | G | INCO (3) |
| HSLA No. 5 | 398 | 5 | 6.0 | - | G, P | CEL (4) |
| HSLA No. 5 | 540 | 5 | 5.4 | - | G, P | CEL (4) |
| HSLA No. 6 | 197 | 2,340 | 1.4 | - | G | Boeing (6) |
| HSLA No. 6 | 402 | 2,370 | 0.9 | 0.9 | U | CEL (4) |
| HSLA No. 7 | 123 | 5,640 | 3.5 | 2.1 | $\mathrm{C}(4), \mathrm{U}$ | INCO (3) |
| HSLA No. 7 | 403 | 6,780 | 1.5 | 0.3 | G | INCO (3) |
| HSLA No. 7 | 751 | 5,640 | 0.8 | 1.3 | G | INCO (3) |
| HSLA No. 7 | 1,064 | 5,300 | 0.8 | 0.6 | U | INCO (3) |
| HSLA No. 7 | 197 | 2,340 | 2.3 | 0.6 | G | INCO (3) |
| HSLA No. 7 | 402 | 2,370 | 1.4 | 1.1 | G | INCO (3) |
| HSLA No. 7 | 181 | 5 | 11.0 | - | G | INCO (3) |
| HSLA No. 7 | 366 | 5 | 8.0 | - | G | INCO (3) |
| HSLA No. 8 | 123 | 5,640 | 3.8 | 2.3 | U | INCO (3) |
| HSLA No. 8 | 403 | 6,780 | 2.3 | 0.3 | G | INCO (3) |
| HSLA No. 8 | 751 | 5,640 | 1.2 | 0.8 | G | INCO (3) |
| HSLA No. 8 | 1,064 | 5,300 | 0.7 | 0.5 | U | INCO (3) |
| HSLA No. 8 | 197 | 2,340 | 1.9 | 0.7 | G | INCO (3) |
| HSLA No. 9 | 123 | 5,640 | 4.3 | 2.1 | C (10) | INCO (3) |
| HSLA No. 9 | 403 | 6,780 | 2.5 | 0.3 | G | INCO (3) |
| HSLA No. 9 | 751 | 5,640 | 1.4 | 1.0 | G | INCO (3) |
| HSLA No. 9 | 1,064 | 5,300 | 0.6 | 0.5 | U | INCO (3) |
| HSLA No. 9 | 197 | 2,340 | 1.6 | 0.6 | G | INCO (3) |
| HSLA No. 10 | 123 | 5,640 | 4.1 | 2.5 | C (9), U | INCO (3) |
| HSLA No. 10 | 403 | 6,780 | 1.8 | 0.5 | G | INCO (3) |
| HSLA No. 10 | 751 | 5,640 | 0.9 | 1.1 | G | INCO (3) |

Continued

Table 5. Continued

| Alloy | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) ${ }^{\text {a }}$ |  | Type of Corrosion ${ }^{c}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $w^{b}$ | $M^{b}$ |  |  |
| HSLA No. 10 | 1,064 | 5,300 | 0.9 | 0.6 | U | INCO (3) |
| HSLA No. 10 | 197 | 2,340 | 2.1 | 0.8 | G | INCO (3) |
| HSLA No. 10 | 402 | 2,370 | 1.5 | 1.2 | G | INCO (3) |
| HSLA No. 10 | 181 | 5 | 11.0 | - | G | INCO (3) |
| HSLA No. 10 | 366 | 5 | 8.0 | - | G | INCO (3) |
| HSLA No. 11 | 123 | 5,640 | 3.4 | 1.7 | U | INCO (3) |
| HSLA No. 11 | 403 | 6,780 | 2.4 | 0.4 | G | INCO (3) |
| HSLA No. 11 | 751 | 5,640 | 1.2 | 0.8 | G | INCO (3) |
| HSLA No. 11 | 1,064 | 5,300 | 0.7 | 0.5 | U | INCO (3) |
| HSLA No. 11 | 197 | 2,340 | 1.8 | 0.6 | G | INCO (3) |
| HSLA No. 12 | 181 | 5 | 8.5 | - | U, P | CEL (4) |
| HSLA No. 12 | 398 | 5 | 4.2 | - | G, P | CEL (4) |
| HSLA No. 12 | 540 | 5 | 4.9 | - | G, P | CEL (4) |
| HSLA No. 12 | 588 | 5 | 4.3 | - | G, P | CEL (4) |
| HS No. 1 | 189 | 5,900 | 2.7 | 1.8 | U | CEL (4) |
| HS No. $1^{2}$ | 189 | 5,900 | 2.7 | 1.8 | U | CEL (4) |
| HS No. $1^{j, k}$ | 189 | 5,900 | 2.6 | 1.6 | U | CEL (4) |
| HS No. 1 | 181 | 5 | 9.9 | - | U | CEL (4) |
| HS No. I | 398 | 5 | 4.7 | - | G, P | CEL (4) |
| HS No. 1 | 540 | 5 | 4.5 | - | G, P | CEL (4) |
| HS No. 1 | 588 | 5 | 4.2 | - | G, P | CEL (4) |
| HS No. 2 | 403 | 6,780 | 14.5 | $9.0 \mathrm{avg}^{l}$ | P | USS (8) |
| HS No. 2, welded | 403 | 6,780 |  |  |  |  |
| Base metal |  |  | 13.5 | 9.0 avg | P | USS (8) |
| Weld metal |  |  | 18.1 | 3.5 avg | P | USS (8) |
| HS No. 2 | 197 | 2,340 | 20.4 | 16.7 avg | P | USS (8) |
| HS No. 2, welded | 197 | 2,340 |  |  |  |  |
| Base metal |  |  | $29.6$ | $25.9 \mathrm{avg}_{\mathrm{l}}^{\mathrm{l}}$ | P | USS (8) |
| Weld metal |  |  |  | $38.9 \mathrm{avg}$ | P | USS (8) |
| HS No. 2 | 181 | 5 | 8.2 | - | U | CEL (4) |
| HS No. 2 | 398 | 5 | 3.5 | - | G, P | CEL (4) |
| HS No. 2 | 588 | 5 | 3.3 | - | G, P | CEL (4) |
| HS No. 3 | 402 | 2,370 | 1.7 | 1.4 | G | CEL (4) |
| HS No. 3 | 398 | 5 | 5.0 | - | U, P | CEL (4) |
| HS No. 3 | 540 | 5 | 3.8 | - | $\mathrm{G}, \mathrm{P}$ | CEL (4) |
| HS No. 3 | 588 | 5 | 4.6 | - | $\mathrm{G}, \mathrm{P}$ | CEL (4) |
| HS No. ${ }^{4}$ | 189 | 5,900 | 2.9 | 1.8 | U | CEL (4) |
| HS No. $4^{1}$ | 189 | 5,900 | 2.3 | 1.5 | U | CEL (4) |
| HS No. $4^{j, k}$ | 189 | 5,900 | 2.5 | 1.7 | U, P | CEL (4) |
| HS No. ${ }^{\text {i }}$ | 189 | 5,900 | 2.3 | 1.9 | U | CEL (4) |
| HS No. $5^{i}$ | 189 | 5,900 | 2.0 | 1.7 | U | CEL (4) |
| HS No. $5^{j}$ | 189 | 5,900 | 1.8 | 1.6 | U | CEL (4) |
| HS No. 6 | 189 | 5,900 | 2.5 | 1.6 | U | CEL (4) |
| HS No. $6^{2}$ | 189 | 5,900 | 2.8 | 1.5 | U | CEL (4) |
| HS No. $6^{\text {J }}$ | 189 | 5,900 | 2.9 | 2.7 | U | CEL (4) |

Continued

Table 5. Continued

| Alloy | Exposure (day) | Depth (ft) | Corrosion Rate (mpy) ${ }^{\text {a }}$ |  | Type of Corrosion ${ }^{\text {C }}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $w^{b}$ | $M^{b}$ |  |  |
| 12\% Ni, maraging | 197 | 2,340 | 2.4 | - | G | Boeing (6) |
| 18\% Ni, maraging | 123 | 5,640 | 3.6 | - | U | NADC (7) |
| 18\% Ni, maraging | 189 | 5,900 | 2.2 | 1.7 | U | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging | 403 | 6,780 | 1.6 | - | U | NADC (7) |
| $18 \% \mathrm{Ni}$, maraging | 197 | 2,340 | 1.6 | - | G | Boeing (6) |
| 18\% Ni, maraging | 197 | 2,340 | 2.9 | - | U | NADC (7) |
| $18 \% \mathrm{Ni}$, maraging | 402 | 2,370 | 1.3 | - | U | NADC (7) |
| 18\% Ni, maraging | 402 | 2,370 | 1.3 | 0.9 | G | CEL (4) |
| 18\% Ni, maraging | 402 | 2,370 | 1.5 | 0.8 | G | INCO (3) |
| $18 \% \mathrm{Ni}$, maraging (as rolied) | 402 | 2,370 | 1.4 | 1.3 | G | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging (machined) | 402 | 2,370 | 1.3 | 1.2 | G | CEL (4) |
| 18\% Ni, maraging | 402 | 2,370 | 3.5 | 2.6 | G | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging | 402 | 2,370 | 2.8 | 1.7 | G | CEL (4) |
| 18\% Ni, maraging | 181 | 5 | 5.4 | - | U, P | CEL (4) |
| 18\% Ni, maraging | 181 | 5 | 10.0 | - | P | INCO (3) |
| $18 \% \mathrm{Ni}$, maraging ${ }^{m}$ | 181 | 5 | 5.8 | - | U | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging ${ }^{\text {b }}$ | 181 | 5 | 5.1 | - | U | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging | 366 | 5 | 7.0 | - | P | INCO (3) |
| 18\% Ni, maraging | 398 | 5 | 3.0 | - | U, P | CEL (4) |
| 18\% Ni, maraging | 588 | 5 | 3.1 | - | $\mathrm{G}, \mathrm{C}, \mathrm{P}$ | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging ${ }^{m}$ | 364 | 5 | 4.0 | - | G, P | CEL (4) |
| 18\% Ni, maraging ${ }_{m}^{m}$ | 723 | 5 | 3.5 | - | G, P | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging ${ }_{b}^{m}$ | 763 | 5 | 4.1 | - | G | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging ${ }^{\text {b }}$ | 364 | 5 | 4.0 | - | P, WB(G) | CEL (4) |
| 18\% Ni, maraging ${ }_{b}$ | 723 | 5 | 3.3 | - | G, P | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging ${ }^{\prime}$ | 763 | 5 | 3.9 | - | G | CEL (4) |
| 1.5\% Ni steel | 123 | 5,640 | 3.5 | 2.7 | U | INCO (3) |
| 1.5\% Ni steel | 403 | 6,780 | 1.7 | 0.8 | G | INCO (3) |
| 1.5\% Ni steel | 751 | 5,640 | 1.0 | 0.5 | G | INCO (3) |
| 1.5\% Ni steel | 1,064 | 5,300 | 0.7 | 0.7 | U | INCO (3) |
| 1.5\% Ni steel | 197 | 2,340 | 1.9 | 0.5 | U | INCO (3) |
| 1.5\% Ni stee! | 402 | 2,370 | 1.5 | 1.2 | U | INCO (3) |
| 1.5\% Ni steel | 181 | 5 | 11.0 | - | G | INCO (3) |
| 1.5\% Ni steel | 366 | 5 | 8.0 | - | G | INCO (3) |
| $3 \% \mathrm{Ni}$ steel | 123 | 5,640 | 3.4 | 3.0 | U | INCO (3) |
| 3\% Ni steel | 403 | 6,780 | 1.9 | 0.4 | C (2), G | INCO (3) |
| 3\% Ni steel | 751 | 5,640 | 0.9 | 0.9 | G | INCO (3) |
| 3\% Ni steel | 1,064 | 5,300 | 0.9 | 0.6 | U | INCO (3) |
| 3\% Ni steel | 197 | 2,340 | 1.7 | 0.4 | G | INCO (3) |
| 3\% Ni steel | 402 | 2,370 | 1.3 | 1.0 | G | INCO (3) |
| 3\% Ni steel | 181 | 5 | 11.0 | - | G | INCO (3) |
| 3\% Ni steel | 366 | 5 | 8.0 | - | G | INCO (3) |
| 5\% Ni steel | 123 | 5,640 | 2.8 | 2.8 | U | INCO (3) |
| 5\% Ni steel | 403 | 6,780 | 2.8 | 0.4 | C (6), G | INCO (3) |
| 5\% Ni steel | 751 | 5,640 | 1.1 | 0.8 | G | INCO (3) |
| 5\% Ni steel | 1,064 | 5,300 | 0.7 | 0.5 | U | INCO (3) |
| 5\% Ni steel | 197 | 2,340 | 1.7 | 0.4 | G | INCO (3) |
| 5\% Ni steel | 402 | 2,370 | 1.3 | 1.1 | U | INCO (3) |

Continued

Table 5. Continued

| Alloy | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) ${ }^{\text {a }}$ |  | Type of Corrosion ${ }^{c}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $w^{b}$ | $\mathrm{M}^{\text {b }}$ |  |  |
| 5\% Ni steel | 181 | 5 | 8.0 | - | G | INCO (3) |
| 5\% Ni steel | 366 | 5 | 7.0 | - | G | INCO (3) |
| 9\% Ni steel | 123 | 5,640 | 5.6 | 5.6 | U | INCO (3) |
| 9\% Ni steel | 403 | 6,780 | 2.9 | 0.5 | C (9), G | INCO (3) |
| 9\% Ni steel | 751 | 5,640 | 1.1 | 4.5 | G | INCO (3) |
| 9\% Ni steel | 1,064 | 5,300 | 4.6 | 1.2 | U | INCO (3) |
| 9\% Ni steel | 197 | 2,340 | 1.9 | 0.4 | G | INCO (3) |
| 9\% Ni steel | 402 | 2,370 | 1.6 | 1.3 | G | INCO (3) |
| 9\% Ni steel | 181 | 5 | 10.0 | - | I-P | INCO (3) |
| 9\% Ni steel | 366 | 5 | 8.0 | - | G | INCO (3) |
| AISI 4130 (100 ksi) | 123 | 5,640 | 2.3 | - | U | NADC (7) |
| AISI 4130 (160 ksi) | 123 | 5,640 | 3.1 | - | U | NADC (7) |
| AISI 4130 (100 ksi) | 403 | 6,780 | 2.7 | - | U | NADC (7) |
| AISI 4130 (100 ksi) | 751 | 5,640 | 2.2 | - | U | NADC (7) |
| AISI 4130 (100 ksi) | 1,064 | 5,300 | 1.3 | - | U | NADC (7) |
| AISI 4130 (100 ksi) | 197 | 2,340 | 2.3 | - | U | NADC (7) |
| AISI 4130 (100 ksi) | 402 | 2,370 | 1.0 | - | U | NADC (7) |
| AISI 4140 | 402 | 2,370 | 1.4 | 1.6 | G, C (12) | Shell (9) |
| AISI 4340 (150 ksi) | 123 | 5,640 | 2.7 | $-$ | U | CEL (4) |
| AISI 4340 (150 ksi) | 403 | 6,780 | 2.2 | 1.7 | U | CEL (4) |
| AISI 4340 (150 ksi) | 403 | 6,780 | 2.2 | 1.5 | U | CEL (4) |
| AISI 4340 | 403 | 6,780 | 1.6 | - | U | NADC (7) |
| AISI 4340 (150 ksi) | 751 | 5,640 | 0.8 | - | U | CEL (4) |
| AISI 4340 (150 ksi) | 197 | 2,340 | 1.9 | 1.3 | U | CEL (4) |
| AISI 4340 (150 ksi) | 197 | 2,340 | 1.6 | 1.8 | U | CEL (4) |
| AISI 4340 | 197 | 2,340 | 4.1 | - | U | NADC (7) |
| AISI 4340 | 402 | 2,370 | 1.0 | - | U | NADC (7) |
| AISI 4340 (150 ksi) | 402 | 2,370 | 1.2 | 1.3 | U | CEL (4) |
| AISI 4340 (200 ksi) | 123 | 5,640 | 2.8 | - | U | CEL (4) |
| AISI 4340 (200 ksi) | 403 | 6,780 | 2.0 | 1.9 | U | CEL (4) |
| AISI 4340 (200 ksi) | 403 | 6,780 | 2.0 | 1.8 | U | CEL (4) |
| AISI 4340 (200 ksi) | 751 | 5,640 | 0.9 | - | U | CEL (4) |
| AISI 4340 (200 ksi) | 197 | 2,340 | 1.4 | 1.4 | U | CEL (4) |
| AISI 4340 (200 ksi) | 197 | 2,340 | 2.1 | 2.2 | U | CEL (4) |
| AISI 4340 (200 ksi) | 402 | 2,370 | 1.4 | 1.4 | U | CEL (4) |
| AISI 502 | 123 | 5,640 | 5.9 | 4.3 | P, C (21) | CEL (4) |
| AISI 502 | 123 | 5,640 | 4.3 | 4.6 | P (12), E, C (24) | INCO (3) |
| AISI 502 | 403 | 6,780 | 2.3 | 2.6 | C (22) | CEL (4) |
| AISI 502 | 403 | 6,780 | 13.2 | 0.4 | $\mathrm{P}, \mathrm{C}(35), \mathrm{G}$ | INCO (3) |
| AISI 502 | 751 | 5,640 | 2.8 | - | E, C (50), P (36) | CEL (4) |
| AISI 502 | 751 | 5,640 | 4.4 | 2.5 | C (PR) | INCO (3) |
| AISI 502 | 1,064 | 5,300 | 2.6 | - | C | CEL (4) |
| AISI 502 | 1,064 | 5,300 | 1.9 | 1.7 | P, C | CEL (4) |
| AISI 502 | 1,064 | 5,300 | 3.0 | 1.1 | C (PR) | INCO (3) |

Table 5. Continued

| Alloy | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) ${ }^{\text {a }}$ |  | Type of Corrosion ${ }^{\text {c }}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $w^{b}$ | $\mathrm{M}^{\text {b }}$ |  |  |
| AISI 502 | 197 | 2,340 | 1.4 | 1.2 | P, C (23) | CEL (4) |
| AISI 502 | 197 | 2,340 | 3.1 | 0.2 | E, C (20) | INCO (3) |
| AISI 502 | 402 | 2,370 | 0.8 | 0.6 | P, C (16) | CEL (4) |
| AISI 502 | 402 | 2,370 | 3.1 | 0.1 | $\mathrm{P}, \mathrm{C}$ (PR) | INCO (3) |
| AISI 502 | 181 | 5 | 7.0 | - | P (18) | CEL (4) |
| AISI 502 | 181 | 5 | 13.0 | - | G | INCO (3) |
| AISI 502 | 366 | 5 | 8.0 | - | G | INCO (3) |
| AISI 502 | 398 | 5 | 4.4 | - | G, P (30) | CEL (4) |
| AISI 502 | 540 | 5 | 4.1 | - | $\mathrm{G}, \mathrm{P}$ | CEL (4) |
| D6AC | 197 | 2,340 | 1.5 | - | G | Boeing (6) |
| Gray cast iron | 123 | 5,640 | 4.2 | 3.0 | U | INCO (3) |
| Gray cast iron | 403 | 6,780 | 1.8 | 1.3 | U | INCO (3) |
| Gray cast iron | 751 | 5,640 | 1.2 | 1.0 | G | INCO (3) |
| Gray cast iron | 1,064 | 5,300 | 0.8 | 0.5 | U | INCO (3) |
| Gray cast iron | 197 | 2,340 | 2.0 | 0.3 | G | INCO (3) |
| Gray cast iron | 402 | 2,370 | 1.7 | 2.0 | U | INCO (3) |
| Gray cast iron | 366 | 5 | 2.6 | - | G | INCO (3) |
| Ni cast iron | 123 | 5,640 | 4.4 | 3.4 | U | INCO (3) |
| Ni cast iron | 403 | 6,780 | 2.9 | 1.5 | U, M | INCO (3) |
| Ni cast iron | 751 | 5,640 | 1.4 | 1.1 | G | INCO (3) |
| Ni cast iron | 1,064 | 5,300 | 0.9 | 1.5 | G | INCO (3) |
| Ni cast iron | 197 | 2,340 | 2.2 | 0.3 | G | INCO (3) |
| Ni cast iron | 402 | 2,370 | 1.5 | 1.5 | U | INCO (3) |
| Ni cast iron | 181 | 5 | 8.5 | - | U | INCO (3) |
| Ni cast iron | 366 | 5 | 7.6 | - | G | INCO (3) |
| Ni-Cr cast iron No. 1 | 123 | 5,640 | 4.3 | 3.3 | U | INCO (3) |
| Ni -Cr cast iron No. 1 | 403 | 6,780 | 1.7 | 1.2 | U | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 1 | 751 | 5,640 | 1.3 | 0.9 | G | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 1 | 1,064 | 5,300 | 0.8 | 0.7 | U | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 1 | 197 | 2,340 | 1.9 | 0.3 | G | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 1 | 402 | 2,370 | 1.8 | 1.4 | U | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 1 | 181 | 5 | 6.7 | - | U | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 1 | 366 | 5 | 5.2 | - | U | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 2 | 123 | 5,640 | 4.3 | 3.7 | U | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 2 | 403 | 6,780 | 1.8 | 1.4 | U | INCO (3) |
| Ni - Cr cast iron No. 2 | 751 | 5,640 | 1.3 | 1.1 | G | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast íron No. 2 | 1,064 | 5,300 | 0.7 | 0.7 | U | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 2 | 197 | 2,340 | 1.9 | 0.3 | G | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 2 | 402 | 2,370 | 1.8 | 1.1 | U | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 2 | 181 | 5 | 8.5 | - | U | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$ cast iron No. 2 | 366 | 5 | 4.9 | - | G | INCO (3) |
| Ductile cast iron No. 1 | 123 | 5,640 | 3.1 | 3.0 | U | INCO (3) |
| Ductile cast iron No. 1 | 403 | 6,780 | 3.4 | 1.0 | G | INCO (3) |
| Ductile cast iron No. 1 | 751 | 5,640 | 1.0 | 0.9 | G | INCO (3) |
| Ductile cast iron No. 1 | 1,064 | 5,300 | 0.6 | 0.7 | U | INCO (3) |
| Ductile cast iron No. 1 | 197 | 2,340 | 1.9 | 0.7 | G | INCO (3) |
| Ductile cast iron No. 1 | 402 | 2,370 | 1.9 | 1.7 | U | INCO (3) |

Table 5. Continued

| Alloy | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) ${ }^{u}$ |  | Type of Corrosion ${ }^{\text {c }}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $w^{b}$ | $\mathrm{m}^{\text {b }}$ |  |  |
| Ductile cast iron No. 1 | 181 | 5 | 10.0 | - | U | INCO (3) |
| Ductile cast iron No. 1 | 366 | 5 | 6.2 | - | CR (24) | INCO (3) |
| Ductile cast iron No. 2 | 123 | 5,640 | 3.9 | 2.9 | U | INCO (3) |
| Ductile cast iron No. 2 | 403 | 6,780 | 2.9 | 0.9 | G, M | INCO (3) |
| Ductile cast iron No. 2 | 751 | 5,640 | 1.0 | 0.8 | G | INCO (3) |
| Ductile cast iron No. 2 | 1,064 | 5,300 | 0.8 | 0.6 | U | INCO (3) |
| Ductile cast iron No. 2 | 197 | 2,340 | 2.3 | 0.5 | G | INCO (3) |
| Ductile cast iron No. 2 | 402 | 2,370 | 1.8 | 1.4 | U | INCO (3) |
| Ductile cast iron No. 2 | 181 | 5 | 10.0 | - | U | INCO (3) |
| Ductile cast iron No. 2 | 366 | 5 | 7.1 | - | G | INCO (3) |
| Silicon cast iron | 123 | 5,640 | $<0.1$ | $<0.1$ | NC | INCO (3) |
| Silicon cast iron | 403 | 6,780 | <0.1 | $<0.1$ | NC | INCO (3) |
| Silicon cast iron | 751 | 5,640 | <0.1 | $<0.1$ | NC | INCO (3) |
| Silicon cast iron | 1,064 | 5,300 | <0.1 | $<0.1$ | NC | INCO (3) |
| Silicon cast iron | 197 | 2,340 | <0.1 | $<0.1$ | NC | INCO (3) |
| Silicon cast iron | 402 | 2,370 | $<0.1$ | $<0.1$ | NC | INCO (3) |
| Silicon cast iron | 181 | 5 | <0.1 | - | E | INCO (3) |
| Silicon cast iron | 366 | 5 | <0.1 | - | ET | INCO (3) |
| Si-Mo cast iron | 123 | 5,640 | <0.1 | <0.1 | NC | INCO (3) |
| Si-Mo cast iron | 403 | 6,780 | <0.1 | $<0.1$ | NC | INCO (3) |
| Si-Mo cast iron | 403 | 6,780 | 0.1 | - | U | NADC (7) |
| Si-Mo cast iron | 751 | 5,640 | $<0.1$ | $<0.1$ | NC | INCO (3) |
| Si-Mo cast iron | 1,064 | 5,300 | $<0.1$ | <0.1 | NC | INCO (3) |
| Si-Mo cast iron | 197 | 2,340 | <0.1 | <0.1 | NC | INCO (3) |
| Si-Mo cast iron | 402 | 2,370 | $<0.1$ | $<0.1$ | NC | INCO (3) |
| Si-Mo cast iron | 402 | 2,370 | <0.1 | - | U | NADC (7) |
| Si -Mo cast iron | 181 | 5 | <0.1 | - | NC | INCO (3) |
| Si-Mo cast iron | 366 | 5 | $<0.1$ | - | ET | INCO (3) |
| Austenitic cast iron, Type 1 | 123 | 5,640 | 2.4 | 2.4 | G | INCO (3) |
| Austenitic cast iron, Type 1 | 403 | 6,780 | 1.0 | 0.2 | U | INCO (3) |
| Austenitic cast iron, Type 1 | 751 | 5,640 | 0.5 | 0.8 | G | INCO (3) |
| Austenitic cast iron, Type 1 | 1,064 | 5,300 | 0.5 | 0.6 | U | INCO (3) |
| Austenitic cast iron, Type 1 | 197 | 2,340 | 1.8 | 1.1 | G | INCO (3) |
| Austenitic cast iron, Type 1 | 402 | 2,370 | 1.5 | 0.6 | U | INCO (3) |
| Austenitic cast iron, Type 1 | 181 | 5 | 4.1 | - - | U | INCO (3) |
| Austenitic cast iron, Type 1 | 366 | 5 | 2.7 | - | U | INCO (3) |
| Austenitic cast iron, Type 2 | 123 | 5,640 | 2.4 | 2.2 | G | INCO (3) |
| Austenitic cast iron, Type 2 | 403 | 6,780 | 2.2 | 0.2 | U | INCO (3) |
| Austenitic cast iron, Type 2 | 751 | 5,640 | 1.5 | 1.6 | G | INCO (3) |
| Austenitic cast iron, Type 2 | 1,064 | 5,300 | 1.4 | 1.0 | G | INCO (3) |
| Austenitic cast iron, Type 2 | 197 | 2,340 | 1.3 | 1.1 | G | INCO (3) |
| Austenitic cast iron, Type 2 | 402 | 2,370 | 1.1 | 0.7 | U | INCO (3) |
| Austenitic cast iron, Type 2 | 181 | 5 | 5.8 | - | U | INCO (3) |
| Austenitic cast iron, Type 2 | 366 | 5 | 2.9 | - | U | INCO (3) |
| Austenitic cast iron, Type 3 | 123 | 5,640 | 1.9 | 1.7 | G | INCO (3) |
| Austenitic cast iron, Type 3 | 403 | 6,780 | 1.8 | <0.1 | U | INCO (3) |
| Austenitic cast iron, Type 3 | 751 | 5,640 | 1.9 | 1.9 | G | INCO (3) |

Continued

Table 5. Continued

| Alloy | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) ${ }^{\text {a }}$ |  | Type of Corrosion ${ }^{\text {c }}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $w^{b}$ | $M^{b}$ |  |  |
| Austenitic cast iron, Type 3 | 1,064 | 5,300 | 1.2 | 0.8 | U | INCO (3) |
| Austenitic cast iron, Type 3 | 197 | 2,340 | 0.8 | 0.7 | G | INCO (3) |
| Austenitic cast iron, Type 3 | 402 | 2,370 | 0.6 | 0.7 | U | INCO (3) |
| Austenitic cast iron, Type 3 | 181 | 5 | 5.0 | - | U | INCO (3) |
| Austenitic cast iron, Type 3 | 366 | 5 | 2.8 | - | U | INCO (3) |
| Austenitic cast iron, Type 4 | 123 | 5,640 | 1.8 | 1.6 | G | INCO (3) |
| Austenitic cast iron, Type 4 | 189 | 5,900 | 2.0 | 1.4 | U | CEL (4) |
| Austenitic cast iron, Type 4 | 403 | 6,780 | 2.0 | 1.3 | U | INCO (3) |
| Austenitic cast iron, Type 4 | 751 | 5,640 | 1.2 | 1.5 | G | INCO (3) |
| Austenitic cast iron, Type 4 | 1,064 | 5,300 | 0.9 | 0.4 | U | INCO (3) |
| Austenitic cast iron, Type 4 | 197 | 2,340 | 0.8 | 0.4 | G | INCO (3) |
| Austenitic cast iron, Type 4 | 402 | 2,370 | 0.9 | 0.7 | G | CEL (4) |
| Austenitic cast iron, Type 4 | 402 | 2,370 | 0.8 | 0.3 | U | INCO (3) |
| Austenitic cast iron, Type 4 | 181 | 5 | 3.8 | - | U | CEL (4) |
| Austenitic cast iron, Type 4 | 181 | 5 | 3.4 | - | U | INCO (3) |
| Austenitic cast iron, Type 4 | 364 | 5 | 2.4 | - | G | CEL (4) |
| Austenitic cast iron, Type 4 | 366 | 5 | 2.4 | - | U | INCO (3) |
| Austenitic cast iron, Type 4 | 723 | 5 | 2.0 | - | G | CEL (4) |
| Austenitic cast iron, Type 4 | 763 | 5 | 2.0 | - | G | CEL (4) |
| Austenitic cast iron, D-2 | 123 | 5,640 | 2.6 | 2.4 | G | INCO (3) |
| Austenitic cast iron, D-2 | 403 | 6,780 | 1.2 | 0.2 | U | INCO (3) |
| Austenitic cast iron, D-2 | 751 | 5,640 | 1.3 | 1.5 | G | INCO (3) |
| Austenitic cast iron, D-2 | 1,064 | 5,300 | 1.1 | 0.4 | U | INCO (3) |
| Austenitic cast iron, D-2 | 197 | 2,340 | 1.2 | 0.2 | G | INCO (3) |
| Austenitic cast iron, D-2 | 402 | 2,370 | 1.1 | 0.5 | U | INCO (3) |
| Austenitic cast iron, D-2 | 181 | 5 | 4.3 | - | G | INCO (3) |
| Austenitic cast iron, D-2 | 366 | 5 | 2.4 | - | G | INCO (3) |
| Austenitic cast iron, D-2b | 123 | 5,640 | 2.1 | 2.0 | G | INCO (3) |
| Austenitic cast iron, D-2b | 403 | 6,780 | 1.6 | 0.1 | U | INCO (3) |
| Austenitic cast iron, D-2b | 751 | 5,640 | 1.2 | 1.3 | G | INCO (3) |
| Austenitic cast iron, D-2b | 1,064 | 5,300 | 1.0 | 0.4 | G | INCO (3) |
| Austenitic cast iron, D-2b | 197 | 2,340 | 1.4 | 0.1 | G | INCO (3) |
| Austenitic cast iron, D-2b | 402 | 2,370 | 0.9 | 0.6 | U | INCO (3) |
| Austenitic cast iron, D-2b | 181 | 5 | 4.1 | - | G | INCO (3) |
| Austenitic cast iron, D-2b | 366 | 5 | 2.7 | - . | G | INCO (3) |
| Austenitic cast iron, D-2c | 189 | 5,900 | 3.3 | 1.5 | U | CEL (4) |
| Austenitic cast iron, D-2c | 402 | 2,370 | 1.8 | 1.2 | U | CEL (4) |
| Austenitic cast iron, D-2c | 181 | 5 | 3.9 | - | U | CEL (4) |
| Austenitic cast iron, D-2c | 364 | 5 | 3.2 | - | G | CEL (4) |
| Austenitic cast iron, D-2c | 723 | 5 | 3.1 | - | U | CEL (4) |
| Austenitic cast iron, D-2c | 763 | 5 | 2.8 | - | U | CEL (4) |
| Austenitic cast iron, D-3 | 123 | 5,640 | 1.9 | 2.2 | G | INCO (3) |
| Austenitic cast iron, D-3 | 402 | 6,780 | 2.7 | 0.4 | G | INCO (3) |
| Austenitic cast iron, D-3 | 751 | 5,640 | 2.1 | 1.9 | G | INCO (3) |
| Austenitic cast iron, D-3 | 1,064 | 5,300 | 1.2 | 0.7 | U | INCO (3) |
| Austenitic cast iron, D-3 | 197 | 2,340 | 0.9 | 0.2 | G | INCO (3) |
| Austenitic cast iron, D-3 | 402 | 2,370 | 0.7 | 0.5 | U | INCO (3) |
| Austenitic cast iron, D-3 | 181 | 5 | 4.3 | - | G | INCO (3) |

Continued

Table 5. Continued

| Alloy | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) ${ }^{\text {a }}$ |  | Type of Corrosion ${ }^{c}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $\mathrm{w}^{b}$ | $\mathrm{M}^{\text {b }}$ |  |  |
| Austenitic cast iron, D-3 | 366 | 5 | 3.2 | - | G | INCO (3) |
| Austenitic cast iron, hardenable | 123 | 5,640 | 2.5 | 2.8 | G | INCO (3) |
| Austenitic cast iron, hardenable | 403 | 6,780 | 1.1 | 0.4 | U | INCO (3) |
| Austenitic cast iron, hardenable | 751 | 5,640 | 0.7 | 0.7 | G | INCO (3) |
| Austenitic cast iron, hardenable | 1,064 | 5,300 | 0.6 | 0.7 | U | INCO (3) |
| Austenitic cast iron, hardenable | 197 | 2,340 | 2.8 | 0.1 | G | INCO (3) |
| Austenitic cast iron, hardenable | 402 | 2,370 | 1.8 | 0.5 | U | INCO (3) |
| Austenitic cast iron, hardenable | 181 | 5 | 4.2 | - | U | INCO (3) |
| Austenitic cast iron, hardenable | 366 | 5 | 2.6 | - | U | INCO (3) |
| Galvanized steel, $1 \mathrm{oz} / \mathrm{ft}^{2}{ }^{n}$ | 189 | 5,900 | 1.9 | 1.6 | U | CEL (4) |
| Galvanized steel, $1 \mathrm{oz} / \mathrm{ft}^{2}$ | 402 | 2,370 | 0.9 | 0.4 | G | CEL (4) |
| Aluminized steel, $1 \mathrm{oz} / \mathrm{ft}^{2} \frac{0}{2}$ | 189 | 5,900 | 0.2 | 0.2 | U | CEL (4) |
| Aluminized steel, $1 \mathrm{oz} / \mathrm{ft}^{2} p$ | 402 | 2,370 | 0.0 | 0.0 | G | CEL (4) |

$a_{\text {mpy }}=$ mils penetration per year calculated from weight loss.
$b_{\mathrm{W}}=$ specimens exposed on sides of structure in seawater; $M=$ specimens exposed in the base of the structure, partially embedded in the bottom sediment.
${ }^{c}$ Symbols signify the following types of corrosion:

| $\mathrm{C}=$ Crevice | $\mathrm{NC}=$ No visible corrosion |
| :--- | :--- |
| $\mathrm{CR}=$ Cratering | $\mathrm{P}=$ Pitting |
| $\mathrm{E}=$ Edge | $\mathrm{PR}=$ Perforation |
| $\mathrm{ET}=$ Etched | $\mathrm{S}=$ Severe |
| $\mathrm{G}=$ General | $\mathrm{U}=$ Uniform |
| $\mathrm{I}=$ Incipient | $\mathrm{WB}=$ Weld bead |
| $\mathrm{M}=$ Corroded at sediment line |  |

Numbers after symbols refer to maximum depth in mils.
$d_{\text {Numbers refer to references at end of report. }}$
${ }^{e}$ Corrosion accelerated below mud line.
$f_{\text {Single crater, }} 12$ mils deep.
$g_{\text {Surface exposure at Francis L. LaQue Corrosion Laboratory, Wrightsville Beach, N.C. }}$
${ }^{b}$ Welded.
${ }^{i}$ Transverse butt weld, weld bead same as plate.
${ }^{j}$ Circular weld bead, $3-\mathrm{in}$. diameter, in center of plate.

${ }^{l}$ Pitting rate, mpy.
${ }^{m}$ Heat treated $900^{\circ} \mathrm{F}, 3 \mathrm{hr}$, air cooled.
${ }^{n}$ Zinc coating completely gone.
${ }^{o}$ Aluminum coating $50 \%$ gone, mottled, bare steel in places.
$p_{\mathrm{W}}=$ no rust, $78 \%$ aluminum coating remaining; $\mathrm{M}=$ no rust, $60 \%$ aluminum coating remaining.

Table 6. Inorganic Coatings on Steels

| Alloy | Exposure (day) | Depth (ft) | Paint Coating | Condition <br> After <br> Exposure ${ }^{a}$ | Source ${ }^{b}$ |
| :---: | :---: | :---: | :---: | :---: | :---: |
| AISI 4130 | 123 | 5,640 | Wash primer, MIL-C-8541 | NPF | NADC (7) |
|  | 403 | 6,780 | Epoxy primer, MIL-P-23377 |  |  |
|  | 1,064 | 5,300 | Epoxy topcoat, MIL-C-22750 |  |  |
|  | 197 | 2,340 |  |  |  |
|  | 402 | 2,370 |  |  |  |
| AISI 4340 | 197 | 2,340 | Wash primer, MIL-C-8541 <br> Epoxy primer, MIL-P-23377 <br> Epoxy topcoat, MIL-C-22750 | NPF | NADC (7) |
|  | 402 | 2,370 |  |  |  |
|  |  |  |  |  |  |
| 18\% Ni, maraging | 123 | 5,640 | Wash primer, MIL-C-8541 | NPF | NADC (7) |
|  | 403 | 6,780 | Epoxy primer, MIL-P-23377 |  |  |
|  | 197 | 2,340 | Epoxy topcoat, M1L-C-22750 |  |  |
|  | 402 | 2,370 |  |  |  |
| HSLA No. 5 | 197 | 2,340 | Wash primer, MIL-C-8541 | NPF | NADC (7) |
|  |  |  | Epoxy primer, MIL-P-23377 |  |  |
|  |  |  | Epoxy topcoat, MLL-C-22750 |  |  |
| HSLA No. 4 | 189 | 5,900 | Zinc rich primer, 8 mils | RS | CEL (4) |
| HSLA No. 5 | 189 | 5,900 |  | RS | CEL (4) |
| HSLA No. 13 | 189 | 5,900 | Zinc rich primer, 8 mils Zinc rich primer, 8 mils |  | CEL (4) |
| HSLA No. 4 | 189 | 5,900 | Wash primer, MHL-C-8514, 1 mil | NPF | CEL (4) |
| HSLA No. 5 | 189 | 5,900 | Zinc rich primer, 8 mils | NPF | CEL (4) |
| HSLA No. 13 | 189 | 5,900 | Epoxy topcoat, 6 mils | NPF | CEL (4) |
| HSLA No. 4 | 189 | 5,900 | Epoxy tar primer, 8 mils Epoxy tar topcoat, 8 mils | NPF | CEL (4) |
| HSLA No. 5 | 189 | 5,900 |  | NPF | CEL (4) |
| HSLA No. 13 | 189 | 5,900 |  | NPF | CEL (4) |
| HSLA No. 4 | 189 | 5,900 | Epoxy tar primer, 8 mils | RS, IPF | CEL (4) |
| HSLA No. 5 | 189 | 5,900 |  | NPF NPF | CEL (4) |
| HSLA No. 13 | 189 | 5,900 | Epoxy tar topcoat, aluminum pigmented, 8 mils |  | CEL (4) |
| HSLA No. 3 | 197 | 2,340 | Epoxy primer, 162-Y-26, W.P. Fuller Co. | GC | Boeing (6) |
| HSLA No. 6 | 197 | 2,340 | Topcoat, Epicote Finish 26-64, Unit | GC | Boeing (6) |
| 12\% Ni, maraging | 197 | 2,340 | Gull gray, National Lead Co. | FPF | Boeing (6) |
| 18\% Ni, maraging | 197 | 2,340 |  | EPF | Boeing (6) |
| D6AC | 197 | 2,340 |  | GC | Boeing (6) |
| HSLA No. 3 | 197 | 2,340 | Polyurethane, laminar X500 | PC | Boeing (6) |
| HSLA No. 6 | 197 | 2,340 | Primer, 4-G-14, green | PB | Boeing (6) |
| 12\% Ni, maraging | 197 | 2,340 | Topcoat, 4-W-1A, white | NPF | Boeing (6) |
| 18\% Ni, maraging | 197 | 2,340 | Magna Coating and Chemical Co. | FPF | Boeing (6) |
| D6AC | 197 | 2,340 |  | GC | Boeing (6) |

${ }^{a}$ Symbols signify the following:

| $\mathrm{GC}=$ Good condition. | $\mathrm{PC}=$ Paint cracked |
| :--- | :--- |
| $\mathrm{FPF}=$ Few spots of paint failure. | $P B=$ Paint blistered |
| $I P F=$ Incipient paint failure. | RS $=$ Rust stains. |

$b_{\text {Numbers indicate references at end of report. }}$

Table 7. Stress Corrosion Tests

| Alloy | $\begin{aligned} & \text { Stress } \\ & (\mathrm{ksi}) \end{aligned}$ | $\begin{aligned} & \text { Percent } \\ & \text { Yield } \\ & \text { Strength } \end{aligned}$ | Exposure (day) | Depth <br> (ft) | Number of Specimens Exposed | Number <br> Failed | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 0.2\% Cu steel | - | 33 | 123 | 5,640 | 3 | 0 | NADC (7) |
| 0.2\% Cu steel | - | 45 | 123 | 5,640 | 3 | 0 | NADC (7) |
| 0.2\% Cu steel | - | 64 | 123 | 5,640 | 3 | 0 | NADC (7) |
| AISI 4130 | 51 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4130, Cd plated | 51 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4130, Ni plated | 51 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4130, Cu plated | 51 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4130 | 51 | 30 | 751 | 5,640 | 3 | 0 | NADC (7) |
| AISI 4130 | 51 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 4130 | 85 | 50 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4130, Cd plated | 85 | 50 | 463 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4130, Ni plated | 85 | 50 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4130, Cu plated | 85 | 50 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4130 | 85 | 50 | 751 | 5,640 | 3 | 0 | NADC (7) |
| AISI 4130 | 127 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4130, Cd plated | 127 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4130, Ni plated | 127 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4130, Cu plated | 127 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4130 | 127 | 75 | 751 | 5,640 | 3 | 0 | NADC (7) |
| AISI 4130 | 127 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 4130, Cd plated | 127 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 4130, Ni plated | 127 | 75 | 197 | 2,340 | 3 | 1 | NADC (7) |
| AISI 4130, Cu plated | 127 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 4140 | 120 | - | 402 | 2,370 | 6 | 0 | Shell (9) |
| AISI 4140, bolts | 90 | - | 402 | 2,370 | 6 | 0 | Shell (9) |
| AISI 4140, bolts | 175 | - | 402 | 2,370 | 6 | $3^{b}$ | Shell (9) |
| AISI 4340 (150 ksi) | 46 | 35 | 123 | 5,640 | 3 | 0 | CEL (4) |
| AISI 4340 (150 ksi) | 46 | 35 | 403 | 6,780 | 3 | 0 | CEL (4) |
| AISI 4340 (150 ksi) | 45 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4340 ( 150 ksi ), Cd plated | 45 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4340 ( 150 ksi ), Ni plated | 45 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Cu plated | 45 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi) | 46 | 35 | 751 | 5,640 | 3 | 0 | CEL (4) |
| AISI 4340 ( 150 ksi ) | 45 | 30 | 751 | 5,640 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi) | 46 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 4340 (150 ksi) | 45 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi) | 45 | 30 | 403 | 2,370 | 3 | 0 | NADC (7) |
| AISI 4340 ( 150 ksi ), Cd plated | 45 | 30 | 403 | 2,370 | 3 | 0 | NADC (7) |
| AISI 4340 ( 150 ksi ), Ni plated | 45 | 30 | 403 | 2,370 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Cu plated | 45 | 30 | 403 | 2,370 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi) | 66 | 50 | 123 | 5,640 | 3 | 0 | CEL (4) |
| AISI 4340 (150 ksi) | 66 | 50 | 403 | 6,780 | 3 | 0 | CEL (4) |
| AISI 4340 (150 ksi) | 66 | 50 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Cd plated | 66 | 50 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Ni plated | 66 | 50 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Cu plated | 66 | 50 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi) | 66 | 50 | 751 | 5,640 | 3 | 0 | CEL (4) |
| AISI 4340 ( 150 ksi ) | 66 | 50 | 751 | 5,640 | 3 | 0 | NADC (7) |
| AlSI 4340 (150 ksi) | 66 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |

Continued

Table 7. Continued.

| Alloy | $\begin{aligned} & \text { Stress } \\ & \text { (ksi) } \end{aligned}$ | $\begin{gathered} \text { Percent } \\ \text { Yield } \\ \text { Strength } \end{gathered}$ | Exposure (day) | Depth <br> (ft) | Number of Specimens Exposed | Number <br> Failed | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| AISI 4340 ( 150 ksi ) | 66 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 4340 (150 ksi) | 66 | 50 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Cd plated | 66 | 50 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 4340 ( 150 ksi ), Ni plated | 66 | 50 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Cu plated | 66 | 50 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 4340 ( 150 ksi ) | 99 | 75 | 123 | 5,640 | 3 | 0 | CEL (4) |
| AISI 4340 (150 ksi) | 99 | 75 | 403 | 6,780 | 3 | 0 | CEL (4) |
| AISI 4340 (150 ksi) | 99 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Cd plated | 99 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Ni plated | 99 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Cu plated | 99 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi) | 99 | 75 | 751 | 5,640 | 3 | 0 | CEL (4) |
| AISI 4340 (150 ksi) | 99 | 75 | 751 | 5,640 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi) | 99 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 4340 (150 ksi) | 99 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Cd plated | 99 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 4340 ( 150 ksi ), Ni plated | 99 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Cu plated | 99 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi) | 99 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 4340 (150 ksi) | 99 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Cd plated | 99 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 4340 ( 150 ksi ), Ni plated | 99 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 4340 (150 ksi), Cu plated | 99 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 4340 (200 ksi) | 65 | 35 | 123 | 5,640 | 3 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 65 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 65 | 35 | 751 | 5,640 | 3 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 65 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 93 | 50 | 123 | 5,640 | 3 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 93 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 93 | 50 | 751 | 5,640 | 3 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 93 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 93 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 139 | 75 | 123 | 5,640 | 3 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 139 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 139 | 75 | 751 | 5,640 | 3 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 139 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 4340 (200 ksi) | 139 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| ASTM A36 | 20 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| ASTM A36 | 30 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| ASTM A387-D | 24 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| ASTM A387-D | 37 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| HSLA No. $1^{c}$ | 38 | 35 | 197 | 2,340 |  | 0 | CEL (4) |
| HSLA No. 1 | 55 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| HSLA No. 1 | 55 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| HSLA No. 1 | 82 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| HSLA No. 1 | 82 | 75 | 402 | 2.370 | 3 | 0 | CEL (4) |
| HSLA No. $4^{\text {d }}$ | - | 75 | 189 | 5,900 | 1 | 0 | CEL (4) |

Continued

Table 7. Continued.

| Alloy | $\begin{aligned} & \text { Stress } \\ & (\mathrm{ksi}) \end{aligned}$ | Percent Yield Strength | Exposure (day) | Depth <br> (ft) | Number of Specimens Exposed | Number Failed | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| HSLA No. $4^{e}$ | - | 75 | 189 | 5,900 | 2 | 0 | CEL (4) |
| HSLA No. $4^{f}$ | - | 75 | 189 | 5,900 | 1 | 0 | CEL (4) |
| HSLA No. 5 | 41 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| HSLA No. 5 | 59 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| HSLA No. 5 | 59 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| HSLA No. 5 | 89 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| HSLA No. 5 | 89 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| HSLA No. $5^{\text {d }}$ | - | 75 | 189 | 5,900 | 2 | 0 | CEL (4) |
| HSLA No. $5^{e}$ | - | 75 | 189 | 5,900 | 2 | 0 | CEL (4) |
| HSLA No. $5^{f}$ | - | 75 | 189 | 5,900 | 1 | 0 | CEL (4) |
| HSLA No. $13{ }^{\text {d }}$ | - | 75 | 189 | 5,900 | 2 | 0 | CEL (4) |
| HSLA No. $13{ }^{e}$ | - | 75 | 189 | 5,900 | 2 | 0 | CEL (4) |
| HSLA No. $13{ }^{f}$ | - | 75 | 189 | 5,900 | 1 | 0 | CEL (4) |
| HS No. $3^{g}$ | 86 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| HS No. $3^{\text {b }}$ | 86 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| HS No. 3 | 129 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| HS No. $3^{\text {b }}$ | 129 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| HS No. 6 | 96 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| HS No. $6^{\text {b }}$ | 96 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| HS No. $6^{6}$ | 143 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| HS No. $6^{h}$. | 143 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| HS No. $6^{2}$ | 96 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| HS No. $6_{i}^{h, i}$ | 96 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| HS No. $6^{i}$ | 143 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| HS No. $6^{h, i}$ | 143 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| AISI 502 | 18 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 502 | 18 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 502 | 27 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 502 | 27 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 18\% Ni, maraging | 83 | 33 | 123 | 5,640 | 3 | 0 | NADC (7) |
| 18\% Ni, maraging | 105 | 42 | 123 | 5,640 | 3 | 0 | NADC (7) |
|  | 158 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| 18\% Ni, maraging ${ }^{2}$ | 158 | 50 | 189 | 5,900 | 3 | 2 | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging ${ }_{\text {b }}$ | 237 | 75 | 189 | 5,900 | 3 | 3 | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging ${ }^{\text {b }}$ | 237 | 75 | 189 | 5,900 | 3 | 3 | CEL (4) |
| 18\% Ni, maraging ${ }_{i, j}$ | - | 75 | 403 | 6,780 | 3 | 3 | NADC (7) |
| 18\% Ni, maraging ${ }^{\text {i, }}$ ] | - | 75 | 403 | 6,780 | 3 | 3 | NADC (7) |
| $18 \% \mathrm{Ni}$, maraging ${ }_{\text {j }}$ | - | 75 | 751 | 5,640 | 3 | 0 | NADC (7) |
| 18\% Ni, maraging ${ }_{\text {a }}$ | 70 | 35 | 197 | 2,340 | 3 | 0 | Boeing (6) |
| $18 \% \mathrm{Ni}$, maraging ${ }_{i}$ | 100 | 50 | 197 | 2,340 | 3 | 0 | Boeing (6) |
| 18\% Ni, maraging ${ }^{2}$ | 150 | 75 | 197 | 2,340 | 3 | 0 | Boeing (6) |
| 18\% Ni, maraging | 75 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 18\% Ni, maraging | 125 | 50 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 18\% Ni, maraging ${ }_{\text {. }}$ | 188 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 18\% Ni, maraging ${ }^{\text {i }}$ | 75 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 18\% Ni, maraging ${ }_{j}{ }^{\text {j }}$ | 125 | 50 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 18\% Ni, maraging ${ }^{\text {3,j }}$ | 188 | 75 | 197 | 2,340 | 3 | 3 | NADC (7) |

Continued

Table 7. Continued.

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure (day) | Depth (ft) | Number of Specimens Exposed | Number Failed | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 18\% Ni, maraging, Cu plated ${ }^{j}$ | 188 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| $18 \% \mathrm{Ni}$, maraging, Ni plated ${ }^{\text {J }}$ | 188 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| $18 \% \mathrm{Ni}$, maraging . | 188 | 75 | 402 | 2,370 | 3 | 3 | NADC (7) |
| 18\% Ni, maraging ${ }^{j}$ | 188 | 75 | 402 | 2,370 | 3 | 3 | NADC (7) |
| 18\% Ni, maraging | 109 | 35 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging | 156 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 18\% Ni, maraging . | 234 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 18\% Ni, maraging. | 109 | 35 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 18\% Ni, maraging ${ }^{j}$. | 156 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $18 \% \mathrm{Ni}$, maraging ${ }^{j}$ | 234 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 12\% Ni, maraging | 63 | 35 | 197 | 2,340 | 3 | 0 | Boeing (6) |
| $12 \% \mathrm{Ni}$, maraging | 90 | 50 | 197 | 2,340 | 3 | 0 | Boeing (6) |
| 12\% Ni, maraging | 135 | 75 | 197 | 2,340 | 3 | 0 | Boeing (6) |

${ }^{a}$ Numbers refer to references at end of report.
$b_{\text {One specimen failed in bottom sediment, two in water. }}$
${ }^{c}$ HSLA $=$ high-strength, low-alloy steel.
$d_{\text {Zinc-rich primer, }} 8$ mils.
${ }^{e}$ Zinc-rich primer, 8 mils plus wash primer (MIL-C-8514), 6 mils plus epoxy topcoat, 6 mils.
$f_{\text {Epoxy tar primer, }} 8$ mils plus epoxy topcoat, 8 mils.
$g_{\mathrm{HS}}=$ high-strength steel.
${ }^{h}$ Exposed in bottom sediment.
${ }^{i}$ TIG welded.
${ }^{j}$ Cracked at edge of heat affected zone.
Table 8. Changes in Mechanical Properties of Irons and Steels Due to Corrosion

| Alloy | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (ksi) | \% Change | Original <br> (ksi) | \% Change | Original (\%) | \% Change |  |
| Wrought iron | 123 | 5,640 | 47 | - | 30 | - | 15 | - | CEL (4) |
| Wrought iron | 403 | 6,780 | 47 | +5 | 30 | +2 | 15 | +118 | CEL (4) |
| Wrought iron | 751 | 5,640 | 47 | +6 | 30 | 0 | 15 | +154 | CEL (4) |
| Wrought iron | 1,064 | 5,300 | 47 | +6 | 30 | +3 | 15 | +113 | CEL (4) |
| Wrought iron | 197 | 2,340 | 47 | - | 30 | - | 15 | - | CEL (4) |
| Wrought iron | 402 | 2,370 | 47 | +2 | 30 | -25 | 15 | +26 | CEL (4) |
| Wrought iron | 181 | 5 | 47 | +7 | 30 | -8 | 15 | +17 | CEL (4) |
| AISI 1010 | 123 | 5,640 | 54 | -1 | 36 | 0 | 42 | +1 | CEL (4) |
| AISI 1010 | 403 | 6,780 | 54 | +1 | 36 | +6 | 42 | -8 | CEL (4) |
| AISI 1010 | 751 | 5,640 | 54 | 0 | 36 | +3 | 42 | -3 | CEL (4) |
| AISI 1010 | 1,064 | 5,300 | 54 | +6 | 36 | +8 | 42 | -5 | CEL (4) |
| AISI 1010 | 197 | 2,340 | 54 | +2 | 36 | +4 | 42 | -4 | CEL (4) |
| AISI 1010 | 402 | 2,370 | 54 | -2 | 36 | -4 | 42 | -3 | CEL (4) |
| AISI 1010 | 181 | 5 | 54 | +3 | 36 | -1 | 42 | -18 | CEL (4) |
| ASTM A36 | 123 | 5,640 | 66 | -1 | 39 | +2 | 39 | +9 | CEL (4) |
| ASTM A36 | 403 | 6,780 | 66 | +2 | 39 | +4 | 39 | -11 | CEL (4) |
| ASTM A36 | 751 | 5,640 | 66 | +1 | 39 | +6 | 39 | -11 | CEL (4) |
| ASTM A36 | 1,064 | 5,300 | 66 | +4 | 39 | +4 | 39 | +14 | CEL (4) |
| ASTM A36 | 197 | 2,340 | 66 | +2 | 39 | +4 | 39 | -1 | CEL (4) |
| ASTM A36 | 402 | 2,370 | 66 | +1 | 39 | -2 | 39 | -4 | CEL (4) |
| ASTM A36 | 181 | 5 | 66 | +1 | 39 | -1 | 39 | -19 | CEL (4) |
| ASTM A387-D | 123 | 5,640 | 76 | +3 | 49 | +4 | 32 | -5 | CEL (4) |
| ASTM A 387-D | 403 | 6,780 | 76 | +6 | 49 | +6 | 32 | -16 | CEL (4) |
| ASTM A387-D | 751 | 5,640 | 76 | +7 | 49 | +7 | 32 | -17 | CEL (4) |
| ASTM A 387-D | 1,064 | 5,300 | 76 | - | 49 | - | 32 | - | CEL (4) |
| ASTM A387-D | 197 | 2,340 | 76 | +4 | 49 | -2 | 32 | -10 | CEL (4) |
| ASTM A $387-$ D | 402 | 2,370 | 76 | +3 | 49 | +6 | 32 | -10 | CEL (4) |
| HSLA No. $1^{b}$ | 123 | 5,640 | 121 | 0 | 110 | +1 | 12 | +42 | CEL (4) |
| HSLA No. 1 | 403 | 6,780 | 121 | +3 | 110 | +3 | 12 | +32 | CEL (4) |
| HSLA No. 1 | 751 | 5,640 | 121 | +2 | 110 | +4 | 12 | +31 | CEL (4) |
| HSLA No. 1 | 1,064 | 5,300 | 121 | +2 | 110 | +3 | 12 | +32 | CEL (4) |
| HSLA No. 1 | 197 | 2,340 | 121 | +2 | 110 | +2 | 12 | +39 | CEL (4) |
| HSLA No. 1 | 402 | 2,370 | 121 | +1 | 110 | -1 | 12 | +30 | CEL (4) |
| HSLA No. 1 | 181 | 5 | 121 | +1 | 110 | +1 | 12 | +13 | CEL (4) |

Table 8. Continued.

| Alhoy | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (kui) | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| IISLA No. 2 | 123 | 5,640 | 100 | +1 | 89 | -2 | 28 | -12 | CEL (4) |
| HSLA No. 2 | 403 | 6,780 | 100 | +4 | 89 | -2 | 28 | -12 | CEL (4) |
| HSLA No. 2 | 751 | 5,640 | 100 | +6 | 89 | +4 | 28 | -3 | CEL (4) |
| IISLA No. 2 | 1,064 | 5,300 | 100 | +6 | 89 | +4 | 28 | +42 | CEL (4) |
| HSLA No. 2 | 197 | 2,340 | 100 | +4 | 89 | +2 | 28 | -9 | CEL (4) |
| HSLA No. 2 | 402 | 2,370 | 100 | +2 | 89 | -3 | 28 | -11 | CEL (4) |
| HSLA No. 2 | 181 | 5 | 100 | +1 | 89 | +3 | 28 | 0 | CEL (4) |
| HiSLA No. 3 | 1,064 | 5,300 | 107 | +10 | 88 | +4 | 30 | +33 | CEL (4) |
| HSLA No. 4 | 123 | 5,640 | 70 | +5 | 52 | +7 | 32 | +24 | CEL (4) |
| HSLA No. 4 | 403 | 6,780 | 70 | +4 | 52 | +4 | 32 | +40 | CEL (4) |
| HSLA No. 4 | 751 | 5,640 | 70 | +4 | 52 | +7 | 32 | +18 | CEL (4) |
| HSLA No. 4 | 1,064 | 5,300 | 70 | +5 | 52 | +6 | 32 | +20 | CEL (4) |
| HSLA No. 4 | 197 | 2,340 | 70 | +6 | 52 | +8 | 32 | +36 | CEL (4) |
| HSLA No. 4 | 402 | 2,370 | 70 | +3 | 52 | -2 | 32 | -33 | CEL (4) |
| HSLA No. 5 | 123 | 5,640 | 125 | +1 | 118 | +2 | 16 | -1 | CEL (4) |
| HSLA No. 5 | 403 | 6,780 | 125 | +4 | 118 | +5 | 16 | -11 | CEL (4) |
| HSLA No. 5 | 751 | 5,640 | 125 | +6 | 118 | +5 | 16 | -6 | CEL (4) |
| HSLA No. 5 | 1,064 | 5,300 | 125 | +6 | 118 | $+6$ | 16 | -2 | CEL (4) |
| IISLA No. 5 | 197 | 2,340 | 125 | +2 | 118 | +2 | 16 | +2 | CEL (4) |
| HSLA No. 5 | 197 | 2,340 | 105 | 0 | 123 | 0 | 17 | -2 | NADC (7) |
| HSLA No. 5 | 402 | 2,370 | 125 | +3 | 118 | +4 | 16 | -3 | CEL (4) |
| HSLA No. 5 | 402 | 2,370 | 123 | 0 | 105 | +12 | 17 | -2 | NADC (7) |
| HSLA No. $5^{c}$ | 402 | 2,370 | 123 | -1 | 105 | 0 | 17 | -77 | NADC (7) |
| HSLA No. $5^{\text {d }}$ | 402 | 2,370 | 123 | +7 | 105 | +13 | 17 | -82 | NADC (7) |
| HSLA No. $5^{e}$ | 402 | 2,370 | 123 | +3 | 105 | +17 | 17 | -4 | NADC (7) |
| HSLA No. $5^{f}$ | 402 | 2,370 | 123 | -1 | 105 | +10 | 17 | -2 | NADC (7) |
| HSLA No. $5^{g}$ | 402 | 2,370 | 123 | -5 | 105 | +4 | 17 | -29 | NADC (7) |
| HSLA No. 6 | 402 | 2,370 | 131 | +2 | 117 | -1 | 16 | -10 | CEL (4) |
| HSLA No. 12 | 181 | 5 | 120 | 0 | 109 | -2 | 26 | +4 | CEL (4) |
| HS No. $1^{b}$ | 181 | 5 | 145 | -2 | 140 | -3 | 22 | +7 | CEL (4) |
| HS No. 2 | 181 | 5 | 194 | -4 | 186 | -5 | 15 | +5 | CEL (4) |
| HS No. 3 | 402 | 2,370 | 169 | 0 | 118 | -6 | 14 | +8 | CEL (4) |

Table 8. Continued.

| Alloy | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{a}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original <br> (\%) | \% Change |  |
| AISI 4130 | 123 | 5,640 | 91 | -22 | 63 | -30 | 30 | -13 | NADC (7) |
| AISI 4130 | 123 | 5,640 | 169 | -44 | - | - | 11 | 0 | NADC (7) |
| AISI 4130 | 403 | 6,780 | 169 | 0 | 148 | -7 | 11 | -41 | NADC (7) |
| AISI 4130 | 751 | 5,640 | 91 | -23 | 63 | -34 | 30 | -22 | NADC (7) |
| AISI 4130 | 1,064 | 5,300 | 91 | -27 | 63 | -37 | - | -31 | NADC (7) |
| AISI 4130 | 197 | 2,340 | 91 | 0 | 63 | 0 | 30 | -11 | NADC (7) |
| AISI 4130 | 402 | 2,370 | 169 | +7 | 148 | -35 | 11 | -41 | NADC (7) |
| AISI 4130 | 402 | 2,370 | 91 | -3 | 63 | +43 | 30 | -36 | NADC (7) |
| AISI $4130^{e}$ | 402 | 2,370 | 91 | -17 | 63 | -21 | 30 | -14 | NADC (7) |
| AISI 4130 | 402 | 2,370 | 91 | -17 | 63 | -23 | 30 | -26 | NADC (7) |
| AISI $4130^{g}$ | 402 | 2,370 | 91 | -17 | 63 | +6 | - | - | NADC (7) |
| AISI 4340 | 123 | 5,640 | 201 | +4 | 185 | +5 | 8 | +43 | CEL (4) |
| AISI 4340 | 123 | 5,640 | 143 | +3 | 132 | +4 | 13 | +28 | CEL (4) |
| AISI 4340 | 402 | 6,780 | 201 | +4 | 185 | +3 | 8 | +25 | CEL (4) |
| AISI 4340 | 402 | 6,780 | 209 | +1 | 190 | +1 | 8 | -2 | CEL (4) |
| AISI 4340 | 402 | 6,780 | 143 | -1 | 132 | -3 | 13 | +27 | CEL (4) |
| AISI 4340 | 402 | 6,780 | 147 | 0 | 136 | -1 | 14 | -6 | CEL (4) |
| AISI 4340 | 751 | 5,640 | 201 | +3 | 185 | +10 | 8 | +30 | CEL (4) |
| AISI 4340 | 751 | 5,640 | 143 | +3 | 132 | +3 | 13 | +28 | CEL (4) |
| AISI 4340 | 197 | 2,340 | 201 | +3 | 185 | +5 | 8 | +49 | CEL (4) |
| AISI 4340 | 197 | 2,340 | 209 | -2 | 190 | -1 | 8 | +11 | CEL (4) |
| AISI 4340 | 197 | 2,340 | 143 | +5 | 132 | +4 | 13 | +29 | CEL (4) |
| AISI 4340 | 197 | 2,340 | 147 | 0 | 136 | 0 | 14 | -3 | CEL (4) |
| AISI 4340 | 197 | 2,340 | 293 | -37 | 213 | -19 | 9 | 0 | NADC (7) |
| AISI 4340 | 402 | 2,370 | 209 | -2 | 190 | 0 | 8 | +3 | CEL (4) |
| AISI 4340 | 402 | 2,370 | 147 | -2 | 136 | -3 | 14 | -5 | CEL (4) |
| AISI 4340 | 402 | 2,370 | 200 | +1 | 184 | +1 | 9 | - | NADC (7) |
| AISI $4340{ }^{e}$ | 402 | 2,370 | 200 | -18 | 184 | -14 | 9 | -27 | NADC (7) |
| AISI $4340^{f}$ | 402 | 2,370 | 200 | -29 | 184 | - | 9 | -48 | NADC (7) |
| AISI $4340{ }^{\text {g }}$ | 402 | 2,370 | 200 | -40 | 184 | - | 9 | -73 | NADC (7) |
| AISI 502 | 123 | 5,640 | 59 | -2 | 36 | +5 | 33 | -34 | CEL (4) |
| AISI 502 | 403 | 6,780 | 59 | -5 | 36 | +1 | 33 | -17 | CEL (4) |
| AISI 502 | 751 | 5,640 | 59 | -8 | 36 | +6 | 33 | -38 | CEL (4) |
| AISI 502 | 1,064 | 5,300 | 59 | -3 | 36 | +3 | 33 | -38 | CEL (4) |
| AISI 502 | 197 | 2,340 | 59 | -1 | 36 | -4 | 33 | -2 | CEL (4) |

Table 8. Continued

${ }^{a}$ Numbers refer to references at end of report.
${ }^{b}$ HSLA $=$ high-strength, low-alloy steel.
${ }^{c} 1 / 4$-in. thick.
$d_{1 / 8 \text {-in. thick. }}$
${ }^{e}$ Cadmium plated.
$f_{\text {Copper plated. }}$
$g_{\text {Nickel plated. }}$
${ }^{b} \mathrm{HS}=$ high-strength steel.
${ }^{i}$ Welded.
$j_{\text {Machined, RMS }} 125$.
${ }^{k}$ As rolled.

Table 9. Effect of Corrosion on Breaking Strengths of Anchor Chains
(chains degreased, 0.75 -inch size)

| Designation | Exposure |  | Breaking Load (lb) |  | Remarks |
| :---: | :---: | :---: | :---: | :---: | :---: |
|  | Days | Depth | Original | Final |  |
| Dilok | 123 | 5,640 | 59,000 | 58,500 | Thin film flaky rust, broke at bottom of socket |
| Dilok | 403 | 6,780 | 59,000 | 64,500 | Thin film flaky rust, broke at bottom of socket |
| Dilok | 751 | 5,640 | 59,000 | 71,000 | Thin film flaky rust, broke at bottom of socket |
| Dilok | 197 | 2,340 | 59,000 | 76,500 | Thin film flaky rust, broke at end of link |
| Welded stud link | 123 | 5,640 | 57,500 | 61,500 | Thin film flaky rust, broke at end of link |
| Welded stud link | 403 | 6,780 | 57,500 | 59,500 | Thin film flaky rust, broke at end of link |
| Welded stud link | 751 | 5,640 | 57,500 | 59,500 | Thin film flaky rust, broke at end of link |
| Welded stud link | 197 | 2,340 | 57,500 | 61,000 | Thin film flaky rust, broke at end of link |

## SECTION 3

## COPPER ALLOYS

The excellent corrosion resistance of copper and its alloys is partially due to its being a reiatively noble metal. However, in many environments, its satisfactory performance depends on the formation of adherent, relatively thin films of corrosion products. In seawater corrosion, resistance depends on the presence of a surface oxidic film throuth which oxygen must diffuse in order for corrosion to continue. This oxide film adjoining the metal is cuprous oxide covered with a mixture of cupric oxy-chloride, cupric hydroxide, basic cupric carbonate, and calcium sulfate. Since oxygen must diffuse through this film for corrosion to occur. it would be expecice that under normal circumstances the corrosion rates would decrease with increase in time of exposure.

Copper alloys corrode uniformly; hence, corrosion rates calculated from weighi losses and reported as mils per year reflect the true condition of the alloys. Therefore, corrosion rates for the copper alloys can be used reliably for design purposes. Hlowever, this does not app!y to those copper base alloys which are susceptible to parting corrosion. (Parting corrosion is defined as the selecrive attack of one or more of the components of a sold solution alloy.) Examp!es of parting corrosion are dezincification, dealuminification, denickelification, desiliconification, etc.

The data on the copper alloys were obtained from the reports given in References 3 through 19 and 23. The copper alloys are separated into the different classes of copper alloys (coppers, brasses, bronzes, and copper-nickel alloys) for comparison and discussion purposes.

The chemical compositions, corrosion rates and types of corrosion, stress corrosion characteristics, and changes in mechanical properties due to corrosion of the coppers are given in Tables 10 through 13. The effects of duration of exposure are shown graphically in Figures 8 and 15.

The chemical compositions, corrosion rates and types of corrosion. stress corrosion characteristics, and changes in mechanical properties due to the corrosion of the brasses are given in Tables 14 through 17. The effects of the duration of exposure are shown graphically in Figures 11 and 15.

The chemical compositions, cerrosion "ater and types of corrosion, stress corrosion characteristics, and changes in mechanical properties due to corrosion of the bronzes are given in Tables 18 through 21 . The effects of the duration of exposure are shown graphically in Figures 12 and 15.

The shemical compositions, comosion ates and types of corrosion, stress corrosion characteristics, and changes in mechanical properties due to corrosion of the copper-nickel alloys are given in Tables 22 through 25 . The effects of the duration of exposure are shown graphically in Figures 13 and 15.

The effects of depth and the effect of the concentration of oxygen in seawater on the corrosion of the copper alloys are shown in Figures 9 and 14.

The effect of the iron content on the corrosion of the copper-nickel alloys is shown in Figure 14.

### 3.1. COPPERS

The chemical compositions of the coppers are given in Table 10. their corrosion rates and rypes of corrosion in Tahie 11, their resistance to stress corrosion cracking in Table 12, and the changes in their mechanical properties due to corrosion in Table 13.

### 3.1.1. Duration of Exposure

The effects of the duration of exposure on the corrosion of copper in seawater at depth, at the surface, and in the bottom sediments are shown graphically in Figure 8. At the surface and at the 6,000 -foot depth, both in the seawater and in the bottom sediment, the corrosion rates decreased with increasing duration of exposure. At the 2,500 -foot depth the corrosion rates in the seawater and in the bottom sediment were essentially constant. Also, the corrosion rates were practically the same at depth as at the surface.

The beryllium-copper alloys behaved very similarly to copper, and their corrosion rates were comparable. Beryllium-copper chain corroded at the
same rates and in the same manner as the alloy in sheet form. Welding by either the TIG or MIG processes as well as aging at either $600^{\circ} \mathrm{F}$ or $800^{\circ} \mathrm{F}$ did not affect the corrosion behavior of the berylliumcopper alloys.

### 3.1.2. Effect of Depth

The effect of depth on the corrosion of copper is shown in Figure 9. The corrosion of copper was not affected by depth, at least to a depth of 6,000 feet.

### 3.1.3. Effect of Concentration of Oxygen

The effect of the concentration of oxygen in seawater on the corrosion of copper is shown in Figure 10. The corrosion of copper was unaffected by the concentration of oxygen in seawater within the range of 0.4 to 5.75 ppm .

### 3.1.4. Stress Corrosion

Copper and beryllium-coppers were exposed in the seawater while stressed at values equivalent to 30 and $75 \%$ of their respective yield strengths for the periods of time and at the depths shown in Table 12. Neither copper nor the beryllium-coppers were susceptible to stress corrosion. Aging at either $600^{\circ} \mathrm{F}$ or $800^{\circ} \mathrm{F}$ did not affect the stress corrosion resistance of beryllium-copper (CDA No. 172).

### 3.1.5. Mechanical Properties

The effects of exposure in seawater on their mechanical properties are given in Table 13. The mechanical properties of the copper and the beryllium-coppers were not significantly affected by exposure in seawater at the surface or at depth.

### 3.1.6. Galvanic Corrosion

Dissimilar metal couples consisting of $1 \times 2$-inch strips of aluminum alloy $7075 \cdot \mathrm{~T} 6$ fastened to $1 \times 6$-inch strips of beryllium-copper alloy (CDA No. 175) with plastic fasteners were exposed in seawater at a depth of 2,500 feet for 402 days. After exposure the 7075-T6 was covered with a heavy uniform layer of corrosion products, while there was a thin layer of
corrosion products on the CDA No. 175 alloy. This indicates that the aluminum was corroding galvanically to protect the beryllium-copper from corroding.

### 3.2. BRASSES (Copper-Zinc Alloys)

The chemical compositions of the brasses are given in Table 14, their corrosion rates and types of corrosion in Table 15, their resistance to stress corrosion cracking in Table 16, and the changes in their mechanical properties due to corrosion in Table 17.

### 3.2.1. Duration of Exposure

The effects of the duration of exposure on the corrosion of the brasses in seawater at depth, at the surface, and in the bottom sediments are shown in Figure 11.

Since the corrosion rates of all the brasses, except those of alloys, CDA No. 280 and 675, were so comparable, the average values for any one time, depth, or environment were used to prepare the curves in Figure 11. The corrosion rate values of alloys CDA No. 280 (Muntz Metal) and CDA No. 675 (manganese bronze A) were considerably higher than those of all the other brasses. These high rates were attributed to the severe parting corrosion (dezincification) which had occurred on these two alloys.

The curves in Figure 11 show the effects of the duration of exposure on the corrosion of the brasses in seawater. The corrosion rates of the brasses decreased slightly with increasing duration of exposure at the surface and at depths of 2,500 and 6,000 feet both in seawater and in the bottom sediments. The corrosion of the brasses was the same in the bottom sediments as in the seawater at the 6,000 -foot depth, but slower in the bottom sediments than in the seawater at the 2,500 -foot depth. The corrosion in seawater was practically the same at the 2,500 -foot depth as at the 6,000 -foot depth. The brasses corroded slightly slower at depth than at the surface.

### 3.2.2. Parting Corrosion (Dezincification)

The alloys attacked by parting corrosion are shown in Table 15. All the brasses, except Arsenical

Admiralty (CDA No. 443), aluminum brass, and nickel brass, were attacked by parting corrosion in degrees varying from slight to severe. The zinc content varied from 10 to $42 \%$ with the severity of parting corrosion generally increasing with the zinc content. Although the Arsenical Admiralty contained about $30 \%$ zinc, the addition of about $0.03 \%$ arsenic rendered it immune to parting corrosion. Because of the $2 \%$ aluminum in aluminum brass and the $8 \%$ nickel in the nickel brass, they were also immune to parting corrosion even though they contained 20 and $40 \%$ zinc, respectively.

### 3.2.3. Effect of Depth

The effect of depth on the corrosion of the brasses after 1 year of exposure in seawater is shown in Figure 9. Although there was a slight tendency for the brasses to corrode more slowly at depth than at the surface, this slight decrease is not significant.

### 3.2.4. Effect of Concentration of Oxygen

The effect of the concentration of oxygen in seawater on the corrosion of the brasses after 1 year of exposure is shown in Figure 10. The corrosion of the brasses increased linearly, but slightly, with increasing oxygen concentration.

### 3.2.5. Stress Corrosion

Two brasses, CDA No. 280 and 443, were exposed in seawater while stressed at values equivalent to 50 and $75 \%$ of their respective yield strengths for the periods of time and at the depths given in Table 16. Neither alloy was susceptible to stress corrosion at either depth, 2,500 or 6,000 feet, after 400 days of exposure.

### 3.2.6. Mechanical Properties

The effects of corrosion on the mechanical properties of three brasses are given in Table 17. The mechanical properties of Arsenical Admiralty were not impaired, while those of Muntz Metal and nickelmanganese bronze were impaired. The degree of impairment increased with the time of exposure at both depths, 2,500 and 6,000 feet. The degree of
impairment in both alloys roughly paralleled the severity of the parting corrosion.

### 3.2.7. Corrosion Products

The corrosion products which formed on cast nickel-manganese bronze during 403 days of exposure at a depth of 6,000 feet were analyzed by X-ray diffraction, spectrographic, infra-red spectrophotometer, and quantitative analyses methods. The corrosion products were composed of cupric chloride ( $\mathrm{CuCl} \mathbf{2}^{\cdot} \cdot 2 \mathrm{H}_{2} \mathrm{O}$ ); copper hydroxychloride $\left(\mathrm{Cu}_{2}(\mathrm{OH})_{3} \mathrm{Cl}\right)$; copper as metal, $35.98 \%$; minor amounts of aluminum, iron, silicon, and sodium; chloride ions as $\mathrm{Cl}, 0.91 \%$; sulfate ions as $\mathrm{SO}_{4}$, $11.53 \%$; small quantities of an organic compound or compounds present due to decomposed algae and vegetative materials.

### 3.3. BRONZES

The chemical compositions of the bronzes are given in Table 18, their corrosion rates and types of corrosion in Table 19, their resistance to stress corrosion in Table 20, and the changes in their mechanical properties due to corrosion in Table 21.

### 3.3.1. Duration of Exposure

The effects of the duration of exposure on the corrosion of the bronzes in seawater at depths, at the surface, and in the bottom sediments are shown in Figure 12.

Since the corrosion rates of all the bronzes, except those of alloys CDA No. 653 and 655 (silicon bronzes), were so comparable, the average values for any one time, depth, or environment were used to prepare the curves in Figure 12. Because the corrosion rates of the silicon bronzes were so much greater than those of the other bronzes, they were not averaged with the others. They are shown in Figure 12 as a separate curve that includes the rates for both depths as well as those for the surface. The average corrosion of the bronzes in seawater and in the bottom sediments was essentially constant with increasing duration of exposure. Also, it was essentially the same at the 2,500 -foot depth as at the

6,000 -foot depth, both in the seawater and in the bottom sediments. The corrosion of the silicon bronzes was greater than that of the other bronzes at depth and at the surface; it decreased with increasing duration of exposure until it was the same as the other bronzes after 1,064 days of exposure. The average corrosion of the bronzes in surface seawater was greater than at depth, and it decreased with increasing duration of exposure.

### 3.3.2. Parting Corrosion

Parting corrosion was found on all the aluminum bronzes (dealuminification) and on the silicon bronzes (coppering) as shown in Table 19. The parting corrosion was most severe on the cast alloy containing 10,11 , and $13 \%$ aluminum. There was much less on the wrought aluminum bronzes with there being very few cases of slight attack on the alloy containing $5 \%$ aluminum and more cases on the alloy containing $7 \%$ aluminum. The parting corrosion on the silicon bronzes occurred only occasionally, but its rating varied from slight to severe.

### 3.3.3. Effect of Depth

The effect of depth on the corrosion of the bronzes after 1 year of exposure in seawater is shown in Figure 9. The bronzes corroded slightly slower at depth than at the surface, but the difference was not considered to be significant. For practical purposes depth does not influence the corrosion of the bronzes.

### 3.3.4. Effect of Concentration of Oxygen

The effect of the concentration of oxygen in seawater on the corrosion of the bronzes after 1 year of exposure is shown in Figure 10. The corrosion of the bronzes increased linearly, but slightly, with increasing oxygen concentration, and at $5.75 \mathrm{ml} / 1$ oxygen they were corroding at the same rate as copper and other copper alloys.

### 3.3.5. Stress Corrosion

Four bronzes, phosphor bronze A, phosphor bronze D, aluminum bronze $5 \%$, and silicon bronze A, were exposed in seawater to determine their
susceptibility to stress corrosion. They were stressed at values equivalent to 35,50 , and $75 \%$ of their respective yield strengths as shown in Table 20. They were not susceptible to stress corrosion for periods of exposure of 400 days at either depth.

### 3.3.6. Mechanical Properties

The effects of corrosion on the mechanical properties of four bronzes are given in Table 21. The mechanical properties of the phosphor bronzes A and D were not affected by exposures at depth. The decreases ( 12,27 , and $29 \%$ ) in the elongation of the aluminum bronze were attributed to parting corrosion. Also, the decrease in the mechanical properties of silicon bronze A after 403 days of exposure in the bottom sediments at a depth of 6,000 feet was attributed to parting corrosion.

### 3.3.7. Corrosion Products

Chemical analyses of the corrosion products removed from aluminum bronze showed the presence of copper oxy-chloride, cupric chloride; major elements, copper and aluminum; minor elements, iron, magnesium, calcium, and silicon; chloride ion, $0.9 \%$, and sulfate ion, $9.0 \%$.

### 3.4. COPPER-NICKEL ALLOYS

The chemical compositions of the copper-nickel alloys are given in Table 22, their corrosion rates and type of corrosion in Table 23, their resistance to stress corrosion in Table 24, and the changes in their mechanical properties due to corrosion in Table 25.

### 3.4.1. Duration of Exposure

The effects of the duration of exposure on the corrosion of the copper-nickel alloys in seawater at depths, at the surface, and in the bottom sediments are shown in Figure 13.

Because the corrosion rates of all the coppernickel alloys were comparable, average values for any one time, depth, or environment were used to construct the curves in Figure 13. The average corrosion at the surface and at the 6,000 -foot depth, both in seawater and in the bottom sediments, decreased
linearly with increasing duration of exposure, while corrosion attack was constant in seawater and in the bottom sediments with increasing duration of exposure at the 2,500-foot depth. Corrosion at the 2,500-foot depth was slightly less rapid than at the surface or the 6,000 -foot depth. Corrosion in surface seawater decreased at a more rapid rate than at depth. The differences in the corrosion rates of the coppernickel alloys in the different environments were so small that, for practical purposes, they can be considered to be the same.

### 3.4.2. Effect of Depth

The effect of depth on the corrosion of the copper-nickel alloys after 1 year of exposure in seawater is shown in Figure 9. Depth exerted no influence on the corrosion of the copper-nickel alloys during 1 year of exposure, at least to a depth of 6,000 feet.

### 3.4.3. Effect of Concentration of Oxygen

The effect of the concentration of oxygen in seawater on the corrosion of the copper-nickel alloys after 1 year of exposure is shown in Figure 10. The corrosion increased slightly with increasing oxygen concentration during 1 year of exposure.

### 3.4.4. Effect of Iron

Copper-nickel alloys with iron contents varying between 0.03 and $5 \%$ were among the alloys in this program. The effects of iron on the corrosion of these alloys after 400 days and 1,064 days of exposure at the 6,000 -foot depth are shown in Figure 14. Generally, the rates of corrosion decreased with increasing iron content.

### 3.4.5. Stress Corrosion

Four copper-nickel alloys were exposed in seawater to determine their susceptibility to stress corrosion under the conditions given in Table 24. They were not susceptible to stress corrosion.

### 3.4.6. Mechanical Properties

The effects of corrosion on the mechanical properties of five copper-nickel alloys are given in Table
25. The mechanical properties were not adversely affected during exposure at the depths and during the times of exposure given in Table 25.

### 3.4.7. Corrosion Products

Chemical analyses of the corrosion products removed from $70 \%$ copper- $30 \%$ nickel-5\% iron exposed for 751 days at a depth of 6,000 feet showed that they were composed of nickel hydroxide $\left(\mathrm{Ni}(\mathrm{OH})_{2}\right)$; cupric chloride $\left(\mathrm{CuCl}_{2}\right)$; major elements, copper and nickel; minor elements, iron, magnesiun, sodium, and traces of silicon, and manganese; chloride ion as $\mathrm{Cl}, 4.77 \%$; sulfate ions as $\mathrm{SO}_{4}, 0.80 \%$; copper as metal, $43.63 \%$.

### 3.5. ALL COPPER ALLOYS

The effects of the duration of exposure on the corrosion of all the copper alloys in seawater at the surface and at the 6,000-foot depth are summarized in Figure 15. Their rates of corrosion decreased essentially linearly with increasing duration of exposure. Their rates were also comparable and were essentially the same after 1,064 days of exposure.

The corrosion of all the copper alloys was not affected by depth as shown in Figure 9.

The corrosion of copper and the silicon bronzes was not influenced by changes in the concentration of oxygen in seawater during 1 year of exposure, while that of the other alloys increased with increasing oxygen concentration, as shown in Figure 10.

None of the copper alloys were susceptible to stress corrosion.

The mechanical properties of copper, the beryllium-copper alloys, copper-nickel alloys, phosphor bronzes A and D, and Arsenical Admiralty brass were not adversely affected by exposure in seawater either at the surface or at depth. Those of $5 \%$ aluminum bronze, silicon bronze A, Muntz Metal, and nickel-manganese bronze were adversely affected.

Aluminum alloy 7075-T6 was galvanically corroded when in contact with beryllium-copper.

All the brasses containing from 10 to $42 \%$ zinc, except Arsenical Admiralty, aluminum brass, nickel brass, all the aluminum bronzes, and the silicon bronzes, were attacked by parting corrosion.

Corrosion products consisted of cupric chloride ( CuCl$)_{2} \cdot \mathrm{H}_{2} \mathrm{O}$ ), copper oxychloride $\left(\mathrm{Cu}_{2}\left(\mathrm{OH}_{3} \mathrm{Cl}\right)\right.$, nickel hydroxide $\left(\mathrm{Ni}(\mathrm{OH})_{2}\right)$, copper, aluminum, nickel, iron, silicon, sodium, magnesium, manganese, calcium, chloride ion, and sulfate ion.

Figure 8. Effect of the duration of exposure on the corrosion of copper in seawater.


Figure 9. Effect of depth on the corrosion of copper alloys after 1 year of exposure in seawater.




Table 10. Chemical Composition of Coppers, Percent by Weight

| Alloy | CDA No. $^{a}$ | Cu | Ni | Be | Co | Source $^{b}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :--- |
| Copper, O free | 102 | 99.96 | - | - | - | CEL (4) |
| Copper, O free | 102 | 99.9 | - | - | - | INCO (3) |
| Copper, O free | 102 | 99.97 | - | - | - | CEL (4) |
| Copper, O free | 102 | 99.9 | - | - | - | MEL (5) |
| $\mathrm{Be-Cu}$ | 170 | 97.06 | - | 2.02 | 0.58 | NADC (7) |
| $\mathrm{Be-Cu}$ | 172 | 97.42 | - | 2.02 | 0.54 | NADC (7) |
| $\mathrm{Be-Cu}$ | 172 | 97.80 | 0.05 | 1.90 | 0.25 | CEL (4) |
| $\mathrm{Be-Cu}$ | 175 | 97.06 | - | 0.61 | 2.4 | NADC (7) |
| $\mathrm{Be-Cu}$ chain $^{c}$ | 825 | remainder | - | 2.0 | 0.5 | CEL (4) |

${ }^{a}$ Copper Development Association alloy number.
${ }^{b}$ Numbers refer to references at end of report.
${ }^{c}$ Cast alloy.

Table 11. Corrosion Rates and Types of Corrosion of Coppers

| Alloy | CDA No. ${ }^{\text {a }}$ | Environment ${ }^{\text {b }}$ | Exposure <br> (day) | Depth <br> (ft) | Corrosion <br> Rate <br> (mpy) | Type of Corrosion ${ }^{c}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Copper, O free | 102 | W | 123 | 5,640 | 1.6 | U | CEL (4) |
| Copper, O free | 102 | W | 123 | 5,640 | 1.5 | U | INCO (3) |
| Copper, O free | 102 | S | 123 | 5,640 | 1.3 | U | CEL (4) |
| Copper, O free | 102 | S | 123 | 5,640 | 1.5 | U | INCO (3) |
| Copper, O free | 102 | W | 123 | 5,640 | 1.9 | U | MEL (5) |
| Copper, O free | 102 | W | 403 | 6,780 | 1.2 | U | CEL (4) |
| Copper, O free | 102 | W | 403 | 6,780 | 1.3 | U | INCO (3) |
| Copper, O free | 102 | S | 403 | 6,780 | 1.1 | U | CEL (4) |
| Copper, O free | 102 | S | 403 | 6,780 | $<0.1$ | U | INCO (3) |
| Copper, O free | 102 | W | 751 | 5,640 | 0.7 | U | CEL (4) |
| Copper, O free | 102 | W | 751 | 5,640 | 1.0 | U | INCO (3) |
| Copper, O free | 102 | S | 751 | 5,640 | 0.7 | U | INCO (3) |
| Copper, O free | 102 | W | 751 | 5,640 | 0.6 | U | MEL (5) |
| Copper, O free | 102 | W | 1,064 | 5,300 | 0.5 | U | CEL (4) |
| Copper, O free | 102 | W | 1,064 | 5,300 | 0.5 | U | INCO (3) |
| Copper, O free | 102 | S | 1,064 | 5,300 | 0.3 | G | INCO (3) |
| Copper, O free | 102 | W | 1,064 | 5,300 | 0.4 | U | MEL (5) |
| Copper, O free | 102 | W | 197 | 2,340 | 0.8 | U | CEL (4) |

Table 11. Continued

| Alloy | CDA No. ${ }^{a}$ | Environment ${ }^{b}$ | Exposure <br> (day) | Depth (ft) | Corrosion Rate (mpy) | Type of Corrosion ${ }^{\text {c }}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Copper, O free | 102 | W | 197 | 2,340 | 1.4 | U | INCO (3) |
| Copper, O free | 102 | S | 197 | 2,340 | 0.2 | U | CEL (4) |
| Copper, O free | 102 | S | 197 | 2,340 | 0.2 | U | INCO (3) |
| Copper, O free | 102 | W | 402 | 2,370 | 0.9 | U | CEL (4) |
| Copper, O free | 102 | W | 402 | 2,370 | 1.4 | U | INCO (3) |
| Copper, O free | 102 | S | 402 | 2,370 | 0.6 | U | CEL (3) |
| Copper, O free | 102 | S | 402 | 2,370 | 0.2 | ET | INCO (3) |
| Copper, O free | 102 | W | 181 | 5 | 1.4 | P (22m) | CEL (4) |
| Copper, O free | 102 | W | 181 | 5 | 1.8 | G | INCO (3) |
| Copper, O free | 102 | W | 366 | 5 | 1.2 | G | INCO (3) |
| Copper, O free ${ }^{e}$ | 102 | W | 386 | 5 | 0.7 | U | MEL (5) |
| Copper, O free | 102 | W | 398 | 5 | 1.1 | G, P (37m) | CEL (4) |
| Copper, O free | 102 | W | 540 | 5 | 0.9 | G, P (22m) | CEL (4) |
| Copper, O free | 102 | W | 588 | 5 | 0.9 | G, P (20m) | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}{ }^{f}$ | 170 | W | 123 | 5,640 | 0.5 | U | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 170 | W | 751 | 5,640 | 0.5 | U | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 170 | W | 197 | 2,340 | 0.0 | NC | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 170 | W | 402 | 2,370 | 0.6 | U | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172^{g}$ | W | 123 | 5,640 | 0.5 | U | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172^{\text {b }}$ | W | 123 | 5,640 | 0.5 | U | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172^{g}$ | W | 751 | 5,640 | 0.5 | U | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172{ }^{\text {b }}$ | W | 751 | 5,640 | 0.5 | U | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172{ }^{\text {g }}$ | W | 197 | 2,340 | 0.2 | U | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172^{\text {b }}$ | W | 197 | 2,340 | 0.2 | U | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172^{g}$ | W | 402 | 2,370 | 0.5 | U | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172^{\text {b }}$ | W | 402 | 2,370 | 0.4 | U | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 172 | W | 402 | 2,370 | 0.6 | U | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}$ | 172 | S | 402 | 2,370 | 0.5 | U | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}$ | 172 | W | 181 | 5 | 0.1 | U | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}$ | 172 | W | 364 | 5 | 1.1 | U | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}$ | 172 | W | 723 | 5 | 0.8 | U | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}$ | 172 | W | 763 | 5 | 0.8 | U | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{i}$ | 172 | W | 402 | 2,370 | 0.5 | $\mathrm{U}^{j}$ | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{i}$ | 172 | S | 402 | 2,370 | 0.5 | $\mathrm{U}^{j}$ | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{2}$ | 172 | W | 181 | 5 | 0.1 | $\mathrm{U}^{j}$ | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{i}$ | 172 | W | 364 | 5 | 1.0 | U | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{i}$ | 172 | W | 723 | 5 | 0.7 | U | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{i}$ | 172 | W | 763 | 5 | 0.8 | U | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{k}$ | 172 | W | 402 | 2,370 | 0.6 | ET | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{k}$ | 172 | S | 402 | 2,370 | 0.5 | ET | CEL (4) |

Table 11. Continued

| Alloy | CDA No. ${ }^{\text {a }}$ | Environment ${ }^{\text {b }}$ | Exposure <br> (day) | Depth <br> (ft) | Corrosion Rate (mpy) | Type of Corrosion ${ }^{\text {c }}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Be}-\mathrm{Cu}^{k}$ | 172 | W | 181 | 5 | 0.1 | ET | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{k}$ | 172 | W | 364 | 5 | 1.1 | U | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{k}$ | 172 | W | 723 | 5 | 0.7 | U | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{k}$ | 172 | W | 763 | 5 | 0.7 | U | CEL (4) |
| Be-Cu chain ${ }^{l}$ | 825 | W | 402 | 2,370 | 0.5 | U | CEL (4) |
| Be-Cu chain ${ }^{l}$ | 825 | S | 402 | 2,370 | 0.4 | U | CEL (4) |
| Be-Cu chain ${ }^{l}$ | 825 | W | 181 | 5 | 0.1 | U | CEL (4) |
| Be-Cu chain ${ }^{l}$ | 825 | W | 364 | 5 | 1.0 | U | CEL (4) |
| Be-Cu chain ${ }^{l}$ | 825 | W | 723 | 5 | 0.8 | U | CEL (4) |
| Be-Cu chain ${ }^{l}$ | 825 | W | 763 | 5 | 0.8 | $\mathrm{P}(30.5 \mathrm{~m}), \mathrm{C}(7 \mathrm{~m})$ | CEL (4) |

${ }^{a}$ Copper Development Association alloy numbers.
${ }^{b_{W}}{ }_{\mathrm{W}}=$ totally exposed in seawater on sides of structure.
$S$ = exposed in base of structure so that the lower portions of the specimens were embedded in the bottom sediments.
${ }^{c}$ Symbols for types of corrosion:
$\mathrm{C}=$ Crevice
$\mathrm{P}=$ Pitting
$\mathrm{ET}=$ Etching
$\mathrm{U}=$ Uniform
$\mathrm{G}=$ General NC $=$ No corrosion
Numbers indicate mils (i.e., $20=20 \mathrm{mils} ; 20 \mathrm{~m}=20$ mils maximum)
${ }^{d}$ Numbers refer to references at end of report.
${ }^{e}$ Francis L. LaQue Corrosion Laboratory, INCO, Wrightsville Beach, N.C.
$f_{\text {Beryllium-copper }}$
$g_{\text {Aged at }} 600^{\circ} \mathrm{F}$
${ }^{h}$ Aged at $800^{\circ} \mathrm{F}$
${ }^{i}$ MIG weld
${ }^{j}$ Uniform, weld bead etched
${ }^{k}$ TIG weld
${ }^{l}$ Cast

Table 12. Stress Corrosion of Coppers

| Alloy | CDA No. ${ }^{\text {a }}$ | Stress <br> (ksi) | Yield Strength (\%) | Exposure <br> (day) | Depth <br> (ft) | Specimens |  | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |  | Exposed | Failed |  |
| Copper, O free | 102 | 11 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Copper, O free | 102 | 11 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{\text {c }}$ | 170 | 50 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 170 | 126 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 170 | 50 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 170 | 126 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172{ }^{\text {d }}$ | 53 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172{ }^{\text {d }}$ | 133 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172{ }^{\text {d }}$ | 53 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172{ }^{\text {d }}$ | 133 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172{ }^{e}$ | 37 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172{ }^{e}$ | 93 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172{ }^{e}$ | 37 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | $172^{e}$ | 93 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 175 | 32 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 175 | 80 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 175 | 32 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 175 | 80 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |

${ }^{a}$ Copper Development Association alloy number.
${ }^{b}$ Numbers refer to references at end of report.
${ }^{c}$ Beryllium-copper
${ }^{d}$ Aged at $600^{\circ} \mathrm{F}$
${ }^{e}$ Aged at $800^{\circ} \mathrm{F}$

Table 13. Changes in Mechanical Properties of Coppers Due to Corrosion

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| Copper, O free | 102 | 123 | 5,640 | 33 | +2 | 14 | -6 | 52 | -2 | CEL (4) |
| Copper, O free | 102 | 403 | 6,780 | 33 | 0 | 14 | +9 | 52 | -6 | CEL (4) |
| Copper, O free | 102 | 751 | 5,640 | 33 | +4 | 14 | +11 | 52 | -6 | CEL (4) |
| Copper, O free | 102 | 197 | 2,340 | 33 | -8 | 14 | -18 | 52 | -4 | CEL (4) |
| Copper, O free | 102 | 402 | 2,370 | 33 | +2 | 14 | +4 | 52 | -7 | CEL (4) |
| Copper, O free | 102 | 181 | 5 | 33 | +4 | 14 | +20 | 52 | -14 | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{\text {c }}$ | 170 | 123 | 5,640 | 188 | 0 | 167 | 0 | 5 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 170 | 751 | 5,640 | 188 | -2 | 167 | -2 | 5 | -6 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 170 | 197 | 2,340 | 188 | +11 | 167 | +10 | 5 | +4 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 170 | 402 | 2,370 | 188 | -7 | 167 | -6 | 5 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 172 | 402 | 2,370 | 176 | -9 | 162 | -8 | 4 | -14 | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}{ }^{\text {d }}$ | 172 | 402 | 2,370 | 161 | -5 | 158 | -2 | 3 | +18 | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}^{e}$ | 172 | 402 | 2,370 | 166 | -1 | 162 | -6 | 3 | -17 | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}$ | 172 | 181 | 5 | 176 | -7 | 162 | -6 | 4 | -29 | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}{ }^{\text {d }}$ | 172 | 181 | 5 | 158 | +3 | 157 | -1 | 4 | -29 | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}{ }^{\text {e }}$ | 172 | 181 | 5 | 166 | +3 | 162 | -4 | 3 | -17 | CEL (4) |
| $\mathrm{Be}-\mathrm{Cu}{ }_{f}$ | 172 | 123 | 5,640 | 196 | 0 | 177 | 0 | 4 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu} f^{\text {f }}$ | 172 | 751 | 5,640 | 196 | -1 | 177 | -2 | 4 | -19 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}{ }^{f}$ | 172 | 197 | 2,340 | 196 | 0 | 177 | 0 | 4 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}{ }^{f}$ | 172 | 402 | 2,370 | 196 | -8 | 177 | -13 | 4 | -26 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}^{\mathrm{g}}$ | 172 | 123 | 5,640 | 144 | -2 | 124 | -7 | 9 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}{ }^{\text {g }}$ | 172 | 751 | 5,640 | 144 | -4 | 124 | -10 | 9 | +26 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}{ }^{\text {g }}$ | 172 | 197 | 2,340 | 144 | +6 | 124 | +10 | 9 | +25 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}{ }^{g}$ | 172 | 402 | 2,370 | 144 | -6 | 124 | -12 | 9 | +26 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 175 | 123 | 5,640 | 121 | 0 | 107 | 0 | 15 | -20 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 175 | 751 | 5,640 | 121 | -4 | 107 | 0 | 15 | -13 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 175 | 197 | 2,340 | 121 | 0 | 107 | 0 | 15 | 0 | NADC (7) |
| $\mathrm{Be}-\mathrm{Cu}$ | 175 | 402 | 2,370 | 121 | -3 | 107 | 0 | 15 | -23 | NADC (7) |

${ }^{a}$ Copper Development Association alloy number.
${ }^{b}$ Numbers refer to references at end of report.
${ }^{c}$ Beryllium-copper.
${ }^{d}{ }_{\text {MIG welded. }}$
${ }^{e}$ TIG welded.
$f_{\text {Aged at } 600^{\circ} \mathrm{F} \text {. }}^{\text {. }}$
$g_{\text {Aged at } 800^{\circ} \mathrm{F} \text {. }}$
Table 14. Chemical Composition of Copper-Zinc Alloys (Brasses), Percent by Weight

| Alloy | CDA No. ${ }^{\text {a }}$ | Cu | Zn | Sn | Ni | AI | Mn | Fe | Other | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Commercial bronze | 220 | 90 | 10 | - | - | - | -- | - | - | INCO (3) |
| Red brass | 230 | 85 | 15 | -- | - | - | - | - | - | INCO (3) |
| Red brass | 230 | 85 | 15 | - | - | - | - | - | - | MEL (5) |
| Commercial brass | 268 | 66.47 | 33.51 | $<0.05$ | - | - | - | 0.02 | $<0.01 \mathrm{~Pb}$ | CEL (4) |
| Yellow brass | 270 | 68.48 | 31.50 | $<0.05$ | - | - | - | 0.02 | $<0.01 \mathrm{~Pb}$ | CEL (4) |
| Yellow brass | 270 | 65.0 | 35.0 | -- | - | - | - | . | - | INCO (3) |
| Muntz metal | 280 | 60.69 | 39.29 | - | - | - | - | $<0.02$ | - | CEL (4) |
| Muntz metal | 280 | 60.0 | 40.0 | - | - | - | - | - | - | INCO (3) |
| Arsenical admiralty | 443 | 71.19 | 27.77 | 1.00 | - | - | - | 0.01 | 0.027 As | CEL (4) |
| Arsenical admiralty | 443 | 70.0 | 29.0 | 1.0 | - | - | - | - | 0.04 As | INCO (3) |
| Naval brass | 464 | 60.46 | 38.74 | 0.69 | - | - | - | 0.03 | 0.08 Pb | CEL (4) |
| Naval brass | 464 | 60.2 | 39.8 | 0.8 | - | - | - | - | -- | MEL (5) |
| Tobin bronze | - | 58.94 | 39.07 | 0.89 | nil | $<0.10$ | - | 1.10 | $<0.05 \mathrm{~Pb}$ | CEL (4) |
| Mn bronze A | 675 | 56.0 | 42.0 | - | - | 1.0 | 0.1 | 1.0 | - | INCO (3) |
| Mn bronze B | 670 | 68.0 | 21.0 | - | - | 5.0 | 4.0 | 2.0 | - | MEL (5) |
| Ni-Mn bronze ${ }^{c}$ | 868 | 54.58 | 34.48 | 0.70 | 3.77 | 1.73 | 3.06 | 1.66 | 0.02 Pb | CEL (4) |
| Al brass | - | 78.0 | 20.0 | - | - | 2.0 | - | - | - | INCO (3) |
| Ni brass | - | 50.0 | 40.0 | - | 8.0 | - | - | 2.0 | - | INCO (3) |
| ${ }^{\text {a }}$ Copper Development Association alloy number. |  |  |  |  |  |  |  |  |  |  |
| ${ }^{b}$ Numbers refer to references at end of report. |  |  |  |  |  |  |  |  |  |  |

Table 15. Corrosion Rates and Types of Corrosion of Copper-Zinc Alloys (Brasses)

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Environment ${ }^{\text {b }}$ | Exposure (day) | Depth <br> (ft) | Corrosion <br> Rate <br> (mpy) | Type of Corrosion ${ }^{c}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Commercial Bronze | 220 | W | 123 | 5,640 | 0.6 | U | INCO (3) |
| Commercial Bronze | 220 | S | 123 | 5,640 | 0.3 | U | INCO (3) |
| Commercial Bronze | 220 | W | 403 | 6,780 | 0.6 | U | INCO (3) |
| Commercial Bronze | 220 | S | 403 | 6,780 | $<0.1$ | U | INCO (3) |
| Commercial Bronze | 220 | W | 751 | 5,640 | 0.6 | C (9) | INCO (3) |
| Commercial Bronze | 220 | S | 751 | 5,640 | 0.4 | C (20) | INCO (3) |
| Commercial Bronze | 220 | W | 1,064 | 5,300 | 0.4 | C (20) | INCO (3) |
| Commercial Bronze | 220 | S | 1,064 | 5,300 | 0.6 | U | INCO (3) |
| Commercial Bronze | 220 | W | 197 | 2,340 | 0.3 | U | INCO (3) |
| Commercial Bronze | 220 | S | 197 | 2,340 | 0.1 | NU-ET | INCO (3) |
| Commercial Bronze | 220 | W | 402 | 2,370 | 0.2 | SL-DZ | INCO (3) |
| Commercial Bronze | 220 | S | 402 | 2,370 | <0.1 | SL-DZ | INCO (3) |
| Commercial Bronze | 220 | W | 181 | 5 | 1.1 | U | INCO (3) |
| Commercial Bronze | 220 | W | 366 | 5 | 1.1 | P (4) | INCO (3) |
| Red Brass | 230 | W | 123 | 5,640 | 1.3 | U | INCO (3) |
| Red Brass | 230 | S | 123 | 5,640 | 1.7 | SL-DZ | INCO (3) |
| Red Brass | 230 | W | 123 | 5,640 | 1.9 | U | MEL (5) |
| Red Brass | 230 | W | 403 | 6,780 | 1.2 | SL-DZ | INCO (3) |
| Red Brass | 230 | S | 403 | 6,780 | 0.4 | GBSL | INCO (3) |
| Red Brass | 230 | W | 751 | 5,640 | 0.9 | SL-DZ | INCO (3) |
| Red Brass | 230 | S | 751 | 5,640 | 0.7 | U | INCO (3) |
| Red Brass | 230 | W | 751 | 5,640 | 0.7 | U | MEL (5) |
| Red Brass | 230 | W | 1,064 | 5,300 | 0.6 | SL-DZ | INCO (3) |
| Red Brass | 230 | S | 1,064 | 5,300 | 0.3 | G | INCO (3) |
| Red Brass | 230 | W | 1,064 | 5,300 | 0.6 | U | MEL (5) |
| Red Brass | 230 | W | 197 | 2,340 | 1.0 | U | INCO (3) |
| Red Brass | 230 | S | 197 | 2,340 | 0.1 | U | INCO (3) |
| Red Brass | 230 | W | 402 | 2,370 | 0.7 | U | INCO (3) |
| Red Brass | 230 | S | 402 | 2,370 | $<0.1$ | ET | INCO (3) |
| Red Brass | 230 | W | 181 | 5 | 1.8 | SL-DZ | INCO (3) |
| Red Brass | 230 | W | 366 | 5 | 1.2 | CR (6) | INCO (3) |
| Red Brass ${ }^{e}$ | 230 | W | 386 | 5 | 0.8 | U | MEL (5) |
| Commercial Brass | 268 | W | 1,064 | 5,300 | 0.8 | S-DZ | CEL (4) |
| Yellow Brass | 268 | W | 1,064 | 5,300 | 0.6 | MO-DZ | CEL (4) |
| Yellow Brass | 270 | W | 123 | 5,640 | 1.4 | U | INCO (3) |
| Yellow Brass | 270 | S | 123 | 5,640 | 1.3 | U | INCO (3) |
| Yellow Brass | 270 | W | 403 | 6,780 | 1.0 | U | INCO (3) |
| Yellow Brass | 270 | S | 403 | 6,780 | 0.2 | GBSL | INCO (3) |
| Yellow Brass | 270 | W | 751 | 5,640 | 2.5 | SL-DZ | INCO (3) |
| Yellow Brass | 270 | S | 751 | 5,640 | 0.6 | SL-DZ | INCO (3) |

Continued

Table 15. Continued.

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Environment ${ }^{\text {b }}$ | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) | Type of Corrosion ${ }^{\text {c }}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Yellow Brass | 270 | W | 1,064 | 5,300 | 0.6 | U | INCO (3) |
| Yellow Brass | 270 | S | 1,064 | 5,300 | 0.5 | U | INCO (3) |
| Yellow Brass | 270 | W | 197 | 2,340 | 0.9 | U | INCO (3) |
| Yellow Brass | 270 | S | 197 | 2,340 | 0.2 | NU-ET | INCO (3) |
| Yellow Brass | 270 | W | 402 | 2,370 | 0.9 | U | INCO (3) |
| Yellow Brass | 270 | S | 402 | 2,370 | 0.1 | ET | INCO (3) |
| Yellow Brass | 270 | W | 181 | 5 | 2.1 | U | INCO (3) |
| Yellow Brass | 270 | W | 366 | 5 | 1.3 | U | INCO (3) |
| Muntz Metal | 280 | W | 123 | 5,640 | 1.6 | SL-DZ | CEL (4) |
| Muntz Metal | 280 | W | 123 | 5,640 | 2.1 | U | INCO (3) |
| Muntz Metal | 280 | S | 123 | 5,640 | 1.3 | SL-DZ | CEL (4) |
| Muntz Metal | 280 | S | 123 | 5,640 | 1.5 | U | INCO (3) |
| Muntz Metal | 280 | W | 403 | 6,780 | 2.6 | SL-DZ | CEL (4) |
| Muntz Metal | 280 | W | 403 | 6,780 | 3.3 | S-DZ | INCO (3) |
| Muntz Metal | 280 | S | 403 | 6,780 | 1.8 | SL-DZ | CEL (4) |
| Muntz Metal | 280 | S | 403 | 6,780 | 0.6 | GBSL | INCO (3) |
| Muntz Metal | 280 | W | 751 | 5,640 | 3.2 | G-DZ | CEL (4) |
| Muntz Metal | 280 | W | 751 | 5,640 | 4.0 | S-DZ | INCO (3) |
| Muntz Metal | 280 | S | 751 | 5,640 | 1.7 | S-DZ | INCO (3) |
| Muntz Metal | 280 | W | 1,064 | 5,300 | 2.3 | S-DZ | INCO (3) |
| Muntz Metal | 280 | S | 1,064 | 5,300 | 0.8 | U | INCO (3) |
| Muntz Metal | 280 | W | 197 | 2,340 | 0.7 | SL-DZ; P (10) | CEL (4) |
| Muntz Metal | 280 | W | 197 | 2,340 | 0.7 | U | INCO (3) |
| Muntz Metal | 280 | S | 197 | 2,340 | 0.5 | SL-DZ; P (5) | CEL (4) |
| Muntz Metal | 280 | S | 197 | 2,340 | <0.1 | SL-DZ ${ }^{f}$ | INCO (3) |
| Muntz Metal | 280 | W | 402 | 2,370 | 0.7 | SL-DZ | CEL (4) |
| Muntz Metal | 280 | W | 402 | 2,370 | 0.7 | SL-DZ | INCO (3) |
| Muntz Metal | 280 | S | 402 | 2,370 | 0.6 | SL-DZ | CEL (4) |
| Muntz Metal | 280 | S | 402 | 2,370 | 0.1 | ET | INCO (3) |
| Muntz Metal | 280 | W | 181 | 5 | 2.4 | DZ | CEL (4) |
| Muntz Metal | 280 | W | 181 | 5 | 3.4 | SL-DZ | INCO (3) |
| Muntz Metal | 280 | W | 366 | 5 | 3.7 | S-DZ | INCO (3) |
| Muntz Metal | 280 | W | 398 | 5 | 3.1 | DZ, P (6) | CEL (4) |
| Muntz Metal | 280 | W | 540 | 5 | 3.4 | DZ, I-P | CEL (4) |
| Muntz Metal | 280 | W | 588 | 5 | 3.3 | DZ, I-P | CEL (4) |
| Arsenical Admiralty | 443 | W | 123 | 5,640 | 1.0 | U | CEL (4) |
| Arsenical Admiralty | 443 | W | 123 | 5,640 | 1.1 | U | INCO (3) |
| Arsenical Admiralty | 443 | S | 123 | 5,640 | 1.0 | U | CEL (4) |
| Arsenical Admiralty | 443 | S | 123 | 5,640 | 1.0 | U | INCO (3) |
| Arsenical Admiralty | 443 | W | 403 | 6,780 | 0.7 | U | CEL (4) |
| Arsenical Admiralty | 443 | W | 403 | 6,780 | 0.8 | U | INCO (3) |

Table 15. Continued.

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Environment ${ }^{\text {b }}$ | Exposure (day) | Depth <br> (ft) | Corrosion <br> Rate <br> (mpy) | Type of Corrosion ${ }^{\text {c }}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Arsenical Admiralty | 443 | S | 403 | 6,780 | 0.8 | U | CEL (4) |
| Arsenical Admiralty | 443 | S | 403 | 6,780 | 0.2 | GBSL | INCO (3) |
| Arsenical Admiralty | 443 | W | 751 | 5,640 | 0.6 | U | CEL (4) |
| Arsenical Admiralty | 443 | W | 751 | 5,640 | 0.7 | U | INCO (3) |
| Arsenical Admiralty | 443 | S | 751 | 5,640 | 0.4 | U | CEL (4) |
| Arsenical Admiralty | 443 | S | 751 | 5,640 | 0.5 | U | INCO (3) |
| Arsenical Admiralty | 443 | W | 1,064 | 5,300 | 0.5 | U | INCO (3) |
| Arsenical Admiralty | 443 | S | 1,064 | 5,300 | 0.5 | U | INCO (3) |
| Arsenical Admiralty | 443 | W | 197 | 2,340 | 0.6 | U | CEL (4) |
| Arsenical Admiralty | 443 | W | 197 | 2,340 | 1.0 | U | INCO (3) |
| Arsenical Admiralty | 443 | S | 197 | 2,340 | 0.2 | U | CEL (4) |
| Arsenical Admiralty | 443 | S | 197 | 2,340 | $<0.1$ | U | INCO (3) |
| Arsenical Admiralty | 443 | W | 402 | 2,370 | 0.6 | U | CEL (4) |
| Arsenical Admiralty | 443 | W | 402 | 2,370 | 0.6 | U | INCO (3) |
| Arsenical Admiralty | 443 | S | 402 | 2,370 | 0.4 | U | CEL (4) |
| Arsenical Admiralty | 443 | S | 402 | 2,370 | 0.1 | ET | INCO (3) |
| Arsenical Admiralty | 443 | W | 181 | 5 | 1.3 | U | CEL (4) |
| Arsenical Admiralty | 443 | W | 181 | 5 | 1.8 | G | INCO (3) |
| Arsenical Admiralty | 443 | W | 366 | 5 | 1.3 | U | INCO (3) |
| Arsenical Admiralty | 443 | W | 608 | 5 | 1.1 | U; I-P | CEL (4) |
| Naval Brass | 464 | W | 123 | 5,640 | 1.2 | U | MEL (5) |
| Naval Brass | 464 | W | 751 | 5,640 | 0.6 | U | MEL (5) |
| Naval Brass | 464 | W | 1,064 | 5,300 | 0.7 | U | MEL (5) |
| Naval Brass | 464 | W | 1,064 | 5,300 | 1.0 | S-DZ | CEL (4) |
| Naval Brass ${ }^{e}$ | 464 | W | 364 | 5 | 0.7 | U | MEL (5) |
| Tobin Bronze | - | W | 1,064 | 5,300 | 0.9 | S-DZ | CEL (4) |
| Mn Bronze A | 675 | W | 123 | 5,640 | 2.9 | EX-DZ | INCO (3) |
| Mn Bronze A | 675 | S | 123 | 5,640 | 2.0 | EX-DZ | INCO (3) |
| Mn Bronze A | 675 | W | 403 | 6,780 | 2.7 | S-DZ | INCO (3) |
| Mn Bronze A | 675 | S | 403 | 6,780 | 0.9 | S-DZ | INCO (3) |
| Mn Bronze A | 675 | W | 751 | 5,640 | 7.2 | S-DZ | INCO (3) |
| Mn Bronze A | 675 | S | 751 | 5,640 | 2.6 | V-S-DZ | INCO (3) |
| Mn Bronze A | 675 | W | 1,064 | 5,300 | 2.0 | S-DZ | INCO (3) |
| Mn Bronze A | 675 | S | 1,064 | 5,300 | 1.2 | S-DZ | INCO (3) |
| Mn Bronze A | 675 | W | 197 | 2,340 | 1.2 | S-DZ | INCO (3) |
| Mn Bronze A | 675 | S | 197 | 2,340 | 0.2 | SL-DZ | INCO (3) |
| Mn Bronze A | 675 | W | 402 | 2,370 | 0.8 | S-DZ | INCO (3) |
| Mn Bronze A | 675 | S | 402 | 2,370 | <0.1 | SL-DZ | INCO (3) |
| Mn Bronze A | 675 | W | 181 | 5 | 4.8 | S-DZ | INCO (3) |
| Mn Bronze A | 675 | W | 366 | 5 | 1.9 | S-DZ | INCO (3) |

Continued

Table 15. Continued.

| Alloy | $\begin{array}{\|l\|l} \text { CDA } \\ \text { No. } \end{array}$ | Environment ${ }^{\text {b }}$ | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) | Type of Corrosion ${ }^{c}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Mn Bronze B | 670 | W | 123 | 5,640 | 1.5 | DZ | MEL (5) |
| Mn Bronze B | 670 | W | 751 • | 5,640 | 3.0 | DZ | MEL (5) |
| Ni-Mn Bronze ${ }^{g}$ | 868 | w | 123 | 5,640 | 0.5 | SL-DZ | CEL (4) |
| Ni-Mn Bronze | 868 | W | 403 | 6,780 | 0.4 | MD-DZ | CEL (4) |
| Ni-Mn Bronze | 868 | S | 403 | 6,780 | 0.5 | MD-DZ | CEL (4) |
| Ni-Mn Bronze | 868 | w | 751 | 5,640 | 2.3 | V-S-DZ | CEL (4) |
| Ni-Mn Bronze | 868 | w | 197 | 2,340 | 0.4 | SL-DZ | CEL (4) |
| Ni-Mn Bronze | 868 | S | 197 | 2,340 | 0.4 | SL-DZ | CEL (4) |
| Ni-Mn Bronze | 868 | W | 402 | 2,340 | 1.6 | SL-DZ | CEL (4) |
| Ni-Mn Bronze | 868 | S | 402 | 2,340 | 2.9 | SL-DZ | CEL (4) |
| Ni-Mn Bronze | 868 | W | 181 | 5 | <0.1 | SL-DZ | CEL (4) |
| Ni-Mn Bronze | 868 | W | 364 | 5 | 0.7 | DZ | CEL (4) |
| Ni-Mn Bronze | 868 | W | 723 | 5 | 2.9 | DZ | CEL (4) |
| Ni-Mn Bronze | 868 | W | 763 | 5 | 3.0 | DZ | CEL (4) |
| Al Brass | - | w | 123 | 5,640 | 0.7 | U | INCO (3) |
| Al Brass | - | S | 123 | 5,640 | 0.5 | U | INCO (3) |
| Al Brass | - | W | 403 | 6,780 | 0.4 | U | INCO (3) |
| Al Brass | - | S | 403 | 6,780 | 0.1 | GBSL | INCO (3) |
| Al Brass | - | W | 751 | 5,640 | 0.3 | U | INCO (3) |
| Al Brass | - | S | 751 | 5,640 | 0.1 | U | INCO (3) |
| Al Brass | - | W | 1,064 | 5,300 | 0.2 | U | INCO (3) |
| Al Brass | - | S | 1,064 | 5,300 | 0.8 | G | INCO (3) |
| Al Brass | - | W | 197 | 2,340 | 0.5 | U | INCO (3) |
| Al Brass | - | S | 197 | 2,340 | $<0.1$ | U | INCO (3) |
| Al Brass | - | W | 402 | 2,370 | 0.3 | U | INCO (3) |
| Al Brass | - | S | 402 | 2,370 | 0.1 | ET | INCO (3) |
| Al Brass | - | W | 181 | 5 | 0.8 | G | INCO (3) |
| Al Brass | - | W | 366 | 5 | 0.4 | P (4) | INCO (3) |
| Ni Brass | - | W | 123 | 5,640 | 1.3 | U | INCO (3) |
| Ni Brass | - | S | 123 | 5,640 | 1.1 | U | INCO (3) |
| Ni Brass | - | W | 403 | 6,780 | 1.3 | U | INCO (3) |
| Ni Brass | - | S | 403 | 6,780 | 0.2 | GBSL | INCO (3) |
| Ni Brass | - | W | 751 | 5,640 | 1.0 | U | INCO (3) |
| Ni Brass | - | S | 751 | 5,640 | 0.7 | U | INCO (3) |
| Ni Brass | - | W | 1,064 | 5,300 | 0.8 | U | INCO (3) |
| Ni Brass | - | S | 1,064 | 5,300 | 0.5 | U | INCO (3) |
| Ni Brass | - | W | 197 | 2,340 | 0.8 | U | INCO (3) |
| Ni Brass | - | S | 197 | 2,340 | <0.1 | NU-ET | INCO (3) |
| Ni Brass | - | w | 402 | 2,370 | 0.7 | U | INCO (3) |
| Ni Brass | - | S | 402 | 2,370 | <0.1 | ET | INCO (3) |

Continued

Table 15. Continued.

| Alloy | CDA <br> No. $^{a}$ | Environment ${ }^{b}$ | Exposure <br> (day) | Depth <br> $(\mathrm{ft})$ | Corrosion <br> Rate <br> $(\mathrm{mpy})$ | Type of <br> Corrosion $^{c}$ | Source $^{d}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Ni Brass | - | W | 181 | 5 | 1.1 | U | INCO (3) |
| Ni Brass | - | W | 366 | 5 | 0.9 | U | $\mathrm{INCO}(3)$ |

${ }^{a}$ Copper Development Association alloy number.
${ }^{b} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimen were embedded in the bottom sediments.
${ }^{\text {c}}$ Symbols for types of corrosion:

| C | $=$ Crevice | MD $=$ Medium |
| :--- | :--- | :--- |
| CR | $=$ Cratering | MO $=$ Moderate |
| DZ | $=$ Dezincification | NU $=$ Nonuniform |
| ET | $=$ Etching | $\mathrm{P}=$ Pitting |
| EX | $=$ Extensive | $\mathrm{S}=$ Severe |
| G | $=$ General | $\mathrm{SL}=$ Slight |
| GBSL | $=$ General below sediment line | $\mathrm{U}=$ Uniform |
| I | $=$ Incipient | $\mathrm{V}=$ Very |

Numbers indicate maximum depth in mils.
${ }^{d}$ Numbers refer to references at end of report.
${ }^{e}$ Francis L. LaQue Corrosion Laboratory, INCO, Wrightsville Beach, N.C.
$f_{\text {At spacer. }}$
${ }^{g}$ Cast alloy.
Table 16. Stress Corrosion of Copper-Zinc Alloys (Brasses)

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. } \end{aligned}$ | Stress <br> (ksi) | Tensile Strength <br> (\%) | Exposure <br> (day) | Depth <br> (ft) | Specimens |  | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |  | Exposed | Failed |  |
| Arsenical admiralty | 443 | 10.0 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Arsenical admiralty | 443 | 14.0 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Arsenical admiralty | 443 | 9.5 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Arsenical admiralty | 443 | 14.3 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Arsenical admiralty | 443 | 9.0 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Arsenical admiralty | 443 | 13.4 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Muntz metal | 280 | 12.0 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Muntz metal | 280 | 18.0 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Muntz metal | 280 | 12.2 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Muntz metal | 280 | 18.3 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Muntz metal | 280 | 17.7 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Muntz metal | 280 | 26.5 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |

${ }^{a}$ Copper Development Association alloy number.
${ }^{b}$ Numbers refer to references at end of report.
Table 17. Changes in Mechanical Properties of Brasses Due to Corrosion

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. } \end{aligned}$ | Exposure (day) | Depth (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Original (ksi) | \% Change | Original <br> (ksi) | \% Change | Original <br> (\%) | \% Change |  |
| Muntz metal | 280 | 123 | 5,640 | 56 | -4 | 24 | -3 | 53 | -6 | CEL (4) |
| Muntz metal | 280 | 403 | 6,780 | 56 | -16 | 24 | -12 | 53 | -25 | CEL (4) |
| Muntz metal | 280 | 751 | 5,640 | 56 | -32 | 24 | -29 | 53 | -31 | CEL (4) |
| Muntz metal | 280 | 197 | 2,340 | 56 | -6 | 24 | -17 | 53 | -6 | CEL (4) |
| Muntz metal | 280 | 402 | 2,370 | 56 | -8 | 24 | -10 | 53 | -15 | CEL (4) |
| Muntz metal | 280 | 181 | 5 | 56 | $-7$ | 24 | -16 | 53 | -13 | CEL (4) |
| Arsenical admiralty | 443 | 123 | 5,640 | 51 | 0 | 19 | +3 | 66 | -2 | CEL (4) |
| Arsenical admiralty | 443 | 403 | 6,780 | 51 | -2 | 19 | -5 | 66 | +1 | CEL (4) |
| Arsenical admiralty | 443 | 751 | 5,680 | 51 | -2 | 19 | -1 | 66 | +1 | CEL (4) |
| Arsenical admiralty | 443 | 197 | 2,340 | 51 | -6 | 19 | -21 | 66 | -3 | CEL (4) |
| Arsenical admiralty | 443 | 402 | 2,370 | 51 | -2 | 19 | -3 | 66 | -1 | CEL (4) |
| Arsenical admiralty | 443 | 181 | 5 | 51 | -2 | 19 | -10 | 66 | -12 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mn}$ bronze | 868 | 123 | 5,640 | 71 | -2 | 31 | -10 | 20 | +25 | CEL (4) |
| Ni-Mn bronze | 868 | 403 | 6,780 | 71 | -33 | 31 | -8 | 20 | -60 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mn}$ bronze | 868 | 751 | 5,640 | 71 | -40 | 31 | -21 | 20 | -58 | CEL (4) |
| Ni-Mn bronze | 868 | 197 | 2,340 | 71 | -3 | 31 | -17 | 20 | +16 | CEL (4) |
| Ni -Mn bronze | 868 | 402 | 2,370 | 71 | -6 | 31 | +53 | 20 | -61 | CEL (4) |
| Ni-Mn bronze | 868 | 181 | 5 | 71 | -10 | 31 | +14 | 20 | -35 | CEL (4) |

"Copper Development Association alloy number.
${ }^{b}$ Numbers refer to references at end of report.

Table 18. Chemical Composition of Copper Alloys (Bronzes), Percent by Weight

| Alloy | CDA No. ${ }^{\text {a }}$ | Cu | Sn | Zn | Ni | Al | Fe | Si | Pb | P | Mn | Source ${ }^{b}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| G bronze ${ }^{c}$ | - | 88.0 | 2.0 | 10.0 | - | - | - | - | - | - | - | INCO (3) |
| Modified G bronze ${ }^{\text {c }}$ | - | 88.0 | 8.0 | 4.0 | - | - | - | - | - | - | - | INCO (3) |
| M bronze ${ }^{c}$ | - | 88.2 | 6.0 | 4.0 | - | - | - | - | 2.0 | - | - | INCO (3) |
| Leaded Sn bronze ${ }^{c}$ | - | 85.0 | 5.0 | 5.0 | - | - | - | - | 5.0 | - | - | INCO (3) |
| Phosphor bronze A | 510 | 94.64 | 4.94 | $<0.10$ | - | - | $<0.05$ | -- | - | 0.26 | - | CEL (4) |
| Phosphor bronze A | 510 | 96.0 | 4.0 | - | - | - | - | - | - | 0.25 | - | INCO (3) |
| Phosphor bronze A | 510 | 95.62 | 4.44 | $<0.10$ | - | - | $<0.05$ | - | - | 0.06 | - | CEL (4) |
| Phosphor bronze D | 524 | 90.00 | 9.23 | $<0.10$ | - | - | $<0.05$ | - | - | 0.17 | - | CEL (4) |
| Al bronze, 5\% | 606 | 95.0 | - | - | - | 5.0 | - | - | - | - | - | INCO (3) |
| Al bronze, 5\% | 606 | 95.11 | - | - | - | 4.76 | $<0.05$ | - | - | - | - | CEL (4) |
| Al bronze, 5\% | 606 | - | - | - | - | - | - | - | - | - | - | Boeing (6) |
| Al bronze, 7\% | 614 | 90.11 | - | 0.15 | - | 6.59 | 3.15 | - | $<0.02$ | - | - | CEL (4) |
| Al bronze, 7\% | 614 | 90.0 | - | - | - | 7.0 | 3.0 | - | - | - | - | INCO (3) |
| Al bronze, 7\% | 614 | 88.0 | - | - | - | 9.0 | 3.0 | - | - | - | - | NADC (7) |
| Al bronze, $10 \%{ }^{c}$ | 953 | 89.0 | - | - | - | 10.0 | 1.0 | - | - | - | - | INCO (3) |
| Al bronze, $11 \%{ }^{\text {c }}$ | 954 | 86.0 | - | - | - | 10.0 | 4.0 | -- | - | - | - | INCO (3) |
| Al bronze, $11 \%^{\text {c }}$ | 954 | 86.5 | - | - | - | 10.0 | 3.5 | - | - | - | - | MEL (5) |
| Al bronze, $13 \%{ }^{\text {c }}$ | - | 83.0 | - | - | - | 13.0 | 4.0 | - | - | - | - | INCO (3) |
| Ni -Al bronze | - | 83.5 | - | - | 2.5 | 10.0 | 4.0 | - | - | - | - | MEL (5) |
| Ni-Al bronze No. 1 | - | 80.0 | - | - | 4.0 | 11.0 | 4.0 | - | - | - | 1.0 | INCO (3) |
| Ni-Al bronze No. 1 | - | 80.0 | - | - | 4.0 | 10.0 | 4.0 | - | - | - | - | MEL (5) |
| Ni-Al bronze No. 2 | - | 80.0 | - | - | 5.0 | 10.0 | 4.0 | - | - | - | 0.5 | INCO (3) |
| Ni-Al bronze No. 3 | - | 80.0 | - | - | 5.0 | 9.0 | 3.5 | - | - | - | 3.0 | INCO (3) |
| Si bronze, 3\% | 653 | 97.0 | - | - | - | - | - | 3.0 | - | - | - | INCO (3) |
| Si bronze A | 655 | 95.49 | - | - | - | - | $<0.02$ | 3.28 | - | - | 1.18 | CEL (4) |
| Si bronze A | 655 | 95.0 | - | - | - | - | - | 3.0 | - | - | 1.0 | INCO (3) |
| Si bronze A | 655 | 96.0 | - | - | - | - | - | 3.0 | - | - | 1.0 | Boeing (6) |
| Ni-Vee bronze ${ }^{\text {c }}$ | - | 88.0 | 5.0 | 2.0 | 5.0 | - | - | - | - | - | - | INCO (3) |
| Ni-Vee bronze B ${ }^{c}$ | - | 87.0 | 5.0 | 2.0 | 5.0 | - | - | 1.0 | - | - | - | INCO (3) |
| Ni-Vee bronze $\mathrm{C}^{\text {c }}$ | - | 80.0 | 5.0 | 5.0 | 5.0 | - | - | 5.0 | - | - | - | INCO (3) |

${ }^{a}$ Copper Development Association alloy number.
${ }^{b}$ Numbers refer to references at end of report.
${ }^{c}$ Cast alloy.

Table 19. Corrosion Rates and Types of Corrosion of Copper Alloys (Bronzes)

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }^{a} \end{aligned}$ | Environment ${ }^{\text {b }}$ | Exposure (day) | Depth <br> (ft) | $\begin{array}{\|c} \hline \text { Corrosion } \\ \text { Rate } \\ \text { (mpy) } \\ \hline \end{array}$ | Type of Corrosion ${ }^{c}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| G Bronze ${ }^{e}$ | - | W | 123 | 5,640 | 0.5 | U | INCO (3) |
| G Bronze | - | S | 123 | 5,640 | 0.3 | U | INCO (3) |
| G Bronze | - | W | 403 | 6,780 | 0.7 | U | INCO (3) |
| G Bronze | - | S | 403 | 6,780 | 0.1 | GBSL | INCO (3) |
| G Bronze | - | W | 751 | 5,640 | 0.7 | U | INCO (3) |
| G Bronze | - | S | 751 | 5,640 | 0.3 | U | INCO (3) |
| G Bronze | - | W | 1,064 | 5,300 | 0.3 | U | INCO (3) |
| G Bronze | - | S | 1,064 | 5,300 | 0.4 | U | INCO (3) |
| G Bronze | - | W | 197 | 2,340 | 0.2 | U | ${ }^{\mathrm{N} C O}$ (3) |
| G Bronze | - | S | 197 | 2,340 | $<0.1$ | I-C | INCO (3) |
| G Bronze | - | w | 402 | 2,370 | 0.3 | U | INCO (3) |
| G Bronze | - | S | 402 | 2,370 | $<0.1$ | ET | INCO (3) |
| G Bronze | - | W | 181 | 5 | 1.3 | G | INCO (3) |
| G Bronze | - | w | 366 | 5 | 1.2 | CR (9) | INCO (3) |
| Modified G Bronze ${ }^{e}$ | - | W | 123 | 5,640 | 0.5 | U | INCO (3) |
| Modified G Bronze | - | S | 123 | 5,640 | 0.3 | U | INCO (3) |
| Modified G Bronze | - | w | 403 | 6,780 | 0.4 | U | INCO (3) |
| Modified G Bronze | - | S | 403 | 6,780 | <0.1 | U | INCO (3) |
| Modified G Bronze | - | W | 751 | 5,640 | 0.7 | U | INCO (3) |
| Modified G Bronze | - | S | 751 | 5,640 | 0.4 | C (19) | INCO (3) |
| Modified G Bronze | - | W | 1,064 | 5,300 | 0.4 | C (18); P | INCO (3) |
| Modified G Bronze | - | S | 1,064 | 5,300 | 0.5 | U | INCO (3) |
| Modified G Bronze | - | W | 197 | 2,340 | 0.3 | U | INCO (3) |
| Modified G Bronze | - | S | 197 | 2,340 | 0.2 | NU-ET | INCO (3) |
| Modified G Bronze | - | W | 402 | 2,370 | 0.3 | U | INCO (3) |
| Modified G Bronze | - | S | 402 | 2,370 | <0.1 | ET | INCO (3) |
| Modified G Bronze | - | W | 181 | 5 | 1.3 | G | INCO (3) |
| Modified G Bronze | - | w | 366 | 5 | 1.0 | CR (7) | INCO (3) |
| M Bronze ${ }^{e}$ | - | W | 123 | 5,640 | 0.5 | U | INCO (3) |
| M Bronze | - | S | 123 | 5,640 | 0.4 | U | INCO (3) |
| M Bronze | - | W | 403 | 6,780 | 0.4 | U | INCO (3) |
| M Bronze | - | S | 403 | 6,780 | <0.1 | U | INCO (3) |
| M Bronze | - | W | 751 | 5,640 | 0.7 | U | INCO (3) |
| M Bronze | - | S | 751 | 5,640 | 0.3 | U | INCO (3) |
| M Bronze | - | W | 1,064 | 5,300 | 0.4 | U | INCO (3) |
| M Bronze | - | S | 1,064 | 5,300 | 0.4 | U | INCO (3) |
| M Bronze | - | W | 197 | 2,340 | 0.4 | U | INCO (3) |
| M Bronze | - | S | 197 | 2,340 | 0.1 | ET | INCO (3) |
| M Bronze | - | W | 402 | 2,370 | 0.3 | U | INCO (3) |
| M Bronze | - | S | 402 | 2,370 | $<0.1$ | ET | INCO (3) |
| M Bronze | - | W | 181 | 5 | 1.6 | G | INCO (3) |
| M Bronze | - | W | 366 | 5 | 1.1 | CR (2) | INCO (3) |
| Leaded Sn Bronze ${ }^{e}$ | - | W | 123 | 5,640 | 0.4 | U | INCO (3) |
| Leaded Sn Bronze | - | S | 123 | 5,640 | 0.2 | U | INCO (3) |
| Leaded Sn Bronze | - | W | 403 | 6,780 | 0.5 | U | INCO (3) |
| Leaded Sn Bronze | - | S | 403 | 6,780 | 0.1 | U | INCO (3) |
| Leaded Sn Bronze | - | w | 751 | 5,640 | 0.6 | U | INCO (3) |
| Leaded Sn Bronze | - | S | 751 | 5,640 | 3.2 | S-G | INCO (3) |
| Leaded Sn Bronze | - | W | 1,064 | 5,300 | 0.4 | U | INCO (3) |

Table 19. Continued.

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Environment ${ }^{b}$ | Exposure (day) | Depth (ft) | Corrosion Rate (mpy) | Type of Corrosion ${ }^{c}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Leaded Sn Bronze | - | S | 1,064 | 5,300 | 0.3 | U | INCO (3) |
| Leaded Sn Bronze | - | W | 197 | 2,340 | 0.5 | U | INCO (3) |
| Leaded Sn Bronze | - | S | 197 | 2,340 | $<0.1$ | NU-ET | INCO (3) |
| Leaded Sn Bronze | - | W | 402 | 2,370 | 0.5 | U | INCO (3) |
| Leaded Sn Bronze | - | S | 402 | 2,370 | $<0.1$ | ET | INCO (3) |
| Leaded Sn Bronze | - | W | 181 | 5 | 1.4 | G | INCO (3) |
| Leaded Sn Bronze | - | W | 366 | 5 | 1.3 | CR (5) | INCO (3) |
| P Bronze A | 510 | W | 123 | 5,640 | 0.6 | U | CEL (4) |
| P Bronze A | 510 | W | 123 | 5,640 | 0.5 | U | INCO (3) |
| P Bronze A | 510 | S | 123 | 5,640 | 0.4 | U | CEL (4) |
| P Bronze A | 510 | S | 123 | 5,640 | 0.4 | U | INCO (3) |
| P Bronze A | 510 | W | 403 | 6,780 | 0.2 | ET | CEL (4) |
| P Bronze A | 510 | W | 403 | 6,780 | 0.3 | U | INCO (3) |
| P Bronze A | 510 | S | 403 | 6,780 | 0.3 | ET | CEL (4) |
| P Bronze A | 510 | S | 403 | 6,780 | 0.1 | GBSL | INCO (3) |
| P Bronze A | 510 | W | 751 | 5,640 | 0.2 | ET | CEL (4) |
| P Bronze A | 510 | W | 751 | 5,640 | 0.3 | U | INCO (3) |
| P Bronze A | 510 | S | 751 | 5,640 | 0.1 | U | INCO (3) |
| P Bronze A | 510 | W | 1,064 | 5,300 | 0.4 | U | CEL (4) |
| P Bronze A | 510 | W | 1,064 | 5,300 | 0.2 | U | INCO (3) |
| P Bronze A | 510 | S | 1,064 | 5,300 | 0.4 | G | INCO (3) |
| P Bronze A | 510 | W | 197 | 2,340 | 0.3 | U | CEL (4) |
| P Bronze A | 510 | W | 197 | 2,340 | 0.4 | U | INCO (3) |
| P Bronze A | 510 | S | 197 | 2,340 | 0.3 | $f$ | CEL (4) |
| P Bronze A | 510 | S | 197 | 2,340 | <0.1 | I-C | INCO (3) |
| P Bronze A | 510 | W | 402 | 2,370 | 0.1 | ET | CEL (4) |
| P Bronze A | 510 | W | 402 | 2,370 | 0.2 | U | INCO (3) |
| P Bronze A | 510 | S | 402 | 2,370 | 0.2 | EWO | CEL (4) |
| P Bronze A | 510 | S | 402 | 2,370 | 0.2 | ET | INCO (3) |
| P Bronze A | 510 | W | 181 | 5 | 1.1 | U | CEL (4) |
| P Bronze A | 510 | W | 181 | 5 | 1.6 | P (4) | INCO (3) |
| P Bronze A | 510 | W | 366 | 5 | 1.3 | CR (5) | INCO (3) |
| P Bronze A | 510 | W | 588 | 5 | 1.3 | CR (15); C (3) | CEL (4) |
| P Bronze A | 510 | W | 608 | 5 | 1.1 | CR (15) | CEL (4) |
| P Bronze D | 524 | W | 123 | 5,640 | 0.5 | U | CEL (4) |
| P Bronze D | 524 | S | 123 | 5,640 | 0.4 | U | CEL (4) |
| P Bronze D | 524 | W | 403 | 6,780 | 0.2 | ET | CEL (4) |
| P Bronze D | 524 | S | 403 | 6,780 | 0.3 | ET | CEL (4) |
| P Bronze D | 524 | W | 751 | 5,640 | 0.3 | U | CEL (4) |
| P Bronze D | 524 | S | 751 | 5,640 | 0.4 | NU | CEL (4) |
| P Bronze D | 524 | W | 197 | 2,340 | 0.4 | U | CEL (4) |
| P Bronze D | 524 | S | 197 | 2,340 | 0.2 | U | CEL (4) |
| P Bronze D | 524 | W | 402 | 2,370 | $<0.1$ | U | CEL (4) |
| P Bronze D | 524 | S | 402 | 2,370 | $<0.1$ | U | CEL (4) |
| P Bronze D | 524 | W | 181 | 5 | 1.1 | NU | CEL (4) |
| P Bronze D | 524 | W | 398 | 5 | 0.9 | CR (4) | CEL (4) |
| P Bronze D | 524 | W | 540 | 5 | 0.7 | CR (2) | CEL (4) |
| P Bronze D | 524 | W | 608 | 5 | 0.7 | CR (7); C (5) | CEL (4) |
| Al Bronze, 5\% | 606 | W | 123 | 5,640 | 0.6 | U | INCO (3) |
| Al Bronze, 5\% | 606 | S | 123 | 5,640 | 0.4 | U | INCO (3) |

Continued

Table 19. Continued.

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{\text {a }} \end{aligned}$ | Environment ${ }^{\text {b }}$ | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) | Type of Corrosion ${ }^{c}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Al Bronze, 5\% | 606 | W | 403 | 6,780 | 0.2 | SL-DA | INCO (3) |
| Al Bronze, 5\% | 606 | S | 403 | 6,780 | $<0.1$ | U | INCO (3) |
| Al Bronze, 5\% | 606 | w | 751 | 5,640 | 0.3 | SL-DA | INCO (3) |
| Al Bronze, 5\% | 606 | S | 751 | 5,640 | 0.2 | V-SL-DA | INCO (3) |
| Al Bronze, 5\% | 606 | W | 1,064 | 5,300 | 0.2 | NU | CEL (4) |
| Al Bronze, 5\% | 606 | W | 1,064 | 5,300 | 0.2 | CR (5) | INCO (3) |
| Al Bronze, 5\% | 606 | S | 1,064 | 5,300 | 0.5 | G | INCO (3) |
| Al Bronze, 5\% | 606 | W | 197 | 2,340 | 0.4 | U | INCO (3) |
| Al Bronze, 5\% | 606 | S | 197 | 2,340 | 0.2 | NU-ET | INCO (3) |
| Al Bronze, 5\% | 606 | W | 197 | 2,340 | 0.2 | U | Boeing (6) |
| Al Bronze, 5\% | 606 | W | 402 | 2,370 | 0.2 | U | INCO (3) |
| Al Bronze, 5\% | 606 | S | 402 | 2,370 | 0.1 | ET | INCO (3) |
| Al Bronze, 5\% | 606 | w | 181 | 5 | 1.1 | G | INCO (3) |
| Al Bronze, 5\% | 606 | w | 366 | 5 | 0.7 | G | INCO (3) |
| Al Bronze, 7\% | 614 |  | 123 | 5,640 | 0.5 | SL-DA | CEL (4) |
| Al Bronze, 7\% | 614 | w | 123 | 5,640 | 0.6 | U | INCO (3) |
| Al Bronze, 7\% | 614 | S | 123 | 5,640 | 0.3 | U | CEL (4) |
| Al Bronze, 7\% | 614 | S | 123 | 5,640 | 0.4 | U | INCO (3) |
| Al Bronze, 7\% | 614 | w | 403 | 6,780 | 0.7 | SL-DA; C (12); P (12) | CEL (4) |
| Al Bronze, 7\% | 614 | w | 403 | 6,780 | 0.2 | U | INCO (3) |
| Al Bronze, 7\% | 614 | S | 403 | 6,780 | 0.7 | SL-DA; C (13); P (16) | CEL (4) |
| Al Bronze, 7\% | 614 | S | 403 | 6,780 | $<0.1$ | SL-DA | INCO (3) |
| Al Bronze, 7\% | 614 | W | 403 | 6,780 | NC | NC | NADC (7) |
| Al Bronze, 7\% | 614 | W | 751 | 5,640 | 0.5 | MD-DA; C (7); P (12) | CEL (4) |
| Al Bronze, 7\% | 614 | w | 751 | 5,640 | 1.5 | G | INCO (3) |
| Al Bronze, 7\% | 614 | S | 751 | 5,640 | 0.2 | V-SL-DA | INCO (3) |
| Al Bronze, 7\% | 614 | W | 1,064 | 5,300 | 0.2 | CR (7) | INCO (3) |
| Al Bronze, 7\% | 614 | S | 1,064 | 5,300 | 0.2 | MO-DA | INCO (3) |
| Al Bronze, 7\% | 614 | W | 197 | 2,340 | 0.3 | U | CEL (4) |
| Al Bronze, 7\% | 614 | w | 197 | 2,340 | 0.3 | U | INCO (3) |
| Al Bronze, 7\% | 614 | S | 197 | 2,340 | 0.1 | U | CEL (4) |
| Al Bronze, 7\% | 614 | S | 197 | 2,340 | 0.2 | ET | INCO (3) |
| Al Bronze, 7\% | 614 | W | 402 | 2,370 | 0.2 | U | CEL (4) |
| Al Bronze, 7\% | 614 | W | 402 | 2,370 | 0.2 | ET | INCO (3) |
| Al Bronze, 7\% | 614 | S | 402 | 2,370 | 0.2 | U | INCO (3) |
| Al Bronze, 7\% | 614 | S | 402 | 2,370 | 0.1 | SL-DA | INCO (3) |
| Al Bronze, 7\% | 614 | w | 181 | 5 | 2.9 | NU-DA | CEL (4) |
| Al Bronze, 7\% | 614 | w | 181 | 5 | 0.8 | G | INCO (3) |
| Al Bronze, 7\% | 614 | w | 366 | 5 | 0.6 | G | INCO (3) |
| Al Bronze, 7\% | 614 | w | 588 | 5 | 0.9 | SL-DA; CR (44); C (20) | CEL (4) |
| Al Bronze, $10 \%{ }^{e}$ | - | W | 123 | 5,640 | 0.7 | SL-DA | INCO (3) |
| Al Bronze, 10\% | - | S | 123 | 5,640 | 0.6 | SL-DA | INCO (3) |
| Al Bronze, 10\% | - | w | 403 | 6,780 | 0.7 | MO-DA | INCO (3) |
| Al Bronze, 10\% | - | S | 403 | 6,780 | $<0.1$ | SL-DA | INCO (3) |
| Al Bronze, 10\% | - | w | 751 | 5,640 | 2.3 | G | INCO (3) |
| Al Bronze, 10\% | - | S | 751 | 5,640 | 0.9 | U; SL-DA | INCO (3) |
| Al Bronze, 10\% | - | W | 1,064 | 5,300 | 0.2 | U | INCO (3) |
| Al Bronze, 10\% | - | S | 1,064 | 5,300 | 0.4 | SL-DA | INCO (3) |
| Al Bronze, 10\% | - | w | 197 | 2,340 | 0.3 | MO-DA | INCO (3) |
| Al Bronze, 10\% | - | S | 197 | 2,340 | 0.2 | MO-DA | INCO (3) |

Table 19. Continued.

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Environment ${ }^{\text {b }}$ | Exposure (day) | Depth (ft) | Corrosion Rate (mpy) | Type of Corrosion ${ }^{c}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Al Bronze, 10\% | - | W | 402 | 2,370 | 0.3 | S-DA | INCO (3) |
| Al Bronze, 10\% | - | S | 402 | 2,370 | $<0.1$ | MO-DA | INCO (3) |
| Al Bronze, 10\% | - | W | 181 | 5 | 2.1 | MO-DA | INCO (3) |
| Al Bronze, 10\% | - | W | 366 | 5 | 1.3 | MO-DA | INCO (3) |
| Al Bronze, $11 \%^{e}$ | - | W | 123 | 5,640 | 0.4 | U | MEL (5) |
| Al Bronze, 11\% | - | W | 123 | 5,640 | 0.5 | V-SL-DA | INCO (3) |
| Al Bronze, 11\% | - | S | 123 | 5,640 | 0.4 | V-SL-DA | INCO (3) |
| Al Bronze, 11\% | - | W | 403 | 6,780 | 0.1 | SL-DA | INCO (3) |
| Al Bronze, 11\% | - | S | 403 | 6,780 | $<0.1$ | SL-DA | INCO (3) |
| Al Bronze, 11\% | - | W | 751 | 5,640 | 0.8 | SL-DA | INCO (3) |
| Al Bronze, 11\% | - | W | 751 | 5,640 | 0.3 | U-P | MEL (5) |
| Al Bronze, 11\% | - | S | 751 | 5,640 | 0.1 | V-SL-DA | INCO (3) |
| Al Bronze, 11\% | - | W | 1,064 | 5,300 | 0.1 | SL-DA | INCO (3) |
| Al Bronze, 11\% | - | S | 1,064 | 5,300 | 0.2 | MO-DA | INCO (3) |
| Al Bronze, 11\% | - | W | 197 | 2,340 | 0.2 | SL-DA ${ }^{g}$ | INCO (3) |
| Al Bronze, 11\% | - | S | 197 | 2,340 | $<0.1$ | SL-DA ${ }^{\circ}$ | INCO (3) |
| Al Bronze, 11\% | - | W | 402 | 2,370 | 0.2 | MO-DA | INCO (3) |
| Al Bronze, 11\% | - | S | 402 | 2,370 | $<0.1$ | SL-DA | INCO (3) |
| Al Bronze, 11\% | - | W | 366 | 5 | 1.1 | U | INCO (3) |
| Al Bronze, 13\% ${ }^{e}$ | - | W | 123 | 5,640 | 0.5 | V-SL-DA | INCO (3) |
| Al Bronze, 13\% | - | S | 123 | 5,640 | 0.5 | V-SL-DA | INCO (3) |
| Al Bronze, 13\% | - | W | 403 | 6,780 | 0.6 | S-DA | INCO (3) |
| Al Bronze, 13\% | - | S | 403 | 6,780 | $<0.1$ | SL-DA | INCO (3) |
| Al Bronze, 13\% | - | W | 751 | 5,640 | 1.9 | S-DA | INCO (3) |
| Al Bronze, 13\% | - | S | 751 | 5,640 | 0.5 | MO-DA | INCO (3) |
| Al Bronze, 13\% | - | W | 1,064 | 5,300 | 0.6 | S-DA | INCO (3) |
| Al Bronze, 13\% | - | S | 1,064 | 5,300 | 0.3 | SL-DA | INCO (3) |
| Al Bronze, 13\% | - | W | 197 | 2,340 | 0.4 | MO-DA | INCO (3) |
| Al Bronze, 13\% | - | S | 197 | 2,340 | $<0.1$ | SL-DA | INCO (3) |
| Al Bronze, $13 \%$ | - | W | 402 | 2,370 | 0.3 | MO-DA | INCO (3) |
| Al Bronze, 13\% | - | S | 402 | 2,370 | $<0.1$ | SL-DA | INCO (3) |
| Al Bronze, 13\% | - | W | 181 | 5 | 2.1 | S-DA | INCO (3) |
| Al Bronze, 13\% | - | W | 366 | 5 | 1.9 | S-DA | INCO (3) |
| Ni-Al Bronze | - | W | 123 | 5,640 | 0.6 | P | MEL (5) |
| Ni-Al Bronze | - | W | 751 | 5,640 | 0.4 | P (21) | MEL (5) |
| Ni-Al Bronze No. 1 | - | W | 123 | 5,640 | 0.5 | U | MEL (5) |
| Ni-Al Bronze No. 1 | - | W | 123 | 5,640 | 0.4 | I-P | INCO (3) |
| Ni-Al Bronze No. 1 | - | S | 123 | 5,640 | 0.2 | I-P | INCO (3) |
| Ni-Al Bronze No. 1 | $\triangle$ | W | 403 | 6,780 | 0.3 | I-P | INCO (3) |
| Ni-Al Bronze No. 1 | - | S | 403 | 6,780 | 0.1 | U | INCO (3) |
| Ni-Al Bronze No. 1 | - | W | 751 | 5,640 | 1.1 | C (16); P | INCO (3) |
| Ni-Al Bronze No. 1 | - | W | 751 | 5,640 | 0.2 | P | MEL (5) |
| Ni-Al Bronze No. 1 | - | S | 751 | 5,640 | 0.3 | P (17) | INCO (3) |
| Ni-Al Bronze No. 1 | - | W | 1,064 | 5,300 | 0.1 | P (8) | INCO (3) |
| Ni-Al Bronze No. 1 | - | S | 1,064 | 5,300 | 1.2 | CR (30) | INCO (3) |
| Ni-Al Bronze No. 1 | - | W | 197 | 2,340 | 0.3 | U | INCO (3) |
| Ni-Al Bronze No. 1 | - | S | 197 | 2,340 | $<0.1$ | U-ET | INCO (3) |

Continued

Table 19. Continued.

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. } \end{aligned}$ | Environment ${ }^{b}$ | Exposure (day) | Depth (ft) | Corrosion Rate (mpy) | Type of Corrosion ${ }^{c}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Ni-Al Bronze No. 2 | - | W | 123 | 5,640 | 0.5 | U | INCO (3) |
| Ni-Al Bronze No. 2 | - | S | 123 | 5,640 | 0.3 | U | INCO (3) |
| Ni-Al Bronze No. 2 | - | W | 403 | 6,780 | 0.2 | I-P | INCO (3) |
| Ni-Al Bronze No. 2 | - | S | 403 | 6,780 | 0.1 | U | INCO (3) |
| Ni-Al Bronze No. 2 | - | W | 751 | 5,640 | 0.5 | SL-DA | INCO (3) |
| Ni-Al Bronze No. 2 | - | S | 751 | 5,640 | 0.2 | C (13) | INCO (3) |
| Ni-Al Bronze No. 2 | - | W | 1,064 | 5,300 | 0.2 | C (5) | INCO (3) |
| Ni-Al Bronze No. 2 | - | S | 1,064 | 5,300 | 0.5 | CR (21) | INCO (3) |
| Ni-Al Bronze No. 2 | - | W | 197 | 2,340 | 0.3 | U | INCO (3) |
| Ni-Al Bronze No. 2 | - | S | 197 | 2,340 | 0.1 | U | INCO (3) |
| Ni-Al Bronze No. 2 | - | W | 402 | 2,370 | 0.2 | U | INCO (3) |
| Ni-Al Bronze No. 2 | - | S | 402 | 2,370 | <0.1 | ET | INCO (3) |
| Ni-Al Bronze No. 2 | - | W | 181 | 5 | 1.0 | C (8) | INCO (3) |
| Ni-Al Bronze No. 2 | - | W | 366 | 5 | 0.4 | U | INCO (3) |
| Ni-Al Bronze No. 3 | - | W | 123 | 5,640 | 0.4 | U | INCO (3) |
| Ni-Al Bronze No. 3 | - | S | 123 | 5,640 | 0.3 | U | INCO (3) |
| Ni-Al Bronze No. 3 | - | W | 403 | 6,780 | 0.2 | U | INCO (3) |
| Ni-Al Bronze No. 3 | - | S | 403 | 6,780 | $<0.1$ | NU-ET | INCO (3) |
| Ni-Al Bronze No. 3 | - | W | 751 | 5,640 | 0.2 | I-P | INCO (3) |
| Ni-Al Bronze No. 3 | - | S | 751 | 5,640 | 0.1 | U | INCO (3) |
| Ni-Al Bronze No. 3 | - | W | 1,064 | 5,300 | $<0.1$ | P (4) | INCO (3) |
| Ni-Al Bronze No. 3 | - | S | 1,064 | 5,300 | 0.2 | CR (10) | INCO (3) |
| Ni-Al Bronze No. 3 | - | W | 197 | 2,340 | 0.3 | U | INCO (3) |
| Ni-Al Bronze No. 3 | - | S | 197 | 2,340 | 0.2 | U | INCO (3) |
| Si Bronze, 3\% | 653 | W | 123 | 5,640 | 1.3 | U | INCO (3) |
| Si Bronze, 3\% | 653 | S | 123 | 5,640 | 1.5 | U | INCO (3) |
| Si Bronze, 3\% | 653 | W | 403 | 6,780 | 1.2 | $\mathrm{MO}-\mathrm{CO}$ | INCO (3) |
| Si Bronze, 3\% | 653 | S | 403 | 6,780 | 0.4 | GBSL | INCO (3) |
| Si Bronze, 3\% | 653 | W | 751 | 5,640 | 1.0 | U | INCO (3) |
| Si Bronze, 3\% | 653 | S | 751 | 5,640 | 0.7 | U | INCO (3) |
| Si Bronze, 3\% | 653 | W | 1,064 | 5,300 | 0.6 | $\mathrm{MO}-\mathrm{CO}$ | INCO (3) |
| Si Bronze, 3\% | 653 | S | 1,064 | 5,300 | 0.4 | SL-CO | INCO (3) |
| Si Bronze, 3\% | 653 | W | 197 | 2,340 | 1.1 | U | INCO (3) |
| Si Bronze, 3\% | 653 | S | 197 | 2,340 | 0.1 | NU-ET | INCO (3) |
| Si Bronze, 3\% | 653 | W | 402 | 2,370 | 1.2 | U | INCO (3) |
| Si Bronze, 3\% | 653 | S | 402 | 2,370 | 0.2 | ET | INCO (3) |
| Si Bronze, 3\% | 653 | W | 181 | 5 | 1.7 | U | INCO (3) |
| Si Bronze, 3\% | 653 | W | 366 | 5 | 1.1 | G | INCO (3) |
| Si Bronze A | 655 | W | 123 | 5,640 | 1.6 | U | CEL (4) |
| Si Bronze A | 655 | W | 123 | 5,640 | 1.4 | U | INCO (3) |
| Si Bronze A | 655 | S | 123 | 5,640 | 1.8 | U | CEL (4) |
| Si Bronze A | 655 | S | 123 | 5,640 | 1.5 | U | INCO (3) |
| Si Bronze A | 655 | W | 403 | 6,780 | 1.2 | CO | CEL (4) |
| Si Bronze A | 655 | W | 403 | 6,780 | 1.2 | U | INCO (3) |
| Si Bronze A | 655 | S | 403 | 6,780 | 1.8 | ET | CEL (4) |
| Si Bronze A | 655 | S | 403 | 6,780 | 0.2 | GBSL | INCO (3) |
| Si Bronze A | 655 | W | 751 | 5,640 | 0.8 | $\mathrm{S}-\mathrm{CO}$ | CEL (4) |
| Si Bronze A | $655$ | W | 751 | 5,640 | $1.4$ | $\mathbf{U}$ | INCO (3) |
| Si Bronze A | 655 | S | 751 | 5,640 | 0.9 | G | INCO (3) |

Continued

Table 19. Continued.

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }^{a} \end{aligned}$ | Environment ${ }^{b}$ | Exposure (day) | Depth (ft) | Corrosion Rate (mpy) | Type of Corrosion ${ }^{c}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Si Bronze A | 655 | W | 1,064 | 5,300 | 0.6 | SL-CO | INCO (3) |
| Si Bronze A | 655 | S | 1,064 | 5,300 | 0.4 | U | INCO (3) |
| Si Bronze A | 655 | W | 197 | 2,340 | 0.8 | U | Boeing (6) |
| Si Bronze A | 655 | W | 197 | 2,340 | 0.9 | U | CEL (4) |
| Si Bronze A | 655 | W | 197 | 2,340 | 1.1 | U | INCO (3) |
| Si Bronze A | 655 | S | 197 | 2,340 | 0.6 | U | CEL (4) |
| Si Bronze A | 655 | S | 197 | 2,340 | 0.2 | U | INCO (3) |
| Si Bronze A | 655 | W | 402 | 2,370 | 1.0 | ET | CEL (4) |
| Si Bronze A | 655 | W | 402 | 2,370 | 0.8 | U | INCO (3) |
| Si Bronze A | 655 | S | 402 | 2,370 | 0.8 | ET | CEL (4) |
| Si Bronze A | 655 | S | 402 | 2,370 | 0.1 | ET | INCO (3) |
| Si Bronze A | 655 | W | 181 | 5 | 1.8 | U | CEL (4) |
| Si Bronze A | 655 | W | 181 | 5 | 1.6 | G | INCO (3) |
| Si Bronze A | 655 | W | 366 | 5 | 1.2 | G | INCO (3) |
| Si Bronze A | 655 | W | 398 | 5 | 1.1 | G | CEL (4) |
| Si Bronze A | 655 | W | 540 | 5 | 2.5 | CR (30); C (15) | CEL (4) |
| Si Bronze A | 655 | W | 588 | 5 | 0.9 | CR (9) | CEL (4) |
| Ni-Vee Bronze $\mathrm{A}^{e}$ | - | W | 123 | 5,640 | 0.7 | U | INCO (3) |
| Ni-Vee Bronze A | - | S | 123 | 5,640 | 0.5 | U | INCO (3) |
| Ni-Vee Bronze A | - | W | 403 | 6,780 | 0.6 | U | INCO (3) |
| Ni-Vee Bronze A | - | S | 403 | 6,780 | 0.3 | U | INCO (3) |
| Ni-Vee Bronze A | - | W | 751 | 5,640 | 2.6 | $\mathrm{S}^{2}$ | INCO (3) |
| Ni-Vee Bronze A | - | S | 751 | 5,640 | 0.4 | U | INCO (3) |
| Ni-Vee Bronze A | - | W | 1,064 | 5,300 | 2.2 | CR (20) | INCO (3) |
| Ni-Vee Bronze A | - | S | 1,064 | 5,300 | 0.3 | U | INCO (3) |
| Ni-Vee Bronze A | - | W | 197 | 2,340 | 0.6 | U | INCO (3) |
| Ni-Vee Bronze A | - | S | 197 | 2,340 | $<0.1$ | NU-ET | INCO (3) |
| Ni-Vee Bronze A | - | W | 402 | 2,370 | 0.4 | U | INCO (3) |
| Ni-Vee Bronze A | - | S | 402 | 2,370 | $<0.1$ | ET | INCO (3) |
| Ni-Vee Bronze A | - | W | 181 | 5 | 2.0 | P (7) | INCO (3) |
| Ni-Vee Bronze A | - | W | 366 | 5 | 1.5 | CR (10) | INCO (3) |
| Ni -Vee Bronze $\mathrm{B}^{e}$ | - | W | 123 | 5,640 | 0.6 | U | INCO (3) |
| Ni-Vee Bronze B | - | S | 123 | 5,640 | 0.4 | U | INCO (3) |
| Ni-Vee Bronze B | - | W | 403 | 6,780 | 0.5 | U | INCO (3) |
| Ni-Vee Bronze B | - | S | 403 | 6,780 | 0.1 | U | INCO (3) |
| Ni-Vee Bronze B | - | W | 751 | 5,640 | 0.5 | U | INCO (3) |
| Ni-Vee Bronze B | - | S | 751 | 5,640 | 0.3 | U | INCO (3) |
| Ni -Vee Bronze B | - | W | 1,064 | 5,300 | 0.3 | U | INCO (3) |
| Ni-Vee Bronze B | - | S | 1,064 | 5,300 | 0.4 | U | INCO (3) |
| Ni-Vee Bronze B | - | W | 197 | 2,340 | 0.6 | U | INCO (3) |
| Ni-Vee Bronze B | - | S | 197 | 2,340 | $<0.1$ | NU-ET | INCO (3) |
| Ni-Vee Bronze B | - | W | 402 | 2,370 | 1.2 | U | INCO (3) |
| Ni-Vee Bronze B | - | S | 402 | 2,370 | $<0.1$ | ET | INCO (3) |
| Ni-Vee Bronze B | - | W | 181 | 5 | 1.8 | P (4) | INCO (3) |
| Ni-Vee Bronze B | - | W | 366 | 5 | 1.3 | CR (6) | INCO (3) |
| Ni-Vee Bronze C ${ }^{e}$ | - | W | 123 | 5,640 | 0.8 | U | INCO (3) |
| Ni-Vee Bronze C | - | S | 123 | 5,640 | 0.5 | U | INCO (3) |
| Ni-Vee Bronze C | - | W | 403 | 6,780 | 0.8 | U | INCO (3) |
| Ni-Vee Bronze C | - | S | 403 | 6,780 | 0.2 | U | INCO (3) |
| Ni-Vee Bronze C | - | W | 751 | 5,640 | 2.0 | G | INCO (3) |

Table 19. Continued.

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Environment ${ }^{b}$ | Exposure (day) | Depth <br> (ft) | $\begin{aligned} & \text { Corrosion } \\ & \text { Rate } \\ & \text { (mpy) } \\ & \hline \end{aligned}$ | Type of Corrosion ${ }^{c}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Ni-Vee Bronze C | - | S | 751 | 5,640 | 0.4 | U | INCO (3) |
| Ni-Vee Bronze C | - | W | 1,064 | 5,300 | 0.5 | U | INCO (3) |
| Ni-Vee Bronze C | - | S | 1,064 | 5,300 | 0.3 | U | INCO (3) |
| Ni-Vee Bronze C | - | W | 197 | 2,340 | 0.8 | U | INCO (3) |
| Ni-Vee Bronze C | - | S | 197 | 2,340 | $<0.1$ | ET | INCO (3) |
| Ni-Vee Bronze C | - | W | 402 | 2,370 | 0.6 | U | INCO (3) |
| Ni-Vee Bronze C | - | S | 402 | 2,370 | 0.1 | ET | INCO (3) |
| Ni-Vee Bronze C | - | w | 181 | 5 | 1.8 | U | INCO (3) |
| Ni-Vee Bronze C | - | W | 366 | 5 | 1.5 | CR (5) | INCO (3) |

${ }^{a}$ Copper Development Association alloy number.
$b_{\mathrm{W}}=$ Totally exposed in seawater on sides of structure ${ }^{\mathrm{S}} \mathrm{S}=$ Exposed in base of structure so that a portion of each specimen was exposed in the bottom sediments.
${ }^{c}$ Symbols for types of corrosion:

| C | $=$ Crevice |
| :--- | :--- |
| CO | $=$ Coppering, a selective attack where copper |
|  | appears on surface similar to dezincification |
| CR | $=$ Cratering |
| DA | $=$ Dealuminification |
| ET | $=$ Etching |
| EWO | $=$ Etched only in the water |
| G | $=$ General |
| GBSL | $=$ General below sediment line |


| I | $=$ Incipient |
| :--- | :--- |
| MD | $=$ Medium |
| MO | $=$ Moderate |
| NU | $=$ Nonuniform |
| P | $=$ Pitting |
| S | $=$ Severe |
| SL | $=$ Slight |
| U | $=$ Uniform |
| V | $=$ Very |

Numbers indicate maximum depth in mils.
${ }^{d}$ Numbers refer to references at end of report.
${ }^{e}$ Cast alloy.
$f_{\text {Pitting in bottom sediment, } 12 \text { mils maximum. }}^{\text {. }}$
$g_{\text {At spacer }}$.
${ }^{b}$ At crevice.
${ }^{i}$ Severe corrosion in small area.

Table 20. Stress Corrosion of Copper Alloys (Bronzes)

| Alloy | $\begin{aligned} & \mathrm{CDA} \\ & \text { No. }{ }^{a} \end{aligned}$ | Stress <br> (ksi) | Tensile Strength <br> (\%) | Exposure (day) | Depth <br> (ft) | Specimens |  | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |  | Exposed | Failed |  |
| Phosphor bronze A | 510 | 12.0 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Phosphor bronze A | 510 | 19.0 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Phosphor bronze A | 510 | 12.5 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Phosphor bronze A | 510 | 18.7 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Phosphor bronze A | 510 | 12.6 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Phosphor bronze A | 510 | 19.0 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Phosphor bronze D | 524 | 10.0 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Phosphor bronze D | 524 | 14.0 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Phosphor bronze D | 524 | 21.0 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Phosphor bronze D | 524 | 9.8 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Phosphor bronze D | 524 | 13.9 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Phosphor bronze D | 524 | 20.9 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Phosphor bronze D | 524 | 16.4 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Phosphor bronze D | 524 | 24.5 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Al bronze, 5\% | 606 | 18.0 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Al bronze, 5\% | 606 | 26.0 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Al bronze, 5\% | 606 | 38.0 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Al bronze, 5\% | 606 | 17.9 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Al bronze, 5\% | 606 | 25.6 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Al bronze, 5\% | 606 | 38.4 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Al bronze, 5\% | 606 | 28.3 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Al bronze, 5\% | 606 | 42.5 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Si bronze A | 655 | 10.0 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Si bronze A | 655 | 14.0 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Si bronze A | 655 | 21.0 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Si bronze A | 655 | 9.6 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Si bronze A | 655 | 13.8 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Si bronze A | 655 | 20.6 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| Si bronze A | 655 | 10.8 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Si bronze A | 655 | 16.2 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |

${ }^{a}$ Copper Development Association alloy number.
${ }^{b}$ Numbers refer to references at end of report.

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | $\begin{gathered} \text { Exposure } \\ \text { (day) } \end{gathered}$ | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{b}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Original <br> (ksi) | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| Phosphor bronze A | 510 | 123 | 5,640 | 51 | 0 | 25 | -3 | 64 | +3 | CEL (4) |
| Phosphor bronze A | 510 | 403 | 6,780 | 51 | +1 | 25 | -1 | 64 | +2 | CEL (4) |
| Phosphor bronze A | 510 | 751 | 5,640 | 51 | +1 | 25 | -1 | 64 | -1 | CEL (4) |
| Phosphor bronze A | 510 | 197 | 2,340 | 51 | -3 | 25 | -13 | 64 | -1 | CEL (4) |
| Phosphor bronze A | 510 | 402 | 2,370 | 51 | -1 | 25 | -3 | 64 | -3 | CEL (4) |
| Phosphor bronze A | 510 | 181 | 5 | 51 | 0 | 25 | -7 | 64 | -2 | CEL (4) |
| Phosphor bronze D | 524 | 123 | 5,640 | 64 | +1 | 28 | 0 | 70 | 0 | CEL (4) |
| Phosphor bronze D | 524 | 403 | 6,780 | 64 | +2 | 28 | $+1$ | 70 | +3 | CEL (4) |
| Phosphor bronze D | 524 | 751 | 5,640 | 64 | +2 | 28 | +3 | 70 | -1 | CEL (4) |
| Phosphor bronze D | 524 | 197 | 2,340 | 64 | 0 | 28 | -3 | 70 | +3 | CEL (4) |
| Phosphor bronze D | 524 | 402 | 2,370 | 64 | +1 | 28 | +1 | 70 | -1 | CEL (4) |
| Phosphor bronze D | 524 | 181 | 5 | 64 | -1 | 28 | +1 | 70 | +4 | CEL (4) |
| Al bronze, 5\% | 606 | 123 | 5,640 | 85 | 0 | 51 | +2 | 45 | -9 | CEL (4) |
| Al bronze, 5\% | 606 | 403 | 6,780 | 85 | -1 | 51 | 0 | 45 | -29 | CEL (4) |
| Al bronze, 5\% | 606 | 751 | 5,640 | 85 | 0 | 51 | +1 | 45 | -27 | CEL (4) |
| Al bronze, 5\% | 606 | 197 | 2,340 | 85 | 0 | 51 | -2 | 45 | -7 | CEL (4) |
| Al bronze, 5\% | 606 | 402 | 2,370 | 85 | 0 | 51 | 0 | 45 | -12 | CEL (4) |
| Al bronze, 5\% | 606 | 181 | 5 | 85 | -3 | 51 | -3 | 45 | -20 | CEL (4) |
| Si bronze A | 655 | 123 | 5,640 | 64 | +2 | 28 | +4 | 61 | -2 | CEL (4) |
| Si bronze A | 655 | 403 | $6,780^{c}$ | 64 | -1 | 28 | -4 | 61 | -1 | CEL (4) |
| Si bronze A | 655 | 403 | $6,780^{d}$ | 64 | -25 | 28 | -18 | 61 | -40 | CEL (4) |
| Si bronze A | 655 | 751 | 5,640 | 64 | -1 | 28 | -5 | 61 | -1 | CEL (4) |
| Si bronze A | 655 | 197 | 2,340 | 64 | -2 | 28 | -14 | 61 | -3 | CEL (4) |
| Si bronze A | 655 | 402 | 2,370 | 64 | -1 | 28 | -3 | 61 | -2 | CEL (4) |
| Si bronze A | 655 | 181 | 5 | 64 | -1 | 28 | -7 | 61 | +3 | CEL (4) |

Table 21. Changes in Mechanical Properties of Copper Alloys (Bronzes) Due to Corrosion

[^2]${ }^{c}$ Totally exposed in water.
${ }^{d}$ Partially embedded in the bottom sediments.
Table 22. Chemical Composition of Copper-Nickel Alloys, Percent by Weight

| Alloy | CDA No. ${ }^{\text {a }}$ | Cu | Ni | Fe | Mn | Zn | Pb | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Cu-Ni, 95-5 | 704 | 91.98 | 6.25 | 1.24 | 0.53 | - | - | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | 89.04 | 9.42 | 1.16 | 0.38 | - | - | CEL (4) |
| Cu-Ni, 90-10 | 706 | 89.0 | 10.0 | 1.4 | 0.5 | - | - | INCO (3) |
| Cu-Ni, 90-10 | 706 | 88.7 | 10.0 | 1.3 | - | - | - | MEL (5) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10^{\text {c }}$ | 962 | 86.0 | 11.0 | 1.4 | 1.3 | - | - | INCO (3) |
| Cu-Ni, 80-20 | 710 | 78.62 | 20.41 | 0.62 | 0.35 | - | - | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | 80.0 | 20.0 | 0.03 | 0.2 | - | - | INCO (3) |
| Cu-Ni, 70-30 | 715 | 68.61 | 30.53 | 0.53 | 0.33 | - | - | CEL (4) |
| Cu-Ni, 70-30 | 715 | 69.0 | 30.0 | 0.6 | 0.4 | - | - | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30$ | 715 | 69.4 | 30.0 | 0.6 | - | - | - | MEL (5) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30$ | 716 | 64.02 | 29.95 | 5.27 | 0.75 | - | - | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 55-45$ | - | 54.0 | 45.0 | 0.1 | 1.0 | - | - | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | 62.0 | 25.0 | - | - | 8.0 | 5.0 | INCO (3) |
| Nickel-silver | 752 | 65.0 | 18.0 | - | - | 17.0 | - | INCO (3) |

${ }^{a}$ Copper Development Association alloy number.
${ }^{b}$ Numbers refer to references at end of report.
${ }^{c}$ Cast alloy.
Table 23. Corrosion Rates and Types of Corrosion of Copper-Nickel Alloys

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Environment ${ }^{\text {b }}$ | Exposure <br> (day) | Depth <br> (ft) | Corrosion <br> Rate <br> (mpy) | Type of Corrosion ${ }^{\text {c }}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Cu-Ni, 95-5 | 704 | W | 123 | 5,640 | 1.5 | U | CEL (4) |
| Cu-Ni, 95-5 | 704 | S | 123 | 5,640 | 1.5 | U | CEL (4) |
| Cu-Ni, 95-5 | 704 | W | 403 | 6,780 | 0.8 | U | CEL (4) |
| Cu-Ni, 95-5 | 704 | S | 403 | 6,780 | 0.8 | U | CEL (4) |
| Cu-Ni, 95-5 | 704 | W | 751 | 5,640 | 0.7 | U | CEL (4) |
| Cu-Ni, 95-5 | 704 | S | 751 | 5,640 | 0.6 | U | CEL (4) |
| Cu-Ni, 95-5 | 704 | W | 197 | 2,340 | 0.9 | U | CEL (4) |
| Cu-Ni, 95-5 | 704 | S | 197 | 2,340 | 0.6 | U | CEL (4) |
| Cu-Ni, 95-5 | 704 | W | 402 | 2,370 | 0.9 | U | CEL (4) |
| Cu-Ni, 95-5 | 704 | S | 402 | 2,370 | 0.8 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | W | 123 | 5,640 | 0.9 | U | MEL (5) |
| Cu-Ni, 90-10 | 706 | W | 123 | 5,640 | 1.6 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | W | 123 | 5,640 | 0.8 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | S | 123 | 5,640 | 1.2 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | S | 123 | 5,640 | 0.8 | U | INCO (3) |
| Cu-Ni, 90-10 | 706 | W | 403 | 6,780 | 0.8 | U | CEL (4) |
| Cu-Ni, 90-10 | 706 | W | 403 | 6,780 | 0.6 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | S | 403 | 6,780 | 0.7 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | S | 403 | 6,780 | $<0.1$ | U | INCO (3) |
| Cu-Ni, 90-10 | 706 | W | 751 | 5,640 | 0.5 | U | MEL (5) |
| Cu-Ni, 90-10 | 706 | W | 751 | 5,640 | 0.7 | U | CEL (4) |
| Cu-Ni, 90-10 | 706 | W | 751 | 5,640 | 0.6 | U | INCO (3) |
| Cu-Ni, 90-10 | 706 | S | 751 | 5,640 | 0.5 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | W | 1,064 | 5,300 | 0.6 | U | MEL (5) |
| Cu-Ni, 90-10 | 706 | W | 1,064 | 5,300 | 0.7 | U | INCO (3) |
| Cu-Ni, 90-10 | 706 | S | 1,064 | 5,300 | 0.2 | G | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | W | 197 | 2,340 | 0.8 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | W | 197 | 2,340 | 0.8 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | S | 197 | 2,340 | 0.5 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | S | 197 | 2,340 | $<0.1$ | U | INCO (3) |

Table 23. Continued.

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Environment ${ }^{\text {b }}$ | Exposure (day) | Depth <br> (ft) | $\begin{aligned} & \text { Corrosion } \\ & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Type of Corrosion ${ }^{c}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | W | 402 | 2,370 | 0.6 | $\mathrm{U}^{e}$ | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | W | 402 | 2,370 | 0.8 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | S | 402 | 2,370 | 0.5 | $\mathrm{U}^{e}$ | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | S | 402 | 2,370 | 0.1 | ET | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | W | 181 | 5 | 1.1 | NU | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | W | 181 | 5 | 0.9 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | W | 366 | 5 | 0.6 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10^{f}$ | 706 | w | 386 | 5 | 0.3 | U | MEL (5) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | W | 608 | 5 | 0.5 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10^{\mathrm{g}}$ | 962 | W | 402 | 2,370 | 0.7 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10^{\text {g }}$ | 962 | S | 402 | 2,370 | 0.1 | ET | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10^{\mathrm{g}}$ | 962 | w | 181 | 5 | 1.1 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10^{\mathrm{g}}$ | 962 | w | 366 | 5 | 0.9 | U | INCO (3) |
| Cu-Ni, 80-20 | 710 | W | 123 | 5,640 | 1.2 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | W | 123 | 5,640 | 1.9 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | S | 123 | 5,640 | 1.3 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | S | 123 | 5,640 | 1.1 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | W | 403 | 6,780 | 1.2 | ET | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | W | 403 | 6,780 | 1.5 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | S | 403 | 6,780 | 1.0 | ET | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | S | 403 | 6,780 | 0.1 | EBSL | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | W | 751 | 5,640 | 0.8 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | W | 751 | 5,640 | 1.3 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | S | 751 | 5,640 | 1.0 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | W | 1,064 | 5,300 | 1.0 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | S | 1,064 | 5,300 | 0.5 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | W | 197 | 2,340 | 0.7 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | W | 197 | 2,340 | 1.1 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | S | 197 | 2,340 | 0.5 | U | CEL (4) |


| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Environment ${ }^{\text {b }}$ | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) | Type of Corrosion ${ }^{c}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | S | 197 | 2,340 | $<0.1$ | SL-ET | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | W | 402 | 2,370 | 0.6 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | W | 402 | 2,370 | 1.1 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | S | 402 | 2,370 | 0.5 | $\mathrm{U}^{b}$ | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | S | 402 | 2,370 | 0.2 | ET | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | W | 181 | 5 | 2.8 | G | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 711 | W | 366 | 5 | 1.9 | G | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 123 | 5,640 | 1.0 | U | MEL (5) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 123 | 5,640 | 1.2 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 123 | 5,640 | 1.3 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | S | 123 | 5,640 | 0.8 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | S | 123 | 5,640 | 0.9 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 403 | 6,780 | 1.2 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 403 | 6,780 | 1.2 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | S | 403 | 6,780 | 1.1 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | S | 403 | 6,780 | 0.2 | G | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 751 | 5,640 | 0.4 | U | MEL (5) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 751 | 5,640 | 0.7 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 751 | 5,640 | 0.9 | G | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | S | 751 | 5,640 | 0.7 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 1,064 | 5,300 | 0.5 | U | MEL (5) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 1,064 | 5,300 | 0.6 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | S | 1,064 | 5,300 | 0.5 | G | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 123 | 2,500 | 0.6 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 197 | 2,340 | 0.7 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 197 | 2,340 | 0.9 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | S | 197 | 2,340 | 0.2 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | S | 197 | 2,340 | $<0.1$ | SL-ET | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 402 | 2,370 | 0.5 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 402 | 2,370 | 0.6 | U | INCO (3) |

Table 23. Continued.

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Environment ${ }^{\text {b }}$ | Exposure (day) | Depth <br> (ft) | Corrosion <br> Rate <br> (mpy) | Type of Corrosion ${ }^{c}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | S | 402 | 2,370 | 0.4 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | S | 402 | 2,370 | 0.1 | ET | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | w | 181 | 5 | 0.5 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 181 | 5 | 0.5 | G | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 366 | 5 | 0.4 | G | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}^{f}$ | 715 | W | 386 | 5 | 0.2 | U | MEL (5) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 398 | 5 | 0.4 | P (7) | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | W | 608 | 5 | 0.3 | I-P | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | W | 123 | 5,640 | 0.2 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | S | 123 | 5,640 | 0.2 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | W | 403 | 6,780 | 0.1 | ET | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | S | 403 | 6,780 | 0.1 | ET | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | W | 751 | 5,640 | 0.5 | C (16); NU-P (16); CO | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | S | 751 | 5,640 | 0.2 | C (14); NU-P (24); CO | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | W | 197 | 2,340 | 0.1 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | S | 197 | 2,340 | 0.1 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | W | 402 | 2,370 | 0.1 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | S | 402 | 2,370 | 0.1 | U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | w | 181 | 5 | 0.8 | I-P; C (5) | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | w | 398 | 5 | 0.7 | CR (17); U | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | W | 608 | 5 | 0.6 | C (13); CR (18) | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 55-45$ | - | W | 123 | 5,640 | 0.7 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 55-45$ | - | S | 123 | 5,640 | 0.7 | U | INCO (3) |
| Cu-Ni, 55-45 | - | W | 403 | 6,780 | 1.2 | U | INCO (3) |
| Cu-Ni, 55-45 | - | S | 403 | 6,780 | <0.1 | SL-ET | INCO (3) |
| Cu-Ni, 55-45 | - | W | 751 | 5,640 | 1.0 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 55-45$ | - | S | 751 | 5,640 | 0.5 | U | INCO (3) |
| Cu-Ni, 55-45 | - | W | 1,064 | 5,300 | 1.0 | G | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 55-45$ | - | S | 1,064 | 5,300 | 0.5 | C to PR (50); S-E | INCO (3) |


| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Environment ${ }^{\text {b }}$ | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) | Type of Corrosion ${ }^{\text {c }}$ | Source ${ }^{d}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Cu-Ni, 55-45 | - | W | 197 | 2,340 | 0.8 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 55-45$ | - | S | 197 | 2,340 | 0.2 | I-C; I-P | INCO (3) |
| Cu-Ni, 55-45 | - | W | 402 | 2,370 | 0.7 | U | INCO (3) |
| Cu-Ni, 55-45 | - | S | 402 | 2,370 | 0.1 | ET | INCO (3) |
| Cu-Ni, 55-45 | - | W | 181 | 5 | 1.8 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}, 55-45$ | - | W | 366 | 5 | 1.2 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | W | 123 | 5,640 | 0.9 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | -- | S | 123 | 5,640 | 0.6 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | W | 403 | 6,780 | 0.8 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | S | 403 | 6,780 | 0.1 | GBSL | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | W | 751 | 5,640 | 0.6 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | S | 751 | 5,640 | 0.5 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | W | 1,064 | 5,300 | 0.5 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | S | 1,064 | 5,300 | 0.3 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | W | 197 | 2,340 | 0.5 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | S | 197 | 2,340 | $<0.1$ | SL-ET | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | W | 402 | 2,370 | 0.4 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | S | 402 | 2,370 | $<0.1$ | ET | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | W | 181 | 5 | 1.0 | U | INCO (3) |
| $\mathrm{Cu}-\mathrm{Ni}-\mathrm{Zn}-\mathrm{Pb}$ | - | W | 366 | 5 | 0.7 | U | INCO (3) |
| Nickel-Silver | 752 | W | 123 | 5,640 | 2.0 | U | INCO (3) |
| Nickel-Silver | 752 | S | 123 | 5,640 | 2.6 | U | INCO (3) |
| Nickel-Silver | 752 | W | 403 | 6,780 | 1.4 | U | INCO (3) |
| Nickel-Silver | 752 | S | 403 | 6,780 | 0.5 | GBSL | INCO (3) |
| Nickel-Silver | 752 | W | 751 | 5,640 | 1.5 | U | INCO (3) |
| Nickel-Silver | 752 | S | 751 | 5,640 | 0.8 | U | INCO (3) |
| Nickel-Silver | 752 | W | 1,064 | 5,300 | 0.6 | U | INCO (3) |
| Nickel-Silver | 752 | S | 1,064 | 5,300 | 0.4 | G | INCO (3) |
| Nickel-Silver | 752 | W | 197 | 2,340 | 1.0 | U | INCO (3) |

Table 23. Continued.

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }^{a} \end{aligned}$ | Environment ${ }^{\text {b }}$ | Exposure (day) | Depth <br> (ft) | Corrosion Rate (mpy) | Type of Corrosion ${ }^{\text {c }}$ | Source ${ }^{\text {d }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Nickel-Silver | 752 | S | 197 | 2,340 | $<0.1$ | SL-ET | INCO (3) |
| Nickel-Silver | 752 | W | 402 | 2,370 | 1.0 | U | INCO (3) |
| Nickel-Silver | 752 | S | 402 | 2,370 | 0.1 | ET | INCO (3) |
| Nickel-Silver | 752 | W | 181 | 5 | 1.1 | U | INCO (3) |
| Nickel-Silver | 752 | W | 366 | 5 | 0.7 | U | INCO (3) |

[^3]Table 24. Stress Corrosion of Copper-Nickel Alloys

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Stress <br> (ksi) | Tensile Strength (\%) | Exposure <br> (day) | Depth <br> (ft) | Specimens |  | Source ${ }^{b}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |  | Exposed | Failed |  |
| Cu-Ni, 95-5 | 704 | 16.0 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}$, 95-5 | 704 | 24.0 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| Cu-Ni, 95-5 | 704 | 16.0 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}$, 95-5 | 704 | 24.0 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}$, 95-5 | 704 | 12.9 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}$, 95-5 | 704 | 19.3 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | 34.4 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Cu-Ni, 90-10 | 706 | 52.0 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | 15.0 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | 15.0 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | 8.9 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | 13.3 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | 13.0 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | 20.0 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | 13.2 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | 19.8 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | 14.0 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | 21.0 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 14.0 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 21.0 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 31.0 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 14.4 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 20.6 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 30.9 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 26.6 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 39.9 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |

${ }^{a}$ Copper Development Association alloy number.
${ }^{b}$ Numbers refer to references at end of report.
Table 25. Changes in Mechanical Properties of Copper-Nickel Alloys Due to Corrosion

| Alloy | $\begin{aligned} & \text { CDA } \\ & \text { No. }{ }^{a} \end{aligned}$ | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| $\mathrm{Cu}-\mathrm{Ni}, 95-5$ | 704 | 123 | 5,640 | 48 | +1 | 32 | +16 | 33 | +2 | CEL (4) |
| Cu-Ni, 95-5 | 704 | 403 | 6,780 | 48 | +2 | 32 | +3 | 33 | -5 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 95-5$ | 704 | 751 | 5,640 | 48 | -1 | 32 | -3 | 33 | +2 | CEL (4) |
| Cu-Ni, 95-5 | 704 | 197 | 2,340 | 48 | -4 | 32 | -9 | 33 | +7 | CEL (4) |
| Cu-Ni, 95-5 | 704 | 402 | 2,370 | 48 | -1 | 32 | -6 | 33 | +4 | CEL (4) |
| Cu-Ni, 90-10 | 706 | 123 | 5,640 | 43 | +3 | 16 | +11 | 42 | 0 | CEL (4) |
| Cu-Ni, 90-10 | 706 | 403 | 6,780 | 43 | +3 | 16 | +6 | 42 | -5 | CEL (4) |
| Cu-Ni, 90-10 | 706 | 751 | 5,640 | 43 | +3 | 16 | +9 | 42 | -2 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | 197 | 2,340 | 43 | -1 | 16 | 0 | 42 | -2 | CEL (4) |
| Cu-Ni, 90-10 | 706 | 402 | 2,370 | 43 | +4 | 16 | +13 | 42 | -6 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 90-10$ | 706 | 181 | 5 | 43 | +4 | 16 | +9 | 42 | -18 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | 123 | 5,640 | 49 | 0 | 21 | -3 | 44 | +2 | CEL (4) |
| Cu-Ni, 80-20 | 710 | 403 | 6,780 | 49 | +2 | 21 | -2 | 44 | -3 | CEL (4) |
| Cu-Ni, 80-20 | 710 | 751 | 5,640 | 49 | +1 | 21 | -2 | 44 | -3 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | 197 | 2,340 | 49 | +1 | 21 | -6 | 44 | +1 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 80-20$ | 710 | 402 | 2,370 | 49 | +1 | 21 | -2 | 44 | -2 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | 123 | 5,640 | 58 | +1 | 26 | -2 | 41 | +3 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | 403 | 6,780 | 58 | -25 | 26 | -1 | 41 | -3 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | 751 | 5,640 | 58 | -25 | 26 | -4 | 41 | -4 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | 197 | 2,340 | 58 | -3 | 26 | -14 | 41 | +1 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | 402 | 2,370 | 58 | +2 | 26 | -3 | 41 | -3 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,0.5 \mathrm{Fe}$ | 715 | 181 | 5 | 58 | +3 | 26 | -9 | 41 | -13 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 123 | 5,640 | 78 | +2 | 41 | +5 | 35 | +1 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 403 | 6,780 | 78 | -1 | 41 | -3 | 35 | -1 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 751 | 5,640 | 78 | +3 | 41 | +5 | 35 | -9 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 197 | 2,340 | 78 | +1 | 41 | -4 | 35 | 0 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 402 | 2,370 | 78 | +7 | 41 | +10 | 35 | -8 | CEL (4) |
| $\mathrm{Cu}-\mathrm{Ni}, 70-30,5.0 \mathrm{Fe}$ | 716 | 181 | 5 | 78 | +3 | 41 | +6 | 35 | -18 | CEL (4) |

[^4]
## SECTION 4

## NICKEL ALLOYS

Nickel and its alloys are passive in moving seawater, but are subject to pitting and concentration cell (crevice) corrosion in stagnant seawater. Their passivity is due to the presence of an impervious oxide layer on their surfaces which breaks down under certain conditions. Fouling organisms, deposits, and crevices which restrict the availability of oxygen to localized areas cause such breakdowns. Where sufficient oxygen is not available to repair the breaks in the protective film, pitting and crevice (concentration cell) corrosion occur. Thus, in seawater, pitting and crevice corrosion are the most prevalent modes of attack.

Corrosion rates calculated from weight losses due to localized corrosion are meaningless because they present an untrue picture of the corrosion behavior of the alloy. Corrosion rates such as mils-per-year connote a uniform thickness of metal lost over a period of time, assuming uniform corrosion. Hence, a very low corrosion rate resulting from a few deep pits or crevice corrosion in one area will present a very misleading picture of the corrosion behavior of an alloy in that particular environment.

The data on the nickel alloys were obtained from the reports given in References 3 through 19 and 23. They are separated into different groups (nickels, nickel-copper-alloys, and nickel alloys) for comparison and discussion purposes.

The chemical compositions, corrosion rates and types of corrosion, and changes in mechanical properties due to corrosion of the nickels are given in Tables 26 to 28; those of the nickel-copper alloys in Tables 29 to 31; those of the nickel alloys in Tables 32 to 34 ; and the resistance to stress corrosion in Table 35.

The effects of depth and the effect of the concentration of oxygen in seawater on the corrosion of the nickels, the nickel-copper alloys, and the nickel alloys are shown in Figures 15 and 16.

### 4.1. NICKELS

The chemical compositions of the nickels are given in Table 26, their corrosion rates and types of
corrosion in Table 27, and the changes in their mechanical properties due to corrosion in Table 28.

### 4.1.1. Duration of Exposure

The corrosion rates and types of corrosion of the seven nickels ( $94 \%$ minimum nickel) are given in Table 27. Pitting, crevice, and edge (on the sheared ends) localized types of corrosion were responsible for practically all the corrosion. The edge corrosion was caused by microcracks and microcrevices that formed during the shearing operation; this illustrates dramatically the corrosion damage that can be caused by this fabricating procedure. Lateral penetration, initiated at a sheared edge, of as much as an inch during 6 months of exposure was found. To prevent this type of corrosion, all deformed metal created during shearing or punching operations must be removed by machining, grinding, or reaming.

Because the corrosion of the nickels was localized, no definite correlation with duration of exposure was possible. However, the severity of pitting and crevice corrosion increased with increasing time of exposure at depth as well as at the surface. Corrosion rates increased with duration of exposure at the 6,000 -foot depth, although they were neither progressive nor constant. In some cases corrosion rates were considerably higher during shorter times of exposure than after longer times of exposure. Corrosion rates at the 2,500 -foot depth were constant with increasing time of exposure.

### 4.1.2. Effect of Depth

The severity and frequency of pitting and crevice corrosion were much greater at the surface than at depth. Also, the average corrosion rates were greater at the surface than at depth, although they did not decrease progressively with increasing depth as shown in Figure 16. The curves in Figure 16 are based on average values for each group of alloys.

### 4.1.3. Effect of Concentration of Oxygen

The severity and frequency of pitting and crevice corrosion, in general, increased with increasing concentration of oxygen in seawater. The average corrosion rates increased progressively, but not constantly, with increasing concentration of oxygen in seawater as shown in Figure 17.

### 4.1.4. Effect of Welding

The weld beads were preferentially corroded when nickel $\mathrm{Ni}-200$ was welded by manual shielded metal-arc welding using welding electrode 141 , and by TIG welding using filler metal 61 . The weld beads were severely pitted when welded with electrode 141. The weld beads and heat-affected zones were perforated when welded with filler metal 61. This preferential attack of the weld bead materials indicates that they were anodic to the parent sheet metal.

### 4.1.5. Mechanical Properties

The effect of exposure on the mechanical properties of nickel Ni-200 is shown in Table 28. The mechanical properties were not affected by exposure at depth for 1,064 days or for 181 days at the surface.

### 4.2. NICKEL-COPPER ALLOYS

The chemical compositions of the nickel-copper alloys are given in Table 29, their corrosion rates and types of corrosion in Table 30, and the changes in their mechanical properties due to corrosion in Table 31.

### 4.2.1. Duration of Exposure

The corrosion rates and types of corrosion of seven nickel-copper alloys are given in Table 30 . Except for the cast alloys 410 and 505 , the predominant types of corrosion were pitting and crevice. At the 6,000 -foot depth there was an overall tendency for the corrosion rates of the cast alloys to decrease with increasing duration of exposure, but
this tendency was neither progressive nor constant. Because the corrosion of the other nickel-copper alloys was localized (pitting and crevice corrosion), no definite correlation with duration of exposure was possible either at depth or at the surface. However, the intensity of pitting and crevice corrosion, in general, increased with increasing duration of exposure both at depth and at the surface.

### 4.2.2. Effect of Depth

The severity and frequency of pitting and crevice corrosion were much greater at the surface than at depth. Also, the average corrosion rates were greater at the surface than at depth, although they did not decrease progressively or constantly with increasing depth, as shown in Figure 16. Although these corrosion rates are unreliable because they are based upon localized corrosion weight losses, they do substantiate the conclusion based upon the frequency and severity of pitting and crevice corrosion.

### 4.2.3. Effect of Concentration of Oxygen

The severity and frequency of pitting and crevice corrosion, in general, increased with increasing concentration of oxygen in seawater. Even though pitting and crevice corrosion were the predominant types for these alloys in seawater, their average corrosion rates calculated from weight losses increased linearly with increasing concentration of oxygen, as shown in Figure 17.

### 4.2.4. Effect of Welding

When Ni-Cu 400 alloy was welded with filler metal 60 by the TIG welding process, the weld beads were severely pitted both in the seawater and in the bottom sediment after exposure for 402 days at a depth of 2,500 feet, but they were corroded uniformly after 181 days of exposure at the surface. Butt welds in $\mathrm{Ni}-\mathrm{Cu} 400$ alloy made by the manual shielded metal-arc process with electrode 190 were attacked by incipient pitting corrosion both in the seawater and in the bottom sediment after 189 days of exposure at a depth of 5,900 feet and by crater corrosion of the weld bead after 540 days of exposure at the surface. Three-inch-diameter, unrelieved,
circular welds in $\mathrm{Ni}-\mathrm{Cu} 400$ alloy by the manual shielded metal-arc process with electrode 190 corroded uniformly both in the seawater and in the bottom sediment after 189 days of exposure at a depth of 5,900 feet. The unrelieved circular welds were tested to determine whether welding stresses would cause any corrosion-induced cracking. When $\mathrm{Ni}-\mathrm{Cu} 400$ alloy was welded by the manual shielded metal-arc process with electrodes 130 and 180, the weld beads were corroded uniformly after 181 days of exposure at the surface and after 402 days of exposure at the 2,500 -foot depth. There was no preferential corrosion when $\mathrm{Ni}-\mathrm{Cu} 400$ was TIG welded with electrode 167 after 402 days of exposure at the 2,500 -foot depth, but the weld bead was selectively attacked and was covered with a deposit of copper after 403 days of exposure at the 6,000 -foot depth [7].

The weld beads in Ni-Cu K-500 alloy made by the manual shielded metal-arc process with electrode 134 were attacked by pitting corrosion of the weld bead and the heat-affected zone after 181 days of exposure at the surface, by crater corrosion of the weld bead after 540 days of exposure at the surface, and by line corrosion at the edge of the weld bead after 402 days of exposure at the 2,500 -foot depth. When Ni-Cu K-500 alloy was TIG welded with filler metal 64 , the weld beads were uniformly corroded after 181 days of exposure at the surface and 402 days of exposure at the 2,500 -foot depth, and the weld beads and the heat-affected zones were attacked by pitting corrosion after 540 days of exposure at the surface.

### 4.2.5. Galvanic Corrosion

When AISI 4130 steel was fastened to $\mathrm{Ni}-\mathrm{Cu} 400$ alloy in a surface area ratio of $1: 2$, the AISI 4130 was severely corroded and the $\mathrm{Ni}-\mathrm{Cu} 400$ was uncorroded after 403 days of exposure at the 6,000 -foot depth [7]. This shows that the steel was being sacrificed to protect the nickel- copper alloy.

### 4.2.6. Crevice Corrosion

$\mathrm{Ni}-\mathrm{Cu} 400$ alloy hardware was attacked by crevice corrosion after 751 days of exposure at the 6,000 -foot depth when in contact with fiberglass [13].

### 4.2.7. Corrosion Products

X-ray diffraction, spectrochemical, and chemical analyses of corrosion products removed from nickelcopper alloys 400 and $\mathrm{K}-500$ showed that they were composed of cupric oxide ( CuO ), nickel oxide ( NiO ), nickel hydroxide $\left(\mathrm{Ni}(\mathrm{OH})_{2}\right)$, cupric chloride $\left(\mathrm{CuCl}_{2}\right)$, copper- oxy-chloride $\left(\mathrm{CuCl}_{2} ; 3 \mathrm{CuO} ; 4 \mathrm{H}_{2} \mathrm{O}\right)$, a trace of nickel sulfide (NiS), and phosphate, chloride, and sulfate ions.

### 4.2.8. Mechanical Properties

The effects of exposure on the mechanical properties of $\mathrm{Ni}-\mathrm{Cu} 400$ and $\mathrm{K}-500$ alloys are shown in Table 31. There were no significant changes due to corrosion of either unwelded or welded alloys.

### 4.3. NICKEL ALLOYS

The chemical compositions of the nickel alloys are given in Table 32, their corrosion rates and types of corrosion in Table 33, and the changes in their mechanical properties due to corrosion in Table 34.

There were no significant weight losses (none greater than 0.1 mpy ) or any visible corrosion on any of the following alloys:
$\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 718$, unwelded and welded
$\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo} 625$, unwelded and welded

## $\mathrm{Ni}-\mathrm{Mo}-\mathrm{CrC}$ and 3

$\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}-\mathrm{Mo} \mathrm{F}$ and G
$\mathrm{Ni}-\mathrm{Cr}-\mathrm{Co} 41$
There were no significant weight losses (none greater than 0.1 mpy ) and only some cases of incipient crevice corrosion on the following alloys:
$\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 804,825 \mathrm{Cb}$, and 901
Ni-Co-Cr 700
$\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}-\mathrm{Mo} \mathrm{X}$
The corrosion resistance of $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825 \mathrm{Cb}$ was better than that of its counterparts 825 and 825 S (sensitized). Alloy 825 was attacked by both pitting and crevice corrosion, and 825 S had only one case of
crevice corrosion. Thus, the addition of small amounts of columbium to alloy 825 improves its corrosion resistance, at least in seawater.

All the nickel alloys corroded essentially the same in the bottom sediments as in the seawater above them.

### 4.3.1. Duration of Exposure

The corrosion rates and types of corrosion of the nickel alloys are given in Table 33. Except for the 12 alloys above, 13 of the remaining 16 alloys were attacked by crevice and pitting corrosion with crevice corrosion being considerably more predominant. $\mathrm{Ni}-\mathrm{Be}$ alloy was attacked by pitting corrosion on the ends of the bars. Ni-Mo-Fe alloys B and 2 were attacked by general corrosion. Because of the crevice and pitting types of corrosion, corrosion rates were meaningless for determining effects of duration of exposure on the corrosion behavior of these alloys. These 14 alloys were: $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}$ alloys 600,610, $\mathrm{X}-750$, and $88, \mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr}$ alloys $800,825,825 \mathrm{~S}$, and 902, Ni-Sn-Zn 23, Ni-Cr alloys $65-35,75$, and $80-20$, Ni -Si alloy D , and $\mathrm{Ni}-\mathrm{Be}$.

### 4.3.2. Effect of Depth

The severity and frequency of crevice and pitting corrosion, in general, of the 16 alloys given in the previous paragraph were much greater at the surface than at depth. Also, the average corrosion rates were greater at the surface than at depth, although they did not decrease progressively or constantly with increasing depth, as shown in Figure 16. Although these corrosion rates are unreliable because they are based upon localized corrosion weight losses, they do substantiate the conclusion based upon the frequency and severity of pitting and crevice corrosion.

### 4.3.3. Effect of Concentration of Oxygen

The severity and frequency of crevice and pitting corrosion of the nickel alloys which corroded significantly, in general, increased with increasing concentration of oxygen in seawater. Their average corrosion rates calculated from weight losses increased asymptotically with increasing concentration of oxygen, as shown in Figure 17.

### 4.3.4. Effect of Weiding

The weld beads in Ni-Cr-Fe 600 alloy, made by the TIG welding process using filler metal 62 , were perforated by line corrosion along their edges after 402 days of exposure at the 2,500 -foot depth, and 540 days of exposure at the surface; the weld bead was attacked by incipient pitting corrosion after 181 days of exposure at the surface.

When Ni-Cr-Fe 600 alloy was TIG welded with filler metal 82 , the weld beads and heat-affected zones were perforated after 402 days of exposure at the 2,500-foot depth; the weld bead was pitted after 540 days of exposure at the surface; and the weld bead was slightly etched after 181 days of exposure at the surface.

The weld beads in Ni-Cr-Fe 600 alloy, made by the manual shielded metal-arc process using electrode 132, were perforated after 402 days of exposure at the 2,500 -foot depth and after 540 days of exposure at the surface. The weld beads were also attacked by tunnel corrosion after 540 days of exposure at the surface.

Weld beads in $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ alloy, made by the manual shielded metal-arc process using electrode 182, were perforated after 181 days of exposure at the surface, but were only etched after 402 days of exposure at the 2,500 -foot depth.

Butt welds in $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 718$ alloy, made by the TIG process using filler metal 718 , were uncorroded after 189 days of exposure in both seawater and bottom sediment at the 6,000-foot depth, in seawater after 402 days of exposure at the 2,500-foot depth, and after 540 days of exposure at the surface. Also, 3-inch-diameter, unrelieved, circular weld beads made by the same process were etched after 189 days of exposure in seawater and in the bottom sediment at the 6,000 -foot depth.

The weld beads in $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}$ X-750 alloy, made by the TIG process using filler metal 69 , were etched after 402 days of exposure at the 2,500-foot depth, but both the weld beads and heat-affected zones were attacked by crater corrosion after 540 days of exposure at the surface. Weld beads in $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}$ X-750 alloy, made by the manual shielded metal-arc process, were perforated and the heat-affected zone was attacked by tunnel corrosion after 402 days of exposure at the 2,500 -foot depth; the heat-affected
zone was perforated by crater corrosion after 540 days of exposure at the surface.

Weld beads in Ni-Fe-Cr 800 alloy, made by the TIG process with filler metal 82 , were perforated by line corrosion along their edges after 402 days of exposure at the 2,500 -foot depth, and both the weld beads and heat-affected zones were attacked by tunnel corrosion after 540 days of exposure at the surface. There was line corrosion along the edge of the weld beads when $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ alloy was welded by the manual shielded metal-arc process using electrode 138 after 402 days of exposure at the 2,500 -foot depth. Both the weld beads and heataffected zones were perforated by corrosion after 540 days of exposure at the surface when $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ alloy was welded by the manual shielded metal-arc process using electrode 182.

Weld beads in Ni-Fe-Cr 825 alloy, made by the TIG welding process with filler metal 65 , were uncorroded after 402 days of exposure at the 2,500-foot depth and after 181 days of exposure at the surface; the weld beads and heat-affected zones were attacked by incipient pitting corrosion after 540 days of exposure at the surface. When butt welds were made by the manual shielded metal-arc process using electrode 135 , the weld beads were uncorroded after 181 days of exposure at the surface and 189 days of exposure in the bottom sediment at the 6,000 -foot depth; there was incipient pitting of the weld bead after 189 days of exposure in the seawater at the 6,000 -foot depth; one end of the weld bead was corroded after 402 days of exposure at the 2,500-foot depth; and there was crater corrosion of the heat-affected zone after 540 days of exposure at the surface. When the weld beads were 3 -inchdiameter unrelieved circles made by the manual shielded metal-arc process, they were uncorroded after 189 days of exposure in seawater and in the bottom sediment at the 6,000-foot depth.

Butt welds and 3 -inch-diameter, unrelieved circular welds in Ni-Cr-Mo 625 alloy, made by the TIG welding process using filler metal 625, were uncorroded after 189 days of exposure at the 6,000 -foot depth and after 588 days of exposure at the surface.

### 4.3.5. Galvanic Corrosion

When AISI 4130 steel was fastened to $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}$ 600 alloy in a surface area ratio of $1: 2$, the 4130 was severely corroded and the $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ alloy was uncorroded after 403 days of exposure at the 6,000 -foot depth [7]. This shows that the 4130 steel was the anodic member of the couple and was being sacrificed to protect the cathodic nickel alloy. When $\mathrm{Ni}-\mathrm{Be}$ alloy was fastened to $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ alloy in a surface area ratio of $1: 2$, the $\mathrm{Ni}-\mathrm{Be}$ was severely attacked with there being a much lesser amount of corrosion on the Ni-Cr-Fe 600 alloy.

### 4.3.6. Mechanical Properties

The effects of exposure on the mechanical properties of five of the nickel alloys are given in Table 34. The mechanical properties of $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ and $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ alloys were not affected. However, there were significant decreases in the elongations of alloys $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$, $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 902$, and $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{HT}$.

### 4.4. STRESS CORROSION

The susceptibility of some of the nickel alloys to stress corrosion is given in Table 35. None of the alloys tested were susceptible to stress corrosion cracking at both the 2,500 -foot and 6,000 -foot depths for exposures of at least 400 days duration.


Figure 16. Effect of depth on the corrosion of nickel alloys after 1 year of exposure.


Figure 17. Effect of concentration of oxygen in seawater on the corrosion of nickel alloys after 1 year of exposure.

Table 26. Chemical Composition of Nickels, Percent by Weight

| Alloy | Ni | C | Mn | Fe | S | Si | Cu | Ti | Other | Source $^{a}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Electrolytic Ni | $99.97+\mathrm{Co}$ | - | - | - | - | - | - | - | - | INCO (3) |
| $\mathrm{Ni}-200$ | 99.5 | 0.05 | 0.29 | 0.04 | 0.006 | 0.07 | 0.02 | - | - | CEL (4) |
| $\mathrm{Ni}-200$ | 99.5 | 0.06 | 0.25 | 0.15 | 0.005 | 0.05 | 0.05 | - | - | CEL (4) |
| $\mathrm{Ni}-200$ | 99.5 | 0.06 | - | - | - | - | - | - | - | INCO (3) |
| $\mathrm{Ni}-201$ | 99.5 | 0.01 | - | - | - | - | - | - | - | INCO (3) |
| $\mathrm{Ni}-211$ | 95.0 | - | 5.0 | - | - | - | - | - | - | INCO (3) |
| $\mathrm{Ni}-270$ | 99.97 | - | - | - | - | - | - | - | - | INCO (3) |
| Ni-210, cast | 95.6 | - | 1.0 | - | - | 2.0 | - | - | - | INCO (3) |
| Ni-301 | 94.0 | - | - | - | - | - | - | - | 4.5 Al | INCO (3) |
| Filler metal 61 | 96.0 | 0.06 | 0.30 | 0.10 | 0.005 | 0.40 | 0.02 | 3.0 | - | CEL (4) |
| Electrode 141 | 96.0 | 0.05 | 0.25 | 0.30 | 0.005 | 0.60 | 0.05 | 2.2 | 0.25 Al | CEL (4) |

[^5]Table 27. Corrosion Rates and Types of Corrosion of Nickels

Table 27. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum Pit Depth (mil) | Crevice Depth (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| Ni -200 | S | 197 | 2,340 | 0.5 | - | - | EX-E ${ }^{\text {d }}$ | - | CEL (4) |
| Ni-200 | S | 197 | 2,340 | $<0.1$ | - | - | NU-ET | - | INCO (3) |
| Ni-200 | W | 402 | 2,370 | 0.6 | - | 3 | C; S-ET | - | CEL (4) |
| Ni 200 | W | 402 | 2,370 | 0.6 | - | 50 | C (PR) | - | INCO (3) |
| Ni -200, welded, electrode 141 | W | 402 | 2,370 | 0.8 | - | - | ET; I-P | S-P | CEL (4) |
| Ni-200, welded, filler metal 61 | W | 402 | 2,370 | 0.6 | - | - | ET; I-P | P (PR) | CEL (4) |
| $\mathrm{Ni}-200$ | S | 402 | 2,370 | 0.5 | - | - | $\mathrm{U}^{f}$ | - | CEL (4) |
| Ni-200 | S | 402 | 2,370 | 0.2 | - | - | I-C; I-P | - | INCO (3) |
| $\mathrm{Ni}-200$, welded, filler metal 61 | S | 402 | 2,370 | 0.7 | - | - | $\mathrm{U}^{f}$ | - | INCO (3) |
| Ni -200 | W | 181 | 5 | 1.9 | 45 | 50 | C; P | - | CEL (4) |
| $\mathrm{Ni}-200$ | W | 181 | 5 | 7.2 | 50 | 50 | C (PR); P (PR) | - | INCO (3) |
| $\mathrm{Ni}-200$ | W | 366 | 5 | 4.5 | 40 | 40 | C (PR); P (PR) | - | INCO (3) |
| Ni-200 | W | 398 | 5 | 1.9 | 125 | - | $\mathrm{P}(\mathrm{PR})$; T | - | CEL (4) |
| $\mathrm{Ni}-200$ | W | 540 | 5 | 1.5 | 125 | - | P (PR); T | - | CEL (4) |
| Ni-200, welded, filler metal 61 | W | 540 | 5 | 1.9 | 125 | - | P (PR); T | WB (PR); HAZ (PR) | CEL (4) |
| Ni-200 | W | 588 | 5 | 1.5 | 125 | - | $\mathrm{P}(\mathrm{PR})$; T | - | CEL (4) |
| Ni -201 | W | 123 | 5,640 | 0.5 | 50 | - | P (PR) | - | INCO (3) |
| Ni-201 | S | 123 | 5,640 | 1.3 | 50 | - | P (PR) | - | INCO (3) |
| Ni -201 | W | 403 | 6,780 | 0.7 | 50 | 50 | C (PR); P (PR) | - | INCO (3) |
| Ni -201 | S | 403 | 6,780 | 0.2 | - | 20 | C | - | INCO (3) |
| Ni-201 | W | 751 | 5,640 | 1.1 | 30 | 30 | C (PR); P (PR) | - | INCO (3) |
| Ni 201 | S | 751 | 5,640 | 0.2 | 30 | 30 | C (PR); P (PR) | - | INCO (3) |
| Ni -201 | W | 1,064 | 5,300 | 0.8 | 50 | 50 | C (PR); P (PR) | - | INCO (3) |
| Ni-201 | S | 1,064 | 5,300 | 0.3 | 50 | 50 | C (PR); P (PR) | - | INCO (3) |
| Ni -201 | W | 197 | 2,340 | 0.5 | - | 50 | C (PR) ; $\mathrm{p}^{g}$ | - | INCO (3) |
| Ni -201 | S | 197 | 2,340 | $<0.1$ | - | - | 1-C | - | INCO (3) |
| Ni-201 | W | 402 | 2,370 | 0.6 | 50 | 50 | C (PR); P (PR) | - | INCO (3) |
| Ni -201 | W | 181 | 5 | 6.7 | 50 | 50 | $\mathrm{C}(\mathrm{PR})$; P (PR) | - | INCO (3) |
| Ni -201 | w | 366 | 5 | 3.6 | 50 | 50 | C (PR); P (PR) | - | INCO (3) |
| Ni-210, cast | W | 123 | 5,640 | 2.0 | 8 | - | P | - | INCO (3) |
| Ni-210, cast | S | 123 | 5,640 | 1.1 | 23 | - | P | - | INCO (3) |
| $\mathrm{Ni}-210$, cast | W | 403 | 6,780 | 7.2 | - | 70 | C | - | INCO (3) |
| Ni -210, cast | S | 403 | 6,780 | 0.3 | - | - | U | - | INCO (3) |
| Ni-210, cast | W | 751 | 5,640 | 3.3 | - | 75 | C | - | INCO (3) |


Table 27. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum Pit Depth (mil) | Crevice Depth (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{b}$ |  |
| Ni-301 | W | 1,064 | 5,300 | 1.8 | - | 50 | C (PR) | - | INCO (3) |
| Ni -301 | S | 1,064 | 5,300 | 1.1 | 35 | 50 | C (PR); $\mathbf{P}$ | - | INCO (3) |
| Ni 301 | W | 197 | 2,340 | 1.1 | - | 50 | C (PR) | - | INCO (3) |
| Ni-301 | S | 197 | 2,340 | $<0.1$ | - | - | 1-C | - | INCO (3) |
| Ni -301 | W | 402 | 2,370 | 0.7 | - | - | SL-E | - | INCO (3) |
| Ni -301 | S | 402 | 2,370 | 0.2 | - | 35 | C (PR) | - | INCO (3) |
| Ni-301 | W | 181 | 5 | 3.8 | 18 | - | P | - | 1NCO (3) |
| Ni -301 | W | 366 | 5 | 4.1 | 40 | 40 | C (PR); P (PR) | - | INCO (3) |

${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure
so that a portion of each specimen was embedded in the bottom sediments.
${ }^{b}$ Symbols for types of corrosion:
$\begin{aligned} \mathrm{P} & =\text { Pitting } \\ \mathrm{PR} & =\text { Perforated } \\ \mathrm{S} & =\text { Severe } \\ \mathrm{SL} & =\text { Slight } \\ \mathrm{T} & =\text { Tunnel } \\ \mathrm{U} & =\text { Uniform } \\ \mathrm{V} & =\text { Very } \\ \mathrm{W} & =\text { Weld bead }\end{aligned}$
Numbers indicate maximum depth in mils.
$=$ Crevice
$=$ Edge
$=$ Etched
$=$ Extensive
$=$ General
$=$ Heat affected zone
$=$ Incipient
$=$ No visible corrosion
C
EX
HAZ
$-\underset{Z}{Z}$
$\mathrm{NU}=$ Nonuniform
${ }^{c}$ Numbers refer to references at end of report.
${ }^{d}$ Sheared edges only.
${ }^{e}$ One pit only.
Portion in sediment bright, uncorroded.
$g_{\text {Elongated pits. }}$

Table 28. Changes in Mechanical Properties of Nickel-200 Due to Corrosion

| Alloy | Exposure <br> (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (ksi) | \% Change | $\begin{aligned} & \text { Original } \\ & (\mathrm{ksi}) \end{aligned}$ | \% Change | Original <br> (\%) | \% Change |  |
| Ni-200 | 123 | 5,640 | 65 | -1 | 18 | -16 | 46 | 0 | CEL (4) |
| Ni-200 | 403 | 6,780 | 65 | +2 | 18 | +1 | 46 | -1 | CEL (4) |
| Ni-200 | 751 | 5,640 | 65 | +1 | 18 | +1 | 46 | -4 | CEL (4) |
| Ni-200 | 1,064 | 5,300 | 65 | 0 | 18 | -10 | 46 | -13 | CEL (4) |
| Ni 200 | 197 | 2,340 | 65 | 0 | 18 | -10 | 46 | -4 | CEL (4) |
| Ni-200 | 402 | 2,370 | 65 | +1 | 18 | +4 | 46 | -5 | CEL (4) |
| Ni-200 | 181 | 5 | 65 | +1 | 18 | +10 | 46 | -4 | CEL (4) |

${ }^{a}$ Numbers refer to references at end of report.

Table 29. Chemical Composition of Nickel-Copper Alloys, Percent by Weight

| Alloy | Ni | Cu | C | Mn | Fe | S | Si | Ti | Other | Source $^{a}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :--- |
| Ni-Cu 400 | 65.17 | 32.62 | 0.11 | 1.06 | 0.90 | 0.007 | 0.10 | - | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 66.00 | 31.50 | 0.12 | 0.90 | 1.35 | 0.005 | 0.15 | - | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 68.02 | 29.25 | 0.12 | 0.99 | 1.52 | 0.010 | $<0.05$ | - | $<0.10 \mathrm{Al}$ | CEL (4) |
| Ni-Cu 400 | 65.90 | 31.75 | 0.14 | 0.94 | 1.07 | 0.010 | 0.19 | - | $<0.10 \mathrm{Al}$ | CEL (4) |
| Ni-Cu 400 | 66.00 | 32.00 | - | 0.90 | 1.40 | - | 0.20 | - | - | INCO (3) |
| Filler metal 60 | 66.00 | 30.50 | 0.03 | 0.35 | 0.10 | 0.005 | 0.50 | 2.20 | - | CEL (4) |
| Electrode 130 | 68.00 | 27.00 | 0.15 | 2.50 | 0.50 | 0.005 | 0.40 | 0.30 | 1.00 Al | CEL (4) |
| Electrode 180 | 63.00 | 28.00 | 0.03 | 5.00 | 0.25 | 0.005 | 0.75 | 0.70 | 0.30 Al | CEL (4) |
| Ni-Cu 402 | 58.00 | 40.00 | - | 0.90 | 1.20 | - | 0.10 | - | - | INCO (3) |
| Ni-Cu 406 | 84.00 | 13.00 | - | 0.90 | 1.40 | - | 0.20 | - | - | INCO (3) |
| Ni-Cu 410 | 66.00 | 31.00 | - | 0.80 | 1.00 | - | 1.60 | - | - | INCO (3) |
| Ni-Cu K-500 | 65.00 | 29.50 | 0.15 | 0.60 | 1.00 | 0.005 | 0.15 | 0.50 | 2.80 Al | CEL (4) |
| Ni-Cu K-500 | 65.00 | 30.00 | - | 0.60 | 1.00 | - | 0.20 | - | 2.80 Al | INCO (3) |
| Ni-Cu 505 | 64.00 | 29.00 | - | 0.80 | 2.00 | - | 4.00 | - | - | INCO (3) |
| Filler metal 64 | 65.00 | 29.50 | 0.15 | 0.60 | 1.00 | 0.005 | 0.15 | 0.50 | 2.80 Al | CEL (4) |
| Electrode 134 | 66.00 | 27.00 | 0.25 | 2.50 | 1.00 | 0.005 | 0.40 | 0.30 | 2.00 Al | CEL (4) |
| Ni-Cu 60 | 65.00 | 30.00 | - | 0.90 | 2.00 | - | 1.00 | - | 1.00 Al | INCO (3) |

${ }^{a}$ Numbers refer to references at end of report.
${ }^{b}$ Cast alloy.
Table 30. Corrosion Rates and Types of Corrosion of the Nickel-Copper Alloys

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth (ft) | Corrosion |  |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum Pit Depth (mil) | Crevice Depth (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| Ni-Cu 400 | W | 123 | 5,640 | 0.8 | - | - | U | - | CEL (4) |
| Ni -Cu 400 | W | 123 | 5,640 | 0.4 | - | - | U | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | W | 123 | 5,640 | 0.5 | - | - | ET; I-C | - | MEL (5) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | S | 123 | 5,640 | 0.5 | - | - | U | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | S | 123 | 5,640 | 0.4 | - | 5 | C; U | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | W | 189 | 5,900 | 0.4 | - | 5 | C; I-P | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | S | 189 | 5,900 | 0.3 | 1 | - | I-C; I-P | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$, long. butt weld, electrode 190 | W | 189 | 5,900 | 0.4 | - | - | I-P | I-P | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$, long. butt weld, electrode 190 | S | 189 | 5,900 | 0.4 | - | - | I-P | I-P | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400,3-\mathrm{in}$. circular weld, electrode 190 | W | 189 | 5,900 | 0.3 | - | - | U | U | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400,3-\mathrm{in}$. circular weld, electrode 190 | S | 189 | 5,900 | 0.3 | - | - | U | U | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | W | 403 | 6,780 | 0.5 | 20 | 10 | C; P; E | $\rightarrow$ | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | W | 403 | 6,780 | 0.8 | - | 40 | C (PR); U | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | S | 403 | 6,780 | 0.4 | 18 | 10 | C; P; E | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | S | 403 | 6,780 | 0.1 | - | 2 | $\mathrm{C} ; \mathrm{U}$ | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | w | 751 | 5,640 | 1.0 | 45 | 45 | C; P; E | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | W | 751 | 5,640 | 3.1 | - | 40 | $\mathrm{C}(\mathrm{PR})$; P | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | W | 751 | 5,640 | 0.1 | - | 13 | C; ET | - | MEL (5) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | S | 751 | 5,640 | 1.3 | - | 40 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | W | 1,064 | 5,300 | 0.8 | - | 6 | $\mathrm{C}_{\mathrm{i}} \mathrm{E}$ | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | w | 1,064 | 5,300 | 0.5 | - | 40 | $\mathrm{C}(\mathrm{PR})$; P | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | S | 1,064 | 5,300 | 0.6 | 47 | 125 | C (PR); P | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | S | 1,064 | 5,300 | 0.6 | - | 40 | C (PR); CR | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | W | 197 | 2,340 | 0.4 | 10 | 11 | C; P | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | w | 197 | 2,340 | 0.4 | - | 7 | C | - | INCO (3) |
| $\mathrm{Ni} \cdot \mathrm{Cu} 400$ | W | 197 | 2,340 | 1.0 | - | - | U | - | NADC (7 |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | S | 197 | 2,340 | 0.3 | - | - | U | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | S | 197 | 2,340 | 0.2 | - | 4 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | W | 402 | 2,370 | 0.4 | 20 | - | P | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | W | 402 | 2,370 | 0.8 | - | 40 | C (PR); P | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | S | 402 | 2,370 | 0.3 | - | - | $\mathrm{E}_{\mathrm{i}} \mathrm{I}-\mathrm{P}$ | - | CEL (4) |
| Ni-Cu 400 | S | 402 | 2,370 | 0.1 | - | - | ET | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 400$, welded, electrode 130 | W | 402 | 2,370 | 0.5 | - | - | I-P | U | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$, welded, electrode 180 | W | 402 | 2,370 | 0.5 | - | - | I-P | U | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$, welded, filler metal 60 | W | 402 | 2,370 | 0.4 | - | - | I-P | S-P | CEL (4) |


Table 30. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum Pit Depth (mil) | Crevice Depth (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{b}$ |  |
| $\mathrm{Ni}-\mathrm{Cu} 406$ | S | 197 | 2,340 | <0.1 | - | - | U-ET | - | INCO (3) |
| Ni-Cu 406 | W | 402 | 2,370 | 0.6 | - | 50 | C (PR) | - | INCO (3) |
| Ni-Cu 406 | S | 402 | 2,370 | 0.1 | - | 30 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 406$ | W | 181 | 5 | 7.5 | 50 | 50 | C (PR); P (PR) | - | INCO (3) |
| Ni-Cu 406 | W | 366 | 5 | 6.0 | 50 | 50 | $\mathrm{C}(\mathrm{PR})$; P (PR) | - | INCO (3) |
| Ni-Cu 410, cast | W | 123 | 5,640 | 0.8 | - | - | U | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 410$, cast | S | 123 | 5,640 | 0.5 | - | - | U | - | INCO (3) |
| Ni-Cu 410, cast | W | 403 | 6,780 | 1.1 | - | - | U | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 410$, cast | S | 403 | 6,780 | <0.1 | - | - | EBSL | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 410$, cast | W | 751 | 5,640 | 0.9 | - | - | U | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 410$, cast | S | 751 | 5,640 | 0.5 | - | - | U | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 410$, cast | W | 1,064 | 5,300 | 0.5 | - | - | G | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 410$, cast | S | 1,064 | 5,300 | 0.4 | - | - | G | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 410$, cast | W | 197 | 2,340 | 0.6 | - | - | U | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 410$, cast | S | 197 | 2,340 | 0.2 | - | - | U | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 410$, cast | W | 402 | 2,370 | 0.4 | - | - | G | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 410$, cast | S | 402 | 2,370 | 0.1 | - | - | I-P | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 410$, cast | W | 181 | 5 | 1.3 | 16 | 14 | C; P | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 410$, cast | W | 366 | 5 | 3.1 | 19 | 30 | $\mathrm{C} ; \mathrm{P}$ | - | INCO (3) |
| Ni-Cu K-500 | W | 123 | 5,640 | 0.4 | - | 9 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} \mathrm{K}-500$ | W | 123 | 5,640 | 0.7 | - | 11 | ET; C | - | MEL (5) |
| Ni-Cu K-500 | S | 123 | 5,640 | 0.7 | - | 11 | C | - | INCO (3) |
| Ni-Cu K-500 | W | 187 | 5,900 | 0.1 | 9 | 16 | C; P | - | CEL (4) |
| Ni-Cu K-500 | S | 187 | 5,900 | 0.2 | 11 | 26 | C; P | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} \mathrm{K}-500$ | W | 403 | 6,780 | 0.3 | - | 18 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu}$ K-500 | S | 403 | 6,780 | <0.1 | - | 2 | C | - | INCO (3) |
| Ni-Cu K-500 | W | 751 | 5,640 | 3.6 | - | 30 | C (PR) | - | INCO (3) |
| Ni-Cu K-500 | W | 751 | 5,640 | 0.4 | 8 | 63 | C (PR); P (PR) | - | MEL (5) |
| Ni-Cu K-500 | S | 751 | 5,640 | 1.5 | - | 30 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu}$ K-500 | W | 1,064 | 5,300 | 0.9 | - | 30 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} \mathrm{K}-500$ | S | 1,064 | 5,300 | 0.7 | -- | - | CR (PR30) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} \mathrm{K}-500$ | W | 197 | 2,340 | 0.4 | - | - | U | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} \mathrm{K}-500$ | S | 197 | 2,340 | 0.2 | - | - | U | - | INCO (3) |
| Ni-Cu K-500 | W | 402 | 2,370 | 0.6 | 38 | 46 | C; P | - | CEL (4) |

Table 30. Continued.

Table 30. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum Pit Depth (mil) | Crevice <br> Depth <br> (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| $\mathrm{Ni}-\mathrm{Cu} 60$ | S | 751 | 5,640 | 1.3 | - | 62 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 60$ | W | 1,064 | 5,300 | 1.4 | - | 62 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 60$ | S | 1,064 | 5,300 | 0.9 | - | 33 | C; CR | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 60$ | W | 197 | 2,340 | 0.5 | - | 17 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cu} 60$ | S | 197 | 2,340 | 0.1 | - | - | U | - | INCO (3) |

${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure
so that the lower portions of the specimen were embedded in the bottom sediments.
${ }^{b}$ Symbols for types of corrosion:
$\begin{array}{ll}1 & =\text { Incipient } \\ \text { NU } & =\text { Nonuniform } \\ \text { P } & =\text { Pitting } \\ \text { PR } & =\text { Perforation } \\ \text { S } & =\text { Severe } \\ \mathrm{U} & =\text { Uniform } \\ \mathrm{WB} & =\text { Weld bead }\end{array}$
${ }^{d}$ Francis L. LaQue Corrosion Laboratory, INCO, Wrightsville Beach, N.C.
${ }^{e}$ Line corrosion at edge of weld bead.
Table 31. Changes in Mechanical Properties of Nickel-Copper Alloys Due to Corrosion

| Alloy | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original <br> (ksi) | \% Change | $\begin{aligned} & \text { Original } \\ & (\mathrm{ksi}) \end{aligned}$ | \% Change | Original (\%) | \% Change |  |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 123 | 5,640 | 75 | +2 | 29 | +1 | 44 | +1 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 403 | 6,780 | 75 | +2 | 29 | +2 | 44 | +2 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 751 | 5,640 | 75 | -2 | 29 | -2 | 44 | -22 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 1,064 | 5,300 | 75 | +3 | 29 | +5 | 44 | -3 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 197 | 2,340 | 75 | +2 | 29 | +1 | 44 | -1 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 197 | 2,340 | 82 | 0 | 41 | +11 | 40 | -14 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 402 | 2,370 | 82 | -1 | 41 | -2 | 40 | -25 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 402 | 2,370 | 75 | +3 | 29 | +6 | 44 | -1 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 402 | 2,370 | 77 | 0 | 29 | 0 | 44 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$, welded, filler metal 60 | 402 | 2,370 | 77 | -3 | 29 | +17 | 44 | $-34^{b}$ | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$, welded, filler metal $60^{\circ}$ | 402 | 2,370 | 77 | +1 | 29 | +14 | 44 | $-26^{\text {b }}$ | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$, welded, electrode 130 | 402 | 2,370 | 77 | 0 | 29 | +14 | 44 | $-15^{b}$ | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$, welded, electrode 180 | 402 | 2,370 | 77 | 0 | 29 | +16 | 44 | $-25^{\text {b }}$ | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 181 | 5 | 75 | +2 | 29 | $+1$ | 44 | -11 | CEL (4) |
| Ni-Cu K-500 | 402 | 2,370 | 98 | +1 | 43 | +4 | 37 | +1 | CEL (4) |

[^6]Table 32. Chemical Composition of Other Nickel Alloys, Percent by Weight

| Alloy | Ni | C | Mn | Fe | S | Si | Cu | Cr | Ti | Mo | Cb | Other | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | 76.00 | 0.04 | 0.20 | 7.20 | 0.007 | 0.20 | 0.10 | 15.8 | - | - | - | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | 75.26 | 0.06 | 0.18 | 7.25 | 0.008 | 0.27 | 0.38 | 16.0 | - | - | - | - | CEL (4) |
| Ni-Cr-Fe 600 | 76.0 | - | - | 7.0 | - | - | - | 16.0 | - | - | - | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$ | 71.0 | - | - | 9.0 | - | 2.0 | - | 16.0 | - | - | - | - | INCO (3) |
| Ni-Cr-Fe 718 | 52.5 | 0.04 | 0.20 | 18.0 | 0.007 | 0.20 | 0.10 | 19.0 | 0.80 | 3.0 | 5.2 | 0.60 Al | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | 73.41 | 0.08 | 0.55 | 6.90 | 0.003 | 0.36 | 0.09 | 14.50 | 2.40 | - | 0.90 | 0.81 Al | CEL (4) |
| Ni-Cr-Fe X750 | 73.0 | - | - | 7.0 | - | - | - | 15.0 | 2.5 | - | - | - | INCO (3) |
| Ni-Cr-Fe X750 | 73.0 | 0.04 | 0.70 | 6.75 | 0.007 | 0.30 | 0.05 | 15.0 | 2.50 | - | 0.85 | 0.80 Al | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | 32.0 | - | 1.0 | 46.0 | - | - | - | 20.0 | - | - | - | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | 32.0 | 0.04 | 0.75 | 46.0 | 0.007 | 0.35 | 0.30 | 20.5 | - | - | - | - | CEL (4) |
| Ni-Fe-Cr 804 | 43.0 | - | - | 25.0 | - | - | - | 29.0 | - | - | - | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | 41.12 | 0.05 | 0.82 | 30.86 | 0.01 | 0.31 | 1.61 | 21.12 | 1.00 | 2.94 | - | 0.14 Al | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | 41.8 | 0.03 | 0.65 | 30.0 | 0.007 | 0.35 | 1.80 | 21.5 | 0.90 | 3.0 | - | 0.15 Al | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | 42.0 | - | - | 30.0 | - | - | 2.0 | 22.0 | - | 3.0 | - | - | INCO (3) |
| Ni-Fe-Cr 825 Cb | 42.0 | - | - | 30.0 | - | - | 2.0 | 22.0 | - | 3.0 | - | - | INCO (3) |
| Ni-Fe-Cr 901 | 43.0 | - | - | 34.0 | - | - | - | 14.0 | - | - | - | - | INCO (3) |
| Ni-Fe-Cr 902 | 42.0 | 0.02 | 0.40 | 48.5 | 0.008 | 0.50 | 0.05 | 5.4 | 2.40 | - | - | 0.65 Al | CEL (4) |
| Ni -Cr-Mo $103{ }^{\text {c }}$ | 67.0 | 0.02 | - | - | - | - | - | 18.0 | - | 14.0 | 0.5 | - | CEL (4) |
| Ni-Cr-Mo 625 | 61.0 | 0.05 | 0.15 | 3.00 | 0.007 | 0.30 | 0.10 | 22.0 | - | 9.0 | 4.0 | - | CEL (4) |
| Ni-Cr-Mo 625 | 63.0 | - | - | - | - | - | - | 22.0 | - | 9.0 | - | - | INCO (3) |
| Ni-Mo-Cr "C" | 55.68 | 0.05 | 0.52 | 6.32 | 0.009 | 0.62 | - | 15.33 | - | 16.71 | - | $\begin{aligned} & 3.53 \mathrm{~W} \\ & 0.96 \mathrm{Co} \\ & 0.26 \mathrm{~V} \\ & 0.010 \mathrm{P} \end{aligned}$ | CEL (4) |
| Ni-Mo-Cr "C" | 60.0 | - | - | 5.0 | - | - | - | 15.0 | - | 16.0 | - | 4.0 W | INCO (3) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} 3$ | 58.0 | - | - | 3.00 | - | - | - | 19.0 | - | 19.0 | - | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | 46.0 | - | - | 1.0 | - | - | - | 15.0 | - | 3.75 | - | $\begin{aligned} & 28.5 \mathrm{Co} \\ & 3.0 \mathrm{Al} \end{aligned}$ | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Co}$ " 41 " | 55.29 | 0.11 | $<0.01$ | 0.33 | - | 0.07 | - | 19.08 | 3.34 | 9.72 | - | 11.47 Co | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Fe}$ " ${ }^{\text {" }}$ | 60.0 | - | - | 5.0 | - | - | - | - | - | 26.0 | - | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Fe} 2$ | 66.0 | - | - | 2.0 | - | - | - | - | - | 30.0 | - | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr}-\mathrm{Mo}^{b}$ | 35.0 | - | - | - | - | - | - | 19.84 | - | 10.0 | - | 35.0 Co | CEL (4) |
| $\mathrm{Co}-\mathrm{Cr}-\mathrm{Ni}-\mathrm{Fe}-\mathrm{Mo}^{\text {c }}$ | 14.96 | 0.05 | 1.96 | 14.60 | - | 0.74 | - | 19.84 | - | 7.14 | - | $\begin{aligned} & 0.058 \mathrm{Al} \\ & 40.46 \mathrm{Co} \\ & 0.07 \mathrm{Be} \end{aligned}$ | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}-\mathrm{Mo}$ " F " | 46.0 | - | - | 21.0 | - | - | - | 22.0 | - | 7.0 | - | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}-\mathrm{Mo}$ " G " | 45.0 | - | - | 20.0 | - | - | 2.00 | 21.0 | - | 7.0 | - | 2.5 Co 1.0 W | INCO (3) |
| $\mathrm{Ni} \mathrm{Cr}-\mathrm{Fe}-\mathrm{Mo}$ " X " | 60.0 | - | - | 19.0 | - | - | - | 22.0 | - | 9.0 | - | 1.0 W | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 88$ | 71.0 | - | - | 7.0 | - | - | - | 10.0 | - | - | - | $\begin{aligned} & 5.0 \mathrm{Sn} \\ & 3.0 \mathrm{Bi} \end{aligned}$ | INCO (3) |
| $\mathrm{Ni}-\mathrm{Sn}-\mathrm{Zn} 23$ | 79.0 | - | 2.0 | - | - | - | - | - | - | - | - | $\begin{aligned} & 8.0 \mathrm{Sn} \\ & 7.0 \mathrm{Zn} \\ & 4.0 \mathrm{~Pb} \end{aligned}$ | INCO (3) |
| $\mathrm{Ni}-\mathrm{Be}$ | 97.55 | - | - | - | - | - | - | - | - | - | - | 1.95 Be | CEL (4) |

Continued

Table 32. Continued.

| Alloy | Ni | C | Mn | Fe | S | Si | Cu | Cr | Ti | Mo | Cb | Other | Source ${ }^{\text {a }}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Ni-Cr 65-35 | 65.0 | - | - | - | - | - | - | 35.0 | - | - | - | - | INCO (3) |
| Ni -Cr 75 | 78.0 | - | - | - | - | - | - | 20.0 | - | - | - | - | INCO (3) |
| Ni Cr 80-20 | 80.0 | - | - | - | - | - | - | 20.0 | - | - | - | - | INCO (3) |
| Ni-Si "D" | 86.0 | - | - | - | - | 10.0 | 3.0 | - | - | - |  |  | INCO (3) |
| Filler metal 62 | 73.5 | 0.04 | 0.20 | 7.50 | 0.007 | 0.30 | 0.03 | 16.0 | - | - | 2.3 | - | CEL (4) |
| Filler metal 65 | 42.0 | 0.03 | 0.80 | 30.0 | 0.008 | 0.30 | 1.70 | 21.0 | 0.90 | 3.0 | - | - | CEL (4) |
| Filler metal 69 | 72.8 | 0.04 | 0.70 | 6.80 | 0.007 | 0.30 | 0.05 | 15.2 | 2.40 | - | 0.85 | 0.80 Al | CEL (4) |
| Filler metal 82 | 72.0 | 0.02 | 3.00 | 1.00 | 0.007 | 0.20 | 0.02 | 20.0 | 0.5 | - | 2.5 | - | CEL (4) |
| Electrode 132 | 74.0 | 0.05 | 0.75 | 8.50 | 0.005 | 0.20 | 0.10 | 14.0 | - | - | 2.0 | 0.1 Ce | CEL (4) |
| Electrode 135 | 38.0 | 0.05 | 0.50 | 31.0 | 0.008 | 0.40 | 1.80 | 19.0 | - | 5.5 | 1.0 | - | CEL (4) |
| Electrode 138 | 38.0 | 0.16 | 1.90 | 27.0 | 0.008 | 0.40 | 0.40 | 28.0 | - | 3.60 | - | 1.0 W | CEL (4) |
| Electrode 182 | 68.0 | 0.05 | 7.50 | 7.50 | 0.005 | 0.60 | 0.10 | 14.0 | - | - | 2.0 | 0.1 Ce | CEL (4) |
| Filler metal 718 | 52.5 | 0.04 | 0.20 | 18.0 | 0.007 | 0.20 | 0.10 | 19.0 | 0.80 | 3.0 | - | - | CEL (4) |

${ }^{a}$ Numbers refer to references at end of report.
$b_{\text {Wire rope and bolts. }}$
${ }^{c_{\text {Wire rope. }}}$
Table 33. Corrosion Rates and Types of Corrosion of Nickel Alloys

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum Pit Depth (mil) | Crevice Depth (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | W | 123 | 5,640 | <0.1 | - | 4 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | S | 123 | 5,640 | <0.1 | - | 3 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | w | 189 | 5,900 | <0.1 | - | 39 | C | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | S | 189 | 5,900 | 0.0 | - | - | I-C | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | W | 403 | 6,780 | 0.1 | - | 23 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | S | 403 | 6,780 | 0.3 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | W | 751 | 5,640 | <0.1 | - | 33 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | S | 751 | 5,640 | <0.1 | - | 4 | C; P | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | W | 1,064 | 5,300 | 0.1 | - | 35 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | W | 1,064 | 5,300 | 0.5 | - | 51 | C; P | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | S | 1,064 | 5,300 | <0.1 | - | 2 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | W | 197 | 2,340 | 0.2 | -- | 15 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | S | 197 | 2,340 | 0.1 | - | 10 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | W | 402 | 2,370 | 0.0 | - | - | SL-ET; I-P | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | W | 402 | 2,370 | 0.1 | - | 28 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$, welded, filler metal $62$ | W | 402 | 2,370 | 0.4 | - | - | U | $\mathrm{WB}(\mathrm{PR})^{\text {d }}$ | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$, welded, filler metal 82 | W | 402 | 2,370 | 0.3 | - | , - | U | WB (PR); ${ }^{\text {( }}$ (PR) $(\mathrm{HAZ})^{d}$ | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$, welded, electrode 132 | W | 402 | 2,370 | 0.3 | - | - | ET | WB (PR) | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$, welded, electrode 182 | W | 402 | 2,370 | <0.1 | - | - | ET | ET | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | S | 402 | 2,370 | 0.1 | - | - | ET; T (PR125) | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | S | 402 | 2,370 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | W | 181 | 5 | 0.2 | 105 | - | E; $\mathrm{P}^{e}$ | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | W | 181 | 5 | 1.7 | 50 | 50 | C (PR); P (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$, welded, filler metal 62 | W | 181 | 5 | <0.1 | - | - | U | I-P | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$, welded, filler metal 82 | W | 181 | 5 | <0.1 | - | - | U | SL-ET | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$, welded, electrode 182 | W | 181 | 5 | 1.3 | - | - | U | WB (PR) | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | W | 366 | 5 | 0.6 | 50 | 50 | C (PR); P (PR) |  | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | W | 540 | 5 | 0.5 | 67 | - | P | - | CEL (4) |

Table 33. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum Pit Depth (mil) | Crevice Depth (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{b}$ |  |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$, welded, filler metal $62$ | W | 540 | 5 | 0.7 | 88 | - | P | WB (PR125) | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$, welded, filler metal 82 | W | 540 | 5 | 0.6 | 77 | - | P | P | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$, welded, electrode 132 | W | 540 | 5 | 0.9 | 60 | - | P | WB (PR); T | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$, welded, electrode $182$ | W | 540 | 5 | 0.9 | 50 | - | P | WB (PR) | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | w | 123 | 5,640 | 0.3 | - | 4 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | S | 123 | 5,640 | 0.1 | - | 2 | C | - | INCO (3) |
| Ni-Cr-Fe 610, cast | W | 403 | 6,780 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | S | 403 | 6,780 | 0.1 | - | - | 1-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | W | 751 | 5,640 | 0.6 | - | 83 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | S | 751 | 5,640 | 0.3 | - | 5 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | W | 1,064 | 5,300 | 0.9 | - | - | I-P | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | S | 1,064 | 5,300 | <0.1 | - | 13 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | W | 197 | 2,340 | 0.2 | - | 2 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | S | 197 | 2,340 | $<0.1$ | - | 3 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | W | 402 | 2,370 | 0.3 | - | 18 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | S | 402 | 2,370 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | W | 181 | 5 | 1.8 | 180 | 23 | C; P(PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 610$, cast | W | 366 | 5 | 1.3 | 55 | 24 | $\mathrm{C} ; \mathrm{P}$ | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 718$ | W | 189 | 5,900 | 0.0 | - | -- | NC | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 718$ | S | 189 | 5,900 | <0.1 | - | - | NC | - | CEL (4) |
| Ni-Cr-Fe 718, longitudinal butt weld, electrode 718 | W | 189 | 5,900 | <0.1 | - | - | NC | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 718$, longitudinal butt weld, electrode 718 | S | 189 | 5,900 | <0.1 | - | - | NC | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 718,3-\mathrm{in}$. circular weld, electrode 718 | W | 189 | 5,900 | <0.1 | - | - | NC | ET | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 718$, 3 -in. circular weld, electrode 718 | S | 189 | 5,900 | $<0.1$ | - | - | NC | ET | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 718$ | w | 402 | 2,370 | <0.1 | - | - | NC | - | CEL (4) |

Table 33. Continued.

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum Pit Depth (mil) | Crevice Depth (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 718$, welded, electrode 718 | W | 402 | 2,370 | 0.0 | - | - | NC | NC | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 718$ | W | 181 | 5 | <0.1 | - | - | NC | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 718$ | W | 540 | 5 | 0.0 | - | - | NC | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 718$, welded, electrode 718 | W | 540 | 5 | 0.0 | - | - | NC | NC | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | W | 123 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | S | 123 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}$ X750 | W | 403 | 6,780 | 0.2 | - | 35 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | S | 403 | 6,780 | <0.1 | - | 3 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | W | 751 | 5,640 | 0.4 | - | 40 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}$ X750 | S | 751 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | W | 1,064 | 5,300 | 0.1 | 25 | 40 | C (PR); P | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | W | 1,064 | 5,300 | 0.1 | - | 47 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | S | 1,064 | 5,300 | <0.1 | - | 9 | C | - | INCO (3) |
| Ni-Cr-Fe X750 | W | 197 | 2,340 | 0.4 | - | 18 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | S | 197 | 2,340 | <0.1 | - | 7 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | W | 402 | 2,370 | 0.1 | - | 16 | C | - | Shell (9) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | W | 402 | 2,370 | 0.1 | - | 17 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | W | 402 | 2,370 | 0.1 | - | - | T | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}$ X750, welded, filler metal 69 | W | 402 | 2,370 | 0.3 | - | - | ET | ET | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$, welded, electrode 718 | W | 402 | 2,370 | 0.2 | - | - | U | T (HAZ55) ; WB; E (PR) | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | S | 402 | 2,370 | 0.4 | - | 16 | C | - | Shell (9) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X750}$ | S | 402 | 2,370 | <0.1 | - | 4 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | S | 402 | 2,370 | <0.1 | - | - | ET | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X750}$ | W | 181 | 5 | 1.4 | 50 | 50 | C (PR); P (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$ | W | 366 | 5 | 0.9 | 50 | 50 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}$ (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}$ X750 | W | 540 | 5 | 0.3 | 130 | 130 | C (PR); P (PR) | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} \mathrm{X} 750$, welded, filler metal 69 | W | 540 | 5 | 0.5 | 130 | - | P (PR) | CR (WB\&HAZ) | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe}$ X750, welded, electrode 718 | W | 540 | 5 | 0.5 | 130 | 130 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}$ (PR) | CR (PR; HAZ) | CEL (4) |

Table 33. Continued.

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum <br> Pit Depth (mil) | Crevice Depth (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | W | 123 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | S | 123 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | W | 403 | 6,780 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | S | 403 | 6,780 | <0.1 | - | 1 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | W | 751 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| Ni -Fe-Cr 800 | S | 751 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | W | 1,064 | 5,300 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | S | 1,064 | 5,300 | <0.1 | - | 6 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | W | 197 | 2,340 | <0.1 | - | - | NC | - | INCO (3) |
| Ni -Fe-Cr 800 | S | 197 | 2,340 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | W | 402 | 2,370 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | W | 402 | 2,370 | 0.0 | - | - | NC | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$, welded, filler metal 82 | W | 402 | 2,370 | <0.1 | - | - | NC | WB; E (PR) | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$, welded, electrode 138 | W | 402 | 2,370 | <0.1 | - | - | NC | WB (E) ${ }^{\text {d }}$ | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | S | 402 | 2,370 | <0.1 | - | - | ET | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | S | 402 | 2,370 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | W | 181 | 5 | <0.1 | - | - | NC | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | W | 181 | 5 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | W | 366 | 5 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$ | W | 540 | 5 | 0.3 | 128 | - | P (PR) | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$, welded, filler metal 82 | W | 540 | 5 | 0.4 | 128 | - | P (PR) | T (WB\&HAZ) | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 800$, welded, electrode 182 | W | 540 | 5 | 0.7 | 128 | - | P (PR) | WB\&HAZ (PR) | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 804$ | W | 123 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 804$ | S | 123 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 804$ | W | 403 | 6,780 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 804$ | S | 403 | 6,780 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 804$ | W | 751 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 804$ | S | 751 | 5,640 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 804$ | W | 1,064 | 5,300 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 804$ | S | 1,064 | 5,300 | <0.1 | - | - | I-C | - | INCO (3) |

Table 33. Continued.

Table 33. Continued.

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum Pit Depth (mil) | Crevice Depth (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | S | 197 | 2,340 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | W | 402 | 2,370 | <0.1 | - | 15 | C; ET | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | W | 402 | 2,370 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$, welded, filler metal 65 | W | 402 | 2,370 | <0.1 | - | - | NC | NC | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe} \cdot \mathrm{Cr} 825$, welded, electrode 135 | W | 402 | 2,370 | <0.1 | - | - | NC | WB ${ }^{f}$ | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | S | 402 | 2,370 | <0.1 | - | 8 | C | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | S | 402 | 2,370 | <0.1 | - | - | 1-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | W | 181 | 5 | <0.1 | $44^{g}$ | 33 | C; P | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | w | 181 | 5 | $<0.1$ | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$, welded, filler metal 65 | W | 181 | 5 | <0.1 | - | - | NC | NC | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$, welded, electrode 135 | W | 181 | 5 | <0.1 | - | - | NC | NC | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | W | 366 | 5 | <0.1 | - | - | NC | - | CEL (4) INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | W | $386{ }^{\text {b }}$ | 5 | <0.1 | 2 | 57 | C; P | - | MEL (5) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | W | 398 | 5 | $<0.1$ | - | - | NC | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | W | 540 | 5 | $<0.1$ | 43 | 24 | C; P | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$, welded, filler metal 65 | W | 540 | 5 | <0.1 | 4 | -- | P | I-P (WB\&HAZ) | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$, welded, electrode 135 | W | 540 | 5 | <0.1 | 6 | - | 1-C; P | CR (HAZ) | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | W | 588 | 5 | <0.1 | 18 | - | P | ) | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | W | 608 | 5 | $<0.1$ | 54 | - | P | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825 \mathrm{~S}^{i}$ | W | 123 | 5,640 | $<0.1$ | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825 \mathrm{~S}$ | S | 123 | 5,640 | <0.1 | -- | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825 \mathrm{~S}$ | w | 403 | 6,780 | $<0.1$ | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825 \mathrm{~S}$ | S | 403 | 6,780 | <0.1 | - | - | 1-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825 \mathrm{~S}$ | W | 751 | 5,640 | $<0.1$ | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825 \mathrm{~S}$ | S | 751 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni} \cdot \mathrm{Fe}-\mathrm{Cr} 825 \mathrm{~S}$ | W | 1,064 | 5,300 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825 \mathrm{~S}$ | S | 1,064 | 5,300 | $<0.1$ | - | 4 | C | - | INCO (3) |
| $\mathrm{Ni} \cdot \mathrm{Fe}-\mathrm{Cr} 825 \mathrm{~S}$ | W | 197 | 2,340 | $<0.1$ | - | - | I-C; $1-\mathrm{P}$ | - | INCO (3) |

Table 33. Continued.

Table 33. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum Pit Depth (mil) | Crevice <br> Depth <br> (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{b}$ |  |
| Ni-Fe-Cr 902 | W | 402 | 2,370 | 1.4 | - | 35 | C $; 1-\mathrm{P}$ | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 902$ | S | 402 | 2,370 | 1.0 | - | 20 | C; I-P | - | CEL (4) |
| Ni -Fe-Cr 902 | W | 181 | 5 | 2.2 | - | 26 | C | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 902$ | W | 364 | 5 | 2.5 | - | 41 | C | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 902$ | W | 723 | 5 | 1.7 | - | 40 | C | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 902$ | W | 763 | 5 | 1.5 | 73 | 125 | C (PR); P | - | CEL (4) |
| Ni-Cr-Mo 625 | w | 189 | 5,900 | 0.0 | - | - | NC | - | CEL (4) |
| Ni-Cr-Mo 625 | S | 189 | 5,900 | 0.0 | - | - | NC | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo} 625$, longitudinal butt weld, electrode 625 | W | 189 | 5,900 | 0.0 | - | - | NC | NC | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo} 625$, longitudinal butt weld, electrode 625 | S | 189 | 5,900 | 0.0 | - | - | NC | NC | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$ 625, $3-\mathrm{in}$. circular weld, electrode 625 | W | 189 | 5,900 | 0.0 | - | - | NC | NC | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$ 625, 3-in. circular weld, electrode 625 | S | 189 | 5,900 | 0.0 | - | - | NC | NC | CEL (4) |
| Ni-Cr-Mo 625 | W | 402 | 2,370 | <0.1 | - | - | NC | - | CEL (4) |
| Ni-Cr-Mo 625 | W | 402 | 2,370 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}$-Mo 625 | S | 402 | 2,370 | <0.1 | - | - | NC | - | INCO (3) |
| Ni-Cr-Mo 625 | w | 181 | 5 | <0.1 | -- | - | NC | - | INCO (3) |
| Ni-Cr-Mo 625 | W | 366 | 5 | <0.1 | - | - | NC | - | INCO (3) |
| Ni-Cr-Mo 625 | W | 398 | 5 | 0.0 | - | - | NC | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$ 625, welded, electrode 625 | W | 398 | 5 | 0.0 | - | - | NC | NC | CEL (4) |
| Ni-Cr-Mo 625 | w | 540 | 5 | 0.0 | - | - | NC | - | CEL (4) |
| Ni-Cr-Mo 625, welded, filler metal 625 | W | 540 | 5 | 0.0 |  | - | NC | NC | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo} 625$, welded, electrode 625 | W | 540 | 5 | 0.0 | - | - | NC | NC | CEL (4) |
| Ni-Cr-Mo 625 | W | 588 | 5 | 0.0 | - | - | NC | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo} 625$, welded, electrode 625 | W | 588 | 5 | 0.0 | - | - | NC | NC | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | W | 123 | 5,640 | <0.1 | - | - | NC | - | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | w | 123 | 5,640 | $<0.1$ | - | - | NC | - | INCO (3) |

Table 33. Continued.

Table 33. Continued.

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth (ft) | Corrosion |  |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum Pit Depth (mil) | Crevice Depth (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} 3$ | S | 751 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| Ni -Mo-Cr 3 | W | 1,064 | 5,300 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} 3$ | S | 1,064 | 5,300 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} 3$ | W | 197 | 2,340 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} 3$ | S | 197 | 2,340 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} 3$ | W | 402 | 2,370 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} 3$ | S | 402 | 2,370 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} 3$ | W | 181 | 5 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} 3$ | w | 366 | 5 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | W | 123 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | S | 123 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | W | 403 | 6,780 | <0.1 | - | - | NC | - | inco (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | S | 403 | 6,780 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | W | 751 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cg}-\mathrm{Cr} 700$ | S | 751 | 5,640 | <0.1 | - | - | NC | - | InCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | w | 1,064 | 5,300 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | S | 1,064 | 5,300 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | W | 197 | 2,340 | <0.1 | - | - | NC | - | INCO (3) |
| Ni-Co-Cr 700 | S | 197 | 2,340 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | W | 402 | 2,370 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | S | 402 | 2,370 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | W | 181 | 5 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr} 700$ | w | 366 | 5 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Co} 41$ | W | 1,064 | 5,300 | 0.0 | - | - | NC | - | CEL (4) |
| Ni -Mo-Fe B | W | 123 | 5,640 | 2.3 | - | - | U | - | INCO (3) |
| Ni -Mo-Fe B | S | 123 | 5,640 | 2.2 | - | - | U | - | INCO (3) |
| Ni -Mo-Fe B | W | 403 | 6,780 | 4.0 | - | -- | U. | - | INCO (3) |
| Ni -Mo-Fe B | S | 403 | 6,780 | 0.6 | - | - | $\mathrm{U}^{j}$ | - | INCO (3) |
| Ni -Mo-Fe B | W | 751 | 5,640 | 2.9 | - | - | G | - | INCO (3) |
| Ni -Mo-Fe B | S | 751 | 5,640 | 1.8 | - | - | G | - | INCO (3) |
| Ni -Mo-Fe B | W | 1,064 | 5,300 | 1.5 | - | - | G | - | INCO (3) |
| Ni -Mo-Fe B | S | 1,064 | 5,300 | 0.8 | - | - | NU | - | INCO (3) |
| Ni -Mo-Fe B | W | 197 | 2,340 | <0.1 | - | - | NU-ET | - | INCO (3) |

Table 33. Continued.

Table 33. Continued.

Table 33. Continued

Table 33. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum Pit Depth (mil) | Crevice Depth (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | W | 123 | 5,640 | <0.1 | - | - | $\mathrm{I}-\mathrm{C}$ | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | S | 123 | 5,640 | <0.1 | - | - | NC | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | W | 403 | 6,780 | 0.4 | - | 40 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | S | 403 | 6,780 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | W | 751 | 5,640 | 0.5 | - | 40 | C (PR); P | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | S | 751 | 5,640 | <0.1 | - | 2 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | W | 1,064 | 5,300 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | S | 1,064 | 5,300 | <0.1 | - | 12 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | W | 197 | 2,340 | 0.3 | - | 40 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | S | 197 | 2,340 | <0.1 | - | 5 | C | - | INCO (3) |
| Ni -Cr, 75 | W | 402 | 2,370 | 0.4 | - | 40 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | S | 402 | 2,370 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | w | 181 | 5 | 1.5 | 50 | 50 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 75$ | w | 366 | 5 | 1.2 | 50 | 50 | $\mathrm{C}(\mathrm{PR})$; $\mathrm{P}(\mathrm{PR})$ | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 80-20$ | W | 123 | 5,640 | <0.1 | - | 2 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 80-20$ | S | 123 | 5,640 | <0.1 | - | - | 1-C | - | INCO (3) |
| Ni -Cr, 80-20 | W | 403 | 6,780 | 0.2 | - | 11 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 80-20$ | S | 403 | 6,780 | <0.1 | - | - | I-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 80-20$ | W | 751 | 5,640 | 0.4 | - | 32 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 80-20$ | S | 751 | 5,640 | <0.1 | - | - | 1-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 80-20$ | W | 1,064 | 5,300 | <0.1 | - | 5 | C | - | INCO (3) |
| Ni-Cr, 80-20 | S | 1,064 | 5,300 | 0.1 | - | 50 | C (PR) | - | INCO (3) |
| Ni-Cr, 80-20 | W | 197 | 2,340 | 0.2 | - | 32 | C (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 80-20$ | S | 197 | 2,340 | <0.1 | - | - | I-C; E | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 80-20$ | W | 402 | 2,370 | 0.2 | - | 18 | C; P | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 80-20$ | S | 402 | 2,370 | $<0.1$ | - | - | 1-C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 80-20$ | W | 181 | 5 | 2.6 | 50 | 50 | C (PR); P (PR) | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}, 80-20$ | W | 366 | 5 | 1.6 | 30 | 30 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | - | INCO (3) |
| Ni -Si D | W | 123 | 5,640 | 1.3 | - | 12 | C | - | INCO (3) |
| Ni -Si D | S | 123 | 5,640 | 1.4 | - | 10 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Si}$ D | W | 403 | 6,780 | 2.4 | - | 5 | C; P | - | INCO (3) |
| Ni -Si D | S | 403 | 6,780 | 1.4 | - | 14 | C | - | INCO (3) |
| Ni -Si D | W | 751 | 5,640 | 1.7 | - | 29 | C | - | INCO (3) |
| $\mathrm{Ni}-\mathrm{Si}$ D | S | 751 | 5,640 | 1.5 | - | 6 | C; ${ }^{\text {P }}$ | - | INCO (3) |

Table 33. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum Pit Depth (mil) | Crevice Depth (mil) | Type ${ }^{\text {b }}$ | Weld ${ }^{b}$ |  |
|  | W | 1,064 | 5,300 | 1.6 | 38 | 42 | C; P | - | INCO (3) INCO (3) |
| $\mathrm{Ni}-\mathrm{Si}$ Ni -Si D | S | 1,064 | 5,300 | 1.1 | 20 | 25 | C; P | - | INCO (3) |
| Ni -Si D Ni -Si D | W | 197 | 2,340 | 0.2 | - | - | 1-C | - | INCO (3) |
| Ni -Si Ni -Si D | S | 197 | 2,340 | <0.1 | - | - | I-C | - | INCO (3) |
| Ni-Si D Ni Si D | W | 402 | 2,370 | 0.5 | - | 14 | C | - | INCO (3) |
| Ni-Si D | S | 402 | 2,370 | 0.2 | - | - | ET | - | INCO (3) |
| Ni -Si D | W | 181 | 5 | 2.5 | 19 | 25 | C; ${ }^{\text {P }}$ | - | INCO (3) |
| Ni-Si D | w | 366 | 5 | 1.9 | 37 | 33 | C; P | - | INCO (3) |

${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure $; \mathrm{S}=$ Exposed in base of structure
so that the lower portions of the specimen were embedded in the bottom sediments.

$$
\mathrm{NU}=\text { Nonuniform }
$$

[^7]${ }^{c}$ Numbers refer to references at end of report.
${ }^{c}$ Numbers refer to references at end of report.
${ }^{d}$ Line corrosion at edge of weld bead.
${ }^{e}$ Only two deep pits.
$f_{\text {One end corroded. }}$
$g_{\text {Under barnacles. }}$

$\begin{aligned} & \mathrm{C}=\text { Crevice } \\ & \mathrm{CR}=\text { Cratering } \\ & \mathrm{E}=\text { Edge } \\ & \mathrm{ET}=\text { Etched } \\ & \mathrm{G}=\text { General } \\ & \mathrm{HAZ}=\text { Heat affec } \\ & \mathrm{I}=\text { Incipient } \\ & \mathrm{NC}=\text { No visible } \\ & \text { Numbers indicate m }\end{aligned}$
$\begin{array}{ll}\mathrm{E} & =\text { Edge } \\ \mathrm{ET} & =\text { Etched } \\ \mathrm{G} & =\text { General } \\ \text { HAZ } & =\text { Heat affected zone } \\ \text { I } & =\text { Incipient } \\ \text { NC } & =\text { No visible corrosion }\end{array}$
${ }^{b}$ Symbols for types of corrosion:
${ }^{j}$ Groove at mud line.
${ }^{k}$ Only one pit.
${ }^{l}$ Pitting only on ends of bars.

$$
\begin{aligned}
\mathrm{P} & =\text { Pitting } \\
\text { PR } & =\text { Perforation } \\
\mathrm{SL} & =\text { Slight } \\
\mathrm{T} & =\text { Tunneling } \\
\mathrm{U} & =\text { Uniform } \\
\text { WB } & =\text { Weld bead }
\end{aligned}
$$

${ }^{h}$ Francis L. LaQue Corrosion Laboratory, INCO, Wrightsville Beach, N.C.
${ }^{i}$ Sensitized by heating 1 hour at $1,200^{\circ}$ F and then air cooling.
Table 34. Changes in Mechanical Properties of Nickel Alloys Due to Corrosion

| Alloy | $\begin{aligned} & \text { Exposure } \\ & \text { (day) } \end{aligned}$ | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $\begin{gathered} \text { Original } \\ (\mathrm{ksi}) \end{gathered}$ | \% Change | Original (ksi) | \% Change | Original <br> (\%) | \% Change |  |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | 197 | 2,340 | 111 | 0 | 51 | +7 | 51 | -30 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | 403 | 2,370 | 111 | -16 | 51 | -14 | 51 | -15 | NADC (7) |
| Ni-Fe-Cr 825 | 123 | 5,640 | 108 | 0 | 52 | +4 | 38 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | 403 | 6,780 | 108 | 0 | 52 | +3 | 38 | -4 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | 751 | 5,640 | 108 | +2 | 52 | +5 | 38 | - 3 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | 197 | 2,340 | 108 | 0 | 52 | +4 | 38 | -6 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | 402 | 2,370 | 108 | +1 | 52 | +4 | 38 | -3 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | 181 | 5 | 108 | +1 | 52 | -2 | 38 | -7 | CEL (4) |
| Ni-Fe-Cr 902 | 402 | 2,370 | 99 | -2 | 40 | -8 | 43 | -49 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 902$ | 181 | 5 | 99 | -1 | 40 | +6 | 43 | -50 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 123 | 5,640 | 121 | +3 | 60 | +5 | 43 | -1 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 403 | 6,780 | 121 | +5 | 60 | +1 | 43 | +15 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 751 | 5,640 | 121 | +4 | 60 | 0 | 43 | +15 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 1,064 | 5,300 | 121 | +3 | 60 | -7 | 43 | +15 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 197 | 2,340 | 121 | +3 | 60 | -1 | 43 | +18 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 402 | 2,370 | 121 | +3 | 60 | +3 | 43 | +12 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 181 | 5 | 121 | +3 | 60 | -1 | 43 | +15 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{H}^{b}$ | 197 | 2,340 | 134 | -2 | 124 | 0 | 12 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{H}$ | 402 | 2,370 | 134 | -3 | 124 | -2 | 12 | -14 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{HT}^{\text {c }}$ | 197 | 2,340 | 248 | -2 | 200 | 0 | 9 | -33 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{HT}$ | 402 | 2,370 | 248 | -5 | 200 | - | 9 | -60 | NADC (7) |

[^8]Table 35. Stress Corrosion of Nickel Alloys

| Alloy | Stress <br> (ksi) | Percent <br> Yield Strength | Exposure <br> (day) | Depth <br> (ft) | Number of Specimens Exposed | Number <br> Failed | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Ni-200 | 13 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Ni-200 | 19 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 14 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 22 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 12 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 31 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 12 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 31 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 12 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cu} 400$ | 31 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cu} \mathrm{K}-500$ | 120 | - | 402 | 2,370 | 3 | 0 | Shell (9) |
| Ni-Cr-Fe 600 | 15 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | 38 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | 15 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | 38 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | 15 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Fe} 600$ | 38 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| Ni-Cr-Fe X-750 | 120 | - | 402 | 2,370 | 3 | 0 | Shell (9) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 21 | 35 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 30 | 50 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 45 | 75 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 21 | 35 | 403 | 6,780 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 30 | 50 | 403 | 6,780 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 45 | 75 | 403 | 6,780 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 21 | 35 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 30 | 50 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 45 | 75 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 21 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 30 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 45 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 30 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | 45 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | 26 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $\mathrm{Ni}-\mathrm{Fe}-\mathrm{Cr} 825$ | 39 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Ni-Be, 1/2 H | 37 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{H}$ | 93 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{H}$ | 37 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |

Continued

Table 35. Continued.

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure <br> (day) | Depth <br> (ft) | Number of <br> Specimens <br> Exposed | Number <br> Failed | Source $^{a}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{H}$ | 93 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{H}$ | 37 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{H}$ | 93 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{HT}$ | 60 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{HT}$ | 60 | 30 | 197 | $2,340^{b}$ | 3 | 0 | NADC (7) |
| $\mathrm{Ni}-\mathrm{Be}, 1 / 2 \mathrm{HT}$ | 60 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |

${ }^{a}$ Numbers refer to references at end of report.
${ }^{b}$ Severe corrosion at sheared edges indicating possible susceptibility to stress corrosion cracking.

## SECTION 5

## STAINLESS STEELS

The corrosion resistance of stainless steels is attributed to a very thin, stable oxide film on the surface of the alloy which results from the alloying of carbon steels with chromium. Chromium, being a passive metal (corrosion resistant), imparts its passivity to steel when alloyed with it in amounts of $12 \%$ or greater. These iron-chromium alloys are very corrosion-resistant in oxidizing environments because the passive film is maintained in most environments when a sufficient amount of oxidizing agent or oxygen is present to repair any breaks in the protective film.

The corrosion resistance of stainless steels is further enhanced by the addition of nickel to the iron-chromium alloys. This group of alloys is popularly known as the 18-8 ( $18 \%$ chromium - $8 \%$ nickel) stainless steels.

In general, oxidizing conditions favor passivity (corrosion resistance), while reducing conditions destroy it. Chloride ions are particularly agressive in destroying this passivity.

Stainless steels usually corrode by pitting and crevice corrosion in seawater. Pits begin by breakdown of the passive film at weak spots or at nonhomogeneities. The breakdown is followed by the formation of an electrolytic cell, the anode of which is a minute area of active metal and the cathode of which is a considerable area of passive metal. The large potential difference characteristic of this "passive-active" cell accounts for considerable flow of current with attendant rapid corrosion (pitting) at the small anode.

Pitting is most likely to occur in the presence of chloride ions (for example, seawater), combined with such depolarizers as oxygen or oxidizing salts. An oxidizing environment is usually necessary for preservation of passivity with accompanying high corrosion resistance; but, unfortunately, it is also a favorable condition for pitting. The oxidizer can often act as a depolarizer for passive-active cells that were established by breakdown of passivity at a specific point or area. The chloride ion in particular can accomplish this breakdown.

Stainless steels can and do pit in aerated seawater (near neutral chloride solution). Pitting is less pronounced in rapidly moving seawater (aerated solution) as compared with partially aerated, stagnant seawater. The flow of seawater carries away corrosion products which would otherwise accumulate at crevices or cracks. It also insures uniform passivity through free access of dissolved oxygen.

As discussed above, stainless steels generally corrode in seawater by pitting and crevice corrosion; therefore, 'as much as 90 to $95 \%$ of the exposed surface can be uncorroded. With such low percentages of the total exposed area affected, corrosion calculated from loss in weight as mils penetration per year (mpy) can give a very misleading picture. The mpy implies a uniform decrease in thickness, which, for stainless steels, is not the case.

A manifestation of pitting corrosion, whose presence and extent is often overlooked, is tunnel corrosion. Tunnel corrosion is also classified by some as edge, honeycomb, or underfilm corrosion. Tunnel corrosion is insidious because of its nature and because many times it is not apparent from the outside surfaces of the object. It starts as a pit on the surface or on an edge and propagates laterally through the material, many times leaving thin films of uncorroded metal on the exposed surfaces.

Another manifestation of localized attack in stainless steels is oxygen concentration cell corrosion in crevices (usually known as crevice corrosion). This type of corrosion occurs underneath deposits of any kind on the metal surface, underneath barnacles, and at the faying surfaces of a joint. The area of stainless steel that is shielded from the surrounding solution becomes deficient in oxygen, thus creating a difference in oxygen concentration between the shielded and unshielded areas. An electrolytic cell is created, with a difference of potential being generated between the high and low oxygen concentration areas (the low oxygen concentration area becomes the anode of the cell).

Low weight losses and corrosion rates accompany these manifestations of corrosion. Thus, the integrity
of a stainless steel structure can be jeopardized if designed solely on the basis of corrosion rates calculated from weight losses rather than on the basis of measured depths of pits, lengths of tunnel corrosion, and depths of crevice corrosion. Pitting, tunneling, and crevice corrosion, can and do penetrate stainless steel rapidly, thus rendering it useless in short periods of time.

Therefore, corrosion rates expressed as mpy calculated from weight losses, maximum pit depths, maximum lengths of tunnel corrosion, maximum depths of crevice corrosion, and types of corrosion are tabulated to provide an overall picture of the corrosion of the stainless steels.

### 5.1. AISI 200 SERIES STAINLESS STEELS

The chemical compositions of the AISI 200 Series stainless steels are given in Table 36, their corrosion rates and types of corrosion in Table 37, their stress corrosion behavior in Table 38, and the effect of exposure on their mechanical properties in Table 39.

The AISI 200 Series stainless steels are 300 Series stainless steels modified by substituting manganese for about one-half of the nickel. This modification does not adversely affect the corrosion resistance of iron-chromium-nickel alloys in many environments.

### 5.1.1. Duration of Exposure

The AISI 201 and 202 alloys were attacked by crevice and pitting types of corrosion both at the surface and at depths in the seawater. There was a tendency for both alloys to be more severely corroded after longer periods of exposure both at the surface and at depth. The bottom sediments were about as corrosive as the seawater above them.

### 5.1.2. Effect of Depth

The corrosion of AISI 201 was approximately the same at the surface as at depth, while that of AISI 202 was less severe at depth than at the surface. However, it was concluded that depth had no definite influence on the corrosion of the AISI 200 Series stainless steels.

### 5.1.3. Effect of Concentration of Oxygen

The effect of changes in the concentration of oxygen in seawater on the corrosion of both AISI 201 and 202 stainless steels was nonuniform. In general, crevice and pitting corrosion were more rapid and severe at the surface than at depth, but there was no definite correlation between corrosion and oxygen concentration.

As is well known, oxygen can and does play a dual role in the corrosion of stainless steels in electrolytes (for example, seawater). An oxidizing environment (presence of oxygen or other oxidizers) is necessary for maintaining the passivity of stainless steels. However, this same oxidizing environment is necessary to initiate and maintain pitting in stainless steels. Oxygen often acts as the depolarizer for passive-active cells created by the breakdown of passivity at a specific point or area. The chloride ion (present in abundance in seawater) is singularly efficient in accomplishing this breakdown. Therefore, this dual role of oxygen can be used to explain the inconsistent and erratic corrosion behavior of stainless steels in seawater.

### 5.1.4. Stress Corrosion

AISI 201 stainless steel was exposed at the depths and for the times shown in Table 38 when stressed at values equivalent to 30 and $75 \%$ of its yield strength to determine its susceptibility to stress corrosion. AISI 201 stainless steel was not susceptible to stress corrosion under the above conditions of test.

### 5.1.5. Mechanical Properties

The effects of exposure on the mechanical properties of AISI 201 and 202 stainless steels are given in Table 39. The mechanical properties were not adversely affected.

### 5.2. AISI 300 SERIES STAINLESS STEELS

The chemical compositions of the AISI 300 Series stainless steels are given in Table 40, their corrosion rates and types of corrosion in Table 41, their stress
corrosion behavior in Table 42, and the effect of exposure on their mechanical properties in Table 43.

The corrosion of the AISI 300 Series stainless steels was very erratic and unpredictable. They were attacked by crevice, pitting, and tunnel types of corrosion, in varying degrees of severity ranging from incipient to perforation of the thickness of the specimens and tunnels extending laterally for a distance of 11 inches ( 11,000 mils) through the specimen. Comparing the intensities of these types of localized corrosion with the corresponding corrosion rates indicates no definite correlation between them.

Two alloys, AISI 317 and 329, were attacked only by incipient crevice corrosion during exposures at all three depths (surface, 2,500 , and 6,000 feet) in seawater for periods ranging from 366 days at the surface to 1,064 days at the 6,000 -foot depth.

Sensitization (heating for 1 hour at $1,200^{\circ} \mathrm{F}$ and cooling in air) rendered AISI 304 and 316 stainless steels more susceptible to corrosion than their unsensitized counterparts.

### 5.2.1. Duration of Exposure

Examination of the data in Table 41 shows that there is no definite or consistent correlation between severity of corrosion or corrosion rates and duration of exposure. For example, at the 6,000 -foot depth in seawater the intensities of pitting and tunnel corrosion were greater after 403 days than after 1,064 days of exposure, the intensity of crevice corrosion was greater after 1,064 days than after 403 days of exposure, and the maximum corrosion rate was greater after 1,064 days than after 403 days of exposure.

### 5.2.2. Effect of Depth

The data in Table 41 show, in general, that the intensities of crevice, pitting, and tunnel corrosion were either about the same or greater at the surface than at the depth. The corrosion rates are in agreement with this conclusion in that those of most of the alloys were greater at the surface than at depth. Based on these data and the above statements, it is concluded that depth in the ocean exerts no significant influence on the corrosion of AISI 300 Series stainless steels.

### 5.2.3. Effect of Concentration of Oxygen

There was no definite correlation between the intensities of pitting, tunnel, and crevice corrosion of the AISI 300 Series stainless steels and changes in the concentration of oxygen in seawater after 1 year of exposure. On the basis of corrosion rates for those alloys which had definite weight losses, the rates increased with increasing concentration of oxygen, but not uniformly.

These data indicate that the corrosion of the AISI 300 Series stainless steels is not proportional to changes in the concentration of oxygen in seawater. The dual role oxygen plays in the corrosion of stainless steels in seawater, as discussed previously, also applies here as an explanation for the erratic behavior of AIS1 300 Series stainless steels.

### 5.2.4. Stress Corrosion

Some of the AISI 300 Series stainless steels were stressed at values ranging from 30 to $80 \%$ of their respective yield strengths. They were exposed in the seawater at depths of 2,500 and 6,000 feet for various periods of time to determine their susceptibility to stress corrosion cracking. These data are given in Table 42.

None of the steels were susceptible to stress corrosion under the conditions of these tests.

### 5.2.5. Mechanical Properties

The effects of exposure on the mechanical properties of some of the 300 Series stainless steels are given in Table 43.

In only two cases were the mechanical properties adversely affected: (1) After 1,064 days of exposure at the 6,000 -foot depth, the tensile and yield strengths and the elongation of AISI 304L were reduced by about $30 \%$. This is attributed to the perforation of the specimen by both crevice and pitting corrosion, and edge and tunnel corrosion. (2) After 402 days of exposure at the 2,500 -foot depth, the tensile and yield strengths of welded, sensitized AISI 316 were reduced by $45 \%$. These reductions are attributed to the effects of welding.

### 5.3. AISI 400 SERIES STAINLESS STEELS

The chamical compositions of the AISI 400 Series stainless steels are given in Table 44, their corrosion rates and type of corrosion in Table 45, their stress corrosion behavior in Table 46, and the effect of exposure on their mechanical properties in Table 47.

The AISI 400 Series stainless steels are those which nominally contain 11 to $27 \%$ chromium. The 400 Series stainless steels are further divided into ferritic and martensitic steels. The ferritic steels are nonhardenable by heat treatment; those in this category in this program were AISI 405, 430, and 446. The martensitic steels are hardenable by heat treatment, and the one in this category in this program was AISI 410.

The corrosion of the AISI 400 Series stainless steels was erratic and was characterized by the localized types of corrosion (crevice, pitting, and tunnel). The intensities of these types varied from none to complete perforation of the thickness of the specimens for the crevice and pitting types and tunnel corrosion extending laterally the entire 12 -inch ( 12,000 mils) length of specimens. There was no correlation between the intensities of these types of localized corrosion and the corresponding corrosion rates calculated from weight losses. The frequencies and intensities of these types of corrosion were also greater for the AISI 400 Series stainless steels than for the AISI 300 Series stainless steels.

### 5.3.1. Duration of Exposure

The data in Table 45 show no correlation between either intensities of the localized types of corrosion or corrosion rates and duration of exposure. Neither one decreased or increased continuously with increasing duration of exposure.

### 5.3.2. Effect of Depth

Depth had no uniform or gradual effect on the corrosion rates of the AISI 400 Series stainless steels, although the rates were lower at depth than at the surface. However, these corrosion rates did not decrease with increasing depth, i.e., they were lower at the 2,500-foot depth than at the 6,000-foot depth for two of the four steels. The intensities of the
localized types of corrosion were either the same or greater at the surface as at depth.

Depth had no definite influence on the corrosion of the AISI 400 Series stainless steels.

### 5.3.3. Effect of Concentration of Oxygen

In general, the corrosion rates of the AISI 400 Series stainless steels were higher at the highest concentration of oxygen (at the surface) than at the lower concentrations. However, the increases were not proportional to the increase in the oxygen concentration except for AISI 410 after 1 year of exposure. The intensities of the localized types of corrosion were not influenced by changes in the concentration of oxygen in the seawater.

In general, changes in the concentration of oxygen in seawater did not exert a major influence on the corrosion of the AISI 400 Series stainless steels.

### 5.3.4. Stress Corrosion

The AISI 400 Series stainless steels were stressed at values ranging from 30 to $75 \%$ of their respective yield strengths. They were exposed in seawater at the 2,500 - and 6,000 -foot depths for various periods of time to determine their susceptibilities to stress corrosion cracking. These data are given in Table 46.

None of the AISI 400 Series stainless steels were susceptible to stress corrosion under the conditions of these tests.

### 5.3.5. Mechanical Properties

The effects of exposure on the mechanical properties of the AISI 400 Series stainless steels are given in Table 47.

In only two cases were the mechanical properties seriously impaired: (1) After 403 days of exposure at the 6,000 -foot depth, the tensile and yield strengths and the elongation of AISI 405 were seriously reduced. (2) After 751 days of exposure at the 6,000-foot depth, the tensile and yield strengths and the elongation of AISI 430 were completely destroyed.

In all other exposures and for the other steels there was no impairment of the mechanical properties.

### 5.3.6. Corrosion Products

The corrosion products taken from one of the corrosion tunnels in AISI Type 430 stainless steel were analyzed by X-ray diffraction, spectrographic analysis, quantitative chemical analysis, and infra-red spectrophotometry and were found to contain amorphous ferric oxide ( $\mathrm{Fe}_{2} \mathrm{O}_{3} \cdot \mathrm{XH}_{2} \mathrm{O}$ ), $\mathrm{Fe}, \mathrm{Cr}, \mathrm{Mn}$, Si , trace of $\mathrm{Ni}, 1.41 \%$ chloride ion, $2.12 \%$ sulfate ion, and a significant amount of phosphate ion.

### 5.4. PRECIPITATION-HARDENING STAINLESS STEEL

The chemical compositions of the precipitationhardening stainless steels are given in Table 48, their corrosion rates and types of corrosion in Table 49, their stress corrosion behavior in Tables 50 and 51 , and the effect of exposure on their mechanical properties in Table 52.

The precipitation-hardening stainless steels differ from the conventional stainless steels (AISI Series 200,300 , and 400 ) in that they can be hardened to very high strength levels by heating the annealed steels to temperatures in the 900 to $1,200^{\circ} \mathrm{F}$ range and then cooling in air.

The corrosion of the precipitation-hardening stainless steels was erratic and was of the crevice, pitting, and tunnel types of localized corrosion with some edge attack. There was no correlation between the intensities of these types of localized corrosion and the corresponding corrosion rates calculated from weight losses. The frequencies and intensities of these types of corrosion were greater for the precipitationhardening stainless steels than for the AISI 300 Series stainless steels.

The $15-7 \mathrm{AMV}$ steels corroded at extremely rapid rates by crevice, pitting, tunnel, and edge corrosion. In many instances, large portions of the specimens had been lost due to corrosion; in other cases tunnel corrosion had extended laterally through the specimens for distances of 11 or 12 inches within a year of exposure. They were considerably more susceptible to corrosion than were the other precipitationhardening stainless steels.

Alloy $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ was nearly free of corrosion after exposure for 366 days at the surface, 402 days at 2,500 feet, and 1,064 days at 6,000 feet.

There was one case of pitting to a depth of 3 mils after 181 days of exposure at the surface. There was incipient crevice corrosion at the surface and at depth except for pitting to a depth of 3 mils after 1,064 days of exposure in the bottom sediment at the 6,000-foot depth.

### 5.4.1. Duration of Exposure

The data in Table 49 show that there was no correlation between the intensities of the localized types of corrosion and duration of exposure. Also, there was no correlation between corrosion rates calculated from weight losses and duration of exposure except for the 15-7AMV steels for which the corrosion rates increased with increasing time of exposure.

### 5.4.2. Effect of Depth

In general, depth had no effect on the corrosion of the precipitation-hardening stainless steels.

### 5.4.3. Effect of Concentration of Oxygen

Changes in the concentration of oxygen in seawater exerted no definite or major influence on the corrosion behavior of the precipitation-hardening stainless steels.

### 5.4.4. Effect of Welding

A 3 -inch-diameter circular, unrelieved weld was made in the center of the $6 \times 12$-inch specimens of some of the alloys to impose residual stresses in the specimens. Others had a 6-inch-long transverse, unrelieved butt weld across the center of the $6 \times 12$-inch specimens. These welds were to determine whether welding affected the corrosion behavior and stress corrosion susceptibilities of the alloys.

Welding did not affect the corrosion behavior of the precipitation-hardening stainless steels.

The effects of residual stresses imposed by welding will be discussed under 5.4.5.

### 5.4.5. Stress Corrosion

The precipitation-hardening stainless steels were stressed at values ranging from 35 to $85 \%$ of their
respective yield strengths. They were exposed in seawater at the surface, 2,500 -, and 6,000 -foot depths for various periods of time to determine their susceptibilities to stress corrosion. Their data are given in Table 50. A 3-inch-diameter circular, unrelieved weld was made in the center of the $6 \times 12$-inch specimens of some alloys to impose residual stresses in them. Transverse, unrelieved butt welds were made in other specimens for the purpose of simulating stresses induced during construction or fabrication. These residual stresses were multiaxial rather than uniaxial as was the case with the specimens with calculated stresses. In addition, values of these residual stresses were indeterminable. These specimens were exposed in seawater under the same conditions as those above. Their data are given in Table 51.

Alloy AISI $630, \mathrm{H} 925$ with a transverse butt weld did not fail by stress corrosion when stressed to $75 \%$ of its yield strength either at the surface or at depth. However, it did fail due to the unrelieved stresses imposed by the circular weld after 403 days of exposure at the 6,000 -foot depth. The crack propagated across the weld bead.

Specimens of transverse, butt-welded AISI 631 ,TH1050 failed when stressed to $50 \%$ of its yield strength and exposed both at the surface and at depth. Specimens with circular welds also failed when exposed at the surface and at depth. At the surface the cracks extended radially from a point inside the circle to the circular weld bead. At depth the crack extended across and around the outside edge of the weld bead.

Specimens of transverse, butt-welded AISI 631 ,RH1050 failed when stressed to $75 \%$ of its yield strength and exposed at the 2,500 -foot depth. Specimens with circular weld beads also failed when exposed at depth. The cracks originated at the outside edge of the weld beads and propagated circumferentially in both directions either at the edge of the weld bead or in the heat-affected zone.

Specimens of alloy AISI $632, \mathrm{RH} 100$ with a transverse butt weld did not fail by stress corrosion when stressed to $75 \%$ of its yield strength and exposed either at the surface or at depth. However, a specimen with a circular weld failed during 402 days of exposure at the 2,500 -foot depth. The origin of the crack was on the outside edge of the weld bead, and it propagated circumferentially in both directions in the heat-affected zone.

Alloys AISI 634,CRT; AISI 635; ASTM XM16,H950 and H1050; AL362,H950 and H1050; and alloy $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ were not susceptible to stress corrosion under the conditions of these tests.

Alloy PH14-8Mo,SRH950 with a transverse butt weld failed by stress corrosion cracking when stressed to $50 \%$ of its yield strength and exposed at depth.

Specimens of 15-7AMV in the A, RH1150, and RH950 tempers failed by stress corrosion cracking when stressed at 35,50 , and $75 \%$ of their respective yield strengths and exposed at depth. Alloy 15-7AMV,RH1 150 failed when exposed at depth due to the stresses imposed by it being squeezed between insulators such that it was slightly bowed. Alloys of 15-7AMV,RH1150 and RH950 failed by stress corrosion when exposed at depth; the cracks originated at unreamed, drilled holes in the specimens.

### 5.4.6. Mechanical Properties

The effects of exposure on the mechanical properties of the precipitating-hardening stainless steels are given in Table 52. Generally, the mechanical properties of the precipitation-hardening stainless steels were adversely affected by exposure in seawater both at the surface and at depth.

### 5.5. MISCELLANEOUS STAINLESS STEELS

Included in this category are the case and specialty stainless steels which could not be included in the other classifications. Their higher nickel contents and the addition of molybdenum are to increase the range of protection of their passive films and to increase their $r$ istance to pitting corrosion. Because these passive films are so much more resistant to destruction, any corrosion $\because$ localized in the form of crevice and pitting.

The chemical compositions of the miscellaneous stainless steels are given in Table 53, their corrosion rates and types of corrosion in Table 54, their stress corrosion behavior in Table 55, and the effect of exposure on their mechanical properties in Table 56.

These alloys were considerably more resistant to corrosion than the other alloys. There were two cases of crevice corrosion at depth of alloy 20 Cb , with the deepest attack being 102 mils. There were also two
cases each of crevice and pitting attack during surface exposure; 21 mils maximum for crevice corrosion, and 24 mils maximum for pitting corrosion.

Alloy $20 \mathrm{Cb}-3$, a modified version of $20 \mathrm{Cb}(4 \%$ higher nickel content), was more resistant to corrosion by seawater and the bottom sediments than 20 Cb . There was only one case of crevice corrosion ( 40 mils deep) at depth.

The corrosion of two cast versions of 20 Cb , $\mathrm{Ni}-\mathrm{Cr}$-Cu-Mo numbers 1 and 2 , was very similar to that of the 20 Cb . There were isolated cases of crevice corrosion, the maximum depth of attack being 27 mils.

There was only incipient crevice corrosion on cast alloy $\mathrm{Ni}-\mathrm{Cr}$-Mo during exposure at the surface and at depth.

Cast alloy $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$ was not susceptible to corrosion by seawater during exposure either at the surface or at depth.

Cast alloy RL-35-100 was attacked by general and uniform types rather than by the localized types of corrosion. The corrosion rates were rather low, the maximum being 0.7 mil per year after 3 years of exposure at the 6,000 -foot depth.

The corrosion behavior of these alloys was not affected by duration of exposure, depth of exposure, or changes in the concentration of oxygen in seawater.

As shown in Table 55 , alloy 20 Cb was not susceptible to stress corrosion in seawater at depth.

The effects of exposure in seawater on the mechanical properties of alloy 20 Cb are given in Table 56. The mechanical properties were not affected.

Table 36. Chemical Compositions of 200 Series Stainless Steels, Percent by Weight

| Alloy | C | Mn | P | S | Si | Ni | Cr | $\mathrm{Fe}^{a}$ | Source $^{b}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :--- |
| AISI 201 | 0.08 | 6.8 | - | - | - | 4.0 | 17.1 | R | INCO (3) |
| AISI 201 | 0.14 | 7.0 | - | 0.009 | - | 4.5 | 16.5 | R | NADC (7) |
| AISI 202 | 0.09 | 7.6 | - | - | - | 4.5 | 17.8 | R | INCO (3) |
| AISI 202 | 0.13 | 7.9 | - | 0.007 | - | 5.2 | 17.0 | R | NADC (7) |

${ }^{a} \mathrm{R}=$ remainder.
${ }^{b}$ Numbers refer to references at end of report.

Table 37. Corrosion Rates and Types of Corrosion of 200 Series Stainless Steels

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | $\begin{aligned} & \text { Crevice } \\ & \text { Depth } \\ & \text { (mils) } \end{aligned}$ | Type ${ }^{\text {b }}$ |  |
| AISI 201 | W | 123 | 5,640 | $<0.1$ | 0 | I | I-C | INCO (3) |
| AISI 201 | W | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | NADC (7) |
| AISI 201 | S | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| AISI 201 | W | 403 | 6,780 | $<0.1$ | 0 | I | 1-C | INCO (3) |
| AISI $201{ }^{\text {d }}$ | W | 403 | 6,780 | - | - | S | S-C | NADC (7) |
| AISI 201 ${ }^{\text {d }}$ | W | 403 | 6,780 | - | - | - | WB (NC) | NADC (7) |
| AISI 201 | S | 403 | 6,780 | <0.1 | 0 | 2 | C | INCO (3) |
| AISI 201 | W | 751 | 5,640 | $<0.1$ | 0 | 1 | I-C | INCO (3) |
| AISI 201 | W | 751 | 5,640 | - | - | S | S-C | NADC (7) |
| AISI 201 | S | 751 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| AISI 201 | W | 1,064 | 5,300 | $<0.1$ | 0 | I | I-C | INCO (3) |
| AISI 201 | S | 1,064 | 5,300 | 0.5 | 50 | 1 | I-C; P (PR) | INCO (3) |
| AISI 201 | W | 197 | 2,340 | $<0.1$ | 0 | 4 | C | INCO (3) |
| AISI 201 | W | 197 | 2,340 | - | - | S | S-C | NADC (7) |
| AISI 201 | S | 197 | 2,340 | $<0.1$ | 0 | 3 | C | INCO (3) |
| AISI 201 | W | 402 | 2,370 | $<0.1$ | 0 | 1 | C | INCO (3) |
| AISI 201 | W | 402 | 2,370 | - | - | S | S-C; WB (S-C) | NADC (7) |
| AISI 201 | S | 402 | 2,370 | $<0.1$ | 0 | 2 | C | INCO (3) |
| AISI 201 | W | 182 | 5 | $<0.1$ | 0 | I | I-C | INCO (3) |
| AISI 201 | W | 366 | 5 | 0.6 | 0 | 0 | S-E | INCO (3) |
| AISI 202 | W | 123 | 5,640 | <0.1 | 0 | 0 | NC | INCO (3) |
| AISI 202 | W | 123 | 5,640 | $<0.1$ | - | - | NC | NADC (7) |
| AISI 202 | S | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| AISI 202 | W | 403 | 6,780 | $<0.1$ | 0 | I | I-C | INCO (3) |
| AISI 202 | S | 403 | 6,780 | $<0.1$ | 0 | I | I-C | INCO (3) |
| AISI 202 | w | 751 | 5,640 | $<0.1$ | 0 | I | I-C | INCO (3) |
| AISI 202 | S | 751 | 5,640 | <0.1 | 0 | I | I-C | INCO (3) |

Table 37. Continued.

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice $^{\text {Depth }}$ (mils) | Type ${ }^{\text {b }}$ |  |
| AISI 202 | w | 1,064 | 5,300 | $<0.1$ |  | 0 | NC | INCO (3) |
| AISI 202 | S | 1,064 | 5,300 | $<0.1$ | 0 | 50 | C (PR) | INCO (3) |
| AlSI 202 | w | 197 | 2,340 | $<0.1$ | 0 | I | I-C | INCO (3) |
| AISI 202 | S | 197 | 2,340 | $<0.1$ | 0 | I | I-C | INCO (3) |
| AlSI 202 | w | 402 | 2,370 | $<0.1$ | 0 | 17 | C | INCO (3) |
| AlSI 202 | S | 402 | 2,370 | $<0.1$ | 17 | 0 | P | 1NCO (3) |
| AlSI 202 | W | 182 | 5 | 0.6 | 50 | 50 | C (PR) ; P (PR) | INCO (3) |
| AlSI 202 | w | 366 | 5 | 0.5 | 50 | 50 | C (PR); P (PR) | INCO (3) |

${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimen were embedded in the bottom sediments.
${ }^{b}$ Symbols for types of corrosion:

| C | $=$ Crevice |
| :--- | :--- |
| I | $=$ Incipient |
| NC | $=$ No visible corrosion |
| P | $=$ Pitting |
| PR | $=$ Perforated |
| S | $=$ Severe |
| WB | $=$ Weld bead |

${ }^{c}$ Numbers refer to references at end of report.
${ }^{d}$ Welded.

Table 38. Stress Corrosion of 200 Series Stainless Steels

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure <br> (day) | Depth <br> (ft) | Number of <br> Specimens <br> Exposed | Number <br> Failed | Source $^{a}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| AISI 201 | 22 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 201 | 55 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 201, sensitized | 55 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 201 | 22 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 201 | 55 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 201, sensitized | 55 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 201 | 22 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 201 | 55 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 201, sensitized | 22 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 201, sensitized | 55 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |

${ }^{a}$ Numbers refer to references at end of report.
Table 39. Changes in the Mechanical Properties of 200 Series Stainless Steels Due to Corrosion

| Alloy | Environment ${ }^{a}$ | Exposure <br> (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{b}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{gathered} \text { Original } \\ (\mathrm{ksi}) \end{gathered}$ | \% Change | Original <br> (ksi) | \% Change | Original <br> (\%) | \% Change |  |
| AISI 201 | W | 123 | 5,640 | 120 | +3 | 73 | 0 | 48 | +6 | NADC (7) |
| AISI 201 | W | 751 | 5,640 | 120 | +2 | 73 | -10 | 48 | +4 | NADC (7) |
| AISI 201 | W | 197 | 2,340 | 120 | 0 | 73 | -4 | 48 | -8 | NADC (7) |
| AISI 201 | W | 402 | 2,370 | 120 | -7 | 73 | -34 | 48 | +35 | NADC (7) |
| AISI 201, welded | W | 402 | 2,370 | 120 | -6 | 73 | -32 | 48 | +29 | NADC (7) |
| AISI 201, sensitized | W | 402 | 2,370 |  | -8 |  | -34 |  | +33 | NADC (7) |
| AISI 202 | W | ' 123 | 5,640 | 106 | +6 | - | 0 | 55 | 0 | NADC (7) |
| AISI 202 | W | 751 | 5,640 | 106 | -8 | - | -31 | - | - | NADC (7) |

[^9]Table 40. Chemical Compositions of 300 Series Stainless Steels, Percent by Weight

| Alloy | C | Mn | P | S | Si | Ni | Cr | Mo | Other | $\mathrm{Fe}^{a}$ | $\mathrm{Source}^{b}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :--- | :--- |
| AISI 301 | 0.11 | 1.17 | 0.025 | 0.021 | 0.34 | 6.73 | 17.4 | - | - | R | CEL (4) |
| AISI 301 | 0.08 | 1.00 | - | 0.008 | - | 7.50 | 17.50 | - | - | R | NADC (7) |
| AISI 302 | 0.06 | 1.05 | 0.020 | 0.013 | 0.60 | 9.33 | 18.21 | - | - | R | CEL (4) |
| AISI 302 | 0.11 | 1.36 | - | - | - | 9.9 | 17.3 | 0.12 | 0.26 Cu | R | INCO (3) |
| AISI 304 | 0.06 | 1.73 | 0.024 | 0.013 | 0.43 | 10.0 | 18.8 | - | - | R | CEL (4) |
| AISI 304 | 0.05 | 1.73 | 0.020 | 0.012 | 0.52 | 9.55 | 18.2 | 0.18 | - | R | CEL (4) |
| AISI 304 | 0.06 | 1.46 | - | - | - | 9.5 | 18.2 | 0.34 | 0.16 Cu | R | INCO (3) |
| AISI 304 | - | - | - | - | - | 10.0 | 19.0 | - | - | R | MEL (5) |
| AISI 304 | 0.07 | - | - | 0.019 | - | 8.90 | 18.2 | 0.30 | 0.19 Cu | R | NADC (7) |
| AISI 304L | 0.03 | 1.24 | 0.028 | 0.023 | 0.68 | 10.20 | 18.7 | - | - | R | CEL (4) |
| AISI 304L | 0.02 | 1.45 | - | - | - | 9.5 | 17.9 | - | - | R | INCO (3) |
| AISI 304L | 0.03 | 1.80 | - | 0.015 | - | 10.00 | 18.90 | - | 0.2 V | R | NADC (7) |
| AISI 309 | 0.1 | 1.60 | - | - | - | 12.7 | 23.3 | - | - | R | INCO (3) |
| AISI 310 | 0.04 | 1.78 | - | - | - | 20.9 | 25.3 | - | - | R | INCO (3) |
| AISI 311 | 0.2 | 2.0 | - | - | - | 19.0 | 25.0 | - | - | R | INCO (3) |
| AISI 316 | 0.06 | 1.61 | 0.021 | 0.016 | 0.40 | 13.60 | 18.3 | 2.41 | - | R | CEL (4) |
| AISI 316 | 0.05 | 1.73 | - | - | - | 13.2 | 17.2 | 2.60 | - | R | INCO (3) |
| AISI 316 | 0.05 | 1.40 | 0.01 | - | - | 12.90 | 16.40 | 2.15 | - | R | NADC (7) |
| AISI 316L | 0.03 | 1.29 | 0.015 | 0.019 | 0.51 | 13.10 | 17.5 | 2.32 | - | R | CEL (4) |
| AISI 316L | 0.02 | 1.31 | 0.013 | 0.015 | 0.47 | 13.70 | 17.9 | 2.76 | - | R | CEL (4) |
| AISI 316L | 0.02 | 1.78 | - | - | - | 13.6 | 17.7 | 2.15 | - | R | INCO (3) |
| AISI 316L |  |  |  |  |  |  |  |  |  |  | NADC (7) |
| AISI 317 | 0.05 | 1.61 | - | - | - | 13.6 | 18.7 | 3.3 | - | R | INCO (3) |
| AISI 321 | 0.06 | 2.0 | - | - | - | 10.5 | 18.5 | - | - | R | INCO (3) |
| AISI 321 | - | 1.37 | - | - | - | 9.85 | 17.12 | - | - | R | NADC (7) |
| AISI 325 | 0.03 | 0.7 | - | - | - | 23.5 | 9.0 | - | - | R | INCO (3) |
| AISI 329 | 0.07 | 0.46 | - | - | - | 4.4 | 27.0 | 1.40 | - | R | INCO (3) |
| AISI 330 | 0.20 | - | - | - | - | 34.5 | 15.0 | - | - | R | INCO (3) |
| AISI 347 | 0.04 | 1.19 | - | - | - | 11.3 | 18.1 | - | - | R | INCO (3) |
| AISI 347 | - | 1.77 | - | - | - | 10.97 | 18.00 | 0.24 | - | R | NADC (7) |

${ }^{a} \mathrm{R}=$ remainder.
${ }^{b}$ Numbers refer to references at end of report.
${ }^{c}$ No values given.


| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ |  |
| AISI 301 | W | 123 | 5,640 | 0.3 | 0 | SL | 1,200 | SL-C; T | CEL (4) |
| AISI 301 | W | 123 | 5,640 | $<0.1$ | 0 | SL | - | SL-C | NADC (7) |
| AISI 301 | S | 123 | 5,640 | 0.7 | 0 | SL | 750 | SL-C; T | CEL (4) |
| AISI 301 | W | 403 | 6,780 | 1.4 | 103 | 15 | 2,450 | $\mathrm{C} ; \mathrm{T} ; \mathrm{P}$ (PR) | CEL (4) |
| AISI 301 | S | 403 | 6,780 | 1.6 | 103 | 25 | 4,000 | C; T; P (PR) | CEL (4) |
| AISI 301 | W | 751 | 5,640 | 1.7 | 103 | 0 | 6,000 | E; T; P (PR) | CEL (4) |
| AISI 301 | S | 751 | 5,640 | 1.0 | 103 | 0 | 10,500 | $\mathrm{E} ; \mathrm{T} ; \mathrm{P}$ (PR) | CEL (4) |
| AISI 301 | W | 1,064 | 5,300 | 0.4 | 103 | 0 | 11,000 | E; T; P (PR) | CEL (4) |
| AISI 301 | S | 1,064 | 5,300 | 1.0 | 103 | 0 | 6,500 | E; T; P (PR) | CEL (4) |
| AISI 301 | W | 197 | 2,340 | 0.3 | 0 | SL | 1,250 | SL-C; T | CEL (4) |
| AISI 301 | S | 197 | 2,340 | 0.3 | 0 | SL | 1,400 | SL-C; T | CEL (4) |
| AISI 301 | W | 402 | 2,370 | 0.5 | 103 | 0 | 2,500 | T; P (PR) | CEL (4) |
| AISI 301 | S | 402 | 2,370 | 0.6 | 103 | 0 | 2,500 | T; P (PR) | CEL (4) |
| AISI 301 | w | 181 | 5 | 1.9 | 103 | 0 | 2,200 | S-E; T; P (PR) | CEL (4) |
| AISI 301 | w | 398 | 5 | 2.3 | 103 | 0 | 1,150 | T; P (PR) | CEL (4) |
| AISI 301 | w | 588 | 5 | 1.7 | 103 | 50 | 1,500 | C; T; P (PR) | CEL (4) |
| AISI 302 | W | 123 | 5,640 | <0.1 | 0 | SL | 0 | SL-C | CEL (4) |
| AISI 302 | W | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 302 | S | 123 | 5,640 | <0.1 | 0 | SL | 0 | SL-C | CEL (4) |
| AISI 302 | S | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 302 | W | 403 | 6,780 | <0.1 | 0 | 18 | 0 | C | CEL (4) |
| AISI 302 | W | 403 | 6,780 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 302 | S | 403 | 6,780 | <0.1 | 0 | 23 | 0 | C | CEL (4) |
| AISI 302 | S | 403 | 6,780 | $<0.1$ | 0 | I | - | I-C | INCO (3) |
| AISI 302 | W | 751 | 5,640 | <0.1 | 0 | 21 | 0 | C | CEL (4) |
| AISI 302 | W | 751 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 302 | S | 751 | 5,640 | <0.1 | 0 | 1 | 6,000 | 1-C; E; T | CEL (4) |

Table 41. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ |  |
| AISI 302 | S | 751 | 5,640 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 302 | w | 1,064 | 5,300 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 302 | S | 1,064 | 5,300 | <0.1 | 50 | 50 | -- | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AISI 302 | W | 197 | 2,340 | <0.1 | 53 | 53 | 1,375 | $\mathrm{C} ;(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | CEL (4) |
| AISI 302 | w | 197 | 2,340 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 302 | S | 197 | 2,340 | <0.1 | 0 | 20 | 250 | C-T | CEL (4) |
| AISI 302 | S | 197 | 2,340 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 302 | W | 402 | 2,370 | <0.1 | 0 | 18 | 6,000 | C-T | CEL (4) |
| AISI 302 | w | 402 | 2,370 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 302 | S | 402 | 2,370 | <0.1 | 0 | 53 | 0 | C | CEL (4) |
| AISI 302 | S | 402 | 2,370 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 302 | w | 181 | 5 | 0.9 | 53 | 0 | 4,950 | $\mathrm{E} ; \mathrm{P}(\mathrm{PR})$; T | CEL (4) |
| AISI 302 | w | 181 | 5 | <0.1 | 15 | 7 | - | C; P | INCO (3) |
| AISI 302 | W | 366 | 5 | <0.1 | I | I | - | I-C; I-P | INCO (3) |
| AISI 302 | w | 398 | 5 | 0.4 | 53 | 53 | 5,400 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{E} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | CEL (4) |
| AISI 302 | w | 588 | 5 | 0.5 | 53 | 53 | 5,500 | C (PR); P (PR); T | CEL (4) |
| AISI 304 | W | 123 | 5,640 | <0.1 | 0 | 0 | 450 | T | CEL (4) |
| AISI 304 | w | 123 | 5,640 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 304 | w | 123 | 5,640 | 1.1 | 0 | 31 | - | C | MEL (5) |
| AISI 304 | S | 123 | 5,640 | <0.1 | 0 | 0 | 0 | NC | CEL (4) |
| AISI 304 | S | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AIS1 304 | W | 403 | 6,780 | 0.5 | 210 | 0 | 2,000 | E; P (PR) ; T | CEL (4) |
| AISI 304 | w | 403 | 6,780 | <0.1 | 0 | I | - | I-C | InCO (3) |
| AISI 304 | W | 403 | 6,780 | - | - | - | - | S-C; WB (NC) | NADC (7) |
| AIS1 304 | S | 403 | 6,780 | $<0.1$ | 0 | 0 | 0 | SL-F. | CEL (4) |
| AISI 304 | S | 403 | 6,780 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 304 | W | 751 | 5,640 | 0.5 | 210 | 0 | 1,600 | P (PR); T | CEL (4) |

Table 41. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum <br> Pit Depth (mils) | Crevice <br> Depth ${ }^{b}$ <br> (mils) | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ |  |
| AISI 304 | W | 751 | 5,640 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AISI 304 | W | 751 | 5,640 | 1.6 | S | 125 | - | C (PR) ; E; S-P | MEL (5) |
| AISI 304 | S | 751 | 5,640 | $<0.1$ | 0 | 13 | - | C | INCO (3) |
| AISI 304 | W | 1,064 | 5,300 | $<0.1$ | 13 | 74 | 1,750 | C; E; P; T | CEL (4) |
| AISI 304 | W | 1,064 | 5,300 | $<0.1$ | 0 | 1 | - | I-C | INCO (3) |
| AISI 304 | W | 1,064 | 5,300 | 1.0 | S | 125 | - | C (PR) ; S-E; S-P | MEL (5) |
| AISI 304 | W | 1,064 | 5,300 | - | - | - | - | S-C; S-T | NADC (7) |
| AISI 304 | S | 1,064 | 5,300 | $<0.1$ | 210 | 29 | 1,500 | C; E; P (PR) ; T | CEL (4) |
| AISI 304 | S | 1,064 | 5,300 | $<0.1$ | 0 | 50 | - | C (PR) | INCO (3) |
| AISI 304 | W | 197 | 2,340 | 0.3 | 0 | 11 | 950 | C; E; T | CEI. (4) |
| AISI 304 | W | 197 | 2,340 | $<0.1$ | 0 | I | - | I-C | INCO (3) |
| AISI 304 | W | 197 | 2,340 | - | - | - | - | S-C | NADC (7) |
| AISI 304 | S | 197 | 2,340 | $<0.1$ | 0 | SL | 1,000 | SL-C; E; T | CEL (4) |
| AISI 304 | S | 197 | 2,340 | $<0.1$ | 0 | 1 | - | I-C | INCO (3) |
| AISI 304 | W | 402 | 2,370 | 0.4 | 210 | 0 | 2,000 | E; P (PR) ; T | CEL (4) |
| AISI 304 | W | 402 | 2,370 | $<0.1$ | 0 | 13 | - | C | INCO (3) |
| AISI 304 | W | 402 | 2,370 | - | - | - | - | S-C; S-T; WB (NC) | NADC (7) |
| AISI 304 | S | 402 | 2,370 | 0.1 | 0 | 0 | 1,250 | S-E; T | CEL (4) |
| AISI 304 | S | 402 | 2,370 | $<0.1$ | 0 | 4 | - | C | INCO (3) |
| AISI 304 | W | 181 | 5 | 1.8 | 210 | 0 | 500 | E; P (PR) ; T | CEL (4) |
| AISI 304 | W | 181 | 5 | 1.3 | 45 | 20 | - | C; P | INCO (3) |
| AISI 304 | W | 366 | 5 | 0.4 | 34 | 33 | - | C; P | INCO (3) |
| AISI 304 | W | $386{ }^{\text {d }}$ | 5 | 0.6 | 63 | 63 | - | C (PR) ; E; P (PR) | MEL (5) |
| AISI 304 | W | 540 | 5 | 0.7 | 42 | 103 | 180 | $C ; P ; T$ | CEL (4) |
| AISI 304 | W | 588 | 5 | 0.5 | 0 | 138 | 110 | C; T | CEL (4) |
| AISI $304{ }^{e}$ | W | 123 | 5,640 | 0.7 | 0 | 50 |  | C (PR) | INCO (3) |
| AISI $304^{\text {e }}$ | S | 123 | 5,640 | 0.8 | 0 | 37 | - | C | INCO (3) |

Table 41. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum <br> Pit Depth (mils) | Crevice <br> Depth ${ }^{b}$ <br> (mils) | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ |  |
| AISI $304^{\circ}$ | W | 403 | 6,780 | 0.7 | 0 | 50 | - | C (PR) | INCO (3) |
| AISI $304^{e}$ | W | 403 | 6,780 | - | 0 | 5 | - | S-C; S-T; WB (NC) | NADC (7) |
| AISI 304 ${ }^{\prime \prime}$ | S | 403 | 6,780 | 0.4 | 0 | 2 | - | C | INCO (3) |
| AISI 304 ${ }^{e}$ | W | 751 | 5,640 | 0.5 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AISI $304^{e}$ | S | 751 | 5,640 | 0.2 | 0 | 50 | - | $\mathrm{C}(\mathrm{PR})$ | INCO (3) |
| AISI 304 ${ }^{e}$ | W | 1,064 | 5,300 | 0.4 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AIS1 $304{ }^{e}$ | W | 1,064 | 5,300 | - |  | - | - | S-C; S-T | NADC (7) |
| AISI $304^{\text {e }}$ | S | 1,064 | 5,300 | 0.2 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AISI $304{ }^{\text {e }}$ | W | 197 | 2,340 | 0.6 | 0 | 50 | - | $\mathrm{C}(\mathrm{PR})$ | INCO (3) |
| AISI $304^{e}$ | W | 197 | 2,340 | - |  | - | - | S-C | NADC (7) |
| AISI $304^{e}$ | S | 197 | 2,340 | $<0.1$ | 0 | 1 | - | I-C | INCO (3) |
| AISI $304{ }^{e}$ | W | 402 | 2,370 | 0.3 | 0 | 50 | - | C (PR) | INCO (3) |
| AISI $304{ }^{e}$ | W | 402 | 2,370 | - |  | - . | - | S-C; S-T; WB (NC) | NADC (7) |
| AISI $304{ }^{e}$ | S | 402 | 2,370 | $<0.1$ | 0 | 17 | - | C | INCO (3) |
| AISI $304{ }^{e}$ | W | 181 | 5 | 1.1 | 50 | 50 | - | $C(P R) ; P(P R)$ | INCO (3) |
| AISI $304{ }^{e}$ | W | 366 | 5 | 1.2 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR})$; $\mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AISI 304 L | W | 123 | 5,640 | 0.2 | 0 | SL | 200 | SL-C; T | CEL (4) |
| AISI 304L | W | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AISI 304L | S | 123 | 5,640 | <0.1 | 0 | SL | 0 | SL-C; T | CEL (4) |
| AISI 304L | S | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | I INCO (3) |
| AISI 304L | W | 403 | 6,780 | $<0.1$ | 115 | 12 | 4,850 | C; E; P (PR); T | CEL (4) |
| AISI 304L | W | 403 | 6,780 | $<0.1$ | 0 | I | - | I-C | INCO (3) |
| AISI 304L | W | 403 | 6,780 | - | - | - | - | S-C; S-T; WB (NC) | NADC (7) |
| AISI 304L | S | 403 | 6,780 | 0.1 | 0 | 29 | 1,000 | C; E; T | CEL (4) |
| AISI 304L | S | 403 | 6,780 | $<0.1$ | 0 | 2 | - | C | INCO (3) |
| AISI 304L | W | 751 | 5,640 | 0.3 | 115 | 0 | 4,000 | P (PR) ; T | CEL (4) |
| AISI 304L | W | 751 | 5,640 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |

Table 41. Continued.

Table 41. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth $^{b}$ (mils) | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ |  |
| AISI 309 | S | 403 | 6,780 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AlSI 309 | W | 751 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 309 | S | 751 | 5,640 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AIS1 309 | w | 1,064 | 5,300 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 309 | S | 1,064 | 5,300 | $<0.1$ | 0 | 1 | - | I-C | INCO (3) |
| AISI 309 | W | 197 | 2,340 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 309 | S | 197 | 2,340 | $<0.1$ | 0 | 1 | - | I-C | INCO (3) |
| AISI 309 | W | 402 | 2,370 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 309 | S | 402 | 2,370 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 309 | w | 181 | 5 | <0.1 | 0 | 33 | - | C | INCO (3) |
| AISI 309 | w | 366 | 5 | $<0.1$ | 0 | 1 | - | 1-C | INCO (3) |
| AISI 310 | W | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AISI 310 | S | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AISI 310 | w | 403 | 6,780 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 310 | S | 403 | 6,780 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 310 | W | 751 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 310 | S | 751 | 5,640 | $<0.1$ | 0 | I | - | I-C | INCO (3) |
| AISI 310 | W | 1,064 | 5,300 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 310 | S | 1,064 | 5,300 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 310 | W | 197 | 2,340 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 310 | S | 197 | 2,340 | <0.1 | 0 | I | - | - C | INCO (3) |
| AISI 310 | w | 402 | 2,370 | <0.1 | 0 | 14 | - | C | INCO (3) |
| AISI 310 | S | 402 | 2,370 | <0.1 | 0 | 2 | - | C | INCO (3) |
| AISI 310 | w | 181 | 5 | $<0.1$ | 0 | 1 | - | I-C | INCO (3) |
| AISI 310 | w | 366 | 5 | $<0.1$ | 0 | 50 | - | C (PR) | INCO (3) |
| AlSI 311 | W | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |

Table 41. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ |  |
| AISI 311 | S | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 311 | W | 403 | 6,780 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 311 | S | 403 | 6,780 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 311 | W | 751 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 311 | S | 751 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 311 | W | 1,064 | 5,300 | $<0.1$ | 0 | 1 | - | I-C | INCO (3) |
| AISI 311 | S | 1,064 | 5,300 | <0.1 | 0 | 5 | - | C | INCO (3) |
| AISI 311 | W | 197 | 2,340 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 311 | S | 197 | 2,340 | $<0.1$ | 0 | 2 | - | C | INCO (3) |
| AISI 311 | W | 402 | 2,370 | <0.1 | 0 | 6 | - | C | INCO (3) |
| AISI 311 | S | 402 | 2,370 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 311 | W | 181 | 5 | 0.5 | 28 | 50 | - | C (PR); P | INCO (3) |
| AISI 311 | w | 366 | 5 | $<0.1$ | 1 | 1 | - | I-C; I-P | INCO (3) |
| AISI 316 | W | 123 | 5,640 | <0.1 | 0 | 0 | 10 | SL-E; SL-T | CEL (4) |
| AISI 316 | W | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 316 | S | 123 | 5,640 | <0.1 | 0 | 0 | 0 | SL-E | CEL (4) |
| AISI 316 | S | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 316 | W | 403 | 6,780 | <0.1 | 0 | 0 | 0 | NC | CEL (4) |
| AISI 316 | w | 403 | 6,780 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 316 | W | 403 | 6,780 | - | - | - | - | C; WB ( NC ) | NADC (7) |
| AISI 316 | S | 403 | 6,780 | <0.1 | 0 | 5 | 0 | C | CEL (4) |
| AISI 316 | S | 403 | 6,780 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 316 | W | 751 | 5,640 | <0.1 | 0 | 0 | 0 | NC | CEL (4) |
| AISI 316 | W | 751 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 316 | S | 751 | 5,640 | <0.1 | 0 | 0 | 0 | NC | CEL (4) |
| AISI 316 | S | 751 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 316 | W | 1,064 | 5,300 | 0.2 | 21 | 0 | 0 | P | CEL (4) |

Table 41. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ |  |
| AISI 316 | W | 1,064 | 5,300 | <0.1 | 0 | 1 | - | C | INCO (3) |
| AISI 316 | S | 1,064 | 5,300 | 0.2 | 0 | I | 0 | 1-C | CEL (4) |
| AISI 316 | S | 1,064 | 5,300 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 316 | w | 197 | 2,340 | <0.1 | 0 | 0 | 0 | NC | CEL (4) |
| AISI 316 | w | 197 | 2,340 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 316 | w | 197 | 2,340 | 0.1 | 0 | SL | 0 | SL-C; SL-R (WB\&HAZ) | NADC (7) |
| AISI 316 | S | 197 | 2,340 | <0.1 | 0 | 0 | 0 | NC | CEL (4) |
| AISI 316 | S | 197 | 2,340 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 316 | W | 402 | 2,370 | 0.1 | 230 | 0 | 500 | E; P (PR); T | CEL (4) |
| AISI 316 | w | 402 | 2,370 | $<0.1$ | 0 | I | - | I-C | INCO (3) |
| AISI 316 | w | 402 | 2,370 | 0.1 | - | - | - | S-C; SC-P; WB (NC) | NADC (7) |
| AISI 316 | S | 402 | 2,370 | <0.1 | 1 | 0 | 0 | I-P | CEL (4) |
| AISI 316 | S | 402 | 2,370 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 316 | W | 181 | 5 | 1.3 | 230 | 0 | 1,250 | E; P (PR) ; T | CEL (4) |
| AISI 316 | W | 181 | 5 | $<0.1$ | 0 | 1 | - | I-C | INCO (3) |
| AISI 316 | W | 366 | 5 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 316 | w | 398 | 5 | 0.4 | 154 | 20 | 1,350 | C; E; P; T | CEL (4) |
| AISI 316 | W | 540 | 5 | 0.3 | 0 | 63 | 70 | C; T | CEL (4) |
| AISI 316 | w | 588 | 5 | 0.2 | 0 | 130 | 1,500 | C; T (PR) | CEL (4) |
| AISI $316^{e}$ | W | 123 | 5,640 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI $316^{e}$ | S | 123 | 5,640 | 0.1 | 0 | 21 | _ | C | INCO (3) |
| AISI $316^{e}$ | W | 403 | 6,780 | <0.1 | 0 | , | - | I-C | InCO (3) |
| AISI $316^{e}$ | W | 403 | 6,780 | - | - | - | _ | MD-C; WB (NC) | NADC (7) |
| AISI $316^{e}$ | S | 403 | 6,780 | $<0.1$ | 0 | 2 | - | C | INCO (3) |
| AISI $316^{e}$ | W | 751 | 5,640 | 1.5 | 50 | 50 | _ | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | InCO (3) |
| AISI $316^{e}$ | S | 751 | 5,640 | 0.3 | 0 | 50 | - | C (PR) | INCO (3) |
| AISI $316^{e}$ | W | 1,064 | 5,300 | $<0.1$ | 0 | 15 | - | C | INCO (3) |

Table 41. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ |  |
| AISI $316^{e}$ | S | 1,064 | 5,300 | <0.1 | 0 | 6 | -- | C | INCO (3) |
| AISI $316^{e}$ | W | 197 | 2,340 | <0.1 | 0 | 8 | - | C | INCO (3) |
| AISI $316^{e}$ | S | 197 | 2,340 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI $316^{e}$ | w | 402 | 2,370 | <0.1 | 0 | 8 | - | C | INCO (3) |
| AISI $316^{e}$ | w | 402 | 2,370 | 0.1 | - | - | - | S-C; SC-P; WB (NC) | NADC (7) |
| AISI $316^{e}$ | S | 402 | 2,370 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI $316^{e}$ | W | 181 | 5 | 0.8 | 7 | 38 | - | C; P | INCO (3) |
| AISI $316^{e}$ | W | 366 | 5 | 0.6 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AISI 316L | W | 123 | 5,640 | <0.1 | 0 | SL | 0 | SL-C | CEL (4) |
| AISI 316L | W | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 316L | S | 123 | 5,640 | <0.1 | 0 | SL | 0 | SL-C | CEL (4) |
| AISI 316L | S | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | - INCO (3) |
| AISI 316L | W | 403 | 6,780 | $<0.1$ | 0 | 0 | 0 | NC | CEL (4) |
| AISI 316L | w | 403 | 6,780 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 316L | W | 403 | 6,780 | <0.1 | - | - | - | C; SL-R; WB (NC) | NADC (7) |
| AISI 316L | S | 403 | 6,780 | <0.1 | 0 | 0 | 0 | NC | CEL (4) |
| AISI 316L | S | 403 | 6,780 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 316L | W | 751 | 5,640 | <0.1 | 0 | 13 | 0 | C | CEL (4) |
| AISI 316L | W | 751 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 316L | S | 751 | 5,640 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AISI 316L | w | 1,064 | 5,300 | <0.1 | 0 | 25 | 0 | C | CEL (4) |
| AISI 316L | W | 1,064 | 5,300 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 316L | S | 1,064 | 5,300 | <0.1 | 0 | 20 | 0 | C | CEL (4) |
| AISI 316L | S | 1,064 | 5,300 | $<0.1$ | 0 | I | - | I-C | INCO (3) |
| AISI 316L | w | 197 | 2,340 | <0.1 | 0 | 18 | 0 | C | CEL (4) |
| AISI 316L | w | 197 | 2,340 | <0.1 | 0 | 8 | - | C | INCO (3) |
| AISI 316L | S | 197 | 2,340 | <0.1 | 0 | 12 | 0 | C | CEL (4) |


| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ |  |
| AISI 316L | S | 197 | 2,340 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 316L | W | 402 | 2,370 | <0.1 | 0 | 0 | 0 | NC | CEL (4) |
| AISI 316L | W | 402 | 2,370 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 316L | w | 402 | 2,370 | - | - | - | - | G; WB (S-L; T) | NADC (7) |
| AISI 316L | S | 402 | 2,370 | <0.1 | 0 | 45 | 0 | C | CEL (4) |
| AISI 316L | S | 402 | 2,370 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 316L | w | 181 | 5 | 0.3 | 125 | 0 | 1,400 | E; P (PR); T | CEL (4) |
| AISI 316L | w | 181 | 5 | <0.1 | 0 | 25 | - | C | INCO (3) |
| AISI 316L | w | 366 | 5 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 316L | w | 398 | 5 | <0.1 | 0 | 0 | 0 | SL-E | CEL (4) |
| AISI 317 | w | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 317 | S | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 317 | w | 403 | 6,780 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 317 | S | 403 | 6,780 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 317 | W | 751 | 5,640 | <0.1 | 0 | 0 | - | NC | InCO (3) |
| AISI 317 | S | 751 | 5,640 | $<0.1$ | 0 | 0 | - | NC | InCO (3) |
| AISI 317 | W | 1,064 | 5,300 | <0.1 | 0 | 1 | - | I-C | InCO (3) |
| AISI 317 | S | 1,064 | 5,300 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 317 | w | 197 | 2,340 | <0.1 | 0 | 1 | - | I-C | InCO (3) |
| AISI 317 | S | 197 | 2,340 | <0.1 | 0 | 1 | - | I-C | InCO (3) |
| AISI 317 | w | 402 | 2,370 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 317 | S | 402 | 2,370 | <0.1 | 0 | 1 | - | I-C | InCO (3) |
| AISI 317 | w | 181 | 5 | <0.1 | 0 | 0 | - | NC | InCO (3) |
| AISI 317 | W | 366 | 5 | $<0.1$ | 0 | 1 | - | I-C | InCO (3) |
| AISI 321 | W | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 321 | W | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | NADC (7) |

Table 41. Continued.

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | $\begin{aligned} & \text { Maximum } \\ & \text { Tunnel Length } \\ & \text { (mils) } \end{aligned}$ | Type ${ }^{\text {b }}$ |  |
| AISI 321 | S | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 321 | W | 403 | 6,780 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 321 | W | 403 | 6,780 | <0.1 | - | - | - | T; HAZ | NADC (7) |
| AISI 321 | S | 403 | 6,780 | $<0.1$ | 0 | 1 | - | I-C | INCO (3) |
| AISI 321 | W | 751 | 5,640 | <0.1 | 0 | 0 | - | NC | 1 NCO (3) |
| AISI 321 | W | 751 | 5,640 | 1.5 | - | - | - | SL-C; SC-R | NADC (7) |
| AISI 321 | S | 751 | 5,640 | $<0.1$ | 0 | I | - | I-C | INCO (3) |
| AISI 321 | W | 1,064 | 5,300 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 321 | W | 1,064 | 5,300 | $<0.1$ | - | - | - | SL-C; SC-R | NADC (7) |
| AISI 321 | S | 1,064 | 5,300 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 321 | W | 197 | 2,340 | $<0.1$ | 0 | I | - | I-C | INCO (3) |
| AlSI 321 | w | 197 | 2,340 | 0.2 | - | - | - | S-C; SL-P; S-T | NADC (7) |
| AISI 321 | S | 197 | 2,340 | $<0.1$ | 0 | 2 | - | C | INCO (3) |
| AISI 321 | w | 402 | 2,370 | 0.2 | 0 | 30 | - | C (PR) | INCO (3) |
| AISI 321 | W | 402 | 2,370 | - | - | - | - | SL-C; SL-R; HAZ (P) | NADC (7) |
| AlSI 321 | S | 402 | 2,370 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 321 | W | 181 | 5 | $<0.1$ | 0 | 2 | - | C | INCO (3) |
| AISI 321 | W | 366 | 5 | $<0.1$ | 22 | 0 | - | P | INCO (3) |
| AISI 325 | W | 123 | 5,640 | 4.2 | 11 | 0 | - | P | INCO (3) |
| AISI 325 | S | 123 | 5,640 | 3.3 | 13 | 0 | - | P | INCO (3) |
| AlSI 325 | W | 403 | 6,780 | 4.6 | 14 | 0 | - | P | INCO (3) |
| AISI 325 | S | 403 | 6,780 | 1.3 | 0 | 3 | - | C | INCO (3) |
| AISI 325 | W | 751 | 5,640 | 2.1 | 0 | 0 | - | G | INCO (3) |
| AISI 325 | S | 751 | 5,640 | 3.5 | 0 | 0 | - | NU | INCO (3) |
| AISI 325 | W | 1,064 | 5,300 | 1.1 | 0 | 0 | - | G | INCO (3) |
| AISI 325 | S | 1,064 | 5,300 | 1.3 | 0 | 0 | - | G | INCO (3) |
| AISI 325 | W | 197 | 2,340 | 2.8 | 12 | 0 | - | P | INCO (3) |

Table 41. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ |  |
| AISI 325 | S | 197 | 2,340 | <0.1 | 0 | 2 | - | C | INCO (3) |
| AISI 325 | W | 402 | 2,370 | 1.9 | 0 | 0 | - | G | INCO (3) |
| AISI 325 | S | 402 | 2,370 | 0.3 | 0 | 0 | - | G | INCO (3) |
| AISI 325 | W | 181 | 5 | 10.0 | 0 | 0 | - | G | INCO (3) |
| AISI 325 | w | 366 | 5 | 6.3 | 16 | 12 | - | C; P | INCO (3) |
| AISI 329 | W | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AISI 329 | S | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 329 | W | 403 | 6,780 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 329 | S | 403 | 6,780 | <0.1 | 0 | 1 | - | 1-C | INCO (3) |
| AISI 329 | W | 751 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 329 | S | 751 | 5,640 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AISI 329 | W | 1,064 | 5,300 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 329 | S | 1,064 | 5,300 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 329 | W | 197 | 2,340 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 329 | S | 197 | 2,340 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 329 | W | 402 | 2,370 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 329 | S | 402 | 2,370 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 329 | W | 181 | 5 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 329 | W | 366 | 5 | $<0.1$ | 0 | 1 | - | I-C | INCO (3) |
| AISI 330 | w | 123 | 5,640 | $<0.1$ | 0 | I | - | I-C | INCO (3) |
| AISI 330 | S | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 330 | W | 403 | 6,780 | <0.1 | 0 | 1 | - | 1-C | INCO (3) |
| AISI 330 | S | 403 | 6,780 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 330 | w | 751 | 5,640 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 330 | S | 751 | 5,640 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 330 | w | 1,064 | 5,300 | <0.1 | 0 | I | - | I-C | INCO (3) |

Table 41. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | $\begin{aligned} & \text { Maximum } \\ & \text { Tunnel Length } \\ & \text { (mils) } \end{aligned}$ | Type ${ }^{\text {b }}$ |  |
| AISI 330 | S | 1,064 | 5,300 | <0.1 | 0 | 3 | - | C | INCO (3) |
| AISI 330 | W | 197 | 2,340 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 330 | S | 197 | 2,340 | <0.1 | 0 | 1 | - | I-C | INCO (3) |
| AISI 330 | W | 402 | 2,370 | $<0.1$ | 0 | 30 | - | C (PR) | INCO (3) |
| AISI 330 | S | 402 | 2,370 | <0.1 | 0 | 2 | - | C | INCO (3) |
| AISI 330 | W | 181 | 5 | 0.8 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AISI 330 | W | 366 | 5 | 0.4 | 50 | 0 | - | P (PR) | INCO (3) |
| AISI 347 | W | 123 | 5,640 | <0.1 | 0 | 10 | - | C | INCO (3) |
| AISI 347 | W | 123 | 5,640 | <0.1 | - | - | - | NC | NADC (7) |
| AISI 347 | S | 123 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 347 | W | 403 | 6,780 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 347 | W | 403 | 6,780 | - | - | - | - | SL-R; WB (SL-R) | NADC (7) |
| AISI 347 | S | 403 | 6,780 | <0.1 | 0 | 3 | - | C | INCO (3) |
| AISI 347 | W | 751 | 5,640 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 347 | W | 751 | 5,640 | - | - | - | - | SL-R | NADC (7) |
| AISI 347 | S | 751 | 5,640 | <0.1 | 0 | 29 | - | C | INCO (3) |
| AISI 347 | W | 1,064 | 5,300 | <0.1 | 1 | I | - | I-C; I-P | INCO (3) |
| AISI 347 | S | 1,064 | 5,300 | <0.1 | 0 | 50 | - | C (PR) | INCO (3) |
| AISI 347 | W | 197 | 2,340 | <0.1 | 0 | 0 | - | NC | INCO (3) |
| AISI 347 | W | 197 | 2,340 | - | - | - | - | S-C; SL-R (WB\&HAZ) | NADC (7) |
| AISI 347 | S | 197 | 2,340 | <0.1 | 0 | I | - | I-C | INCO (3) |
| AISI 347 | W | 402 | 2,370 | $<0.1$ | 0 | I | - | I-C | INCO (3) |
| AISI 347 | W | 402 | 2,370 | - | - | - | - | SL-C; SL-R | NADC (7) |
| AISI 347 | S | 402 | 2,370 | $<0.1$ | 0 | 2 | - | C | INCO (3) |
| AISI 347 | W | 181 | 5 | 0.8 | 50 | 50 | - | C (PR); P (PR) | INCO (3) |
| AISI 347 | W | 366 | 5 | 0.7 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |

Table 41. Continued.
${ }^{a} \mathrm{~V}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimens were embedded in the bottom sediments.
${ }^{b}$ Symbols for types of corrosion:
$\begin{aligned} \mathrm{C} & =\text { Crevice } \\ \mathrm{E} & =\text { Edge } \\ \mathrm{HAZ} & =\text { Heat-affe } \\ \mathrm{I} & =\text { Incipient } \\ \mathrm{L} & =\text { Line } \\ \mathrm{MD} & =\text { Medium } \\ \mathrm{NC} & =\text { No visible } \\ \mathrm{NU} & =\text { Nonunifo }\end{aligned}$
Numbers indicate maximum depth or length in mils.
${ }^{c}$ Numbers refer to references at end of report.
${ }^{d}$ Francis L. LaQue Corrosion Laboratory, INCO, Wrightsville Beach, NC
${ }^{e}$ Sensitized, heated 1 hr at $1,200^{\circ} \mathrm{F}$, air cooled.

Table 42. Stress Corrosion of 300 Series Stainless Steels

| Alloy | Stress (ksi) | Percent <br> Yield <br> Strength | Exposure (day) | Depth (ft) | Number of Specimens Exposed | Number Failed | Source ${ }^{a}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| AISI 301 | 24 | 62 | 123 | 5,640 | 3 | 0 | NADC (7) |
| AISI 301 | 32 | 82 | 123 | 5,640 | 3 | 0 | NADC (7) |
| AISI 301 | 62 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 301 | 94 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 302 | 22 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 302 | 33 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 304 | 14 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 304 | 35 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 304, sensitized | 14 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 304, sensitized | 35 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 304 | 14 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 304 | 35 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 304, sensitized | 35 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 304 | 14 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 304, sensitized | 14 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 304 | 16 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 304 | 35 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 304, sensitized | 35 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 304 | 25 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 304L | 14 | 35 | 123 | 5,640 | 3 | 0 | CEL (4) |
| AISI 304L | 20 | 50 | 123 | 5,640 | 3 | 0 | CEL (4) |
| AISI 304L | 30 | 75 | 123 | 5,640 | 3 | 0 | CEL (4) |
| AISI 304L | 13 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 304L, sensitized | 13 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 304L | 20 | 50 | 403 | 6,780 | 3 | 0 | CEL (4) |
| AISI 304L | 32 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 304L, sensitized | 32 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 304L | 30 | 75 | 403 | 6,780 | 3 | 0 | CEL (4) |
| AISI 304L | 14 | 35 | 751 | 5,640 | 3 | 0 | CEL (4) |
| AISI 304L | 20 | 50 | 751 | 5,640 | 3 | 0 | CEL (4) |
| AISI 304L | 30 | 75 | 751 | 5,640 | 3 | 0 | CEL (4) |
| AISI 304L | 13 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 304L | 20 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 304L | 32 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 304L, sensitized | 32 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 304L | 30 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 304L | 13 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 304L | 20 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 304L | 32 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 304L, sensitized | 32 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 304L | 30 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |

Continued

Table 42. Continued.

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure (day) | Depth <br> (ft) | Number of Specimens Exposed | Number Failed | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| AISI 316 | 14 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 316, sensitized | 14 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 316 | 35 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 316, sensitized | 35 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 316 | 14 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 316 | 35 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 316, sensitized | 35 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 316 | 14 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 316, sensitized | 14 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 316 | 18 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 316 | 35 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 316, sensitized | 35 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 316 | 27 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 316L | 17 | 35 | 123 | 5,640 | 3 | 0 | CEL (4) |
| AISI 316L | 24 | 50 | 123 | 5,640 | 3 | 0 | CEL (4) |
| AISI 316L | 36 | 75 | 123 | 5,640 | 3 | 0 | CEL (4) |
| AISI 316L | 14 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 316L | 17 | 35 | 403 | 6,780 | 3 | 0 | CFL (4) |
| AISI 316L | 24 | 50 | 403 | 6,780 | 3 | 0 | CEL (4) |
| AISI 316L | 35 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AlSI 316L, sensitized | 35 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 316L | 36 | 75 | 403 | 6,780 | 3 | 0 | CEL (4) |
| AISI 316L | 17 | 35 | 751 | 5,640 | 3 | 0 | CEL (4) |
| AISI 316L | 24 | 50 | 751 | 5,640 | 3 | 0 | CEL (4) |
| AISI 316L | 36 | 75 | 751 | 5,640 | 3 | 0 | CEL (4) |
| AISI 316L | 17 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 316L | 24 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 316L | 36 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 316L | 24 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 316L | 36 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 321 | 10 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 321 | 25 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 321, sensitized | 25 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 321 | 10 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 321 | 25 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 321, sensitized | 25 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 321 | 10 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 321 | 25 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 321, sensitized | 25 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |

Continued

Table 42. Continued

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure <br> (day) | Depth <br> $(\mathrm{ft})$ | Number of <br> Specimens <br> Exposed | Number <br> Failed | Source $^{\boldsymbol{a}}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| AISI 347 | - | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 347 | - | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 347, sensitized | - | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 347 | - | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 347 | - | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 347, sensitized | - | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 347 | - | 30 | 402 | 2,340 | 3 | 0 | NADC (7) |
| AISI 347 | - | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 347, sensitized | - | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |

[^10]|  | $\begin{aligned} & 0 \\ & U \\ & 0 \\ & 0 \\ & 0 \end{aligned}$ |  |
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| $\begin{aligned} & \text { ت} \\ & \text { E } \\ & E \\ & 0 \\ & \frac{1}{5} \\ & E \end{aligned}$ |  |  |
| $\stackrel{\grave{0}}{\stackrel{\rightharpoonup}{2}}$ |  |  <br>  |

Table 43．Continued．

|  | $\begin{gathered} \text { E } \\ \text { U } \\ 0 \\ 0 \\ i \end{gathered}$ |  |
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| $\stackrel{\text { io }}{0}$ |  |  |

Table 43. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| Alsi 316 | W | 181 | 5 | 82 | +2 | 37 | +14 | 55 | -8 | CEL (4) |
| AISI 316 | W | 365 | 5 | 82 | +1 | 37 | +9 | 55 | +1 | CEL (4) |
| Alst 316 L | S | 123 | 5,640 | 86 | -1 | 48 | -2 | 48 | +2 | CEL (4) |
| AISI 316L. | W | 403 | 6,780 | 86 | +1 | 48 | 0 | 48 | -1 | CEL (4) |
| AlSi 316L | S | 403 | 6,780 | 86 | +1 | 48 | +5 | 48 | -1 | CEL (4) |
| AIS1 3161. | W | 751 | 5,640 | 86 | +1 | 48 | +1 | 48 | +3 | CEL (4) |
| ASSI 316L | S | 751 | 5,640 | 86 | 0 | 48 | 0 | 48 | 0 | CEL (4) |
| AIS1 316L | w | 197 | 2,340 | 86 | +1 | 48 | -3 | 48 | +2 | CEL (4) |
| AIS1 316L | S | 197 | 2,340 | 86 | +1 | 48 | -2 | 48 | +1 | CEL (4) |
| AISi 316L, welded | w | +02 | 2,370 | - | -9 | - | -9 |  | - | NADC (7) |
| AISI 316L, sensitized | w | 402 | 2,370 | - | 0 | - | 0 | - | 0 | NADC (7) |
| AISI 316L. | W | 402 | 2,370 | 86 | 0 | 48 | -1 | 48 | -4 | CEL (4) |
| AISI 3161. | S | 402 | 2,370 | 86 | +1 | 48 | +2 | 48 | -4 | CEL (4) |
| AISI 316L | W | 181 | 5 | 86 | 0 | 48 | -2 | 48 | -8 | CEL (4) |
| AISI 316L | W | 365 | 5 | 86 | +2 | 48 | +3 | 48 | +5 | CEL (4) |
| AISI 321 | W | 123 | 5,640 | 83 | -1 | - | 0 | 50 | 0 | NADC (7) |
| AISI 321 | w | 1,064 | 5,300 | 83 | -37 | - | -61 | 50 | +18 | NADC (7) |
| AISI 321 | w | 1,064 | 5,300 | 80 | +13 | 33 | +3 | 53 | +6 | NADC (7) |
| AlSI 321 | w | 197 | 2,340 | 80 | +8 | 33 | +15 | 53 | -9 | NADC (7) |
| AISI $3+7$ | W | 123 | 5,640 | 94 | 0 | - | 0 | 46 | +9 | NADC (7) |
| AISI $3+7$ | w | 751 | 5,640 | 94 | +2 | - | - | 46 | 0 | NADC (7) |

${ }^{4} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimen were embedded in the bottom sediments.
${ }^{b}$ Numbers refer to references at end of report.
${ }^{c}$ Welded and then sensitized.
Table 44. Chemical Compositions of 400 Series Stainless Steels, Percent by Weight

| Alloy | C | Mn | P | S | Si | Ni | Cr | Mo | Al | Other | $\mathrm{Fe}^{\text {a }}$ | Source ${ }^{b}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| AISI 405 | 0.06 | 0.41 | 0.025 | 0.019 | 0.56 | - | 12.6 | - | 0.26 | - | R | CEL (4) |
| AISI 405 | 0.05 | 0.85 | 0.017 | 0.015 | 0.57 | - | 12.46 | - | 0.21 | - | R | CEL (4) |
| AISI 405 | 0.05 | 0.62 | 0.014 | 0.011 | 0.27 | - | 14.5 | - | 0.27 | - | R | CEL (4) |
| AISI 410 | 0.13 | 0.43 | 0.019 | 0.005 | 0.45 | 0.010 | 12.30 | - | - | - | R | CEL (4) |
| AISI 410 | 0.13 | 0.4 | - | - | - | 0.2 | 12.1 | - | - | - | R | INCO (3) |
| AISI 410 | 0.15 | 0.30 | - | - | - | - | 12.10 | 0.03 | - | $<0.09 \mathrm{~V}$ | R | NADC (7) |
| AISI 430 | 0.07 | 0.47 | 0.029 | 0.011 | 0.36 | - | 16.4 | - | - | - | R | CEL (4) |
| AISI 430 | 0.05 | 0.46 | 0.012 | 0.010 | 0.36 | 0.12 | 16.5 | - | - | - | R | CEL (4) |
| AISI 430 | 0.06 | 0.4 | - | - | - | - | 17.7 | -- | - | - | R | INCO (3) |
| AISI 430 | 0.06 | 0.40 | - | 0.013 | - | $<0.25$ | 15.00 | - | - | $<0.20 \mathrm{~V}$ | R | NADC (7) |
| AISI 446 | 0.15 | 0.8 | - | - | - | 0.2 | 30.0 | - | - | - | R | INCO (3) |

${ }^{a} \mathrm{R}=$ remainder.
${ }^{b}$ Numbers refer to references at end of report.
Table 45. Corrosion Rates and Types of Corrosion of 400 Series Stainless Steels

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum Pit Depth (mils) | $\begin{aligned} & \text { Crevice } \\ & \text { Depth } \\ & \text { (mils) } \end{aligned}$ | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ |  |
| AISI 405 | w | 123 | 5,640 | 2.4 | 0 | 0 |  | E; T | CEL (4) |
| AISI 405 | S | 123 | 5,640 | 2.0 | 0 | 0 |  | E; T | CEL (4) |
| Alsi 405 | W | 403 | 6,780 | 3.9 | 0 | 0 | 2,000 | E; T | CEL (4) |
| AlSt 405 | S | 403 | 6,780 | 2.4 | 0 | 0 | 2,500 | E; T | CEL (4) |
| AlSI 405 | W | 751 | 5,640 | 2.8 | 262 | 0 | 500 | E; P (PR); T | CEL (4) |
| AISI 405 | W | 1,064 | 5,300 | 1.5 | 262 | 262 | 12,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | CEL (4) |
| AISI 405 | S | 1,064 | 5,300 | 1.1 | 262 | 262 | 6,250 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | CEL (4) |
| AISI 405 | W | 197 | 2,340 | 2.1 | 0 | 0 | 125 | $\mathrm{E}_{;} \mathrm{T}$ | CEL (4) |
| AISI 405 | S | 197 | 2,340 | 1.4 | 0 | 0 | 380 | E; T | CEL (4) |
| AISI 405 | W | 402 | 2,370 | 1.8 | 40 | 15 | 0 | C; P | CEL (4) |
| AISI 405 | S | 402 | 2,370 | 2.0 | 52 | 11 | 0 | $\mathrm{C}_{\mathrm{i}} \mathrm{P}$ | CEL (4) |
| AISI 405 | W | 588 | 5 | 4.5 | 124 | 250 | 0 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}$ | CEL (4) |
| AISI 410 | W | 123 | 5,640 | 0.2 | 40 | SL | 2,950 | SL-C; P (PR); T | CEL (4) |
| AISI 410 | W | 123 | 5,640 | 3.1 | 50 | 0 | - | P (PR) | INCO (3) |
| AISI 410 | W | 123 | 5,640 | $<0.1$ | - | SL | - | SL-C | N^DC (7) |
| AISI 410 | S | 123 | 5,640 | 0.3 | 40 | SL | 1,600 | SL-C; P (PR); T | CEL (4) |
| AISI 410 | S | 123 | 5,640 | 2.3 | 50 | 0 | - | C (PR) | INCO (3) |
| AISI 410 | W | 403 | 6,780 | 0.2 | 40 | 40 | 6,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | CEL (4) |
| AISI 410 | W | 403 | 6,780 | 1.9 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AISI 410 | W | 403 | 6,780 |  | - | S | EX | S-C; EX-T | NADC (7) |
| AISI 410 | S | 403 | 6,780 | 1.0 | 40 | 40 | 8,850 | C ( PR ) ; P (PR) ; T | CEL (4) |
| AISI 410 | S | 403 | 6,780 | $<0.1$ | 0 | 6 |  | C | 1 NCO (3) |
| AISI 410 | W | 751 | 5,640 | 0.5 | 40 | 40 | 10,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | CEL (4) |
| AISI 410 | W | 751 | 5,640 | 4.4 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AISI 410 | W | 751 | 5,640 | - |  | S | EX | S-C; EX-T | NADC (7) |
| AISI 410 | S | 751 | 5,640 | 0.4 | 40 | 40 | 12,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | CEL (4) |
| AlSI 410 | W | 1,064 | 5,300 | 1.7 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | $1 \mathrm{NCO}(3)$ |
| AISI 410 | S | 1,064 | 5,300 | 0.9 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AISI 410 | W | 197 | 2,340 | 0.3 | 0 | 10 | 7,000 | C; T | CEL (4) |
| AISI 410 | W | 197 | 2,340 | 1.3 | 50 | 0 | $\checkmark$ | C (PR) | INCO (3) |
| AISI 410 | W | 197 | 2,340 | - | - | S | 3.500 | S-C; T | NADC (7) |
| AISI 410 | S | 197 | 2,340 | 0.2 | 0 | 15 | 7,000 | C; E; T | CEL (4) |
| AISI 410 | S | 197 | 2,340 | 0.2 | 0 | 18 | - | C | InCO (3) |
| AISI 410 | W | 402 | 2,370 | 0.5 | 40 | 40 | 6,400 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | CEL (4) |

Table 45．Continued

|  | $\begin{aligned} & \text { Ü } \\ & \text { B } \\ & \text { in } \end{aligned}$ |  |
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Table 45. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth (mils) | $\begin{aligned} & \text { Maximum } \\ & \text { Tunnel Length } \\ & \text { (mils) } \end{aligned}$ | Type ${ }^{\text {b }}$ |  |
| Alst +30 | S | 402 | 2,370 | 0.4 | 137 | 0 | 11,500 | E; P (PR); ${ }^{\text {T }}$ | CEL (4) |
| AISI 430 | S | 402 | 2,370 | 0.2 | 0 | 8 | - | C | INCO (3) |
| Nsist +30 | W | 181 | 5 | 1.7 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AISI 430 | W | 366 | 5 | 1.1 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AlSI 430 | W | 540 | 5 | 0.7 | 50 | 50 | 4,450 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | CEL (4) |
| AISI +30 | W | 588 | 5 | 0.9 | 50 | 50 | 3,900 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | CEL (4) |
| AISI $4+6$ | W | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AISI 446 | S | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AISI 446 | W | 403 | 6,780 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AISI $4+6$ | S | 403 | 6,780 | $<0.1$ | 0 | I | - | 1-C | INCO (3) |
| Alsi 446 | W | 751 | 5,640 | <0.1 | 0 | 0 | - | SL-ET | INCO (3) |
| AlSI 446 | S | 751 | 5,640 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AISI 446 | W | 1,064 | 5,300 | $<0.1$ | 0 | 0 | - | NC | INCO (3) |
| AlSI 4.46 | S | 1,064 | 5,300 | $<0.1$ | 0 | 1 | - | - -C | INCO (3) |
| AISI 446 | w | 197 | 2,340 | $<0.1$ | 0 | 1 | - | I-C | INCO (3) |
| AISI 446 | S | 197 | 2,340 | $<0.1$ | 0 | 2 | - | C | INCO (3) |
| AISI 446 | w | 402 | 2,370 | $<0.1$ | 0 | 1 | - | I-C | INCO (3) |
| AISI 446 | S | 402 | 2,370 | $<0.1$ | 0 | 2 | - | C | INCO (3) |
| AISI 446 | W | 181 | 5 | 0.3 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | INCO (3) |
| AlSI 446 | w | 366 | 5 | 0.6 | 50 | 50 | - | $\mathrm{C}(\mathrm{PR})$; $\mathrm{P}(\mathrm{PR})$ | INCO (3) |

[^11]Table 46. Stress Corrosion of 400 Series Stainless Steels

| Alloy | Stress <br> $(\mathrm{ksi})$ | Percent <br> Yield <br> Strength | Exposure <br> (day) | Depth <br> $(\mathrm{ft})$ | Number of <br> Specimens <br> Exposed | Number <br> Failed | Source $^{a}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :--- |
| AISI 405 | 43 | 75 | 403 | 6,780 | 3 | 0 | CEL (4) |
| AISI 405 | 20 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 405 | 29 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 405 | 43 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 405 | 43 | 75 | 402 | 2,340 | 3 | 0 | CEL (4) |
| AISI 410 | 47 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 410 | 116 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 410 | 47 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 410 | 116 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 410 | 47 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 410 | 24 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 410 | 116 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 410 | 36 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 410 | 120 | - | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 430 | - | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 430 | - | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| AISI 430 | - | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 430 | - | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| AISI 430 | - | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| AISI 430 | 27 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 430 | - | 75 | 402 | 2,370 | 3 | NADC (7) |  |
| AISI 430 | 41 | 75 | 402 | 2,370 | 3 | CEL (4) |  |

[^12]Table 47. Changes in Mechanical Properties of 400 Series Stainless Steels Due to Corrosion

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{b}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| AISI 405 | S | 123 | 5,640 | 77 | +1 | 59 | -2 | 25 | +3 | CEL (4) |
| AISI 405 | w | 403 | 6,780 | 77 | -33 | 59 | -25 | 25 | -61 | CEL (4) |
| AISI 405 | S | 403 | 6,780 | 77 | +2 | 59 | -2 | 25 | -7 | CEL (4) |
| AISI 405 | w | 751 | 5,640 | 77 | 0 | 59 | -3 | 25 | -13 | CEL (4) |
| AISI 405 | W | 197 | 2,340 | 77 | +2 | 59 | -1 | 25 | -1 | CEL (4) |
| AISI 405 | S | 197 | 2,340 | 77 | +2 | 59 | -2 | 25 | -2 | CEL (4) |
| AISI 405 | w | 402 | 2,370 | 77 | -21 | 59 | -9 | 25 | -39 | CEL (4) |
| AISI 405 | S | 402 | 2,370 | 77 | +3 | 59 | -1 | 25 | -3 | CEL (4) |
| AISI 410 | w | 123 | 5,640 | 183 | -52 | 155 | 0 | 15 | 0 | NADC (7) |
| AISI 410 | S | 123 | 5,640 | 80 | 0 | 49 | -3 | 31 | +8 | CEL (4) |
| AISI 410 | W | 403 | 6,780 | 80 | +2 | 49 | +2 | 31 | -8 | CEL (4) |
| AISI 410 | S | 403 | 6,780 | 80 | +2 | 49 | -2 | 31 | -4 | CEL (4) |
| AIS1 410 | W | 751 | 5,640 | 80 | +4 | 49 | -4 | 31 | -8 | CEL (4) |
| AISI 410 | S | 751 | 5,640 | 80 | +2 | 49 | -5 | 31 | -3 | CEL (4) |
| AISI 410 | W | 197 | 2,340 | 201 | -26 | 155 | +8 | 9 | -11 | NADC (7) |
| AISI 410 | W | 197 | 2,340 | 80 | +1 | 49 | -4 | 31 | -3 | CEL (4) |
| AISI 410 | S | 197 | 2,340 | 80 | +1 | 49 | -3 | 31 | -3 | CEL (4) |
| AISI 410 | W | 402 | 2,370 | 201 | 0 | 155 | - | 9 | +3 | NADC (7) |
| AlSI 410 | W | 402 | 2,370 | 80 | +1 | 49 | -2 | 31 | -9 | CEL (4) |
| AISI 410 | S | 402 | 2,370 | 80 | +1 | 49 | -2 | 31 | -7 | CEL (4) |
| AISI 410 | w | 181 | 5 | 80 | +2 | 49 | -9 | 31 | -13 | CEL (4) |
| AISI 430 | w | 123 | 5,640 | 72 | +5 | 54 | -4 | 29 | -6 | CEL ( 4 ) |
| AISI 430 | w | 123 | 5,640 | 73 | -1 | 55 | -2 | 28 | -1 | CEL ( + ) |
| AISI 430 | w | 403 | 6,780 | 72 | +12 | 54 | +6 | 29 | -22 | CEL (4) |
| AISI 430 | S | 403 | 6,780 | 72 | +9 | 54 | 0 | 29 | -6 | CEL (4) |
| AISI 430 | W | 751 | 5,640 | 72 | -75 | 54 | $-100$ | 29 | -100 | CEL (4) |
| AISI 430 | w | 1,064 | 5,300 | 72 | +10 | 54 | +4 | 29 | -16 | CEL (4) |
| AISI 430 | S | 1,064 | 5,300 | 72 | -4 | 54 | 0 | 29 | -42 | CEL (4) |
| AISI 430 | W | 197 | 2,340 | 72 | +4 | 54 | +1 | 29 | -8 | CEL (4) |
| AISI 430 | S | 197 | 2,340 | 72 | +6 | 54 | +3 | 29 | -15 | CEL ( 4 ) |
| AISI 430 | W | 197 | 2,340 | 73 | +3 | 55 | +3 | 28 | -10 | CEL (4) |
| AISI 430 | S | 197 | 2,340 | 73 | +3 | 55 | 0 | 28 | -9 | CEL (4) |
| AISI 430 | w | 402 | 2,370 | 72 | +5 | 54 | +5 | 29 | -18 | CEL (4) |
| AISI 430 | W | 402 | 2,370 | 73 | +3 | 55 | +2 | 28 | -20 | CEL (4) |
| AISI 430 | S | 402 | 2,370 | 73 | +3 | 55 | +1 | 28 | -18 | CEL (4) |

$a_{\mathrm{W}}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimen were embedded in the bottom sediment.
$b_{\text {Numbers refer to references at end of report. }}$.
Table 48. Chemical Compositions of Precipitation-Hardening Stainless Steels, Percent by Weight

| Alloy | C | Mn | P | S | Si | Ni | Cr | Mo | Al | Cu | Other | $\mathrm{Fe}^{a}$ | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| AISI 630 | 0.03 | 0.24 | 0.017 | 0.011 | 0.59 | 4.17 | 15.92 | - | - | 3.23 | 0.24 Cb | R | CEL (4) |
| AISI 630 | - | 1.00 | - | - | - | 4.00 | 17.30 | - | - | 3.00 | $<0.04 \mathrm{~V}$ | R | NADC (7) |
| AISI 631 | 0.07 | 0.48 | 0.017 | 0.018 | 0.42 | 7.42 | 17.12 | - | 1.19 | - | - | R | CEL (4) |
| AISI 631 | - | 0.69 | - | - | - | 7.70 | 17.40 | 0.30 | - | - | - | R | NADC (7) |
| AISI 632 | 0.07 | 0.50 | 0.016 | 0.016 | 0.28 | 7.19 | 15.05 | 2.19 | 1.11 | - | - | R | CEL (4) |
| AISI $632^{\text {c }}$ |  |  |  |  |  |  |  |  |  |  |  |  | NADC (7) |
| AISI 633 | - | - | - | - | - | 4.0 | 17.0 | 3.0 | - | - | - | R | INCO (3) |
| AISI 633 | - | - | - | - | - | 4.0 | 14.0 | 3.0 | - | - | - | R | MEL (5) |
| AISI 634 | 0.12 | 0.77 | 0.021 | 0.009 | 0.34 | 6.42 | 15.35 | 2.73 | - | - | 0.095 N | R | CEL (4) |
| AISI 635 | 0.05 | 0.54 | 0.011 | 0.006 | 0.57 | 6.89 | 16.8 | - | 0.14 | - | 0.64 Ti | R | CEL (4) |
| AISI 635 | 0.05 | 0.56 | 0.026 | 0.009 | 0.74 | 6.80 | 16.8 | - | 0.09 | - | 0.79 Ti | R | CEL (4) |
| AISI XM16 | 0.03 | 0.50 | - | - | 0.50 | 8.50 | 12.00 | - | - | 1.50 | $\begin{aligned} & 1.15 \mathrm{Ti} \\ & 0.50 \mathrm{Cb}+\mathrm{Ta} \end{aligned}$ | R | CEL (4) |
| PH14-8 Mo | 0.04 | 0.36 | 0.004 | 0.002 | 0.34 | 8.12 | 14.71 | 2.25 | 1.21 | - | - | R | CEL (4) |
| 15-7 AMV | 0.27 | 0.67 | 0.020 | 0.012 | 0.84 | 7.4 | 15.0 | 2.42 | 1.55 | - | 0.16 V | R | CEL (4) |
| AL362 | 0.03 | 0.30 | 0.015 | 0.015 | 0.20 | 6.50 | 14.50 | - | - | - | 0.80 Ti | R | CEL (4) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | - | - | - | - | - | 14.0 | 16.0 | 2.0 | - | 3.0 | - | R | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | 0.07 | 14.3 | 0.021 | 0.03 | 0.67 | 0.27 | 18.4 | - | - | - | 0.48 N | R | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | - | 15.0 | - | - | - | 0.5 | 18.0 | - | - | - | 0.5 N | R | INCO (3) |

[^13]${ }^{b}$ Numbers refer to references at end of report.
${ }^{c}$ No values given.
Table 49. Corrosion Rates and Types of Corrosion of Precipitation-Hardening Stainless Steels

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | $\begin{aligned} & \text { Maximum } \\ & \text { Tunnel Length } \\ & \text { (mils) } \end{aligned}$ | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| AISI 630, A | W | 197 | 2,340 | 0.4 | - | - | - | NU | - | Bocing (6) |
| AIS1 630, H925 ${ }^{\text {d }}$ | W | 403 | 6,780 | <0.1 | 0 | 0 | 0 | NC | NC | CEL (4) |
| AISI 630, H925 ${ }_{\text {e }}$ | W | 403 | 6,780 | $<0.1$ | 0 | 0 | 0 | NC | $T$; WB (PR) | CEL (4) |
| AISI 630, H925 ${ }^{d}$ | S | 403 | 6,780 | 0.4 | 0 | 0 | 4,500 | T | T; WB (PR) | CEL (4) |
| AISI 630, H925 ${ }_{\text {d }}$ | S | 403 | 6,780 | $<0.1$ | 0 | 0 | 1,000 | T | T; HAZ (PR) | CEL (4) |
| AISI 630, H925 ${ }^{\text {d }}$ | W | 197 | 2,340 | 0.1 | 0 | 0 | 0 | NC | NC | CEL (4) |
| AISI 630, H925 ${ }^{\text {e }}$ | W | 197 | 2,340 | $<0.1$ | 0 | 0 | 0 | NC | NC | CEL (4) |
| AISI 630, H925 ${ }^{\text {d }}$ | S | 197 | 2,340 | 0.2 | 0 | 0 | 0 | NC | WB (U) | CEL (4) |
| AISI 630, H925 ${ }_{\text {d }}$ | S | 197 | 2,340 | 0.2 | 0 | 0 | 1,200 | T | NC | CEL (4) |
| AISI 630, H925 ${ }^{\text {d }}$ | w | 402 | 2,370 | $<0.1$ | 0 | 0 | 0 | NC | NC | CEL (4) |
| AISI 630, H925 ${ }^{\text {e }}$ d | w | 402 | 2,370 | $<0.1$ | 0 | 0 | 0 | NC | NC | CEL (4) |
| AISI 630, H925 ${ }^{\text {d }}$ | S | 402 | 2,370 | 0.6 | 112 | 0 | 1,750 | $\mathrm{E}_{;} \mathrm{P}(\mathrm{PR})$; T | NC | CEL (4) |
| AISI 630, H925 ${ }^{\text {d }}$ | S | 402 | 2,370 | $<0.1$ | 0 | 0 | 0 | NC | NC | CEL (4) |
| AISI 630, H925 ${ }^{\text {d }}$ | w | 181 | 5 | 3.0 | 112 | 112 | 1,800 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{E}_{\mathrm{i}} \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL. (4) |
| AISI 630, H925 ${ }^{\text {e }}$ | W | 181 | 5 | 3.2 | 112 | 112 | 1,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{E} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL (4) |
| AlSI 630, H925 ${ }^{\text {e }}$ | W | 398 | 5 | 1.4 | 112 | 112 | 0 | C (PR); E; P (PR) | T: WB\&HAZ (PR) | CEL (4) |
| AISI 631, A | W | 123 | 5,640 | - | - | - | - | SC-P | - | NADC (7) |
| AISI 631, TH1050 ${ }^{\text {d }}$ | w | 403 | 6,780 | 0.2 | 0 | 0 | 1.750 | E; T | NC | CEL (4) |
| AISI 631, TH1050 ${ }^{e}{ }_{d}$ | w | 403 | 6,780 | 1.1 | 125 | 0 | 1,000 | E; P (PR) ; T | T; WB (PR) | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {d }}$ | S | 403 | 6,780 | 0.6 | 125 | 0 | 1,500 | $\mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | SCC | CEL (4) |
| AISI 631, TH1050 ${ }^{e}$ | S | 403 | 6,780 | 1.6 | 0 | 125 | 4,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{S}-\mathrm{E} ; \mathrm{T}$ | NC | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {d }}$ | W | 197 | 2,340 | 0.4 | 125 | 125 | 1,250 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$; T | NC | CEL (4) |
| AISI 631, THio50 ${ }^{\text {f }}$ | w | 197 | 2,340 | 0.1 | 125 | 125 | 750 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL (4) |
| AISI 631, A d | w | 197 | 2,340 | - | - | S | S | S-C; S-T | - | NADC (7) |
| AISI 631, TH1050 ${ }^{\text {d }}$ | S | 197 | 2,340 | 0.3 | 125 | 125 | 1,500 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL (4) |
| AISI 631, TH1050 ${ }^{e}$ | S | 197 | 2,340 | $<0.1$ | 125 | 125 | 375 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {d }}$ | W | 402 | 2,370 | 0.4 | 125 | 0 | 3,750 | E; P (PR) ; ${ }^{\text {T }}$ | I-P | CEL (4) |
| AISI 631, TH1050 ${ }_{d}^{e}$ | W | 402 | 2,370 | 0.6 | 125 | 0 | 3,250 | $E ; P(P R) ; T$ | 1.P | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {d }}$ | S | 402 | 2,370 | 0.6 | 125 | 0 | 3,500 | E; P (PR) ; T | I-P | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {d }}$ | S | 402 | 2,370 | 0.5 | 125 | 0 | 4,250 | E; P (PR) ; T | I-P | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {d }}$ | W | 181 | 5 | 4.4 | 125 | 125 | 4,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{E} ; \mathrm{P}(\mathrm{PR})$; T | NC | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {e }}$ | W | 181 | 5 | 2.7 | 125 |  | $\begin{aligned} & 1,000 \\ & 1,000 \end{aligned}$ | $\mathrm{C}(\mathrm{PR}) ; \mathrm{E} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | WB\&HAZ (PR) | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {d }}$ | W | 398 | 5 | 1.9 | 125 | 125 | 2,600 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | SCC | CEL (4) |

Table 49. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth (mils) | $\begin{aligned} & \text { Maximum } \\ & \text { Tunnel Length } \\ & \text { (mils) } \end{aligned}$ | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| AISI 631, RH1050 ${ }^{\text {d }}$ | W | 403 | 6,780 | 0.4 | 125 | 125 | 3,000 | C (PR); P (PR); T | SCC | CEL (4) |
| AISI 631, RH1050 ${ }_{\text {e }}$ | W | 403 | 6,780 | 1.0 | 1 | 125 | 2,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{I}-\mathrm{P} ; \mathrm{T}$ | I-P | CEL (4) |
| AISI 631, RH1050 | S | 403 | 6,780 | 1.3 | 1 | 125 | 6,000 | C (PR) ; S-E; I-P; T; X | $\mathrm{I}-\mathrm{P}$ | CEL (4) |
| AISI 631, RH1050 ${ }_{\text {d }}$ | S | 403 | 6,780 | 0.8 | 125 | 0 | 1,500 | $\mathrm{P}(\mathrm{PR}) ; \mathrm{T} ; \mathrm{X}$ | 1-P | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {d }}$ | W | 197 | 2,340 | 0.3 | 1 | 0 | 1,375 | I-P; T; SCC | SCC | CEL (4) |
| AISI 631, RH1050 ${ }_{d}$ | w | 197 | 2,340 | 0.6 | 125 | 0 | 1,500 | $\mathrm{P}(\mathrm{PR})$; T | T; HAZ (PR) | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {d }}$ | S | 197 | 2,340 | 0.6 | 0 | 125 | 1,000 | C (PR); T | NC | CEL (4) |
| AISI 631, RH1050 ${ }_{\text {d }}$ | S | 197 | 2,340 | 0.3 | 125 | 0 | 1,250 | P (PR); T | NC | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {d }}$ | W | 402 | 2,370 | $<0.1$ | I | 0 | 1,250 | I-P; T | I-P | CEL (4) |
| AISI 631, RH1050 ${ }_{\text {e }}$ | W | 402 | 2,370 | 0.7 | 125 | 0 | 2,500 | P (PR); T | I-P | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {d }}$ | S | 402 | 2,370 | $<0.1$ | 125 | 0 | 3,000 | P (PR); T | SCC | CEL (4) |
| AISI 631, RH1050 ${ }_{\text {e }}$ | S | 402 | 2,370 | 0.5 | 125 | 0 | 5,800 | P (PR); T | T; HAZ (PR) | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {d }}$ | w | 181 | 5 | 2.9 | 125 | 125 | 2,500 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{S}-\mathrm{E} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {e }}$ | W | 181 | 5 | 3.3 | 125 | 125 | 1,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{S}-\mathrm{E} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {d }}$ | W | 403 | 6,780 | 2.1 | 125 | 125 | 2,850 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$; T | NC | CEL (4) |
| AISI 632, RH1100 ${ }^{e}{ }^{\text {d }}$ | W | 403 | 6,780 | 1.5 | 125 | 125 | 2,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {d }}$ | S | 403 | 6,780 | 1.1 | 125 | 125 | 2,100 | $\mathrm{C}(\mathrm{PR}) ; \mathbf{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {e }}$ | S | 403 | 6,780 | 1.2 | 125 | 125 | 5,500 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {d }}$ | W | 197 | 2,340 | 0.6 | 125 | 96 | 750 | $\mathrm{C} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL (4) |
| AISI 632, RH1100 ${ }_{\text {e }}$ | W | 197 | 2,340 | 0.2 | 125 | 0 | 500 | E; $\mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {d }}$ | S | 197 | 2,340 | 0.5 | 125 | 0 | 750 | E; P (PR) ; T | NC | CEL (4) |
| AISI 632, RH1100 ${ }^{e}$ | S | 197 | 2,340 | 0.3 | 125 | 0 | 650 | E; P (PR) ; T | NC | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {d }}$ | w | 402 | 2,370 | 0.9 | 125 | 0 | 1,400 | $\mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | I-P | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {e }}$, | W | 402 | 2,370 | 0.7 | 125 | 0 | 1,000 | P (PR) ; T | NC | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {d }}$ | S | 402 | 2,370 | 0.7 | 1 | 0 | 1,000 | 1-P; T | SCC | CEL (4) |
| AISI 632, RH1100 ${ }^{e}$ | S | 402 | 2,370 | 0.6 | 125 | 0 | 1,000 | P (PR) ; T | 1-P; HAZ (T) | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {d }}$ | W | 181 | 5 | 3.7 | 125 | 0 | 1,150 | S-E; P (PR) ; T | NC | CEL (4) |
| AISI 632, RH1100 ${ }^{e}$ | W | 181 | 5 | 3.7 | 125 | 0 | 1,000 | S-E; P (PR); T | NC | CEL (4) |
| AISI 632, RH1100 ${ }^{e}$ | W | 398 | 5 | 1.8 | 125 | 125 | 750 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | NC | CEL (4) |
| AISI 633 | W | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | - | INCO (3) |
| AISI 633 | W | 123 | 5,640 | 1.4 | 0 | 43 | - | C | - | MEL (5) |
| AISI 633 | S | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | - | INCO (3) |
| AISI 633 | W | 403 | 6,780 | $<0.1$ | 0 | 1 | - | I-C | - | INCO (3) |


| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | $\begin{aligned} & \text { Maximum } \\ & \text { Tunnel Length } \\ & \text { (mils) } \end{aligned}$ | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| AISI 633 | S | 403 | 6,780 | $<0.1$ | 0 | 0 | - | NC | - | INCO (3) |
| AISI 633 | W | 751 | 5,640 | $<0.1$ | 0 | 0 | - | NC | - | INCO (3) |
| AISI 633 | W | 751 | 5,640 | 6.3 | 0 | 63 | - | C (PR) | - | MEL (5) |
| AISI 633 | S | 751 | 5,640 | $<0.1$ | 0 | I | - | 1-C | - | INCO (3) |
| AISI 633 | W | 1,064 | 5,300 | $<0.1$ | 0 | 0 | - | NC | - | INCO (3) |
| AISI 633 | W | 1,064 | 5,300 | 2.8 | 0 | 63 | - | C (PR) | - | MEL (5) |
| AISI 633 | S | 1,064 | 5,300 | $<0.1$ | 0 | 1 | - | C | - | INCO (3) |
| AISI 633 | W | 197 | 2,340 | $<0.1$ | 0 | 1 | - | I-C | -- | INCO (3) |
| AISI 633 | S | 197 | 2,340 | $<0.1$ | 0 | 1 | - | 1-C | - | INCO (3) |
| AISI 633 | W | 402 | 2,370 | $<0.1$ | 0 | I | - | I-C | - | INCO (3) |
| AISI 633 | S | 402 | 2,370 | $<0.1$ | 0 | I | - | I-C | - | INCO (3) |
| AISI 633 | W | 181 | 5 | $<0.1$ | 0 | I | - | I-C | - | INCO (3) |
| AISI 633 | w | 366 | 5 | $<0.1$ | 0 | 1 | - | 1-C | - | INCO (3) |
| AISI 634, CRT | w | 123 | 5,640 | $<0.1$ | 0 | 0 | 0 | NC | - | CEL (4) |
| AISI 634, CRT | w | 403 | 6,780 | 0.2 | 0 | 50 | 0 | C | - | CEL (4) |
| AISI 634, CRT | S | 403 | 6,780 | $<0.1$ | 0 | 30 | 0 | C | - | CEL (4) |
| AISI 634, CRT | w | 751 | 5,640 | 0.0 | 0 | 0 | 0 | NC | - | CEL (4) |
| AISI 634, CRT | W | 197 | 2,340 | 0.2 | I | I | 0 | I-C; I-P | - | CEL (4) |
| AISI 634, CRT | S | 197 | 2,340 | $<0.1$ | 1 | 0 | 0 | I-P | - | CEL (4) |
| AlSI 634, CRT | W | 402 | 2,370 | 0.4 | I | 38 | 0 | $\mathrm{C} ; \mathrm{I}-\mathrm{P}$ | - | CEL (4) |
| AISI 634, CRT | S | 402 | 2,370 | 0.1 | I | 20 | 0 | C; I-P | - | CEL (4) |
| AISI 635 | w | 123 | 5,640 | <0.1 | 0 | SL | 0 | SL-C | - | CEL (4) |
| AISI 635 | S | 123 | 5,640 | $<0.1$ | 0 | 0 | 0 | NC | - | CEL (4) |
| AISI 635 | W | 403 | 6,780 | 0.2 | 0 | 20 | 0 | C | - | CEL (4) |
| AISI 635 | S | 403 | 6,780 | 0.1 | 0 | 11 | 0 | C | $\cdots$ | CEL (4) |
| AISI 635 | W | 751 | 5,640 | $<0.1$ | 0 | 0 | 0 | NC | - | CEL (4) |
| AISI 635 | W | 1,064 | 5,300 | $<0.1$ | 0 | 50 | 0 | C | - | CEL (4) |
| AISI 635 | S | 1,064 | 5,300 | $<0.1$ | 0 | 0 | 0 | NC | - | CEL (4) |
| AISI 635 | w | 197 | 2,340 | $<0.1$ | 0 | 0 | 0 | NC | - | CEL (4) |
| AISI 635 | S | 197 | 2,340 | <0.1 | 0 | 0 | 0 | NC | - | CEL (4) |
| AISI 635 | w | 402 | 2,370 | 0.3 | 0 | 275 | 0 | C (PR) | - | CEL (4) |
| AISI 635 | S | 402 | 2,370 | $<0.1$ | 0 | 0 | 0 | NC | - | CEL (4) |
| AISI 635 | w | 181 | 5 | 0.9 | 40 | 275 | 1,000 | C (PR); E; P; T | - | CEL (4) |

Table 49. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{\text {b }}$ (mils) | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ | Weid ${ }^{\text {b }}$ |  |
| AISI 635 | W | 398 | 5 | 0.6 | 40 | 40 | 1,200 | C; E; P; T | - | CEL (4) |
| ASTM XM16, H950 | W | 189 | 5,900 | - | - | - | - | 1-C; SPP | - | CEL (4) |
| ASTM XM16, H950 ${ }^{\text {d }}$ | W | 189 | 5,900 | - | - | - | - | I-C; SPP | SC-P | CEL (4) |
| ASTM XM16, H1050 | W | 189 | 5,900 | - | - | - | - | I-C; SPP | - | CEL (4) |
| ASTM XM16, H1050 ${ }^{\text {d }}$ | W | 189 | 5,900 | - | - | - | - | I-C; SPP | F-P (WB) | CEL (4) |
| PH148Mo, SRH950 ${ }^{\text {d }}$ | W | 403 | 6,780 | 0.4 | I | 0 | 3,000 | I-P; T | I-P | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {e }}$ | W | 403 | 6,780 | $<0.1$ | 1 | 0 | 0 | I-P | 1-P | CEL (4) |
| PH14-8MO, SRH950 ${ }^{\text {d }}$ | S | 403 | 6,780 | 0.2 | 120 | 0 | 2,000 | P (PR) ; T | HAZ (PR120) | CEL (4) |
| PH14-8Mo, SRH950 ${ }_{\text {e }}$ | S | 403 | 6,780 | $<0.1$ | 1 | 0 | 0 | I-P | I-P | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {d }}$ | W | 197 | 2,340 | 0.3 | 0 | 1 | 500 | I-C; T | NC | CEL (4) |
| PH14-8Mo, SRH950 ${ }_{\text {e }}$ | W | 197 | 2,340 | $<0.1$ | 1 | 0 | 0 | I-P | NC | CEL (4) |
| PH148MO, SRH950 ${ }^{\text {d }}$ | S | 197 | 2,340 | $<0.1$ | 0 | 1 | 0 | I-C | NC | CEL (4) |
| PH14-8Mo, SRH950 ${ }_{d}^{e}$ | S | 197 | 2,340 | <0.1 | 0 | 0 | 0 | NC | NC | CEL (4) |
| PH14.8Mo, SRH950 ${ }^{\text {d }}$ | W | 402 | 2,370 | 0.5 | I | 120 | 2,000 | C (PR) ; I-P; T | 1-P | CEL (4) |
| PH148Mo, SRH950 ${ }^{e}$ | W | 402 | 2,370 | <0.1 | I | 0 | 0 | 1-P | I-P | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {d }}$ | S | 402 | 2,370 | 0.3 | 120 | 0 | 3,000 | P (PR) ; T | I-P | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{e}$ | S | 402 | 2,370 | $<0.1$ | 1 | 0 | 0 | I-P | I-P | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {d }}$ | W | 181 | 5 | 3.1 | 120 | 0 | 2,400 | S-E; P (PR); T | NC | CEL (4) |
| PH148Mo, SRH950 ${ }^{\text {e }}$ | W | 181 | 5 | 1.7 | 120 | 0 | 1,400 | S-E; P (PR); T | WB (PR90) | CEL (4) |
| 15-7 AMV, A | W | 123 | 5,640 | $<0.1$ | 48 | 48 | 750 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
| 15-7 AMV, A | S | 123 | 5,640 | 0.4 | 48 | 48 | 1,800 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$; T | - | CEL (4) |
| 15-7 AMV, A | W | 403 | 6,780 | 1.2 | 48 | 48 | 4,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T} ; \mathrm{X}$ | - | CEL (4) |
| 15-7 AMV, A | S | 403 | 6,780 | 0.8 | 48 | 48 | 2,500 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T} ; \mathrm{X}$ | - | CEL (4) |
| 15-7 AMV, A | W | 751 | 5,640 | 2.7 | 48 | 48 | 6,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T} ; \mathrm{X}$ | - | CEL (4) |
| 15-7 AMV, A | S | 751 | 5,640 | 2.4 | 48 | 48 | 6,000 | C ( PR ) $; \mathrm{P}$ ( PR ) $; \mathrm{T} ; \mathrm{X}$ | - | CEL (4) |
| 15-7 AMV, A | W | 1,064 | 5,300 | 3.0 | 48 | 48 | 0 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{X}$ | - | CEL (4) |
| 15-7 AMV, A | S | 1,064 | 5,300 | 0.9 | 48 | 48 | 0 | $\mathrm{C}(\mathrm{PR})$; P (PR) ; X | - | CEL (4) |
| 15-7 AMV, A | W | 197 | 2,340 | 0.4 | 48 | 48 | 2,000 | C (PR); E; P (PR) ; T | - | CEL (4) |
| 15-7 AMV ${ }^{\text {, }}$ A | S | 197 | 2,340 | 0.3 | 48 | 48 | 4,000 | C (PR); E; P (PR) ; T | - | CEL (4) |
| 15-7 AMV, A | W | 402 | 2,370 | 0.5 | 48 | 48 | 7,300 | C (PR) ; E; P (PR) ; T | - | CEL (4) |
| 15-7 AMV, A | S | 402 | 2,370 | 0.5 | 48 | 48 | 13,800 | C (PR) ; E; P (PR) ; T; X | - | CEL (4) |

Table 49. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth (ft) | Corrosion |  |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum Pit Depth (mils) | Crevice Depth ${ }^{\text {b }}$ (mils) | Maximum Tunnel Length (mils) | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| 15-7 AMV, RH1150 | w | 123 | 5,640 | 0.1 | 45 | 45 | 0 | C (PR) ; P (PR) | - | CEL (4) |
| 15-7 AMV, RH1150 | S | 123 | 5,640 | 0.1 | 45 | 45 | 1,150 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
| 15-7 AMV, RH1150 | W | 403 | 6,780 | 1.2 | 45 | 45 | 5,750 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
| 15-7 AMV, RH1150 | S | 403 | 6,780 | 1.2 | 45 | 45 | 4,500 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
| 15-7 AMV, RH1150 | W | 751 | 5,640 | 1.5 | 45 | 45 | 6,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T} ; \mathrm{X}$ | - | CEL (4) |
| 15-7 AMV, RH1150 | S | 751 | 5,640 | 1.2 | 45 | 45 | 6,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T} ; \mathrm{X}$ | - | CEL (4) |
| 15-7 AMV, RH1150 | W | 1,064 | 5,300 | 1.1 | 45 | 45 | 5,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
| 15-7 AMV, RH1150 | S | 1,064 | 5,300 | 0.9 | 45 | 45 | 6,000 | $\mathrm{C}(\mathrm{PR}) ; \mathbf{P}(\mathrm{PR}) ; \mathbf{T}$ | - | CEL (4) |
| 15-7 AMV, RH1150 | w | 197 | 2,340 | 0.4 | 45 | 45 | 3,300 | $\mathrm{C}(\mathrm{PR}) ; \mathbf{P}(\mathrm{PR}) ; \mathbf{T}$ | - | CEL (4) |
| 15-7 AMV, RH1150 | S | 197 | 2,340 | 0.7 | 45 | 45 | 1,150 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
| 15-7 AMV, RH1150 | w | 402 | 2,370 | 1.0 | 45 | 45 | 8,750 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{E} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T} ; \mathrm{X}$ | - | CEL (4) |
| 15-7 AMV, RH1150 | S | 402 | 2,370 | 0.7 | 45 | 45 | 4,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{E} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
| 15-7 AMV, RH950 | W | 123 | 5,640 | 0.7 | 48 | 48 | 0 | C (PR); P (PR) | - | CEL (4) |
| 15-7 AMV, RH950 | S | 123 | 5,640 | 1.3 | 48 | 48 | 2,900 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T} ; \mathrm{X}$ | - | CEL (4) |
| 15-7 AMV, RH950 | W | 403 | 6,780 | 1.0 | 48 | 48 | 3,750 | $\mathrm{C}(\mathrm{PR})$; P (PR) ; T ; X | - | CEL (4) |
| 15-7 AMV, RH950 | S | 403 | 6,780 | 1.2 | 48 | 48 | 5,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T} ; \mathrm{X}$ | - | CEL (4) |
| 15-7 AMV, RH950 | W | 751 | 5,640 | 1.6 | 48 | 48 | 6,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{E} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
| 15-7 AMV, RH950 | W | 1,064 | 5,300 | 2.0 | 48 | 48 | 6,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T} ; \mathrm{X}$ | - | CEL (4) |
| 15-7 AMV, RH950 | S | 1,064 | 5,300 | 0.9 | 48 | 48 | 6,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T} ; \mathrm{X}$ | - | CEL (4) |
| 15-7 AMV, RH950 | w | 197 | 2,340 | 0.8 | 48 | 48 | 1,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{E} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
| 15-7 AMV, RH950 | S | 197 | 2,340 | 0.7 | 48 | 48 | 1,750 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{E} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
| 15-7 AMV, RH950 | W | 402 | 2,370 | 0.6 | 48 | 48 | 4,650 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{E} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
| 15-7 AMV, RH950 | S | 402 | 2,370 | 0.6 | 48 | 48 | 6,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{E} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
|  | W | 189 | 5,900 | - | - | - | - | 1-C; SPP | - | CEL (4) |
| AL362, H950 ${ }^{\text {d }}$ | W | 189 | 5,900 | - | - | - | - | I-C; SPP | SPP (WB) | CEL (4) |
| $\mathrm{AL} 362,^{\text {H1050 }}{ }_{\text {d }}$ | W | 189 | 5,900 | - | - | - | - | 1-C; SPP | - | CEL (4) |
| AL362, H1050 ${ }^{\text {d }}$ | W | 189 | 5,900 | -- | - | - | - | S-C; SPP | D-P (WB) | CEL (4) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}$-Cu-Mo | W | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | - | INCO (3) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | S | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | - | INCO (3) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | W | 403 | 6,780 | $<0.1$ | 0 | 1 | - | I-C | - | INCO (3) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | S | 403 | 6,780 | $<0.1$ | 0 | 1 | - | I-C | - | INCO (3) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | W | 751 | 5,640 | $<0.1$ | 0 | 1 | - | I-C | - | INCO (3) |

Table 49. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | $\begin{aligned} & \text { Exposure } \\ & \text { (day) } \end{aligned}$ | Depth <br> (ft) | Corrosion |  |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | $\begin{aligned} & \text { Maximum } \\ & \text { Tunnel Length } \\ & \text { (mils) } \end{aligned}$ | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | S | 751 | 5,640 | $<0.1$ | 0 | 1 | - | I-C | - | INCO (3) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | W | 1,064 | 5,300 | $<0.1$ | 0 | I | - | I-C | - | INCO (3) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | S | 1,064 | 5,300 | $<0.1$ | 0 | 3 | - | C | - | INCO (3) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | W | 197 | 2,340 | $<0.1$ | 0 | I | - | I-C | - | INCO (3) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | S | 197 | 2,340 | $<0.1$ | 0 | 1 | - | I-C | - | INCO (3) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | W | 402 | 2,370 | $<0.1$ | 0 | 1 | - | I-C | - | INCO (3) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}$-Cu-Mo | S | 402 | 2,370 | $<0.1$ | 0 | 1 | - | I-C | - | INCO (3) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | W | 181 | 5 | $<0.1$ | 3 | 0 | -- | P | - | INCO (3) |
| $17 \mathrm{Cr}-14 \mathrm{Ni}-\mathrm{Cu}-\mathrm{Mo}$ | W | 366 | 5 | $<0.1$ | 0 | I | - | 1-C | - | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 123 | 5,640 | $<0.1$ | 0 | 0 | 0 | NC | - | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | w | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | - | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 123 | 5,640 | 0.3 | 0 | 115 | 0 | C (PR) | - | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 123 | 5,640 | $<0.1$ | 0 | 0 | - | NC | - | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 403 | 6,780 | 0.5 | 115 | 0 | 2,750 | P (PR) ; T | - | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 403 | 6,780 | $<0.1$ | 0 | I | - | I-C | - | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 403 | 6,780 | 0.6 | 115 | 115 | 2,000 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$; T | - | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 403 | 6,780 | $<0.1$ | 0 | 3 | - | C | - | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 751 | 5,640 | 0.2 | 115 | 24 | 2,800 | C; P (PR) ; T | -- | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 751 | 5,640 | $<0.1$ | 0 | 0 | - | NC | - | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 751 | 5,640 | $<0.1$ | 0 | 0 | 0 | NC | - | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 751 | 5,640 | $<0.1$ | 0 | 3 | - | C | - | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 1,064 | 5,300 | 0.2 | 115 | 0 | 2,250 | P (PR) ; T | - | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | w | 1,064 | 5,300 | $<0.1$ | 0 | 0 | - | NC | - | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 1,064 | 5,300 | 0.3 | 115 | 0 | 4,000 | P (PR) ; T | - | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 1,064 | 5,300 | 0.4 | 62 | 62 | - | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR})$ | - | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 197 | 2,340 | 0.8 | 115 | 109 | 1,250 | $\mathrm{C} ; \mathrm{P}(\mathrm{PR}) ; \mathrm{T}$ | - | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 197 | 2,340 | $<0.1$ | 0 | 1 | - | I-C | - | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 197 | 2,340 | 0.4 | 115 | 21 | 1,000 | $\mathrm{C} ; \mathrm{P}(\mathrm{PR})$; $\mathbf{T}$ | - | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 197 | 2,340 | $<0.1$ | 0 | 13 | - | C | - | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | w | 402 | 2,370 | 0.8 | 115 | 0 | 2,000 | P (PR) ; T | -- | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | w | 402 | 2,370 | 1.1 | 0 | 62 | - | C (PR) | - | INCO (3) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 402 | 2,370 | 1.0 | 115 | 0 | 3,000 | P (PR); T | - | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 402 | 2,370 | $<0.1$ | 0 | 1 | - | $\mathrm{I}-\mathrm{C}$ | - | INCO (3) |

Table 49. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum Pit Depth (mils) | Crevice Depth (mils) | $\qquad$ | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| $18 \mathrm{Cr}-1+\mathrm{Mn}-0.5 \mathrm{~N}$ <br> $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W w | 181 366 | 5 5 | 4.9 2.6 | 50 50 | 50 50 | - | $\begin{aligned} & \mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) \\ & \mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) \end{aligned}$ | - | $\begin{aligned} & \text { INCO (3) } \\ & \text { INCO (3) } \end{aligned}$ |

[^14][^15]Table 50. Stress Corrosion of Precipitation-Hardening Stainless Steels, Calculated Stresses

| Alloy | Stress (ksi) | Percent <br> Yield <br> Strength | Exposure (day) | Depth (ft) | Number of Specimens Exposed | Number Failed ${ }^{a}$ | Source ${ }^{b}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| AISI $630, \mathrm{H925}{ }^{\text {c }}$ | 65 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| AISI 630, $\mathrm{H} 925^{\text {c }}$ | 93 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| AISI 630, H925 ${ }^{\text {c }}$ | 139 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| AISl 630, H925 ${ }^{\text {c }}$ | 65 | 35 | 197 | 2,340 | 2 | 0 | CEL (4) |
| AlS1 630, H925 ${ }^{\text {c }}$ | 93 | 50 | 197 | 2,340 | 2 | 0 | CEL (4) |
| AISI 630, H925 ${ }^{\text {c }}$ | 139 | 75 | 197 | 2,340 | 2 | 0 | CEL (4) |
| AlSl $630, \mathrm{H} 925{ }^{\text {c }}$ | 93 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AIS1 630, H925 ${ }^{\text {c }}$ | 139 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AlSI 630, H925 ${ }^{\text {c }}$ | 65 | 35 | 70 | 5 | 3 | $3^{d}$ | CEL (4) |
| AISI 630, H925 ${ }^{\text {c }}$ | 93 | 50 | 364 | 5 | 3 | $1(89)^{d}$ | CEL (4) |
| AIS1 630, H925 ${ }^{\text {c }}$ | 139 | 75 | 364 | 5 | 3 | $2(70,233)^{d}$ | CEL (4) |
| AISI 630, RH950 | 61 | 39 | 123 | 5,640 | 3 | 0 | NADC (7) |
| AISI 630, RH950 | 86 | 55 | 123 | 5,640 | 3 | 0 | NADC (7) |
| AlSI 630, RH9 50 | 134 | 86 | 123 | 5,640 | 3 | 0 | NADC (7) |
| AISI 631, TH1050 ${ }_{C}$ | 66 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| AISI 631, TH1050 ${ }_{c}$ | 94 | 50 | 403 | 6,780 | 2 | 1 | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {c }}$ | 141 | 75 | 403 | 6,780 | 2 | 1 | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {c }}$ | 66 | 35 | 197 | 2,340 | 2 | 0 | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {c }}$ | 94 | 50 | 197 | 2,340 | 2 | 0 | CEL (4) |
| AiSI 631, TH1050 ${ }^{\text {c }}$ | 141 | 75 | 197 | 2,340 | 2 | 0 | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {C }}$ | 94 | 50 | . 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 631, TH1050 ${ }^{c}$ | 141 | 75 | - 402 | 2,370 | 3 | 1 | CEL (4) |
| AISI 631, TH1050 ${ }^{c}$ | 66 | 35 | 364 | 5 | 3 | 0 | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {c }}$ | 94 | 50 | 364 | 5 | 3 | $1(253){ }^{e}$ | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {c }}$ | 141 | 75 | 364 | 5 | 3 | $1(253){ }^{\text {f }}$ | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {c }}$ | 69 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {c }}$ | 98 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| AISI 631, RH1050 ${ }^{c}$ | 147 | 75 | 403 | 6,780 | 2 | $1^{d}$ | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {c }}$ | 69 | 35 | 197 | 2,340 | 2 | 0 | CEL (4) |
| AISI 631, RH1050 ${ }_{c}^{C}$ | 98 | 50 | 197 | 2,340 | 2 | 0 | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {c }}$ | 147 | 75 | 197 | 2,340 | 2 | 0 | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {c }}$ | 98 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {c }}$ | 147 | 75 | 402 | 2,370 | 3 | 2 | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {c }}$ | 69 | 35 | 364 | 5 | 3 | 0 | CEL (4) |
| AISI 631, $\mathrm{RH} 1050^{\text {c }}$ | 98 | 50 | 364 | 5 | 3 | 0 | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {c }}$ | 147 | 75 | 364 | 5 | 3 | $1(90)^{d}$ | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {c }}$ | 64 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| AISI 632, RH1 $100^{c}$ | 92 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| AlSI 632, RH1 $100^{c}$ | 138 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| AISI 632, RH1 $100^{\text {c }}$ | 64 | 35 | 197 | 2,340 | 2 | 0 | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {c }}$ | 92 | 50 | 197 | 2,340 | 2 | 0 | CEL (4) |
| AISI 632, $\mathrm{RH} 1100^{\text {c }}$ | 138 | 75 | 197 | 2,340 | 2 | 0 | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {c }}$ | 92 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 632, RH1 $100^{\text {c }}$ | 138 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AlSI 632, RH1100 ${ }^{\text {c }}$ | 64 | 35 | 364 | 5 | 3 | 0 | CEL (4) |
| AISI 632, RH1 $100^{\text {c }}$ | 92 | 50 | 364 | 5 | 3 | $1(322)^{g}$ | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {c }}$ | 138 | 75 | 364 | 5 | 3 | $3(56,70,133)^{g}$ | CEL (4) |

Continued

Table 50. Continued.

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure (day) | Depth (ft) | Number of Specimens Exposed | Number Failed ${ }^{a}$ | Source ${ }^{b}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| AISI 634, CRT | 108 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 634, CRT | 162 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 635 | 68 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 635 | 97 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 635 | 145 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| AISI 635 | 92 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| AISI 635 | 138 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| ASTM XM16, H950 | - | 75 | 189 | 5,900 | 2 | 0 | CEL (4) |
| ASTM XM16, H1050 | - | 75 | 189 | 5,900 | 1 | 0 | CEL (4) |
| ASTM XM16, H1050 | - | 75 | 189 | 5,900 | 1 | 0 | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{c}$ | 75 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {c }}$ | 107 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {c }}$ | 161 | 75 | 403 | 6,780 | 2 | 1 | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {c }}$ | 75 | 35 | 197 | 2,340 | 2 | 0 | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {c }}$ | 107 | 50 | 197 | 2,340 | 2 | 0 | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {c }}$ | 161 | 75 | 197 | 2,340 | 2 | 0 | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {c }}$ | 107 | 50 | 402 | 2,370 | 3 | 1 | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\circ}$ | 161 | 75 | 402 | 2,370 | 3 | 1 | CEL (4) |
| PH14-8MO, SRH950 ${ }^{\text {c }}$ | 75 | 35 | 364 | 5 | 3 | $3(322){ }^{\text {d }}$ | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {C }}$ | 107 | 50 | 364 | 5 | 3 | 3 (322) $^{d}$ | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {c }}$ | 161 | 75 | 364 | 5 | 3 | 0 | CEL (4) |
| 15-7 AMV, A | 20 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| 15-7 AMV, A | 29 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| 15-7 AMV, A | 43 | 75 | 403 | 6,780 | 2 | 1 | CEL (4) |
| 15-7 AMV, A | 20 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| 15-7 AMV, A | 29 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| 15-7 AMV, A | 43 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| 15-7 AMV, A | 29 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 15-7 AMV, A | 43 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 15-7 AMV, RH1150 | 55 | 35 | 123 | 5,640 | 3 | 0 | CEL (4) |
| 15-7 AMV, RH1150 | 79 | 50 | 123 | 5,640 | 3 | 0 | CEL (4) |
| 15-7 AMV, RH1150 | 119 | 75 | 123 | 5,640 | 3 | 0 | CEL (4) |
| 15-7 AMV, RH1150 | 55 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| 15-7 AMV, RH1150 | 79 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| 15-7 AMV, RH1150 | 119 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| 15-7 AMV, RH1150 | 55 | 35 | 751 | 5,640 | 3 | $3{ }^{i}$ | CEL (4) |
| 15-7 AMV, RH1150 | 79 | 50 | 751 | 5,640 | 3 | $3^{2}$ | CEL (4) |
| 15-7 AMV, RH1150 | 119 | 75 | 751 | 5,640 | 3 | $3^{2}$ | CEL (4) |
| 15-7 AMV, RH1150 | 79 | 50 | 197 | 2,340 | 3 | 2 | CEL (4) |
| 15-7 AMV, RH1150 | 119 | 75 | 197 | 2,340 | 3 | 3 | CEL (4) |
| 15-7 AMV, RH1150 | 79 | 50 | 402 | 2,370 | 3 | 1 | CEL (4) |
| 15-7 AMV, RH1150 | 119 | 75 | 402 | 2,370 | 3 | 1 | CEL (4) |
| 15-7 AMV, RH950 | 77 | 35 | 123 | 5,640 | 3 | $1{ }^{j}$ | CEL (4) |
| 15-7 AMV, RH950 | 110 | 50 | 123 | 5,640 | 3 | 3 | CEL (4) |
| 15-7 AMV, RH950 | 165 | 75 | 123 | 5,640 | 3 | $3^{k}$ | CEL (4) |
| 15-7 AMV, RH950 | 77 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| 15-7 AMV, RH950 | 110 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| 15-7 AMV, RH950 | 165 | 75 | 403 | 6.780 | 2 | 2 | CEL (4) |

Continued

Table 50. Continued.

| Alloy | $\begin{aligned} & \text { Stress } \\ & (\mathrm{ksi}) \end{aligned}$ | Percent <br> Yield <br> Strength | Exposure (day) | Depth <br> (ft) | Number of Specimens Exposed | Number <br> Failed ${ }^{a}$ | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 15-7 AMV, RH950 | 77 | 35 | 751 | 5,640 | 3 | 0 | CEL (4) |
| 15-7 AMV, RH950 | 110 | 50 | 751 | 5,640 | 3 | 3 | CEL (4) |
| 15-7 AMV, RH950 | 165 | 75 | 751 | 5,640 | 3 | 3 | CEL (4) |
| 15-7 AMV, RH950 | 77 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| 15-7 AMV, RH950 | 110 | 50 | 197 | 2,340 | 3 | 21 | CEL (4) |
| 15-7 AMV, RH950 | 165 | 75 | 197 | 2,340 | 3 | 2 | CEL (4) |
| 15-7 AMV, RH950 | 110 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 15-7 AMV, RH950 | 165 | 75 | 402 | 2,370 | 3 | 3 | CEL (4) |
| AL362, H950 | - | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $\mathrm{AL} 362, \mathrm{H} 950^{b}$ | - | 75 | 189 | 5,900 | 6 | 0 | CEL (4) |
| AL362, H 1050 | - | 75 | 189 | 5,900 | 5 | 0 | CEL (4) |
| AL362, H1050 ${ }^{\text {b }}$ | - | 75 | 189 | 5,900 | 6 | 0 | CEL (4) |
| AL362, H1050 | - | 75 | 189 | 5,900 | 1 | 0 | CEL (4) |
| AL362, $\mathrm{H} 1050{ }^{\text {m }}$ | - | 75 | 189 | 5,900 | 1 | 0 | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | 41 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | 61 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |

${ }^{a}$ Numbers in parentheses indicate days to failure.
$b_{\text {Numbers refer to references at end of report. }}$
${ }^{c}$ Transverse butt weld at midength of specimens.
${ }^{d}$ Failed by crevice corrosion at anvils of stress jig.
${ }^{e}$ Broke at edge of weld bead.
$f_{\text {Broke at junction of heat-affected zone and sheet metal. }}$
$g_{\text {Crevice corrosion at bolt hole, released tension. }}$
${ }^{\prime}$ Painted, zinc-rich primer, 8 mils.
${ }^{i}$ Specimens were missing when structure was retrieved.
$j_{\text {Incipient crack in one specimen. }}$
${ }^{k}$ One specimen broke prior to exposure in the seawater.
${ }^{l}$ Painted, wash primer (MIL-C-8514) + red lead epoxy primer + epoxy topcoat, 7 mils.
${ }^{m}$ Painted, wash primer (MIL-C-8514) + epoxy primer + epoxy topcoat, 7 mils.

| Alloy | Exposure (day) | Depth (ft) | Type of Residual Stress | Remarks |
| :---: | :---: | :---: | :---: | :---: |
| AISI 630, H925 | 403 | 6,780 | Unrelieved circular weld | $\mathrm{SCC}^{a}$ propagated across weld bead |
| AISI 631, TH1050 | 398 | 5 | Unrelieved circular weld | SCC radially in three directions to circular weld bead |
| AISI 631, TH1050 | 403 | 6,780 | Unrelieved circular weld | SCC propagated across and around weld bead |
| AISI 631, RH11050 | 197 | 2,340 | Unrelieved circular weld | SCC, origin on outside of weld bead, propagated into heat-affected zone, circumferentially in both directions around weld |
| AISI 631, RH1050 | 402 | 2,370 | Unrelieved circular weld | SCC, origin on outside of weld bead, propagation in both directions around weld bead at edge of heataffected zone |
| AlSI 631, RH1050 | 403 | 6,780 | Unrelieved circular weld | SCC, origin on outside of weld bead and propagated in both directions around outside of weld bead |
| AISI 632, RH1100 | 402 | 2,370 | Unrelieved circular weld | SCC, origin on outside edge of weld bead, propagated in both directions around weld in heat-affected zone |
| 15-7 AMV, RH1150 | 123 | 5,640 | Imposed by insulator | SCC due to squeeze by insulators on sides of panel |
| 15-7 AMV, RH1150 | 1,064 | 5,300 | Unreamed drilled hole | SCC, origin at unreamed hole, not deburred |
| 15-7 AMV, RH950 | 1,064 | 5,300 | Unreamed drilled hole | SCC, origin at unreamed hole, not deburred |

${ }^{a} \mathrm{SCC}=$ stress corrosion cracking failure.
Table 51. Stress Corrosion of Precipitation-Hardening Stainless Steels, Residual Stresses
Table 52. Changes in Mechanical Properties of the Precipitation-Hardening Stainless Steels Due to Corrosion

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Original (ksi) | \% Change | Original <br> (ksi) | \% Change | Original <br> (\%) | \% Change |  |
| AISI 630, H925 ${ }^{\text {c }}$ | W | 403 | 6,780 | 192 | -6 | 185 | -6 | 3 | -17 | CEL (4) |
| AISI 630, $11925{ }^{\text {c }}$ | S | 403 | 6,780 | 192 | -9 | 185 | -7 |  | -33 | CEL (4) |
| AISI $630, H 925^{\text {c }}$ | W | 402 | 2,370 | 192 | -8 | 185 | -11 | 3 | -17 | CEL (4) |
| AISI 630, $19225^{\text {c }}$ | S | 402 | 2,370 | 192 | -7 | 185 | -9 | 3 | -27 | CEL (4) |
| AISI 630, H925 ${ }^{\text {c }}$ | W | 181 | 5 | 192 | -7 | 185 | -11 | 3 | +7 | CEL (4) |
| AlSI 630, H925 ${ }^{\text {c,d }}$ | W | 365 | 5 | 192 | - | 185 | - | 3 | - | CEL (4) |
| AlSI 631, A | W | 197 | 2,340 | 134 | -8 | 58 | -22 | 37 | -3 | NADC (7) |
| AISI 631, TH1050 ${ }^{\text {c }}$ | w | 403 | 6,780 | 197 | -25 | 188 | -60 | 3 | +77 | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {c }}$ | S | 403 | 6,780 | 197 | -24 | 188 | -54 | 3 | +57 | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {c }}$ | W | 402 | 2,370 | 197 | -22 | 188 | -56 | 3 | +40 | CEL (4) |
| AISI 631, TH1050 ${ }^{\text {c }}$ | S | 402 | 2,370 | 197 | -21 | 188 | -57 | 3 | $+57$ | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {c }}$. | W | 403 | 6,780 | 207 | -29 | 197 | -59 | 5 | -7 | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {c }}$ | S | 403 | 6,780 | 207 | -21 | 197 | -54 | 5 | +9 | CEL (4) |
| AISI 631, RH1050 ${ }^{\text {c }}$ | W | 402 | 2,370 | 207 | -24 | 197 | -60 | 5 | -7 | CEL (4) |
| AIS1 632, RH1 $100^{\text {c }}$ | W | 403 | 6,780 | 192 | -22 | 183 | -53 | 3 | +33 | CEL (4) |
| AISI 632, RH1100 ${ }^{c}$ | S | 403 | 6,780 | 192 | -24 | 183 | -54 | 3 | +43 | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {c }}$ | W | 402 | 2,370 | 192 | -20 | 183 | -55 | 3 | +43 | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {c }}$ | S | 402 | 2,370 | 192 | -19 | 183 | -54 | 3 | +5 | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {c }}$ | W | 181 | 5 | 192 | -15 | 183 | -54 | 3 | +73 | CEL (4) |
| AISI 632, RH1100 ${ }^{\text {c }}$ | W | 365 | 5 | 192 | -15 | 183 | -54 | 3 | +72 | CEL (4) |
| AISI 634, CRT | W | 123 | 5,640 | 229 | +4 | 215 | -11 | 19 | -1 | CEL (4) |
| AISI 634, CRT | W | 403 | 6,780 | 229 | +4 | 215 | -11 | 19 | +2 | CEL (4) |
| AISI 634, CRT | S | 403 | 6,780 | 229 | +4 | 215 | -8 | 19 | +4 | CEL (4) |
| AISI 634, CRT | W | 751 | 5,640 | 229 | +8 | 215 | -7 | 19 | +4 | CEL (4) |
| AISI 634, CRT | W | 197 | 2,340 | 229 | +3 | 215 | -11 | 19 | +4 | CEL (4) |
| AISI 634, CRT | S | 197 | 2,340 | 229 | +3 | 215 | -9 | 19 | +4 | CEL (4) |
| AISI 635 | W | 123 | 5,640 | 191 | +11 | 184 | +11 | 14 | +7 | CEL (4) |
| AIS1 635 | W | 403 | 6,780 | 191 | +12 | 184 | +12 | 14 | -32 | CEL (4) |
| AISI 635 | S | 403 | 6,780 | 191 | +11 | 184 | +10 | 14 | -33 | CEL (4) |
| AISI 635 | W | 751 | 5,640 | 191 | +8 | 184 | +8 | 14 | -26 | CEL (4) |
| AIS1 635 | W | 1,064 | 5,300 | 199 | +2 | 193 | +2 | 9 | -26 | CEL (4) |
| AISI 635 | S | 1,064 | 5,300 | 199 | +4 | 193 | +5 | 9 | -24 | CEL (4) |
| AISI 635 | W | 197 | 2,340 | 191 | +1 | 184 | +2 | 14 | -18 | CEL (4) |


| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{gathered} \text { Original } \\ \text { (ksi) } \end{gathered}$ | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| AISI 035 | s | 197 | 2,340 | 191 | +2 | 184 | +4 | 14 | -20 | CEL (4) |
| AISt 635 | w | 402 | 2,370 | 191 | -2 | 184 | -6 | 14 | -7 | CEL (4) |
| Alst 635 | S | 402 | 2,370 | 191 | +7 | 184 | +5 | 14 | -30 | CEL (4) |
| A1S1 635 | w | 181 | 5 | 191 | +4 | 184 | +5 | 14 | -38 | CEL (4) |
| AISI 635 | w | 365 | 5 | 191 | +1 | 184 | +2 | 14 | -5 | CEL (4) |
| P114-8MO, SRH1950 ${ }^{\circ}$ | w | 403 | 6,780 | 229 | -32 | 214 | -53 | 2 | +40 | CEL (4) |
| $\mathrm{PH} 11+8 \mathrm{Mo}, \mathrm{SRH1959}{ }^{\text {c }}$ | S | 403 | 6,780 | 229 | -30 | 214 | -49 | 2 | +150 | CEL (4) |
| Pllit-8.10, SRH959 ${ }^{\text {c }}$ | w | 402 | 2,370 | 229 | -30 | 214 | -54 | 2 | +100 | CEL (4) |
| PH14-8Mo, SRH950 ${ }^{\text {c }}$ | S | 402 | 2,370 | 229 | -32 | 214 | -57 | 2 | +50 | CEL (4) |
| PH14-8MO, SRH950 ${ }^{\text {c }}$ | w | 181 | 5 | 229 | -23 | 214 | -52 | 2 | +100 | CEL (4) |
| 15-7 Amiv, A | w | 123 | 5,640 | 129 | -8 | 57 | +3 | 27 | -50 | CEL (4) |
| 15-7 AMV, A | w | 403 | 6,780 | 129 | +9 | 57 | -2 | 27 | +38 | CEL (4) |
| 15-7 AMV, A | s | 403 | 6,780 | 129 | +10 | 57 | 0 | 27 | +64 | CEL (4) |
| 15-7 AMV, A | w | 751 | 5,640 | 129 | +6 | 57 | +2 | 27 | +2 | CEL (4) |
| 15-7 AmV, A | S | 751 | 5,640 | 129 | +4 | 57 | +5 | 27 | +3 | CEL (4) |
| 15-7 AMV, A | w | 197 | 2,340 | 129 | 0 | 57 | -10 | 27 | +39 | CEL (4) |
| 15-7 AMV, A | S | 197 | 2,340 | 129 | 0 | 57 | -10 | 27 | +43 | CEL (4) |
| 15-7 AMV, A | w | 402 | 2,370 | 129 | +8 | 57 | +1 | 27 | +34 | CEL (4) |
| 15-7 AMV, A | S | 402 | 2,370 | 129 | +8 | 57 | +1 | 27 | +44 | CEL (4) |
| 15-7 AMV, RH1150 | s | 123 | 5,640 | 190 | -2 | 158 | +1 | 8 | -16 | CEL (4) |
| 15-7 AMV, RH1150 | w | 403 | 6,780 | 190 | +4 | 158 | +7 | 8 | -50 | CEL (4) |
| 15-7 AMV, RH1150 | s | 403 | 6,780 | 190 | +3 | 158 | +9 | 8 | -37 | CEL (4) |
| 15-7 AMV, RH1150 | w | 751 | 5,640 | 190 | -3 | 158 | -28 | 8 | -31 | CEL (4) |
| 15-7 AMV, RH1150 | S | 751 | 5,640 | 190 | +5 | 158 | +8 | 8 | -4 | CEL (4) |
| 15-7 AMV, RH1150 | w | 1,064 | 5,300 | 196 | +2 | 166 | +2 | 8 | -10 | CEL (4) |
| 15-7 AMV, RH1150 | S | 1,064 | 5,300 | 196 | +2 | 166 | +3 | 8 | -3 | CEL (4) |
| 15-7 AMV, RH1150 | W | 197 | 2,340 | 190 | +2 | 158 | +3 | 8 | -1 | CEL (4) |
| 15-7 ANV, RH1150 | S | 197 | 2,340 | 190 | +2 | 158 | +5 | 8 | -10 | CEL (4) |
| 15-7 AMV, RH1150 | w | 402 | 2,370 | 190 | +5 | 158 | +8 | 8 | -7 | CEL (4) |
| 15-7 AMV, RH1150 | S | 402 | 2,370 | 190 | +4 | 158 | +4 | 8 | -4 | CEL (4) |
| 15-7 AMV, RH950 | S | 123 | 5,6+0 | 248 | -19 | 220 | - | 2 | -4 | CEL (4) |
| 15-7 AMV, RH950 | w | 403 | 6,780 | 248 | -9 | 220 | +1 | 2 | -100 | CEL (4) |
| 15-7 AMV, RH950 | S | 403 | 6,780 | 248 | -2 | 220 | +5 | 2 | -88 | CEL (4) |
| 15-7 AMV, RH950 | w | 751 | 5,640 | $2+8$ | -1 | 220 | +2 | 2 | - | CEL (4) |

Table 52. Continued

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| 15-7 AMV, RH950 | W | 197 | 2,340 | 248 | -3 | 220 | 0 | 2 | -18 | CEL (4) |
| 15-7 AMV, RH950 | S | 197 | 2,340 | 248 | -2 | 220 | +2 | 2 | -18 | CEL (4) |
| 15-7 AMV, RH950 | W | 402 | 2,370 | 248 | +2 | 220 | +2 | 2 | -32 | CEL (4) |
| 15-7 AMV, RH950 | S | 402 | 2,370 | 248 | -2 | 220 | +1 | 2 | -18 | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 123 | 5,640 | 126 | 0 | 81 | -3 | 52 | +6 | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 403 | 6,780 | 126 | +2 | 81 | +6 | 52 | -3 | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 403' | 6,780 | 126 | 0 | 81 | +2 | 52 | -1 | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 751 | 5,640 | 126 | 0 | 81 | +5 | 52 | -2 | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 751 | 5,640 | 126 | -1 | 81 | +1 | 52 | -2 | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 1,064 | 5,300 | 126 | -2 | 81 | +4 | 52 | -4 | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 1,064 | 5,300 | 126 | -27 | 81 | -31 | 52 | -36 | CEL. (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 197 | 2,340 | 126 | -2 | 81 | +5 | 52 | -4 | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 197 | 2,340 | 126 | -1 | 81 | +3 | 52 | +2 | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | W | 402 | 2,370 | 126 | +1 | 81 | +5 | 52 | -11 | CEL (4) |
| $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | S | 402 | 2,370 | 126 | -1 | 81 | +3 | 52 | -9 | CEL (4) |

[^16]Table 53. Chemical Compositions of Miscellaneous Stainless Steels, Percent by Weight

| Alloy | C | Mn | P | S | Si | Ni | Cr | Mo | Cu | Other | $\mathrm{Fe}^{\text {a }}$ | Source ${ }^{b}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 20 Cb | 0.04 | 0.79 | 0.018 | 0.004 | 0.67 | 28.38 | 19.80 | 2.06 | 3.11 | $0.77 \mathrm{Cb}+\mathrm{Ta}$ | R | CEL (4) |
| 20 Cb | 0.05 | 0.82 | - | - | 0.70 | 28.43 | 20.09 | 2.32 | 3.37 | $0.83 \mathrm{Cb}+\mathrm{Ta}$ | R | CEL (4) |
| 20 Cb | - | - | - | - | - | 33.0 | 20.0 | 2.5 | 3.5 | - | R | INCO (3) |
| $20 \mathrm{Cb}-3^{C}$ | 0.07 | 2.0 | 0.035 | 0.035 | 1.0 | 34.0 | 20.0 | 2.5 | 3.5 | $\mathrm{Cb}+\mathrm{Ta}, 8 \mathrm{XC}$ | R | CEL (4) |
| $20 \mathrm{Cb}-3$ | - | - | - | - | - | 34.0 | 20.0 | 2.3 | 3.4 | - | R | INCO (3) |
| Ni-Cr-Cu-Mo No. 1, cast | - | - | - | - | - | 30.0 | 20.0 | 2.5 | 4.0 | - | R | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo}$ No. 2, cast | - | - |  | - | - | 30.0 | 20.0 | 2.5 | 3.5 | - | R | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$, cast | - | - | - | - | - | 24.0 | 19.0 | 3.0 | - | - | R | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$-Si, cast | - | - | - | - | 1.0 | 23.0 | 21.0 | 5.0 | - | - | R | INCO (3) |
| RL-35-100, cast | - | 1.0 | - | - | - | 31.0 | 23.0 | 9.0 | - | - | R | INCO (3) |

${ }^{a} \mathrm{R}=$ remainder.
$b_{\text {Numbers refer to references at end of report. }}$.
${ }^{c}$ Typical analysis.

Table 54. Corrosion Rates and Types of Corrosion of Miscellaneous Stainless Steels

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | Type ${ }^{\text {b }}$ |  |
| 20 Cb | W | 123 | 5,640 | 0 | 0 | 0 | NC | CEL (4) |
| 20 Cb | S | 123 | 5,640 | 0 | 0 | 0 | NC | CEL (4) |
| 20 Cb | w | 403 | 6,780 | 0 | 0 | 0 | NC | CEL (4) |
| 20 Cb | S | 403 | 6,780 | $<0.1$ | 0 | 102 | C | CEL (4) |
| 20 Cb | W | 751 | 5,640 | $<0.1$ | 0 | 26 | C | CEL (4) |
| 20 Cb | S | 751 | 5,640 | 0 | 0 | 0 | NC | CEL (4) |
| 20 Cb | w | 1,064 | 5,300 | 0 | 0 | 0 | NC | CEL (4) |
| 20 Cb | S | 1,064 | 5,300 | 0 | 0 | 0 | NC | CEE (4) |
| 20 Cb | W | 197 | 2,340 | $<0.1$ | 0 | I | I-C | CEL (4) |
| 20 Cb | S | 197 | 2,340 | $<0.1$ | 0 | 1 | I-C | CEL (4) |
| 20 Cb | W | 402 | 2,370 | $<0.1$ | 0 | 0 | NC | CEL (4) |
| 20 Cb | S | 402 | 2,370 | $<0.1$ | 0 | 0 | NC | CEL (4) |
| 20 Cb | W | 181 | 5 | <0.1 | 0 | 5 | C; SL-E | CEL (4) |
| 20 Cb | w | 398 | 5 | $<0.1$ | 14 | 0 | SL-E; P | CEL (4) |
| 20 Cb | w | 540 | 5 | $<0.1$ | 24 | 0 | P | CEL (4) |
| 20 Cb | w | 588 | 5 | $<0.1$ | 0 | 21 | C | CEL (4) |
| $20 \mathrm{Cb}-3$ | w | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $20 \mathrm{Cb}-3$ | S | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $20 \mathrm{Cb}-3$ | W | 189 | 5,900 | $<0.1$ | I | I | I-C; I-P | CEL (4) |
| $20 \mathrm{Cb}-3$ | S | 189 | 5,900 | $<0.1$ | 0 | 40 | C | CEL (4) |

Table 54. Continued.

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth (ft) | Corrosion |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum <br> Pit Depth (mils) | Crevice <br> Depth ${ }^{b}$ <br> (mils) | Type ${ }^{\text {b }}$ |  |
| $20 \mathrm{Cb}-3$ | W | 403 | 6,780 | $<0.1$ | 0 | I | I-C | INCO (3) |
| $20 \mathrm{Cb}-3$ | S | 403 | 6,780 | $<0.1$ | 0 | I | I-C | INCO (3) |
| $20 \mathrm{Cb}-3$ | W | 751 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $20 \mathrm{Cb}-3$ | S | 751 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $20 \mathrm{Cb}-3$ | W | 1,064 | 5,300 | $<0.1$ | 0 | I | I-C | INCO (3) |
| $20 \mathrm{Cb}-3$ | S | 1,064 | 5,300 | $<0.1$ | 0 | 1 | I-C | INCO (3) |
| $20 \mathrm{Cb}-3$ | W | 197 | 2,340 | $<0.1$ | 0 | I | H C | INCO (3) |
| $20 \mathrm{Cb}-3$ | S | 197 | 2,340 | $<0.1$ | 0 | 1 | I-C | INCO (3) |
| $20 \mathrm{Cb}-3$ | W | 402 | 2,370 | $<0.1$ | 1 | 0 | I-P | INCO (3) |
| $20 \mathrm{Cb}-3$ | S | 402 | 2,370 | $<0.1$ | 0 | I | $\mathrm{I}-\mathrm{C}$ | INCO (3) |
| $20 \mathrm{Cb}-3$ | W | 181 | 5 | $<0.1$ | 0 | I | I-C | INCO (3) |
| $20 \mathrm{Cb}-3$ | W | 366 | 5 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| Ni-Cr-Cu-Mo No. 1, cast | W | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo} \mathrm{No}. \mathrm{1}$, | S | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo}$ No. 1, cast | W | 403 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| Ni-Cr-Cu-Mo No. 1, cast | S | 403 | 6,780 | <0.1 | 0 | 0 | NC | INCO (3) |
| Ni-Cr-Cu-Mo No. 1, cast | W | 751 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo}$ No. 1, cast | S | 751 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo}$ No. 1, cast | W | 1,064 | 5,300 | $<0.1$ | 0 | I | $\mathrm{I}-\mathrm{C}$ | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo}$ No. 1, cast | S | 1,064 | 5,300 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo}$ No. 1, cast | W | 197 | 2,340 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| Ni-Cr-Cu-Mo No. 1, cast | S | 197 | 2,340 | $<0.1$ | 0 | I | I-C | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo} \mathrm{No}. \mathrm{1}$, | W | 402 | 2,370 | $<0.1$ | 0 | 8 | C | INCO (3) |
| Ni-Cr-Cu-Mo No. 1, cast | S | 402 | 2,370 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| Ni-Cr-Cu-Mo No. 1, cast | W | 181 | 5 | 0.5 | 0 | 20 | C | INCO (3) |
| Ni-Cr-Cu-Mo No. 1, cast | W | 366 | 5 | $<0.1$ | 0 | 1 | I-C | INCO (3) |
| Ni-Cr-Cu-Mo No. 2, cast | W | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo}$ No. 2, cast | S | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo} \mathrm{No}. \mathrm{2}$, | W | 403 | 6,780 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo} \mathrm{No}. \mathrm{2}$, | S | 403 | 6,780 | $<0.1$ | 0 | 1 | I-C | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo}$ No. 2, cast | W | 751 | 5,640 | $<0.1$ | 0 | I | I-C | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo} \mathrm{No}. \mathrm{2}$, | S | 751 | 5,640 | $<0.1$ | 0 | 0 | NC | 1 NCO (3) |
| Ni -Cr-Cu-Mo No. 2, cast | W | 1,064 | 5,300 | 0.1 | 0 | 3 | C | INCO (3) |
| Ni-Cr-Cu-Mo No. 2, cast | S | 1,064 | 5,300 | 0.1 | 0 | 5 | C | INCO (3) |
| Ni-Cr-Cu-Mo No. 2, cast | W | 197 | 2,340 | 0.1 | 0 | 1 | I-C | INCO (3) |
| Ni-Cr-Cu-Mo No. 2, cast | S | 197 | 2,340 | $<0.1$ | 0 | I | I-C | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo} \mathrm{No}. \mathrm{2}$, | W | 402 | 2,370 | 0.2 | 3 | 0 | P | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo} \mathrm{No}. \mathrm{2}$, | S | 402 | 2,370 | $<0.1$ | 0 | 1 | 1-C | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Cu}-\mathrm{Mo} \mathrm{No}. \mathrm{2}$, | W | 181 | 5 | 0.1 | 0 | 18 | C | INCO (3) |
| Ni-Cr-Cu-Mo No. 2, cast | W | 366 | 5 | 0.1 | 0 | 27 | C | INCO (3) |
| Ni-Cr-Mo, cast | W | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$, cast | S | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$, cast | W | 403 | 6,780 | $<0.1$ | 0 | I | 1-C | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$, cast | S | 403 | 6,780 | $<0.1$ | 0 | I | I-C | INCO (3) |
| Ni-Cr-Mo, cast | W | 751 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| Ni-Cr-Mo, cast | S | 751 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| Ni-Cr-Mo, cast | W | 1,064 | 5,300 | $<0.1$ | 0 | 0 | NC | INCO (3) |

Table 54. Continued.

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | Crevice Depth ${ }^{b}$ (mils) | Type ${ }^{\text {b }}$ |  |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$, cast | S | 1,064 | 5,300 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$, cast | w | 197 | 2,340 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$, cast | S | 197 | 2,340 | $<0.1$ | 0 | 1 | $\mathrm{I}-\mathrm{C}$ | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$, cast | W | 402 | 2,370 | $<0.1$ | 0 | I | I-C | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$, cast | S | 402 | 2,370 | <0.1 | 0 | 1 | I-C | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$, cast | W | 181 | 5 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}$, cast | W | 366 | 5 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | W | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | S | 123 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | W | 403 | 6,780 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | S | 403 | 6,780 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | W | 751 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | S | 751 | 5,640 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | W | 1,064 | 5,300 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | S | 1,064 | 5,300 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | W | 197 | 2,340 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | S | 197 | 2,340 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | W | 402 | 2,370 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| Ni -Cr-Mo-Si, cast | S | 402 | 2,370 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | w | 181 | 5 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| $\mathrm{Ni}-\mathrm{Cr}-\mathrm{Mo}-\mathrm{Si}$, cast | S | 366 | 5 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| RL-35-100, cast | W | 123 | 5,640 | 0.3 | 0 | 0 | G | INCO (3) |
| RL-35-100, cast | S | 123 | 5,640 | 0.5 | 0 | 0 | G | INCO (3) |
| RL-35-100, cast | W | 403 | 6,780 | $<0.1$ | 0 | 0 | G | INCO (3) |
| RL-35-100, cast | S | 403 | 6,780 | 0.3 | 0 | 0 | G | INCO (3) |
| RL-35-100, cast | W | 751 | 5,640 | $<0.1$ | 0 | 0 | U-ET | INCO (3) |
| RL-35-100, cast | S | 751 | 5,640 | 0.3 | 0 | 0 | ET | INCO (3) |
| RL-35-100, cast | w | 1,064 | 5,300 | 0.7 | 0 | 0 | G | INCO (3) |
| RL-35-100, cast | S | 1,064 | 5,300 | 0.1 | 0 | 0 | G | INCO (3) |
| RL-35-100, cast | W | 197 | 2,340 | $<0.1$ | 0 | 0 | NC | INCO (3) |
| RL-35-100, cast | S | 197 | 2,340 | $<0.1$ | 0 | 0 | NU-ET | INCO (3) |

${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimen were embedded in the bottom sediments.
${ }^{b}$ Symbols for types of corrosion:

| C | $=$ Crevice | NC | $=$ No visible corrosion |
| :--- | :--- | :--- | :--- |
| E | $=$ Edge | $\mathrm{NU}=$ Nonuniform |  |
| ET | $=$ Etched | P | $=$ Pitted |
| G | $=$ General | SL | $=$ Slight |
| I | $=$ Incipient | U | $=$ Uniform |

[^17]Table 55. Stress Corrosion of Miscellaneous Stainless Steels

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure <br> (day) | Depth <br> $(\mathrm{ft})$ | Number of <br> Specimens <br> Exposed | Number <br> Failed | Source $^{\boldsymbol{a}}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 20 Cb | 16 | 35 | 123 | 5,640 | 3 | 0 | CEL (4) |
| 20 Cb | 24 | 50 | 123 | 5,640 | 3 | 0 | CEL (4) |
| 20 Cb | 35 | 75 | 123 | 5,640 | 3 | 0 | CEL (4) |
| 20 Cb | 24 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| 20 Cb | 35 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| 20 Cb | 16 | 35 | 751 | 5,640 | 3 | 0 | CEL (4) |
| 20 Cb | 24 | 50 | 751 | 5,640 | 3 | 0 | CEL (4) |
| 20 Cb | 35 | 75 | 751 | 5,640 | 3 | 0 | CEL (4) |
| 20 Cb | 24 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| 20 Cb | 35 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| 20 Cb | 24 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 20 Cb | 35 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |

${ }^{a}$ Numbers refer to references at end of report.

Table 56. Changes in Mechanical Properties of Miscellaneous Stainless Steels

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {b }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| 20 Cb | S | 123 | 5,640 | 92 | +5 | 47 | +19 | 40 | -5 | CEL (4) |
| 20 Cb | W | 403 | 6,780 | 92 | +2 | 47 | +4 | 40 | -3 | CEL (4) |
| 20 Cb | S | 403 | 6,780 | 92 | +1 | 47 | +1 | 40 | -2 | CEL (4) |
| 20 Cb | W | 751 | 5,640 | 92 | +1 | 47 | 0 | 40 | -2 | CEL (4) |
| 20 Cb | S | 751 | 5,640 | 92 | +2 | 47 | +6 | 40 | -5 | CEL (4) |
| 20 Cb | S | 1,064 | 5,300 | 92 | +4 | 47 | +9 | 40 | -3 | CEL (4) |
| 20 Cb | W | 197 | 2,340 | 92 | -1 | 47 | -1 | 40 | +4 | CEL (4) |
| 20 Cb | S | 197 | 2,340 | 92 | 0 | 47 | +1 | 40 | -5 | CEL (4) |
| 20 Cb | w | 402 | 2,370 | 92 | 0 | 47 | -1 | 40 | -7 | CEL (4) |
| 20 Cb | S | 402 | 2,370 | 92 | +4 | 47 | +11 | 40 | -12 | CEL (4) |
| 20 Cb | w | 181 | 5 | 92 | +2 | 47 | +5 | 40 | -4 | CEL (4) |
| 20 Cb | W | 365 | 5 | 92 | -2 | 47 | -4 | 40 | +1 | CEL (4) |

${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimen were embedded in the bottom sediments.
${ }^{b}$ Numbers refer to references at end of report.

## SECTION 6

## ALUMINUM ALLOYS

The resistance of aluminum and its alloys to corrosion is due to a relatively chemically inert film of aluminum oxide which forms on its surface. As long as this oxide film remains intact the good corrosion resistance is preserved. In oxidizing environments where a sufficient amount of oxidizing agent or oxygen is present to repair any breaks in this protective film, the corrosion resistance of the aluminum alloys is maintained. The usual corrosion protection (passive) film that forms on aluminum in waters at temperatures below $70^{\circ} \mathrm{C}$ is bayerite $\left(\beta-\mathrm{Al}_{2} \mathrm{O}_{3}-3 \mathrm{H}_{2} \mathrm{O}\right)$.

In general, oxidizing conditions favor the preservation of this passive film, while reducing conditions destroy it. Chloride ions are particularly agressive in destroying this passive film.

When aluminum is immersed in water, the oxide film thickens much more rapidly than it does in air. The rate of growth decreases with time and reaches a limiting thickness which depends on the temperature, the oxygen content of the water, the ions present, and the pH . In seawater this naturally formed protective film breaks down more readily, and its repair and growth are retarded by the chloride ion.

The corrosion of aluminum alloys in seawater is usually of the pitting and crevice types. Pits begin by breakdown of the protective film at weak spots or at nonhomogeneities. The breakdown is followed by the formation of an electrolytic cell, the anode of which is a minute area of active metal and the cathode of which is a considerable area of passive metal. The large potential difference of this "passive-active" cell accounts for the considerable flow of current with its attendant rapid corrosion at the small anode (pitting).

Pitting is most likely to occur in the presence of chloride ions (for example, in seawater), combined with such cathodic depolarizers as oxygen or oxidizing salts. An oxidizing environment is usually necessary for preservation of a passive protective film with accompanying high corrosion resistance, but, unfortunately, it is also a condition for the occurrence of pitting. The oxidizer can often act as a depolarizer for "passive-active" cells established by
the breakdown of passivity at a specific point or area. The chloride ion in particular can accomplish this breakdown.

As discussed above, aluminum alloys generally corrode in seawater by pitting and crevice corrosion; therefore, as much as 90 to $95 \%$ of the exposed surface can be uncorroded. With such low percentages of the total exposed area affected, corrosion rates calculated from weight losses as mils penetration per year (mpy) can give a very misleading picture. The mpy implies an uniform decrease in thickness, which for aluminum alloys is not the case.

Another manifestation of localized attack in aluminum alloys is oxygen concentration cell corrosion in crevices (usually known as crevice corrosion). This type of corrosion occurs underneath deposits of any kind on the metal surface, underneath barnacles, and at the faying surfaces of joints. The area of the aluminum alloys which is shielded from the surrounding solution becomes deficient in oxygen, thus creating a difference in oxygen concentration between the shielded and unshielded areas. An electrolytic cell is created with a difference in electrical potential being generated between the high and low oxygen concentration areas; the low concentration area becomes the anode of the cell. Corrosion occurs at the small anodic area and, because the cathodic area is much larger, the rate of attack is considerably greater than if no such cell were present.

There are two other types of localized corrosion often found in aluminum alloys: intergranular and exfoliation. Intergranular (intercrystalline) attack is selective corrosion of grain boundaries or closely adjacent regions without appreciable attack of the grains or crystals themselves. Exfoliation is a lamellar form of corrosion, resulting from a rapid lateral attack along grain boundaries or striations within the grains parallel to the metal surface. This directional attack results in a leafing action, aggravated by the voluminous corrosion products that causes the uncorroded strata to be split apart.

Low weight losses and low corrosion rates accompany these manifestations of localized
corrosion. Thus, the integrity of an aluminum alloy structure will be jeopardized if designed solely on the basis of corrosion rates calculated from weight losses rather than on the basis of measured depths of pits and depths of crevice corrosion. Pitting and crevice corrosion can, and do, penetrate aluminum alloys rapidly in seawater, thus rendering them useless in short periods of time.

Therefore, corrosion rates expressed as mils penetration per year calculated from weight losses, maximum pit depths, maximum depths of crevice corrosion and other type of corrosion are tabulated to provide an overall picture of the corrosion of the aluminum alloys.

### 6.1. 1000 SERIES ALUMINUM ALLOYS (99.00\% MINIMUM ALUMINUM)

The chemical compositions of the 1000 Series aluminum alloys are given in Table 57, their corrosion rates and type of corrosion in Table 58, their stress corrosion behavior in Table 59, and the effect of exposure on their mechanical properties in Table 60.

The 1000 Series aluminum alloys contain a minimum of $99 \%$ aluminum and are considered unalloyed aluminums.

The 1000 Series aluminum alloys corroded by the localized types of corrosion, pitting, and crevice.

### 6.1.1. Duration of Exposure

The corrosion rates of 1100 alloy decreased with increasing duration of exposure at the surface, in seawater at the 2,500 -foot depth, and in the bottom sediments at the 6,000 -foot depth, while the reverse occurred in the bottom sediments at the 2,500 -foot depth and in the seawater at the 6,000 -foot depth. There was no correlation between the severity of crevice corrosion and duration of exposure. The same was true for the severity of pitting corrosion, except at the surface where the maximum depth of pitting corrosion increased with increasing duration of exposure over a period of 1 year.

The corrosion of 1180 alloy was comparable to that of the 1100 alloy.

### 6.1.2. Effect of Depth

The corrosion rates of 1100 alloy increased with increasing depth after 1 year of exposure. However, there were no correlations between maximum depths of pitting and crevice corrosion and corrosion rates. In general, pitting and crevice corrosion were more severe at depth than at the surface.

There was no definite effect of depth on the corrosion of 1000 Series aluminum alloys.

### 6.1.3. Effect of Concentration of Oxygen

Changes in the concentration of oxygen in seawater had no definite or consistent effect on the corrosion of 1100 aluminum alloy. In general, after 1 year of exposure the corrosion rates and severity of crevice corrosion were greater at the lower oxygen concentrations, while the severity of pitting corrosion was greatest at the highest oxygen concentration.

### 6.1.4. Stress Corrosion

Alloys 1100 and 1180 were exposed at the 2,500-foot depth for 402 days when stressed at values equivalent to 50 and $75 \%$ of their respective yield strengths (Table 59) to determine their susceptibilities to stress corrosion. They were not susceptible to stress corrosion under the conditions of the test.

### 6.1.5. Mechanical Properties

The effects of exposure on the mechanical properties of 1100 and 1180 alloys are given in Table 60. Their mechanical properties were not affected by exposure in seawater at the 2,500 -foot depth for 402 days.

### 6.2. 2000 SERIES ALUMINUM ALLOYS (ALUMINUM-COPPER ALLOYS)

The chemical compositions of the 2000 Series aluminum alloys are given in Table 61, their corrosion rates and type of corrosion in Table 62, their stress corrosion behavior in Table 63, and the effect of exposure on their mechanical properties in Table 64.

The 2000 Series aluminum alloys contain copper as the chief alloying element. Copper is one of the most important alloying metals for aluminum because of its appreciable solubility and its strengthening effect.

The 2000 Series alloys corroded by pitting, crevice, intergranular, and exfoliation types of corrosion.

### 6.2.1. Duration of Exposure

There was no definite or consistent correlation between corrosion rates and types of corrosion of the 2000 Series alloys and duration of exposure.

### 6.2.2. Effect of Depth

In general, corrosion rates were greater, and pitting, crevice, and intergranular corrosion were more severe at depth than at the surface after 1 year of exposure. Thus, seawater at depth is more aggressive to the 2000 Series aluminum alloys than is seawater at the surface.

### 6.2.3. Effect of Concentration of Oxygen

The effect of changes in the concentration of oxygen in seawater on the corrosion behavior of the 2000 Series alloys was inconsistent and erratic except for alloy $2219-\mathrm{T} 81$. The corrosion rates, maximum depths of pits, and maximum depths of crevice corrosion decreased with increasing oxygen concentration, but not linearly, after 1 year of exposure. This behavior of alloy $2219-\mathrm{T} 81$ shows that the concentration of oxygen in seawater exerts considerable influence on the corrosion of this alloy.

### 6.2.4. Stress Corrosion

The 2000 Series aluminum alloys were exposed at the depths and for the times shown in Table 63 when stressed at values equivalent to 30,50 , or $75 \%$ of their respective yield strengths to determine their susceptibilities to stress corrosion. They were not susceptible to stress corrosion under the test conditions.

### 6.2.5. Other Types of Corrosion

Alloys 2014-T3, 2014-T6, 2024-T3, 2024-T81, 2219-T81, and 2219-T87 were attacked by intergranular corrosion. Alloys 2014-T3, 2024-T3, 2024-T6, 2024-T81, and 2219-T81 were attacked by the exfoliation type of corrosion.

### 6.2.6. Welding

Welding did not affect the corrosion behavior of aluminum alloys 2024-T3 and 2219-T81.

### 6.2.7. Mechanical Properties

The effects of exposure on the mechanical properties of the 2000 Series aluminum alloys are given in Table 64. The mechanical properties of the 2000 Series alloys were impaired except for those of alloy Alclad 2024-T3.

### 6.3. 3000 SERIES ALUMINUM ALLOYS (ALUMINUM-MANGANESE ALLOYS)

The chemical compositions of the 3000 Series aluminum alloys are given in Table 65, their corrosion rates and types of corrosion in Table 66, their stress corrosion behavior in Table 67, and the effect of exposure on their mechanical properties in Table 68.

The chief alloying element of the 3000 Series aluminum alloys is manganese. Manganese is added to aluminum in amounts above $1 \%$ to increase its strength.

The 3000 Series alloys corroded chiefly by the crevice and pitting types of localized corrosion. There was also some blistering of the Alclad 3003 alloy.

### 6.3.1. Duration of Exposure

The corrosion rates of alloys 3003 and Alclad 3003 neither increased nor decreased uniformly with increasing duration of exposure, except for Alclad 3003 at the 2,500-foot depth. At this depth the corrosion rates decreased with increasing duration of exposure. In general, the severity of pitting and crevice corrosion was greater after the longer times of exposure.

The corrosion behavior of the 3000 Series alloys was erratic and unpredictable with regard to duration of exposure.

### 6.3.2. Effect of Depth

After 1 year of exposure the corrosion rates and maximum depths of pits increased with increasing depth, but not linearly. Alclad 3003 did not behave in this manner. In other words, the corrosion behavior of alloy 3003 appears to be depth (pressure) dependent in that it increased in severity with increasing depth.

### 6.3.3. Effect of Concentration of Oxygen

The corrosion rates, maximum pit depths, and maximum depths of crevice corrosion on alloys 3003 and Alclad 3003 due to changes in the concentration of oxygen in seawater were erratic.

### 6.3.4. Stress Corrosion

Alloy 3003 -H14 was not susceptible to stress corrosion when stressed at values equivalent to 50 and $75 \%$ of its yield strength and exposed at the 2,500-foot depth for 402 days as given in Table 67.

### 6.3.5. Corrosion Products

Corrosion products from alloy 3003 -H14 were analyzed by X-ray diffraction, spectrographic analysis, quantitative chemical analysis, and infra-red spectrophotometry. The qualitative results were: amorphous $\mathrm{Al}_{2} \mathrm{O}_{3} \cdot \mathrm{XH}_{2} \mathrm{O}, \mathrm{NaCl}, \mathrm{SiO}_{2}, \mathrm{Al}, \mathrm{Na}, \mathrm{Si}, \mathrm{Mg}$, $\mathrm{Fe}, \mathrm{Cu}, \mathrm{Ca}, \mathrm{Mn}, 3.58 \%$ chloride ion, $18.77 \%$ sulfate ion, and considerable phosphate ion.

### 6.3.6. Mechanical Properties

The effects of exposure on the mechanical properties of alloys $3003-\mathrm{H} 14$, Alclad $3003-\mathrm{H} 12$, and Alclad 3003-H14 are given in Table 68. In general, the mechanical properties of alloys $3003-\mathrm{H} 14$ and Alclad 3003 -H12 were adversely affected by exposure at depth.

### 6.4. 5000 SERIES ALUMINUM ALLOYS (ALUMINUM-MAGNESIUM ALLOYS)

The chemical compositions of the 5000 Series aluminum alloys are given in Table 69, their corrosion rates and types of corrosion in Table 70, their stress corrosion behavior in Table 71, and the effect of exposure on their mechanical properties in Table 72.

Aluminum is alloyed with magnesium to form an important class of nonheat-treatable alloys (5000 Series). Their utility and importance are based on their resistance to corrosion, high strength without heat treatment, and good weldability.

The 5000 Series aluminum alloys corroded chiefly by the crevice and pitting types of localized corrosion. Other types of corrosion found were: blistering, crater, edge, intergranular, line, and exfoliation.

### 6.4.1. Duration of Exposure

The general effect of duration of exposure on the corrosion of the 5000 Series alloys was erratic and nonuniform. The corrosion rates and the maximum depths of pitting or crevice corrosion neither increased nor decreased consistently with increasing duration of exposure; in many cases, the behavior was erratic.

### 6.4.2. Effect of Depth

After 1 year of exposure the average corrosion rates of all the 5000 Series alloys increased with depth, but not linearly. Also, the maximum depths of pits of all the alloys increased linearly with depth. The maximum depth of crevice corrosion of all the alloys increased with depth, but not consistently. The corrosion behavior of the 5000 Series aluminum alloys appears to be more uniformly affected by depth than by duration of exposure or changes in the concentration of oxygen in seawater.

### 6.4.3. Effect of Concentration of Oxygen

The cor-osion rates of alloy $5086-\mathrm{H} 34$ increased linearly with increasing concentration of oxygen in
seawater, but the slope of the line was very small (1 to 25 ). However, such relationships were not found for the maximum depths of pitting and crevice corrosion. The pit depths were a maximum at the highest oxygen concentration, and the maximum depth of crevice corrosion was at the intermediate oxygen concentration.

The corrosion rates of alloy 5456-H321 decreased linearly with increasing concentration of oxygen in seawater, but the slope of the line was very small (1 to 10). However, no correlations were possible between maximum depth of pitting and crevice corrosion.

The corrosion rates and changes in the maximum depths of pits and crevice corrosion of the other 5000 Series aluminum alloys were erratic and inconsistent with respect to changes in the concentration of oxygen in seawater. Changes in the concentration of oxygen in seawater did not exert a constant or uniform influence on the corrosion behavior of the 5000 Series aluminum alloys. This behavior, like that of the stainless steels and some nickel alloys, can be attributed to the dual role oxygen can play with regard to alloys which depend upon passive films for their corrosion resistance.

### 6.4.4. Stress Corrosion

Some 5000 Series aluminum alloys were exposed at the depths and for the times given in Table 71 when stressed at values equivalent to 30,50 , or $75 \%$ of their respective yield strengths to determine their susceptibilities to stress corrosion. They were not susceptible to stress corrosion under the test conditions.

### 6.4.5. Other Types of Corrosion

Alloys $5052-\mathrm{H} 32$ and $5456-\mathrm{H} 34$ were attacked by the exfoliation type of corrosion. Alloys 5083-H113, $5086-\mathrm{H} 32$, and $5086-\mathrm{H} 34$ were attacked by intergranular corrosion.

### 6.4.6. Welding

Welding did not affect the corrosion behavior of alloys $5083-\mathrm{H} 113,5086-\mathrm{H} 34$, and $5454-\mathrm{H} 32$.

### 6.4.7. Corrosion Products

Corrosion products from alloy 5086 were analyzed by X-ray diffraction, spectrographic analysis, quantitative chemical analysis, and infra-red spectrophotometry. The qualitative results were: amorphous $\mathrm{Al}_{2} \mathrm{O}_{3} \cdot \mathrm{XH}_{2} \mathrm{O}, \mathrm{NaCl}, \mathrm{SiO}_{2}, \mathrm{Al}, \mathrm{Na}, \mathrm{Mg}$, $\mathrm{Cu}, \mathrm{Fe}, \mathrm{Si}, \mathrm{Ti}, 5.8 \%$ chloride ion, $26.2 \%$ sulfate ion, and considerable phosphate ion.

### 6.4.8. Mechanical Properties

The effects of exposure on the mechanical properties of the 5000 Series aluminum alloys are given in Table 72. The mechanical properties of the following alloys were adversely affected by exposure: 5456-H321 after 123 days of exposure at the 6,000 -foot depth; $5052-\mathrm{H} 32,5083-\mathrm{H} 113$, and $5456-\mathrm{H} 34$ after 403 days of exposure at the 6,000-foot depth; and $5456-\mathrm{H} 321$ and $5456-\mathrm{H} 34$ after 751 days of exposure at the 6,000 -foot depth. The mechanical properties of the above alloys after exposures for different times at different depths and of the other alloys were not adversely affected by exposure at depth in the seawater.

### 6.5. 6000 SERIES ALUMINUM ALLOYS (ALUMINUM-MAGNESIUM-SILICON ALLOYS)

The chemical compositions of the 6000 Scries aluminum alloys are given in Table 73, their corrosion rates and types of corrosion in Table 74, their stress corrosion behavior in Table 75, and the effects of exposure on their mechanical properties in Table 76.

The aluminum-magnesium-silicon system is the basis for a major class of heat-treatable aluminumbase alloys. They combine many desirable characteristics, including moderately high strength and good resistance to corrosion.

There was only one 6000 Series alloy (6061) in this program. Alloy 6061 corroded chiefly by the crevice and pitting types of localized corrosion. Also, there was some intergranular corrosion.

### 6.5.1. Duration of Exposure

The corrosion rates of 6061 at the surface and at the 6,000 -foot depth decreased with duration of
exposure, but not uniformly, while those at the 2,500-foot depth increased with duration of exposure. However, the maximum depths of pitting and crevice corrosion increased with increasing duration of exposure at the surface and at depths of 2,500 and 6,000 feet.

### 6.5.2. Effect of Depth

Although the corrosion rates and the maximum depths of pitting and crevice corrosion were greater at depth than at the surface, these increases did not increase uniformly with increasing, depth. Depth exerted no uniform influence on the corrosion behavior of alloy 6061 .

### 6.5.3. Effect of Concentration of Oxygen

The corrosion rates and maximum depths of pitting and crevice corrosion decreased with increasing concentration of oxygen in seawater. The maximum depths of crevice corrosion decreased linearly with increasing oxygen concentration. The corrosion rates and maximum depths of pitting decreased constantly, but not uniformly, with depth.

### 6.5.4. Stress Corrosion

Alloy 6061-T6 was exposed at the depths and for the times given in Table 75 when stressed at values equivalent to 30 and $75 \%$ of its yield strength to determine its susceptibility to stress corrosion. Alloy 6061-T6 was not susceptible to stress corrosion under the test conditions.

### 6.5.5. Welding

The corrosion of alloy 6061-T6 was adversely affected by welding. Alloy 6061 was attacked by intergranular corrosion in the "as-welded" condition.

### 6.5.6. Mechanical Properties

The effects of exposure on the mechanical properties of alloy 6061-T6 are given in Table 76. The mechanical properties of 6061-T6 were adversely affected by exposure in seawater. Those specimens which had been welded and which had been attacked by intergranular corrosion were the most seriously affected.

### 6.6. 7000 SERIES ALUMINUM ALLOYS (ALUMINUM-ZINC-MAGNESIUM ALLOYS)

The chemical compositions of the 7000 Series aluminum alloys are given in Table 77, their corrosion rates and types of corrosion in Table 78, their stress corrosion behavior in Table 79, and the effect of exposure on their mechanical properties in Table 80.

Combinations of zinc and magnesium in aluminum provide a class of heat-treatable alloys, some of which develop the highest strengths presently known for commercial aluminum-base alloys. The addition of copper to the aluminum-zinc-magnesium system, together with small but important amounts of chromium and manganese, results in the highest strength, heat-treatable, aluminum-base alloys commercially available.

The 7000 Series alloys were attacked by crevice, edge, exfoliation, intergranular, and pitting types of corrosion. Corrosion of the Alclad alloys was by shallow pitting and crevice corrosion, slight blistering, and general corrosion.

Because of the erratic behavior of the 7000 Series aluminum alloys during exposure in seawater at depth, it was impossible to find any correlation between their corrosion behavior and duration of exposure, effect of depth, or the effect of changes in the concentration of oxygen in seawater.

A practical case of unusual corrosion on an aluminum alloy was encountered with the Alclad 7178-T6 aluminum alloy buoys used in the installation of the STU structures. During the retrieval of STU 1-3 after 123 days of exposure, the buoy, which was 300 feet below the surface, was found to be corroded. White corrosion products on the bottom hemisphere covered areas where the cladding alloy had corroded through to the core material. The top hemisphere was blistered, the blisters being as large as 2 inches in diameter and 0.75 inch high with a hole in the top of each blister. The hole in the top of the blister indicates the origin of the failure: originally a pinhole in the cladding alloy existed where seawater gained access to the interface between the cladding alloy and the core alloy. When this blister was sectioned to inspect the corrosion underneath, it was found to be filled with white crystalline aluminum oxide corrosion products. It appeared that seawater penetrated the cladding alloy at a defect, or a pit was initiated at a particle of a cathodic metal (probably
iron), and the corrosion was then concentrated at the interface between the two alloys (cladding alloy and core alloy). The thickness of the remaining Alclad layer indicated that it had not been sacrificed to protect the core alloy as was its intended function. On the other hand, the selective corrosion of the Alclad layer on the bottom hemisphere and the uncorroded core material showed that, in this case, the cladding alloy was being sacrificed to protect the core material as intended.

When an attempt was made to repair these buoys for reuse by grinding off all traces of corrosion prior to painting, it was found that the corrosion had penetrated along the interface between the cladding alloy and the core alloy for considerable distances from the edges of the blisters and the edges of the holes where the cladding alloy layer had been sacrificed. Polished transverse sections taken from the buoy through these corroded areas corroborated the indications found from grinding operations. Metallurgical examinations showed that the corroded paths were, in fact, entirely in the cladding alloy, with a thin diffusion layer of material between the corrosion path and the core material.

Blistering of Alclad aluminum alloys such as encountered with these Alclad 7178-T6 spheres was very unusual. Blistering due to corrosion and the rapid rate of sacrifice of Alclad layers had not been encountered previously by the author and other investigators in surface seawater applications. Because of this unique blistering one of the spheres was sent to the Research Laboratories of the Aluminum Company of America where an investigation was made to determine the mechanism of this behavior.

Wei [15] showed that there was preferential diffusion of zinc over copper from the core alloy into this interfacial zone. The high zinc and low copper contents of this interfacial zone rendered it anodic to both the cladding and core alloys. Selective attack was inevitable once corrosion reached this anodic diffusion zone.

That this type of blistering has been encountered on buoys at depths from 300 to 6,800 feet emphasizes the fact that there is some factor present which either is more influential at depth or is not present at the surface. The fact that this thin anodic zone is probably present in all Alclad 7178-T6 products and, as such, is not blistered during surface
seawater exposures indicates that the seawater environments at depths of 300 feet and greater differ from the seawater environments at the surface, at least with respect to the corrosion behavior of this alloy.

### 6.6.1. Stress Corrosion

The 7000 Series aluminum alloys were exposed at the depths and for the times given in Table 79 when stressed at values equivalent to 30,50 , and $75 \%$ of their respective yield strengths to determine their susceptibilities to stress corrosion. Alloys 7075-T6, 7079-T6, Alclad 7079-T6, and 7178-T6, failed by stress corrosion cracking.

### 6.6.2. Corrosion Products

Corrosion products from alloy 7079-T6 were analyzed by X-ray diffraction, spectographic analysis, quantitative chemical analysis, and infra-red spectrophotometry. The qualitative results were: amorphorous $\mathrm{Al}_{2} \mathrm{O}_{3} \cdot \mathrm{XH}_{2} \mathrm{O}, \mathrm{NaCl}, \mathrm{Al}$ metal, $\mathrm{Al}, \mathrm{Cu}, \mathrm{Mg}$, $\mathrm{Mn}, \mathrm{Zn}, \mathrm{Na}, \mathrm{Ca}$, traces of Ti and $\mathrm{Ni}, 2.82 \%$ chloride ion, $16.7 \%$ sulfate ion, and considerable phosphate ion.

### 6.6.3. Mechanical Properties

The effects of exposure on the mechanical properties of the 7000 Series aluminum alloys are given in Table 80. The mechanical properties of alloys 7002-T6, 7039-T6, 7075-T6, 7075-T64, 7075-173, 7079-T6, and 7178-T6 were adversely affected.

Table 57. Chemical Composition of 1000 Series Aluminum Alloys, Percent by Weight

| Alloy | Gage <br> (in.) | Si | Fe | Cu | Mn | Zn | $\mathrm{Al}^{a}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 1100 | - | - | - | - | - | - | 99.0 |
| $1100-0$ | - | $b$ | $b$ | 0.14 | 0.03 | - | R |
| $1100-\mathrm{H} 14$ | 0.050 | 0.14 | 0.55 | 0.14 | - | 0.06 | R |
| 1180 | 0.050 | 0.06 | 0.08 | 0.002 | 0.002 | - | R |

${ }^{a} \mathrm{R}=$ remainder.
${ }^{b} \mathrm{Si}+\mathrm{Fe}=0.57$

Table 58. Corrosion Rates and Types of Corrosion of 1000 Series Aluminum Alloys

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth (mils) | $\begin{aligned} & \text { Crevice } \\ & \text { Depth } \\ & \text { (mils) } \end{aligned}$ | Type ${ }^{\text {b }}$ |  |
| 1100-H14 | W | 123 | 5,640 | 2.0 | 39 | 0 | P | INCO (3) |
| 1100-H14 | S | 123 | 5,640 | 3.1 | 41 | 0 | P | INCO (3) |
| 1100-H14 | W | 403 | 6,780 | 4.2 | 0 | 62 | C (PR) | INCO (3) |
| 1100-H14 | S | 403 | 6,780 | 1.3 | 62 | 62 | C (PR); P (PR) | INCO (3) |
| 1100-H14 | W | 751 | 5,640 | 4.5 | 0 | 62 | C (PR) | INCO (3) |
| 1100-H14 | S | 751 | 5,640 | 2.0 | 0 | 62 | C (PR) | INCO (3) |
| 1100-H14 | W | 1,064 | 5,300 | 3.0 | 0 | 62 | C (PR) | INCO (3) |
| 1100-H14 | w | 1,064 | 5,300 | 1.8 | S-P | S-C | $\mathrm{S}-\mathrm{C} ; \mathrm{S}-\mathrm{P}$ | CEL (4) |
| $1100-\mathrm{H} 14$ | S | 1,064 | 5,300 | 1.0 | 0 | 62 | C (PR) | INCO (3) |
| 1100-H14 | W | 197 | 2,340 | 5.6 | 0 | 62 | C (PR) | INCO (3) |
| 1100-H14 | W | 197 | 2,340 | $<0.1$ | 23 | 0 | E; P | REY (14) |
| 1100-H14 | S | 197 | 2,340 | <0.1 | I | 1 | ${ }_{\text {I-C; }}$ I-P | INCO (3) |
| 1100-H14 | W | 402 | 2,340 | 1.6 | 0 | 62 | C (PR) | INCO (3) |
| 1100-H14 | W | 402 | 2,370 | 0.9 | 1 | 50 | C (PR); I-P | CEL (4) |
| 1100-H14 | S | 402 | 2,370 | 0.5 | 26 | 26 | C; P | INCO (3) |
| 1100-H14 | S | 402 | 2,370 | 0.5 | I | 50 | C (PR) ; I-P | CEL (4) |
| 1100-H14 | w | 181 | 5 | 1.4 | 0 | S | S-C | INCO (3) |
| 1100-H14 | W | 366 | 5 | 0.6 | 13 | 13 | C; P | INCO (3) |
| 1180-H14 | w | 402 | 2,370 | 1.0 | 1 | 50 | C (PR); I-P | CEL (4) |
| 1180 | w | 402 | 2,370 | $<0.1$ | 41 | 0 | E; S-P | REY (14) |
| 1180-H14 | S | 402 | 2,370 | 0.8 | 1 | 50 | C (PR) ; I-P | CEL (4) |

${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimen were embedded in the bottom sediments.
${ }^{b}$ Symbols for types of corrosion:
$\mathrm{C}=$ Crevice
$\mathrm{P} \quad=$ Pitting
$\mathrm{E}=$ Edge
PR = Perforated
I $=$ Incipient
$\mathrm{S}=$ Severe
${ }^{c}$ Numbers refer to references at end of report.

Table 59. Stress Corrosion of 1000 Series Aluminum Alloys

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure <br> (day) | Depth <br> $(\mathrm{ft})$ | Number of <br> Specimens <br> Exposed | Number <br> Failed | Source $^{a}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $1100-\mathrm{H} 14$ | 8 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $1100-\mathrm{H14}$ | 12 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 1180 | 6 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 1180 | 10 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |

${ }^{a}$ Numbers refer to references at end of report.

Table 60. Changes in Mechanical Properties of 1000 Series Aluminum Alloys Due to Corrosion

| Alloy | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| 1100-H14 | 402 | 2,370 | 19 | -2 | 18 | +3 | 7 | +14 | CEL (4) |
| 1180 | 402 | 2,370 | 15 | +1 | 14 | +4 | 11 | +1 | CEL ( + ) |

${ }^{a}$ Numbers refer to references at end of report.

Table 61. Chemical Composition of 2000 Series Aluminum Alloys, Percent by Weight

| Alloy | Gage <br> (in.) | Si | Fe | Cu | Mn | Mg | Cr | Ni | Zn | Ti | Other | $\mathrm{Al}^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 2014-Te | - | 0.80 | 0.34 | 3.90 | 0.70 | 0.50 | 0.01 | 0.01 | 0.03 | - | - | R |
| 2014-T6 | 0.050 | 0.91 | 0.53 | 4.23 | 0.80 | 0.32 | 0.03 | 0.01 | 0.08 | 0.02 | - | R |
| 2014-T6 | - | 0.84 | 0.35 | 4.43 | 0.80 | 0.66 | - | - | - | - | - | R |
| 2024 | 0.064 | - | - | 4.3 | 0.6 | 1.5 | - | - | - | - | - | R |
| $\begin{aligned} & 2024-\mathrm{T} 3+\mathrm{T} 81 \\ & \text { Alclad 2024-T3 } \end{aligned}$ | - | 0.20 | 0.20 | 4.50 | 0.80 | 1.50 | 0.02 | 0.04 | 0.06 | 0.01 | - | R |
| 2219-T81 | 0.064 | 0.20 | 0.30 | 6.3 | 0.30 | $<0.02$ | - | - | 0.10 | 0.06 | $\begin{aligned} & 0.10 \mathrm{~V} \\ & 0.17 \mathrm{Zr} \end{aligned}$ | R |
| 2219-T81 | - | 0.05 | 0.15 | 4.00 | 0.10 | 0.05 | 0.05 | 0.10 | 0.05 | 0.05 | - | R |
| 2219-T87 | 0.040 | 0.08 | 0.12 | 6.54 | 0.26 | $<0.02$ | $<0.02$ | $<0.02$ | 0.03 | 0.02 | $\begin{aligned} & 0.10 \mathrm{~V} \\ & 0.15 \mathrm{Zr} \end{aligned}$ | R |
| 2219-T87 | 0.040 | - | - | 6.3 | 0.30 | - | - | - | - | 0.06 | - | R |

${ }^{a} \mathrm{R}=$ remainder.
$b_{\text {No analysis given. }}$
Table 62. Corrosion Rates and Types of Corrosion of 2000 Series Aluminum Alloys

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum <br> Pit Depth <br> (mils) | Crevice Depth (mils) | Type ${ }^{\text {b }}$ | Weld ${ }^{\text {b }}$ |  |
| 2014-T3 | W | 123 | 5,640 | - | 7 | 0 | B; IG; P; XF | - | NADC (7) |
| 2014-T3 | W | 751 | 5,640 | - | PR | 0 | P (PR) | - | NADC (7) |
| 2014-T6 | W | 403 | 6,780 | - | S | S | S-C; S-P | - | NADC (7) |
| 2014-T6 | W | 1,064 | 5,300 | - | S | SH | SH-C; S-P | - | NADC (7) |
| 2014-T6 | S | 1,064 | 5,300 | - | PR | - | P (PR) | - | NADC (7) |
| 2014-T6 | W | 197 | 2,340 | - | 10 | 0 | IG; U-P; L | - | NADC (7) |
| 2014-T6 | W | 197 | 2,340 | $<0.1$ | 20 | 0 | P | - | REY (14) |
| 2014-T6 | W | 402 | 2,370 | 5.4 | 27 | 50 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{IG} ; \mathrm{P}$ | - | CEL (4) |
| 2014-T6 | S | 402 | 2,370 | 3.5 | 25 | 50 | C (PR); IG; P | - | CEL (4) |
| 2024-0 | W | 123 | 5,640 | 5.2 | 28 | 0 | P | - | INCO (3) |
| 2024-0 | S | 123 | 5,640 | 3.4 | 27 | 0 | P | - | INCO (3) |
| 2024-0 | W | 403 | 6,780 | 6.2 | 62 | 0 | P (PR) | - | INCO (3) |
| 2024-0 | S | 403 | 6,780 | 1.2 | 0 | 55 | S-C | - | INCO (3) |
| 2024-0 | W | 751 | 5,640 | 2.8 | 65 | 65 | C (PR); P (PR) | - | INCO (3) |
| 2024.0 | S | 751 | 5,640 | 2.6 | 65 | 0 | P (PR) | - | INCO (3) |
| 2024-0 | W | 1,064 | 5,300 | 1.9 | 62 | 62 | C (PR); P (PR) | - | INCO (3) |
| 2024-0 | S | 1,064 | 5,300 | 1.6 | 55 | 0 | S-P | - | INCO (3) |
| 2024-0 | W | 197 | 2,340 | 3.1 | 19 | 0 | S-E; P | - | INCO (3) |
| 2024-0 | S | 197 | 2,340 | $<0.1$ | I | 1 | I-C; I-P | - | INCO (3) |
| 2024-0 | W | 402 | 2,370 | 3.0 | 62 | 62 | C (PR); P (PR) | - | INCO (3) |
| 2024-0 | S | 402 | 2,370 | 0.8 | 0 | 0 | E | - | INCO (3) |
| 2024-0 | W | 181 | 5 | 3.8 | 32 | 32 | C; P | - | InCO (3) |
| 2024-0 | w | 366 | 5 | 4.1 | 34 | 34 | C; P | - | INCO (3) |
| 2024-T3 | W | 403 | 6,780 | - | 39 | 0 | S-E; S-P | - | NADC (7) |
| 2024-T3, anodized | W | 751 | 5,640 | - | 0 | 0 | G | - | NADC (7) |
| 2024-T3 | W | 1,064 | 5,300 | 1.9 | - | - | C; P; XF | - | CEL (4) |
| 2024-T3 ${ }^{e}$ | W | 197 | 2,340 | - | 30 | 0 | SL-B; IG; U-P | HAZ (IG) | NADC (7) |
| 2024-T3 | W | 402 | 2,370 | - | 39 | 0 | E; L; P | - | NADC (7) |
| Alclad 2024-T3 | W | 123 | 5,640 | - | 1 | 0 | I-P | - | NADC (7) |
| Alclad 2024-T3 | W | 751 | 5,640 | - | 0 | 0 | G | - | NADC (7) |
| 2024-T6 | W | 403 | 6,780 | - | - | - | L; D-P; XF | - | NADC (7) |


| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth ${ }^{b}$ (mils) | Crevice Depth ${ }^{b}$ (mils) | Type ${ }^{b}$ | Weld ${ }^{b}$ |  |
| 2024-T81 | W | 123 | 5,640 | - | 40 | 0 | B; EX-IG; CR; P | - | NADC (7) |
| 2024-T81 | w | 403 | 6,780 | - | 62 | 0 | S-P | - | NADC (7) |
| 2024-T81, anodized | W | 403 | 6,780 | - | 0 | 0 | XF | - | NADC (7) |
| 2024-T81, painted | W | 403 | 6,780 | - | 0 | 0 | NPF | - | NADC (7) |
| 2024-T81 | w | 402 | 2,370 | - | 39 | S | S-C; E; D-P; EX-XF | - | NADC (7) |
| 2024-T81, anodized | W | 402 | 2,370 | - | 0 | MD | MD-C; EX-XF | - | NADC (7) |
| 2219-781 ${ }_{\text {d }}$ | W | 123 | 5,640 | 2.0 | 16 | 15 | C; IG; G-P | - | CEL (4) |
| 2219-T81 ${ }^{\text {d }}$ | W | 123 | 5,640 | - | 40 | 0 | EX-IG; ${ }^{\text {; }}$; XF | NC | NADC (7) |
| 2219-T81 | S | 123 | 5,640 | 1.9 | 20 | 10 | C; IG; G-P | - | CEL (4) |
| 2219-T81 | W | 403 | 6,780 | 3.6 | 35 | 38 | C; E; IG; G-P | - | CEL (4) |
| 2219-T81 | S | 403 | 6,780 | 3.1 | 62 | 34 | C; E; IG; P (PR) | - | CEL (4) |
| 2219-T81 ${ }_{\text {d }}$ | W | 751 | 5,640 | 2.4 | 30 | 24 | C; E; IG; G-P | - | CEL (4) |
| 2219-T81 ${ }^{\text {d }}$ | w | 751 | 5,640 | - | PR | 0 | $\mathrm{P}(\mathrm{PR})$ | NC | NADC (7) |
| 2219-T81 | w | 1,064 | 5,300 | 1.6 | 47 | 47 | C; SL-E; IG; G-P | - | CEL (4) |
| 2219-T81 | S | 1,064 | 5,300 | 1.5 | 48 | 0 | IG; G-P | - | CEL (4) |
| 2219-T81 | W | 197 | 2,340 | 2.6 | 25 | 0 | SL-E; IG; G-P | - | CEL (4) |
| 2219-T81 | S | 197 | 2,340 | 2.0 | 32 | 0 | E; IG; G-P | - | CEL (4) |
| 2219-T81 | W | 402 | 2,370 | 4.5 | 78 | 69 | S-C; S-E; IG; S-P | - | CEL (4) |
| 2219-T81 | S | 402 | 2,370 | 2.0 | 50 | 66 | S-C; S-E; IG; S-P | - | CEL (4) |
| 2219-T81 | W | 181 | 5 | 3.5 | 24 | 0 | IG; G-P | - | CEL (4) |
| 2219-T81 | w | 398 | 5 | 2.5 | 26 | 0 | P | - | CEL (4) |
| 2219-T81 | W | 540 | 5 | 1.4 | 48 | 32 | C; P | - | CEL (4) |
| 2219-T81 | W | 588 | 5 | 4.4 | 62 | 43 | C; E; P | - | CEL (4) |
| 2219-T87 | W | 197 | 2,340 | $<0.1$ | 12 | 0 | P | - | REY (14) |
| 2219-T87 | w | 402 | 2,370 | 1.7 | 20 | 40 | C (PR); IG; U-P | - | CEL (4) |
| 2219-T87 | S | 402 | 2,370 | 1.1 | 16 | 40 | C (PR); IG; U-P | - | CEL (4) |

[^18]
$=$ Slight
$=$ Uniform
$=$ Exfolia
$\cdots \longmapsto \stackrel{a}{\infty}$
Table 62. Continued.

| PR | $=$ Perforated |
| ---: | :--- |
| S | $=$ Severe |
| SH | $=$ Shallow |

[^19]Table 63. Stress Corrosion of 2000 Series Aluminum Alloys

| Alloy | $\begin{aligned} & \text { Stress } \\ & (\mathrm{ksi}) \end{aligned}$ | $\begin{aligned} & \text { Percent } \\ & \text { Yield } \\ & \text { Strength } \end{aligned}$ | Exposure (day) | Depth <br> (ft) | Number of Specimens Exposed | Number Failed | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 2014-T6 | 17 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 2014-T6 | 41 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 2014-T6 | 31 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 2014-T6 | 46 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 2024-T3 | 15 | 30 | 403 | 6,780 | 3 | $b$ | NADC (7) |
| 2024-T3 | 38 | 75 | 403 | 6,780 | 3 | $b$ | NADC (7) |
| 2024-T3 | 15 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 2024-T3 | 38 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 2024-T81 | 15 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 2024-T81 | 38 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 2024-T81, anodized | 15 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 2024-T81, anodized | 38 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 2219-T81 | 25 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 2219-T81 | 38 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 2219-T87 | 25 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 2219-T87 | 38 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |

[^20]| Alloy | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original <br> (\%) | \% Change |  |
| 2014-T3 | 123 | 5,640 | 65 | -54 | - | - | 20 | -90 | NADC (7) |
| 2014-T6 | 123 | 5,640 | 72 | -16 | 55 | +8 | 20 | -90 | NADC (7) |
| 2014-T6 | 751 | 5,640 | 72 | -39 | 55 | -22 | 20 | -94 | NADC (7) |
| 2014-T6 | 1,064 | 5,300 | 72 | -19 | 55 | -40 | 20 | -95 | NADC (7) |
| 2014-T6 | 197 | 2,340 | 72 | -20 | 55 | -4 | 20 | -91 | NADC (7) |
| 2014-T6 | 402 | 2,370 | 68 | -38 | 62 | -28 | 11 | -92 | CEL (4) |
| 2024-T3 ${ }^{\text {3 }}$ | 403 | 6,780 | 62 | -35 | 42 | -10 | 17 | -89 | NADC (7) |
| 2024-T3 | 751 | 5,640 | 62 | +3 | 42 | -4 | 17 | -24 | NADC (7) |
| 2024-T3 ${ }^{\text {b }}$ | 197 | 2,340 | 62 | -100 | 42 | -100 | 17 | -100 | NADC (7) |
| 2024-T3 | 402 | 2,370 | 62 | -26 | 42 | -3 | 17 | -96 | NADC (7) |
| Alclad 2024-T3 | 123 | 5,640 | 62 | 0 | 42 | 0 | 17 | +12 | NADC (7) |
| 2024-T81 | 123 | 5,640 | 72 | -49 | - | - | 10 | -95 | NADC (7) |
| 2024-T81 | 403 | 6,780 | 72 | -10 | - | - | 10 | -58 | NADC (7) |
| 2024-T81 | 402 | 2,370 | 72 | -3 | - | - | 10 | -55 | NADC (7) |
| 2219-T81 ${ }^{\text {c,d }}$ | 123 | 5,640 | 62 | +5 | 50 | +2 | 8 | +38 | NADC (7) |
| 2219-T81 | 123 | 5,640 | 66 | -3 | 50 | -1 | 11 | -42 | CEL (4) |
| 2219-T81 | 403 | 6,780 | 66 | -19 | 50 | -12 | 11 | -52 | CEL (4) |
| 2219-T81 | 751 | 5,640 | 66 | -18 | 50 | -14 | 11 | -62 | CEL (4) |
| 2219-T81 | 1,064 | 5,300 | 66 | -22 | 50 | -16 | 11 | -57 | CEL (4) |
| 2219-T81 | 197 | 2,340 | 66 | -5 | 50 | -6 | 11 | -30 | CEL (4) |
| 2219-T81 | 402 | 2,370 | 66 | -15 | 50 | -14 | 11 | -62 | CEL (4) |
| 2219-T81 | 181 | 5 | 65 | -17 | 50 | -13 | 12 | -62 | CEL (4) |
| 2219-T87 | 402 | 2,370 | 61 | -27 | 51 | -29 | 10 | -77 | CEL (4) |

${ }^{a}$ Numbers refer to references at end of report.
${ }^{b}$ Too severely corroded to machine into tensile specimens.
${ }^{c}$ Welded, 2319 electrode, TIG process.
${ }^{d}$ Transverse properties.
Table 65. Chemical Composition of 3000 Series Aluminum Alloys, Percent by Weight

| Alloy | Gage <br> (in.) | Si | Fe | Cu | Mn | Mg | Cr | Ni | Zn | Ti | $\mathrm{Al}^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 3003 | - | - | - | - | 1.2 | - | - | - | - | - | R |
| 3003 | - | 0.15 | 0.45 | 0.15 | 1.25 | - | - | - | 0.05 | - | R |
| $3003-\mathrm{H} 12$ \& H14 | - | 0.10 | 0.20 | 0.05 | 1.10 | 0.03 | 0.01 | 0.01 | 0.02 | 0.01 | R |
| 3003-H14 | 0.125 | 0.60 | 0.70 | 0.20 | 1.25 | - | - | - | 0.10 | - | R |
| 3003-H14 | 0.063 | 0.20 | 0.58 | 0.13 | 1.05 | $<0.01$ | $<0.01$ | $<0.01$ | $<0.01$ | - | R |
| 3003-H14 | 0.063 | 0.15 | 0.50 | 0.10 | 1.30 | 0.02 | - | - | 0.07 | - | R |
| 3003-H24 | - | 0.10 | 0.48 | 0.16 | 1.10 | - | - | - | 0.08 | - | R |
| Alclad 3003-H12 | 0.125 |  |  |  |  |  |  |  |  |  |  |
| Core |  | 0.06 | 0.70 | 0.20 | 1.25 | - | - | - | 0.10 | - | R |
| Cladding |  | $b$ | $b$ | 0.10 | 0.10 | 0.10 | - | - | 1.0 | - | R |
| Alclad 3003-H14 | - | 0.10 | 0.20 | 0.05 | 1.10 | 0.03 | 0.01 | 0.01 | 0.02 | 0.01 | R |

[^21]Table 66. Corrosion Rates and Types of Corrosion of 3000 Series Aluminum Alloys

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum <br> Pit Depth ${ }^{b}$ (mils) | Crevice Depth (mils) | Type ${ }^{\text {b }}$ |  |
| 3003-H14 | W | 123 | 5,640 | 0.5 | 27 | 32 | C; P | CEL (4) |
| 3003 | W | 123 | 5,640 | 0.6 | 0 | 28 | C | INCO (3) |
| 3003-H14 | S | 123 | 5,640 | 1.9 | 55 | 68 | C; E; P | CEL (4) |
| 3003 | S | 123 | 5,640 | 3.6 | 0 | 50 | C (PR) | INCO (3) |
| 3003-H14 | W | 403 | 6,780 | 3.9 | 125 | 66 | S-C; P (PR) | CEL (4) |
| 3003 | W | 403 | 6,780 | 3.8 | 0 | 50 | C (PR) | INCO (3) |
| 3003-H14 | S | 403 | 6,780 | 3.7 | 125 | 52 | S-C; P (PR) | CEL (4) |
| 3003 | S | 403 | 6,780 | 1.3 | 0 | 50 | C (PR) | INCO (3) |
| 3003-H14 | W | 751 | 5,640 | 2.3 | 125 | 125 | C (PR); P (PR) | CEL (4) |
| 3003 | W | 751 | 5,640 | 3.0 | 0 | 40 | C (PR) | INCO (3) |
| 3003-H14 | S | 751 | 5,640 | 2.5 | 125 | 125 | C (PR); P (PR) | CEL (4) |
| 3003 | S | 751 | 5,640 | 1.8 | 0 | 40 | C (PR) | INCO (3) |
| 3003-H14 | W | 1,064 | 5,300 | 2.0 | 125 | 125 | C (PR); S-E; P (PR) | CEL (4) |
| 3003 | W | 1,064 | 5,300 | 2.8 | - | - | $d$ | INCO (3) |
| 3003-H14 | S | 1,064 | 5,300 | 1.9 | 125 | 0 | EX-E; P (PR) | CEL (4) |
| 3003 | S | 1,064 | 5,300 | 1.0 | 0 | 50 | C (PR) | INCO (3) |
| 3003-H14 | w | 197 | 2,340 | $<0.1$ | 17 | 0 | E; P | REY (14) |
| 3003-H14 | w | 197 | 2,340 | 2.4 | 48 | 28 | C; S-E; P | CEL (4) |
| 3003 | W | 197 | 2,340 | 1.4 | 0 | 40 | C (PR) | INCO (3) |
| 3003-H14 | S | 197 | 2,340 | 1.6 | 55 | 25 | C; E; S-P | CEL (4) |
| 3003 | S | 197 | 2,340 | $<0.1$ | 1 | 1 | I-C; I-P | INCO (3) |
| 3003-H14 | W | 402 | 2,370 | 1.4 | 91 | 93 | S-C; S-E; D-P | CEL (4) |
| 3003 | w | 402 | 2,370 | 1.1 | 0 | 40 | C (PR) | INCO (3) |
| 3003-H14 | S | 402 | 2,370 | 1.7 | 115 | 70 | S-C; D-P | CEL (4) |
| 3003 | S | 402 | 2,370 | 0.5 | 0 | 50 | C (PR) | INCO (3) |
| 3003-H14 | w | 181 | 5 | 1.1 | 33 | 0 | E; P | CEL (4) |
| 3003 | W | 181 | 5 | 1.0 | 1 | 0 | $1-\mathrm{P}$ | INCO (3) |
| 3003 | W | 366 | 5 | 0.6 | 1 | 0 | I-P | INCO (3) |
| 3003-H14 | w | 398 | 5 | 1.0 | 21 | 0 | P | CEL (4) |
| 3003-H14 | w | 540 | 5 | 0.3 | 34 | 75 | C; P | CEL (4) |
| 3003-H14 | w | 588 | 5 | 2.0 | 65 | 0 | P | CEL (4) |
| Alclad 3003-H14 | W | 123 | 5,640 | - | I | 0 | I-P | NADC (7) |
| Alclad 3003-H12 | W | 123 | 5,640 | 0.2 | 18 | 15 | B; C; SL-E; $\mathrm{P}^{e}$ | CEL (4) |
| Alclad 3003 | W | 123 | 5,640 | 2.7 | 0 | 0 | G | INCO (3) |
| Alclad 3003-H12 | S | 123 | 5,640 | 2.8 | 20 | 0 | B; C; SL-E ${ }^{e}$ | CEL (4) |
| Alclad 3003 | S | 123 | 5,640 | 2.6 | 0 | 0 | G | INCO (3) |
| Alclad 3003-H12 | W | 403 | 6,780 | 0.4 | 13 | 14 | C; SL-E; P | CEL (4) |
| Alclad 3003 | W | 403 | 6,780 | 2.5 | 0 | 0 | G | INCO (3) |
| Alclad 3003-H12 | S | 403 | 6,780 | 0.2 | 14 | 13 | C; SL-E; P | CEL (4) |
| Alclad 3003 | S | 403 | 6,780 | 0.4 | 0 | 0 | $f$ | INCO (3) |
| Alclad 3003-H14 | W | 751 | 5,640 | - | 0 | 0 | SL-G | NADC (7) |
| Alclad 3003-H12 | w | 751 | 5,640 | 0.3 | 13 | 13 | C; E; P | CEL (4) |
| Alclad 3003 | W | 751 | 5,640 | 1.4 | 0 | 0 | G | INCO (3) |
| Alclad 3003-H12 | S | 751 | 5,640 | 2.4 | 14 | 14 | C; E; $\mathbf{P}^{e}$ | CEL (4) |
| Alclad 3003 | S | 751 | 5,640 | 1.5 | 0 | 0 | U | INCO (3) |
| Alclad 3003-H12 | W | 1,064 | 5,300 | 0.5 | 20 | 13 | C; $\mathrm{p}^{g}$ | CEL (4) |
| Alclad 3003 | W | 1,064 | 5,300 | 1.5 | 0 | 0 | U | INCO (3) |

Continued

Table 66. Continued.

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum <br> Pit Depth ${ }^{b}$ (mils) | Crevice Depth ${ }^{b}$ (mils) | Type ${ }^{\text {b }}$ |  |
| Alclad 3003-H12 | S | 1,064 | 5,300 | 0.8 | 16 | 13 | $\mathrm{C} ; \mathbf{P}^{b}$ | CEL (4) |
| Alclad 3003 | S | 1,064 | 5,300 | 0.6 | 0 | 0 | U | INCO (3) |
| Alclad 3003-H12 | w | 197 | 2,340 | 2.2 | 15 | 13 | $\mathrm{C}_{\mathrm{i}} \mathrm{P}^{e}$ | CEL (4) |
| Alclad 3003 | W | 197 | 2,340 | 2.3 | 0 | 0 | G | INCO (3) |
| Alclad 3003-H12 | S | 197 | 2,340 | 1.1 | 14 | 13 | C; $\mathrm{P}^{2}$ | CEL (4) |
| Alclad 3003 | S | 197 | 2,340 | <0.1 | 2 | 2 | C; P | INCO (3) |
| Alclad 3003-H12 | w | 402 | 2,370 | 2.2 | 14 | 15 | $\mathrm{C}_{;} \mathrm{P}^{j}$ | CEL (4) |
| Alclad 3003 | W | 402 | 2,370 | 1.6 | 0 | 0 | $k$ | INCO (3) |
| Alclad 3003-H12 | S | 402 | 2,370 | 1.8 | 13 | 14 | C; $\mathrm{P}^{\text {I }}$ | CEL (4) |
| Alclad 3003 | S | 402 | 2,370 | 0.4 | 0 | 0 | $m$ | INCO (3) |
| Alclad 3003-1112 | W | 181 | 5 | 1.0 | 1 | 0 | I-P | CEL (4) |
| Alclad 3003 | W | 181 | 5 | 1.0 | 2 | 0 | $\mathrm{N}-\mathrm{P}$ | INCO (3) |
| Alclad 3003 | W | 366 | 5 | 0.5 | 2 | 0 | N-P | INCO (3) |
| Alclad 3003-H12 | w | 398 | 5 | 1.1 | 16 | 0 | P | CEL (4) |
| Alclad 3003-H12 | w | 540 | 5 | 0.3 | 16 | 0 | P | CEL (4) |
| Alclad 3003-1112 | W | 588 | 5 | 1.8 | 17 | 0 | P | CEL (4) |

${ }^{a} \mathrm{~W}=$ Totally exposed in sewater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimens were embedded in the bottom sediments.
$b_{\text {Symbols }}$ for types of corrosion:

| B | $=$ Blisters | N | $=$ Numerous |
| :---: | :---: | :---: | :---: |
| C | = Crevice | P | $=$ Pitting |
| D | $=$ Deep | PR | $=$ Perforated |
| E | $=$ Edge | S | $=$ Severe |
| EX | = Extensive | SL | = Slight |
| G | $=$ General | U | = Uniform |
| 1 | $=$ Incipient |  |  |

${ }^{\text {Numbers refer to references at end of report. }}$
$d_{\text {About }} 40 \%$ of specimen missing.
${ }^{e}$ Large area of cladding gone.
$f_{\text {Nonuniform cladding loss. }}$
$g_{\text {Nonuniform cladding loss, }} 18 \%$ gone, one area 7 sq , in.
${ }^{b}$ Nonuniform cladding loss, $13 \%$ gone, one area 5 sq . in.
${ }^{i}$ One 7 -sq.-in. area of cladding gone from portion in water; cladding gone on 2 -in.-high strip across bottom portion in bottom sediment.
$j_{20 \%}$ cladding gone and incipient pitting in denuded area.
${ }^{k} 60 \%$ of cladding gone.
${ }^{l} 18 \%$ of cladding gone, incipient pitting in denuded area; cladding gone on 2 -in.-high strip across bottom portion embedded in bottom sediment.
${ }^{m} 20 \%$ of cladding gone.

Table 67. Stress Corrosion of 3000 Series Aluminum Alloys

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure <br> (day) | Depth <br> $(\mathrm{ft})$ | Number of <br> Specimens <br> Exposed | Number <br> Failed | Source $^{a}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $3003-\mathrm{H} 14$ | 6 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $3003-\mathrm{H} 14$ | 9 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |

${ }^{a}$ Numbers refer to references at end of report.

Table 68. Changes in Mechanical Properties of 3000 Series Aluminum Alloys Due to Corrosion

| Alloy | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{a}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| 3003-H14 | 123 | 5,640 | 22 | -4 | 20 | -8 | 13 | -43 | CEL (4) |
| 3003-H14 | 403 | 6,780 | 22 | -48 | 20 | -50 | 13 | -77 | CEL (4) |
| 3003-H14 | 751 | 5,640 | 21 | -20 | 21 | -11 | 4 | +25 | NADC (7) |
| 3003-H14 | 751 | 5,640 | 22 | -24 | 20 | -26 | 13 | -72 | CEL (4) |
| 3003-H14 | 1,064 | 5,300 | 22 | -3 | 20 | -18 | 13 | -20 | CEL (4) |
| 3003-H14 | 197 | 2,340 | 22 | -6 | 20 | -10 | 13 | +15 | CEL (4) |
| 3003-H14 | 402 | 2,370 | 22 | +6 | 20 | +11 | 13 | +5 | CEL (4) |
| 3003-H14 | 181 | 5 | 23 | +4 | 21 | -1 | 18 | -1 | CEL (4) |
| Alclad 3003-H12 | 123 | 5,640 | 19 | -2 | 18 | -1 | 14 | -42 | CEL (4) |
| Alclad 3003-H12 | 403 | 6,780 | 19 | +3 | 18 | -1 | 14 | 0 | CEL (4) |
| Alclad 3003-H12 | 751 | 5,640 | 19 | -3 | 18 | -4 | 14 | -19 | CEL (4) |
| Alclad 3003-H12 | 1,064 | 5,300 | 19 | +3 | 18 | +1 | 14 | -12 | CEL (4) |
| Alclad 3003-H12 | 197 | 2,340 | 19 | +1 | 18 | +2 | 14 | +7 | CEL (4) |
| Alclad 3003-H12 | 402 | 2,370 | 19 | +1 | 18 | 0 | 14 | -14 | CEL (4) |
| Alclad 3003-H12 | 181 | 5 | 19 | +4 | 18 | +1 | 14 | +2 | CEL (4) |
| Alclad 3003-H14 | 123 | 5,640 | 21 | 0 | - | - | 4 | +100 | NADC (7) |

[^22]Table 69. Chemical Composition of 5000 Series Aluminum Alloys, Percent by Weight

| Alloy | Gage (in.) | Si | Fe | Cu | Mn | Mg | Cr | Ni | Zn | Ti | $\mathrm{Al}^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 5050-H134 | 0.050 | 0.18 | 0.64 | 0.04 | - | 1.19 | - | - | - | - | R |
| 5052 | - | - | - | - | - | 2.5 | 0.25 | - | - | - | R |
| 5052-0, H32, H34 | - | 0.10 | 0.17 | 0.02 | 0.03 | 2.20 | 0.22 | -- | - | 0.01 | R |
| 5052-H22 | - | b | b | 0.05 | $<0.01$ | 2.50 | 0.23 | - | 0.07 | - | R |
| 5052-H134 | 0.050 | 0.13 | 0.30 | 0.02 | - | 2.31 | 0.22 | - | -- | - | R |
| 5083 | - | - | - | 0.15 | 0.6 | 4.5 | - | - | - | - | R |
| 5083-H113 | 0.500 | 0.40 | 0.40 | 0.10 | 0.65 | 4.5 | 0.15 | - | 0.25 | 0.15 | R |
| 5454 | - | - | - | -- | 0.03 | 1.0 | 0.02 | - | - | - | R |
| 5454-H32 | 0.162 | ' | c | 0.10 | 0.75 | 2.7 | 0.13 | - | 0.25 | 0.20 | R |
| BA28-1/4 H (5456) | 0.050 | 0.18 | 0.32 | 0.036 | 0.26 | 5.08 | $<0.02$ | $<0.02$ | 0.05 | - | R |
| $5456-\mathrm{H} 321+\mathrm{H} 343$ | 0.125 | $c$ | $c$ | 0.10 | 0.75 | 5.0 | 0.13 | - | 0.25 | 0.20 | R |
| 5456-H321 | - | - | - | 0.15 | 0.7 | 5.0 | 0.15 | - | - | - | R |
| $5456-\mathrm{H} 327+\mathrm{H} 343$ | - | 0.10 | 0.16 | 0.10 | 0.80 | 5.0 | 0.07 | - | 0.01 | 0.05 | R |
| 5456-1134 | - | 0.10 | 0.30 | 0.08 | 0.77 | 5.0 | 0.07 | - | - | - | R |
| 5086 |  | - | - |  | 0.3 | 4.0 | 0.15 | - | - | - | R |
| 5086-H112 | $3 \mathrm{in} . \times 3 \mathrm{in}$. $\mathrm{x} 1 / 2 \mathrm{in}$. L | - | - | - | 0.45 | 4.0 | 0.15 | -- | - | - | R |
| 5086-H32 | 0.500 | 0.15 | 0.25 | 0.05 | 0.32 | 3.75 | 0.12 | - | 0.12 | 0.01 | R |
| 5086-H34 | 0.125 | 0.40 | 0.50 | 0.10 | 0.45 | 4.0 | 0.15 | - | 0.25 | 0.15 | R |
| 5086-H34 | - | 0.70 | 0.10 | 0.12 | 0.90 | 4.00 | 0.05 | 0.10 | 0.04 | 0.01 | R |

[^23]Table 70. Corrosion Rates and Types of Corrosion of 5000 Series Aluminum Alloys

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Maximum Pit Depth ${ }^{b}$ (mils) | Crevice Depth <br> (mils) | Type ${ }^{\text {b }}$ |  |
| 5050-H34 | W | 402 | 2,370 | <0.1 | 4 | 0 | P | REY (14) |
| 5050-H34 | W | 402 | 2,370 | 0.1 | 1 | 31 | C; I-P | CEL (4) |
| 5050-H34 | S | 402 | 2,370 | 0.2 | 1 | 31 | C; I-P | CEL (4) |
| 5052-0 | W | 123 | 5,640 | 3.7 | 0 | 65 | C (PR) | INCO (3) |
| 5052-0 | S | 123 | 5,640 | 4.0 | 0 | 65 | C (PR) | INCO (3) |
| 5052-0 | W | 403 | 6,780 | 4.5 | 0 | 62 | C (PR) | INCO (3) |
| 5052-0 | S | 403 | 6,780 | 1.2 | 0 | 62 | C (PR) | INCO (3) |
| 5052-0 | W | 751 | 5,640 | 2.2 | 65 | 65 | C (PR); P (PR) | INCO (3) |
| 5052-0 | S | 751 | 5,640 | 3.2 | 0 | 62 | C (PR) | INCO (3) |
| 5052-0 | W | 1,064 | 5,300 | 3.1 | 0 | 65 | C (PR) | INCO (3) |
| 5052-0 | S | 1,064 | 5,300 | 1.5 | 0 | 62 | C (PR) | INCO (3) |
| 5052-0 | W | 197 | 2,340 | 1.8 | 0 | 65 | C (PR) | INCO (3) |
| 5052-0 | S | 197 | 2,340 | $<0.1$ | 1 | I | I-C; I-P | INCO (3) |
| 5052-0 | W | 402 | 2,370 | 0.4 | 0 | 20 | C | INCO (3) |
| 5052-0 | S | 402 | 2,370 | 0.3 | 0 | 0 | S-B at C | INCO (3) |
| 5052-0 | W | 181 | 5 | 1.2 | I | 0 | I-P | INCO (3) |
| 5052-0 | W | 366 | 5 | 0.6 | 5 | 5 | $\mathrm{C}: \mathrm{P}$ | INCO (3) |
| 5052-H22 | W | 1,064 | 5,300 | 0.4 | -- | -- | - | CEL (4) |
| 5052-H32 | W | 403 | 6,780 | - | 39 | 0 | E-XF; P | NADC (7) |
| 5052-H32 | W | 197 | 2,340 | - | U | 0 | U-P | NADC (7) |
| 5052-H32 | S | 197 | 2,340 | $<0.1$ | 26 | 0 | L-P | NADC (7) |
| 5052-H32 | W | 402 | 2,370 | - | F | 0 | E-XF; L; F-D-P | NADC (7) |
| 5052-H34 | W | 123 | 5,640 | <0.1 | S | 0 | S-P | NADC (7) |
| 5052-H34 | W | 123 | 5,640 | - | PP | MO | MO-Ci PP | Bell (13) |
| 5052-H34 | W | 751 | 5,640 | - | 0 | 0 | SH-CR | NADC (7) |
| 5052-H34 | W | 751 | 5,640 | - | SH | MO | MO-C; SH-P | Bell (13) |
| 5052-H34 | S | 751 | 5,640 | - | SH | S | S-C; SH-P | Bell (13) |
| 5052-H34 | W | 197 | 2,340 | $<0.1$ | 12 | 0 | P | REY (14) |
| 5052-H34 | W | 402 | 2,370 | 0.2 | I | 34 | C; I-P | CEL (4) |
| 5052-H34 | S | 402 | 2,370 | 0.2 | I | 50 | C (PR) ; I-P | CEL (4) |
| 5083-H113 ${ }^{\text {d }}$ | W | 123 | 5,640 | <0.1 | 28 | 0 | $\mathrm{p}^{e}$ | CEL (4) |
| 5083-H113 ${ }^{\text {d }}$ | S | 123 | 5,640 | 0.9 | 65 | 0 | F-P | CEL (4) |

Table 70．Continued．

|  | U |  |
| :---: | :---: | :---: |
| $\begin{aligned} & \text { 苛 } \\ & \text { 葡 } \end{aligned}$ | ${ }^{\infty}$ |  |
|  | 苞号忽 |  |
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|  | 秃菦 | 人 <br>  |
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|  | － |  |

Table 70. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth (ft) | Corrosion |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum Pit Depth ${ }^{b}$ (mils) | Crevice Depth ${ }^{\text {b }}$ (mils) | Type ${ }^{\text {b }}$ |  |
| 5086-H32 | W | 189 | 5,900 | 0.1 | 33 | 15 | C; SC-P | CEL (4) |
| 5086-H32 | S | 189 | 5,900 | 0.2 | 23 | 4 | C; SC-P | CEL (4) |
| 5086-H32 ${ }_{j}$ | W | 360 | 4,200 | 0.1 | 47 | 0 | E; $P$ | CEL (4) |
| 5086-H32 ${ }^{\text {J }}$ | S | 360 | 4,200 | $<0.1$ | 29 | 0 | B; P | CEL (4) |
| 5086-H32 | W | 197 | 2,340 | $<0.1$ | 5 | 0 | P | REY (14) |
| 5086-H32 | W | 402 | 2,370 | 0.4 | I | 18 | C; I-P | CEL (4) |
| 5086-H32 | S | 402 | 2,370 | 0.5 | 1 | 18 | Cif $\mathrm{I}-\mathrm{P}$ | CEL (4) |
| 5086-H32 | W | 181 | 5 | 1.0 | 1 | 0 | IG (E) I - P | CEL (4) |
| 5086-H32 | W | 398 | 5 | 0.4 | 20 | 0 | P | CEL (4) |
| 5086-H32 | W | 588 | 5 | 0.2 | 22 | I | 1-C; P | CEL (4) |
| $5086-\mathrm{H} 34_{k}$ | w | 123 | 5,640 | 0.1 | 0 | 0 | ET | CEL (4) |
| $5086-\mathrm{H} 34^{k}$ | W | 123 | 5,640 | - | - | - | NC-SD | NADC (7) |
| 5086-H34 | S | 123 | 5,640 | 1.4 | 13 | 50 | S-C; E; P | CEL (4) |
| 5086-H34 | W | 189 | 5,900 | 0.2 | 3 | 0 | SL-P | CEL (4) |
| 5086-H34 | S | 189 | 5,900 | 0.7 | 55 | 126 | C (PR); E; S-P | CEL (4) |
| 5086-H34 | W | 403 | 6,780 | 0.6 | I | 53 | S-C; I-P | CEL (4) |
| 5086-H34 | S | 403 | 6,780 | 0.8 | 4 | 48 | S-C; P | CEL (4) |
|  | W | 751 | 5,640 | 2.0 | 72 | 60 | S-C; E; S-P | CEL (4) |
| 5086-H34 ${ }^{k}$ | W | 751 | 5,640 | - | PR | 0 | P (PR); L (HAZ\&WB) | NADC (7) |
| 5086-H34 | W | 1,064 | 5,300 | 0.9 | 73 | 69 | S-C; S-E; S-P | CEL (4) |
| $5086-\mathrm{H} 34{ }_{k}$ | S | 1,064 | 5,300 | 1.2 | 75 | 50 | S-C; S-E; S-P | CEL (4) |
| $5086-\mathrm{H} 34^{k}$ | W | 360 | 4,200 | $<0.1$ | 47 | 0 | E; ${ }^{\text {P }}$ | CEL (4) |
| 5086-H34 ${ }^{k}$ | S | 360 | 4,200 | $<0.1$ | 51 | 0 | $\mathbf{E} ; \mathbf{P}^{m}$ | CEL (4) |
| 5086-H34 | W | 197 | 2,340 | 0.7 | 29 | 1 | I-C; SL-E; F-P | CEL (4) |
| 5086-H34 | S | 197 | 2,340 | 1.1 | 53 | 38 | S-C; S-E; S-P | CEL (4) |
| 5086-H34 | W | 402 | 2,370 | 0.6 | I | 5 | C; I-P | CEL (4) |
| 5086-H34 | S | 402 | 2,370 | 1.3 | 17 | 48 | S-C; P | CEL (4) |
| 5086-H34 | W | 181 | 5 | 1.2 | 6 | 0 | IG; P | CEL (4) |
| 5086-H34 | W | 398 | 5 | 0.8 | 27 | 1 | I-C; P | CEL (4) |
| 5086-H34 | W | 540 | 5 | 0.3 | 0 | 1 | I-C; ET | CEL (4) |
| 5086-H34 | W | 588 | 5 | 1.6 | 47 | 43 | S-C; S-P | CEL (4) |
| $5086-\mathrm{H} 112^{n}$ | S | $189$ | 5,900 | $0.1$ | $21$ | $32$ | C; SC-P | CEL (4) |
| $5086-\mathrm{H} 112^{\text {n,j }}$ | W | 360 | 4,200 | 0.1 | 20 | 117 | S-C; P | CEL (4) |

Table 70. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Maximum <br> Pit Depth (mils) | Crevice Depth (mils) | Type ${ }^{\text {b }}$ |  |
| 5086-H112 ${ }^{n, j}$ | S | 360 | 4,200 | $<0.1$ | 20 | 64 | S-C; P | CEL (4) |
| 5086-H1112 ${ }^{\prime \prime}$ | W | 181 | 5 | 1.1 | 1 | 0 | I-P | CEL (4) |
| 5454 | W | 402 | 2,370 | 0.4 | 0 | 28 | C | INCO (3) |
| 5454 | S | 402 | 2,370 | 0.6 | 0 | 80 | C (PR) | INCO (3) |
| 5454 | W | 181 | 5 | 1.0 | 1 | 0 | I-P | INCO (3) |
| 5454 | w | 366 | 5 | 0.5 | 0 | 0 | ET | INCO (3) |
| $545+\mathrm{H} 32{ }^{\text {d }}$ d | W | 123 | 5,640 | 0.1 | 7 | 0 | ${ }_{\mathrm{p}}{ }^{p}$ | CEL (4) |
| $5454-\mathrm{H} 32^{\text {d }}$ | S | 123 | 5,640 | 1.1 | 49 | 0 | E; S-P; SL (WB) | CEL (4) |
| 5454-H32 ${ }_{\text {d }}$ | W | 403 | 6,780 | 0.9 | 38 | 0 | MO-E; MO-P | CEL (4) |
| 5454-H132 ${ }^{\text {d }}$ | W | 403 | 6,780 | 1.7 | 64 | 0 | E; P; L (HAZ) | CEL (4) |
| $5454-\mathrm{H} 32{ }_{\text {d }}$ | S | 403 | 6,780 | 0.5 | 37 | 0 | MO-E; MO-P | CEL (4) |
| $5454-\mathrm{H} 32{ }^{\text {d }}$ | S | 403 | 6,780 | 0.6 | 42 | 0 | E; P; L (HAZ) | CEL (4) |
| 5454-H32 ${ }^{\text {d }}$ | w | 751 | 5,640 | 0.9 | 65 | 0 | E; D-P; SL (WB) | CEL (4) |
| 5454-H32 ${ }_{\text {d }}$ | W | 197 | 2,340 | 0.7 | 24 | 0 | SL-E; P | CEL (4) |
| $5454-\mathrm{H} 32 \mathrm{~d}$ | W | 197 | 2,340 | 0.3 | 41 | 0 | MO-P; P (HAZ) | CEL (4) |
| 5454-H32 ${ }_{\text {d }}$ | S | 197 | 2,340 | 0.6 | 16 | 0 | SL-E; P | CEL (4) |
| $5454-\mathrm{H} 32{ }^{\text {d }}$ | S | 197 | 2,340 | 0.6 | 49 | S | S-C; MO-P; P (HAZ) | CEL (4) |
| $5454-\mathrm{H} 32{ }^{\text {d }}$ | W | 402 | 2,370 | 0.3 | 1 | 39 | C; I-P | CEL (4) |
| $5454-\mathrm{H} 32{ }^{\text {d }}$ | W | 402 | 2,370 | 0.6 | 42 | 0 | $\mathrm{P} ; \mathrm{P}(\mathrm{WB})^{q}$ | CEL (4) |
|  | S | 402 | 2,370 | 0.3 | I | 33 | $\mathrm{C} ; \mathrm{I}-\mathrm{P}$ | CEL (4) |
| ${ }^{5454} 4 \mathrm{H}_{3}{ }^{\text {d }}$ d | S | 402 | 2,370 | 0.7 | 22 | 0 | $\mathrm{P}^{r}$ | CEL (4) |
| $5454-\mathrm{H32}{ }^{\text {d }}$ | W | 398 | 5 | 0.5 | 8 | 0 | P | CEL (4) |
| $5454-\mathrm{H} 32{ }^{\text {d }}$ | W | 540 | 5 | 0.3 | I | 0 | I-P | CEL (4) |
| 5454-H32 ${ }^{\text {d }}$ | W | 588 | 5 | 0.7 | 39 | 0 | P; P (WB\&HAZ) | CEL (4) |
| BA28-1/4 H (5456) | w | 197 | 2,340 | $<0.1$ | 2 | 0 | P | REY (14) |
| 5456-H32 | W | 402 | 2,370 | 0.6 | 0 | 5 | C; E | CEL (4) |
| 5456-H32 | S | 402 | 2,370 | 0.7 | 0 | 21 | C; E | CEL (4) |
| 5456-H321 | w | 123 | 5,640 | $<0.1$ | 0 | SL | SL-C;SL-ET | CEL (4) |
| 5456-H321 | W | 123 | 5,640 | 1.3 | 5 | 5 | C; P | MEL (5) |
| 5456-H321 | S | 123 | 5,640 | 2.1 | 49 | 33 | C; E; S-P | CEL (4) |
| 5456-H321 | W | 403 | 6,780 | 1.0 | I | 50 | S-C; E; I-P | CEL (4) |
| 5456-H321 | S | 403 | 6,780 | 0.8 | 44 | 32 | C; E; F-P | CEL (4) |

Table 70．Continued．

| Alloy | Environment ${ }^{\text {a }}$ | Exposure （day） | Depth <br> （ft） | Corrosion |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate （mpy） | Maximum Pit Depth ${ }^{b}$ （mils） | Crevice Depth （mils） | Type ${ }^{\text {b }}$ |  |
| 5456－H321 | W | 751 | 5，640 | 2.0 | 53 | 25 | C；E；F－D－P | CEL（4） |
| 5456－H321 | W | 751 | 5，640 | 2.2 | 17 | 23 | C；EX－E；P | CEL（4） |
| 5456－H321 | S | 751 | 5，640 | 2.2 | 50 | 45 | S－C；E；F－D－P | CEL（4） |
| 5456－H321 | W | 1，064 | 5，300 | 1.0 | 43 | 37 | C；EX－E；F－D－P | CEL（4） |
| $5456-\mathrm{H} 321$ | S | 1，064 | 5，300 | 1.5 | 36 | 46 | S－C；EX－E；F－P | CEL（4） |
| 5456－H321 | W | 197 | 2，340 | 2.7 | 17 | 36 | C；E；F－P | CEL（4） |
| 5456－H321 | S | 197 | 2，340 | 2.6 | 43 | 18 | C；E；F－D－P | CEL（4） |
| $5456-\mathrm{H} 321$ | W | 402 | 2，370 | 1.1 | 41 | 44 | S－C；E；D－P | CEL（4） |
| 5456－H321 | S | 402 | 2，370 | 1.2 | 14 | 32 | C；E；P | CEL（4） |
| 5456－H321 | w | 181 | 5 | 1.2 | 12 | － 0 | $P$ | CEL（4） |
| 5456－H321 | W | 398 | 5 | 0.6 | 16 | 0 | P | CEL（4） |
| 5456－H321 | W | 540 | 5 | 0.9 | 14 | 8 | $\mathrm{C} ; \mathrm{P}$ | CEL（4） |
| 5456－H34 | W | 123 | 5，640 | － | 31 | 0 | CR（PR）；S－E； P | NADC（7） |
| 5456－H34 | W | 403 | 6，780 | － | PR | 0 | P（PR）；CR | NADC（7） |
| 5456－H34 | S | 403 | 6，780 | － | － | － | XF | NADC（7） |
| 5456－H34 | W | 751 | 5，640 | － | PR | － | CR；P（PR）；XF；WB（C；S－CR；XF） | NADC（7） |
| 5456－H34 | W | 1，064 | 5，300 | － | 32 | PR | C（PR）；P；S－XF | NADC（7） |
| 5456－H34 | W | 1，064 | 5，300 | － | 63 | I | I－C；E；D－P | CEL（4） |
| 5456－H34 | S | 197 | 2，340 | $<0.1$ | 12 | S | S－C；CR（PR）；E；P | NADC（7） |
| 5456－H343 | W | 123 | 5，640 | 1.0 | 25 | I | I－C； $\mathrm{P}^{\text {P }}$ | CEL（4） |
| 5456－H343 | W | 123 | 5，640 | － | S | 0 | S－E；CR（30）；S－P | CEL（4） |
| 5456－H343 | W | 403 | 6，780 | 0.2 | 1 | 28 | C；I－P | CEL（4） |
| 5456－H343 | S | 403 | 6，780 | 0.2 | 1 | 28 | C；I－P | CEL（4） |
| 5456－H343 | W | 751 | 5，640 | 1.1 | 0 | 64 | C（PR）；SL－E | CEL（4） |
| 5456－H343 | W | 1，064 | 5，300 | 0.1 | 35 | 35 | C；E；F－P | CEL（4） |
| 5456－H343 | W | 197 | 2，340 | 0.4 | 0 | I | I－C；E | CEL（4） |
| 5456－H343 | S | 197 | 2，340 | 0.3 | 0 | I | I－C；E | CEL（4） |

${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure； $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the
specimens were embedded in the boctom sediments．
$=$ Slight
$=$ Uniform
$=$ Weld bead
$=$ Exfoliated
Table 70. Continued.
SL
U
WB
XF

$$
\begin{array}{llll}
\mathrm{D} & =\text { Deep } & \mathrm{MO} & =\text { Moderate } \\
\mathrm{E} & =\text { Edge } & \mathrm{NC} & =\text { No visible corrosion } \\
\mathrm{ET} & =\text { Etched } & \mathrm{P} & =\text { Pitting } \\
\mathrm{EX} & =\text { Extensive } & \mathrm{PP} & =\text { Pin-point pitting } \\
\mathrm{F} & =\text { Few } & \mathrm{PR} & =\text { Perforated } \\
\mathrm{HAZ} & =\text { Heat-affected zone } & \mathrm{S} & =\text { Severe } \\
\text { Numbers in parentheses indicate maximum depth in mils. }
\end{array}
$$

Numbers refer to references at end of report.
${ }^{4}$ Welded with 5052 rod.
'Two measurable pits.
Transverse butt weld, 5183 wire, M1G process.
${ }^{g}$ Scattered pitting, heavier in sediment than in water; shallow interconnected pitting in heat-affected zone parallel to weld bead.
${ }^{6}$ One measurable pit.
${ }^{i}$ Line of pits at edge of weld bead; 8 sq. in. area in bottom sediment reduced in thickness from 250 to 71 mils. ${ }^{j}$ Tongue-of-the Ocean, Atlantic Ocean, $4.5^{\circ} \mathrm{C}, 5.18 \mathrm{ml} / \mathrm{l}$ oxygen.
${ }^{k}$ Welded, 5566 rod, TIG process.
6 -sq.-in. area in sediment reduced in thickness by 60 mils. ${ }^{n} 0.5$-in.-wide undercut pits.
${ }^{n} 3 \times 3 \times 1 / 2-\mathrm{in}$. angle
$P_{\text {Six measurable pits. }}$
${ }^{q}$ Cracked in weld bead.
${ }^{r} 4$-sq-in. area in sediment reduced in thickness by 9.4 mils.
Table 71. Stress Corrosion of 5000 Series Aluminum Alloys

| Alloy | $\begin{gathered} \text { Stress } \\ (\mathrm{ksi}) \end{gathered}$ | Percent <br> Yield <br> Strength | Exposure (day) | Depth <br> (ft) | Number of Specimens Exposed | Number Failed | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 5050-H34 | 11 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 5050-H34 | 17 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 5052-H32 | 8 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| 5052-H32 | 21 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| 5052-H32 | 8 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 5052-H32 | 21 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 5052-H32 | 8 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 5052-H32 | 21 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 5052-H34 | 15 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 5052-H34 | 22 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 5083-H113 ${ }^{b}$ | 15 | 75 | 403 | 6,780 | 3 | 0 | CEL (4) |
| $5083-\mathrm{H} 113^{b}$ | 15 | 75 | 197 | 6,780 | 3 | 0 | CEL (4) |
| $5083-\mathrm{H} 113^{b}$ | 15 | 75 | 402 | 6,780 | 3 | 0 | CEL (4) |
| 5086-H32 | 15 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 5086-H32 | 23 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 5454-H32 ${ }^{\text {b }}$ | 12 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $5454-\mathrm{H} 32^{\text {b }}$ | 12 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $5454-\mathrm{H} 32{ }^{\text {b }}$ | 12 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 5456-H32 | 20 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 5456-H32 | 30 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |

Table 72. Changes in Mechanical Properties of 5000 Series Aluminum Alloys Due to Corrosion

| Alloy | Exposure (day) | Depth (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original <br> (\%) | \% Change |  |
| 5050-H34 | 402 | 2,370 | 28 | $+2$ | 22 | +11 | 8 | +25 | CEL (4) |
| 5052-H32 | 403 | 6,780 | 34 | 0 | 27 | -4 | 11 | -54 | NADC (7) |
| 5052-H32 | 197 | 2,340 | 34 | -3 | 27 | 0 | 11 | -3 | NADC (7) |
| 5052-H32 | 402 | 2,370 | 34 | 0 | 27 | -4 | 11 | -18 | NADC (7) |
| $5052-\mathrm{H} 34{ }^{\text {b }}$ | 123 | 5,640 | 35 | 0 | 26 | +3 | 11 | $+9$ | NADC (7) |
| 5052-H34 | 751 | 5,640 | 34 | -64 | 27 | -58 | 11 | +95 | NADC (7) |
| 5052-H34 | 402 | 2,370 | 37 | +4 | 30 | +7 | 9 | +34 | CEL (4) |
| 5083-H113 ${ }^{\text {c }}$ | 123 | 5.640 | 42 | +4 | 20 | -1 | 13 | +26 | CEL (4) |
| 5083-H113 | 403 | 6,780 | 50 | -15 | 38 | -6 | 14 | -30 | CEL (4) |
| $5083-\mathrm{H} 113^{\text {c }}$ | 403 | 6,780 | 42 | +4 | 20 | -5 | 13 | +23 | CEL (4) |
| 5083-H113 ${ }^{\text {c }}$ | 751 | 5.640 | 42 | +5 | 20 | -8 | 13 | +32 | CEL (4) |
| 5083-H113 | 197 | 2,340 | 50 | -1 | 38 | -1 | 14 | +3 | CEL (4) |
| 5083-H113 ${ }^{\text {C }}$ | 197 | 2,340 | 42 | +6 | 20 | 0 | 13 | $+21$ | CEL (4) |
| 5083-H113 | 402 | 2,370 | 50 | -1 | 38 | 0 | 14 | +37 | CEL (4) |
| 5083-H113 ${ }^{\text {C }}$ | 402 | 2.370 | 42 | +2 | 20 | -1 | 13 | +17 | CEL (4) |
| 5083-H113 | 181 | 5 | 48 | -2 | 35 | -11 | 19 | +16 | CEL (4) |
| $5083-\mathrm{H} 113^{\text {c }}$ | 181 | 5 | 41 | +9 | 22 | +11 | 13 | +8 | CEL (4) |
| 5086-H32 | 402 | 2,370 | 44 | -5 | 30 | -6 | 14 | +33 | CEL (4) |
| 5086-H32 | 181 | 5 | 46 | -4 | 32 | 0 | 17 | +41 | CEL (4) |
| $5086-\mathrm{H} 34{ }^{\text {d,e }}$ | 123 | 5,640 | 49 | -3 | 35 | +8 | 13 | -23 | NADC (7) |
| 5086-H34 | 123 | 5,640 | 48 | -1 | 37 | +2 | 12 | -1 | CEL (4) |
| 5086-H34 | 403 | 6,780 | 48 | 0 | 37 | -1 | 12 | -5 | CEL (4) |
| 5086-H34 | 751 | 5,640 | 48 | -7 | 37 | -5 | 12 | -9 | CEL (4) |
| 5086-H34 | 1,064 | 5,300 | 48 | -2 | 37 | -4 | 12 | -3 | CEL (4) |
| 5086-H34 | 197 | 2,340 | 48 | -5 | 37 | -6 | 12 | +4 | CEL (4) |
| 5086-H34 | 402 | 2,370 | 48 | -3 | 37 | -2 | 12 | -3 | CEL (4) |
| 5086-H34 | 181 | 5 | 48 | 0 | 37 | +2 | 12 | -9 | CEL (4) |
| 5086-H112 ${ }^{f}$ | 181 | 5 | 47 | 0 | 29 | -3 | 16 | -2 | CEL (4) |
| 5454-H32 ${ }^{\text {c }}$ | 123 | 5,640 | 35 | -1 | 16 | +10 | 14 | -11 | CEL (4) |
| 5454-H32 | 403 | 6,780 | 41 | -2 | 31 | -1 | 13 | -22 | CEL (4) |
| $5454-\mathrm{H} 32{ }^{\text {c }}$ | 403 | 6,780 | 35 | -2 | 16 | +23 | 14 | -7 | CEL (4) |
| $5454-\mathrm{H} 32{ }^{\text {c }}$ | 751 | 5,640 | 35 | -16 | 16 | +19 | 14 | -36 | CEL (4) |
| 5454-H32 | 197 | 2,340 | 41 | 0 | 31 | -1 | 13 | -3 | CEL (4) |
| $5454-\mathrm{H} 32^{\text {c }}$ | 197 | 2.340 | 35 | -3 | 16 | -7 | 14 | -3 | CEL (4) |
| 5454-H32 | 402 | 2,370 | 41 | +4 | 31 | +2 | 13 | $+27$ | CEL (4) |
| 5454-H32 | 402 | 2,370 | 35 | -4 | 16 | +6 | 14 | -6 | CEL (4) |
| 5456 | 402 | 2,370 | 50 | -7 | 36 | +1 | 15 | $-7$ | CEL (4) |
| 5456-H321 | 123 | 5,640 | 56 | -10 | 39 | -5 | 14 | -24 | CEL (4) |
| 5456-H321 | 403 | 6.780 | 56 | -1 | 39 | -6 | 14 | +22 | CEL (4) |
| 5456-H321 | 751 | 5,640 | 56 | -21 | 39 | -33 | 14 | -30 | CEL (4) |
| 5456-H321 | 1,064 | 5,300 | 56 | -4 | 39 | -8 | 14 | -6 | CEL (4) |
| 5456-11321 | 197 | 2,340 | 56 | -4 | 39 | -10 | 14 | 0 | CEL (4) |
| 5456-H321 | 402 | 2,370 | 56 | -1 | 39 | -5 | 14 | +9 | CEL (4) |
| 5456-11321 | 181 | 5 | 56 | -1 | 39 | -6 | 14 | +9 | CEL (4) |

Table 72. Continued.

| Alloy | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{a}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| 5456-H34 | 123 | 5,640 | 61 | -6 | 43 | +5 | 13 | -34 | NADC (7) |
| 5456-H34 | 403 | 6,780 | 58 | -14 | 50 | -21 | 3 | -80 | NADC (7) |
| 5456-H34 | 751 | 5,640 | 58 | -14 | 50 | -44 | 3 | -71 | NADC (7) |
| 5456-H34 | 1,064 | 5,300 | 58 | -14 | 50 | -14 | 3 | +94 | NADC (7) |
| 5456-H34 | 197 | 2,340 | 58 | -14 | 50 | -8 | 3 | +170 | NADC (7) |
| 5454-H34 ${ }^{\text {g }}$ | 403 | 6,780 | 58 | -11 | 50 | -14 | 3 | +196 | NADC (7) |
| 5456-H343 | 123 | 5,640 | 34 | +10 | - | - | 6 | +33 | NADC (7) |
| 5456-H343 | 123 | 5,640 | 58 | 0 | 46 | +2 | 9 | +27 | CEL (4) |
| 5456-H343 | 403 | 6,780 | 58 | -2 | 46 | -6 | 9 | +54 | CEL (4) |
| 5456-H343 | 751 | 5,640 | 58 | -2 | 40 | -1 | 9 | +27 | CEL (4) |
| 5456-H343 | 1,064 | 5,300 | 58 | -1 | 40 | -3 | 9 | +30 | CEL (4) |
| 5456-H343 | 197 | 2,340 | 58 | -3 | 40 | -1 | 9 | +22 | CEL (4) |

${ }^{a}$ Numbers refer to references at end of report.
${ }^{6}$ Transverse properties.
${ }^{c}$ Transverse butt weld with 5052 rod.
${ }^{d}$ Transverse butt weld, 5566 electrode, TIG process.
${ }^{e}$ Mechanical properties transverse to direction of weld.
$f_{3 \text { in. } \times 3 \text { in. } \times 1 / 2 \text { in. angle. }}$
$g_{\text {Anodized. }}$

Table 73. Chemical Composition of 6000 Series Aluminum Alloys, Percent by Weight

| Alloy | Gage <br> (in.) | Si | Fe | Cu | Mn | Mg | Cr | Ni | Zn | Ti | $\mathrm{Al}^{a}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 6061 | - | - | - | 0.25 | - | 1.0 | 0.28 | - | - | - | R |
| $6061-\mathrm{T} 6$ | 0.125 | 0.60 | 0.70 | 0.27 | 0.15 | 1.0 | 0.25 | - | 0.25 | 0.15 | R |
| $6061-\mathrm{T} 6$ | - | 0.6 | - | 0.25 | - | 1.0 | 0.25 | - | - | - | R |
| $6061-\mathrm{T} 6$ | - | 0.50 | 0.30 | 0.20 | 0.05 | 0.95 | 0.15 | - | 0.06 | 0.04 | R |

${ }^{a} \mathrm{R}=$ remainder.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | $\begin{aligned} & \text { Maximum } \\ & \text { Pit Depth } \\ & \text { (mils) } \end{aligned}$ | Crevice <br> Depth ${ }^{b}$ <br> (mils) | Type ${ }^{\text {b }}$ |  |
| 6061 | W | 402 | 2,370 | 1.2 | 0 | 32 | C (PR) | INCO (3) |
| 6061 | S | 402 | 2,370 | 0.7 | 0 | 32 | $C$ (PR) | INCO (3) |
| 6061 | W | 181 | 5 | 1.2 | 5 | 0 | P | INCO (3) |
| 6061 | W | 366 | 5 | 0.9 | 11 | 11 | C; P | INCO (3) |
| 6061-T6 | W | 123 | 5,640 | $<0.1$ | 5 | 10 | $\mathrm{C} ; \mathrm{P}^{d}$ | CEL (4) |
| $6061-\mathrm{T} 6^{\text {e }}$ | W | 123 | 5,640 | - | 13 | 0 | EX-IG; SC-P | NADC (7) |
| $6061-\mathrm{T} 6^{e, f}$ | W | 123 | 5,640 | - | SH | 0 | SH-P; WB\&HAZ (B; EX-IG; P (30)) | NADC (7) |
| 6061-T6 | W | 123 | 5,640 | 5.2 | 27 | 21 | C; P | MEL (5) |
| 6061-T6 ${ }^{\text {g }}$ | W | 123 | 5,640 | - | - | - | F-PP | BELL (13) |
| $6061-\mathrm{T} 6^{\text {b }}$ | W | 123 | 5,640 | 0 | 0 | 0 | NC | BELL (13) |
| 6061-T6 | S | 123 | 5,640 | 1.3 | 32 | 32 | C; P | CEL (4) |
| 6061-T6 | W | 189 | 5,900 | 0.1 | 3 | 0 | SC-P | CEL (4) |
| 6061-T6 | S | 189 | 5,900 | 0.1 | 33 | 18 | C; SC-P ${ }^{i}$ | CEL (4) |
| 6061-T6 | W | 403 | 6,780 | 1.0 | 58 | 55 | S-C; S-P | CEL (4) |
| 6061-T6 | W | 403 | 6,780 | - | - | - | MO-CR | NADC (7) |
| 6061-T6 | S | 403 | 6,780 | 0.7 | 60 | 51 | S-C; S-P | CEL (4) |
| 6061-T6 | W | 751 | 5,640 | 1.1 | 65 | 65 | S-C; E; S-P | CEL (4) |
| $6061-\mathrm{T} 6^{e}$ | W | 751 | 5,640 | - | - | - | CR (PR) | NADC (7) |
| 6061-T6 | W | 751 | 5,640 | 2.6 | 63 | 63 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{S}-\mathrm{E} ; \mathrm{P}(\mathrm{PR})$ | MEL (5) |
| 6061-T6 ${ }^{\text {g }}$ | W | 751 | 5,640 | - | PR | 0 | $\mathrm{P}(\mathrm{PR}) ; \mathrm{HAZ}(\mathrm{PR})$ | BELL (13) |
| $6061-\mathrm{T} 6^{\text {b }}$ | W | 751 | 5,640 | - | PR | 0 | $\mathrm{P}(\mathrm{PR})$ | BELL (13) |
| 6061-T6 | S | 751 | 5,640 | 1.6 | 72 | 75 | S-C; E; S-P | CEL (4) |
| 6061-T6 | W | 1,064 | 5,300 | 0.8 | 77 | 66 | S-C; E; S-P | CEL (4) |
| 6061-T6 | S | 1,064 | 5,300 | 1.0 | 77 | 0 | E; S-P | CEL (4) |
| 6061-T6 | W | 197 | 2,340 | 1.2 | 47 | 42 | C; P | CEL (4) |
| 6061-T6 ${ }^{\text {e }}$ | W | 197 | 2,340 |  | 36 | 0 | IG; P | NADC (7) |
| 6061-T6 | S | 197 | 2,340 | 1.3 | 46 | I | I-C; P | CEL (4) |

Table 74. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{c}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | $\begin{aligned} & \text { Maximum } \\ & \text { Pit Depth } \\ & \text { (mils) } \end{aligned}$ | Crevice <br> Depth ${ }^{b}$ <br> (mils) | Type ${ }^{\text {b }}$ |  |
| 6061-T6 | W | 402 | 2,370 | 2.0 | 75 | 66 | S-C; S-P | CEL (4) |
| 6061-T6 | W | 402 | 2,370 | - | S | 0 | S-P | NADC (7) |
| 6061-T6 | S | 402 | 2,370 | 1.2 | 76 | 60 | S-C; S-P | CEL (4) |
| 6061-T6 | W | 181 | 5 | 1.0 | 1 | 8 | C; I-P | CEL (4) |
| 6061-T6 | W | 398 | 5 | 0.7 | 16 | 0 | E; P | CEL (4) |
| 6061-T6 | W | 540 | 5 | 0.3 | 23 | 1 | 1-C; P | CEL (4) |

${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimens were embedded in the bottom sediments.
${ }^{b}$ Symbols for types of corrosion:
$\mathrm{MO}=$ Moderate
$\mathrm{NC}=$ No visible corrosion
$=$ Pitting
$\mathrm{PR}=$ Perforated
$\mathrm{S}=$ Severe
$\begin{aligned} \mathrm{SC} & =\text { Scattered } \\ \mathrm{SH} & =\text { Shallow }\end{aligned}$
$\mathrm{WB}=$ Weld Bead

Table 75. Stress Corrosion of 6000 Series Aluminum Alloys

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure <br> (day) | Depth <br> (ft) | Number of Specimens Exposed | Number Failed | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 6061-T6 | 12 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| 6061-T6 | 30 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| 6061-T6 | 12 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 6061-T6 | 30 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 6061-T6 | 12 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 6061-T6 | 30 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |

${ }^{a}$ Numbers refer to references at end of report.

Table 76. Changes in Mechanical Properties of 6000 Series Aluminum Alloys Due to Corrosion

| Alloy | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original <br> (\%) | \% Change |  |
| 6061-T6 ${ }_{\text {b }}$ | 123 | 5,640 | 48 | -7 | 41 | -2 | 17 | -58 | CEL (4) |
| 6061-T6 ${ }^{\text {b }}$ | 123 | 5,640 | 44 | -43 | 38 | -43 | 12 | -85 | NADC (7) |
| 6061-T6 ${ }^{\text {b,c }}$ | 123 | 5,640 | 44 | -100 | 38 | -100 | 12 | -100 | NADC (7) |
| 6061-T6 | 403 | 6,780 | 48 | -11 | 41 | -19 | 17 | -58 | CEL (4) |
| 6061-T6 | 751 | 5,640 | 48 | -24 | 41 | -41 | 17 | -73 | CEL (4) |
| 6061-T6 | 1,064 | 5,300 | 48 | -11 | 41 | -9 | 17 | -61 | CEL (4) |
| 6061-T6 | 197 | 2,340 | 48 | -5 | 41 | 0 | 17 | -42 | CEL (4) |
| 6061-T6 | 197 | 2,340 | 44 | -2 | 38 | +3 | 12 | -57 | NADC (7) |
| 6061-T6 | 402 | 2,370 | 48 | -15 | 41 | -9 | 17 | -70 | CEL (4) |
| 6061-T6 | 181 | 5 | 48 | 0 | 41 | +2 | 17 | -9 | CEL (4) |

${ }^{a}$ Numbers refer to references at end of report.
${ }^{b}$ Transverse butt weld, 6061 electrode, TIG process.
${ }^{c}$ Transverse welds excessively corroded; not possible to obtain tensile specimens.
Table 77. Chemical Composition of 7000 Series Aluminum Alloys, Percent by Weight

Table 78. Corrosion Rates and Types of Corrosion of 7000 Scries Aluminum Alloys

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | $\begin{aligned} & \text { Maximum } \\ & \text { Pit Depth } b \end{aligned}$ (mils) | Crevice Depth (mils) | Type ${ }^{\text {b }}$ |  |
| 7002.T0 | W | 123 | 5,6+0 | - | 13 | 0 | E; 1G; SII-P; XF | NADC (7) |
| 7002-T6, welded ${ }^{\text {d }}$ | w | 123 | 5,640 |  | PR | 0 | S-E; MO-IG; SH-P; WB (PR); XF | NADC (7) |
| $7002-T 6^{d_{c} c^{\prime}}$ | w | 123 | 5,640 | - | PR | 0 | E; IG; SH-P; WB (PR) ; XF | NADC (7) |
| 7002-T6 | W | 403 | 6,780 | 1.2 | 62 | 0 | $\mathrm{E}_{;} \mathrm{P}(\mathrm{PR})$ | CEL (4) |
| 7002-T6 | W | 403 | 6,780 |  | 1 | 0 | I-P | NADC (7) |
| 7002-T\% | S | 403 | 6,780 | 3.2 | 62 | 0 | E; P (PR) | CEL (4) |
| 7002-T\% | W | 751 | 5,640 | - | 0 | 0 | NC | NADC (7) |
| 7002-T6, anodized | w | 751 | 5,640 | $\cdots$ | 0 | 0 | E | NADC (7) |
| $7002 \cdot \mathrm{~T} 6$, welded ${ }^{\text {d }}$ | W | 751 | 5,6+0 | - | 0 | 0 | WB (HIC) | NADC (7) |
| 7002-T6, welded ${ }^{\text {d }}$ | w | 197 | 2,340 | - | 0 | 0 | E; IG; U; WB (S; IG) | NADC (7) |
| 7002-T6 | W | 197 | 2,340 | $<0.1$ | 15 | 1 | P ; XF | REY (14) |
| $7002-\mathrm{T} 6$ | w | 197 | 2,340 | $<0.1$ | 0 | 0 | NC | Boeing (6) |
| 7002-T6 | w | 402 | 2,370 | 1.6 | 30 | 62 | C (PR); P | CEL (4) |
| 7002-T6, welded ${ }^{\text {d }}$ | W | 402 | 2,370 | - | SII | 0 | E; SII-P; WB\&HAZ (S) | NADC (7) |
| 7002-T6 | S | 402 | 2,370 | 1.9 | 5 | 62 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}^{f}$ | CEL (4) |
| Alclad 7002-T6 | w | 403 | 6.780 | 1.4 | 5 | 0 | $\mathrm{p} g$ | CEL (4) |
| Alclad 7002-T6 | w | 403 | 6,780 |  | 0 | 0 | G | NADC (7) |
| Alclad 7002-T6 | w | 403 | 6,780 | 0.2 | 4 | 0 | $\mathrm{p}^{2}$ | CEL (4) |
| Alclad 7002-T6 | w | 751 | 5,640 | - | 0 | 0 | G | NADC (7) |
| Alclad 7002-T6 | w | 197 | 2,340 | $<0.1$ | 6 | 0 | $\mathrm{p}^{2}$ | REY (14) |
| Alclad 7002-T6 | w | 402 | 2.370 | 0.8 | 5 | 5 | C; P | CEL (4) |
| Alclad 7002-T6 | w | 402 | 2,370 | - | 0 | 0 | SL-B | NADC (7) |
| Alclad 7002-T6 | S | 402 | 2,370 | 1.0 | 5 | 5 | $\mathrm{C}_{\mathrm{i}} \mathrm{P}^{\prime}$ | CEL (4) |
| 7106 | w | 197 | 2,340 | $<0.1$ | 0 | 0 | NC | Boeing (6) |
| X7106-W51, welded ${ }^{k}$ | W | 197 | 2,340 | - |  |  | NC | Boeing (6) |
| 7039-T6 | w | 123 | $5.6+0$ | 0.7 | 3 | 0 | $\mathrm{P} ; \mathrm{XF}$ | CEL ( $\psi$ ) |
| 7039-T6 | S | 123 | 5.640 | 2.1 | 38 | 2 | C; Pi XF | CEL (4) |
| 7039-T6 | w | +03 | 6,780 | - | - | - | EX-XF | CEL (4) |
| 7039-T6 | S | 403 | 6.780 | - | - | - | EX-XF | CEL (4) |
| 7039-T6 | W | 751 | 5,640 |  |  | -- | EX-XF | CEL (4) |
| 7039-T6 | S | 751 | 5,640 | - | - |  | EX-XF | CEL (4) |
| 7039-T6 | W | 197 | 2,340 | 1.9 | 27 | 0 | P: XF | CEL (4) |

Table 78. Continued.

| Alloy | Environment ${ }^{\text {t }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & (m p y) \end{aligned}$ | $\begin{aligned} & \text { Maximum } \\ & \text { Pit Depth } b \\ & \text { (mils) } \end{aligned}$ | Crevice Depth (mils) | Type ${ }^{\text {b }}$ |  |
| 7039 | W | 197 | 2,3+0) | $<0.1$ | 0 | 0 | NC: | Boeing (6) |
| 7039-T6 | S | 197 | 2,340 | 1.2 | 61 | 0 | D-P; XI: | CEL (4) |
| 7039-T6 | W | 402 | 2,370 | - | - | - | EX-XF | CEL (4) |
| 7039-16 | S | 402 | 2,370 | - | - | - | EX-XF | CEL (4) |
| 7039-T64 | W | 189 | 5,900 | 0.2 | 1 | 14 | C; I-P | CEL (4) |
| 7039-T64, welded ${ }^{\text {l }}$ | W | 189 | 5,900 | 0.3 | 41 | 1 | 1-C; $\mathrm{P} ; \mathrm{L}(11 \mathrm{AZ})$ | CEL ( 4 ) |
| 7039-T64 , | S | 189 | 5,900 | 0.3 | 40 | 45 | C; SC-P | (ELL (4) |
| 7039-T64, welded | S | 189 | 5,900 | 0.3 | 47 | 1 | 1-C; P; P (HAZ) | CEL (t) |
| 7039-T64 , | W | 181 | 5 | 1.1 | 0 | 0 | ET | CEL. (t) |
| 7039-T64, welded ${ }^{\text {l }}$ | W | 181 | 5 | 1.2 | 1 | 0 | 1-P | CEL (4) |
| 7039-T64 | W | 398 | 5 | 1.1 | 22 | 1 | I-C; P | CEL (4) |
| 7039-T6t, welded ${ }^{\text {l }}$ | W | 398 | 5 | 0.5 | 0 | 0 | P (WB\&HAZ) | CEL (4) |
| 7039-T64 | W | 540 | 5 | 0.3 | 6 | 3 | C; ${ }^{\text {P }}$ | CEL (4) |
| 7039-T64, welded | w | 540 | 5 | 0.3 | 18 | 1 | I-C; P (HAZ) | CEL (4) |
| 7039-T64, welded ${ }^{\text {l }}$ | W | 588 | 5 | 0.3 | 26 | 1 | H-C; P (HAZ) | CEL (4) |
| 7075-T6 | W | 123 | 5,640 | - | 8 | 0 | E; IG; L; G-P | NADC (7) |
| 7075-T6 | W | 403 | 6,780 | - | PR | 0 | $\mathrm{P}(\mathrm{PR})$ | NADC (7) |
| 7075-T6 | S | +03 | 6,780 | - | PR | 0 | P (PR) | NADC (7) |
| 7075-T6A | W | 403 | 6,780 | - | PR | 0 | $\mathrm{P}(\mathrm{PR}) ; \mathrm{XF}$ (E) | NADC (7) |
| 7075-T6 | w | 751 | 5,640 | - | PR | - | C; $\mathrm{P}(\mathrm{PR})$; XF (E) | NADC (7) |
| 7075-T6A | W | 751 | 5,640 | - | PR | 0 | $\mathrm{P}(\mathrm{PR}) ; \mathrm{XF}$ (E) | NADC (7) |
| 7075-T6 | W | 1,06+ | 5,300 | - | PR | 0 | P (PR) ; S-XF (E) | NADC (7) |
| 7075-T6A | w | 1,064 | 5,300 | - | PR | 0 | $\mathrm{P}(\mathrm{PR})$; XF ( E$)$ | NADC (7) |
| 7075-T6 | W | 197 | 2,340 | - | 31 | - | C; IG;P;S-XF (E) | NADC (7) |
| 7075-T6 | S | 197 | 2,340 | - | SL | - | C; IG; SL-P; XF (E) | NADC (7) |
| 7075-T6A | W | 197 | 2,340 | - | 44 | 0 | SL-IG; SC-P | NADC (7) |
| 7075-T6A | S | 197 | 2,340 | - | 44 | 0 | SL-IG; N-P | NADC (7) |
| 7075-T6 | W | 402 | 2,370 | - | SL | 0 | S-E; SL-P | NADC (7) |
| 7075-T6 | S | 402 | 2,370 | - | SL | 0 | S-E; SL-P | NADC (7) |
| 7075-T6A | w | 402 | 2,370 | - | PR | 0 | P (PR) | NADC (7) |
| 7075-T6A | S | 402 | 2,370 | - | PR | 0 | P (PR) | NADC (7) |


| Alloy | Enviromment ${ }^{\text {a }}$ | Fxposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & (m p y) \end{aligned}$ | $\begin{aligned} & \text { Maximum } \\ & \text { Pit Depth }{ }^{b} \\ & \text { (mils) } \end{aligned}$ | Crevice Depth (mils) | Type ${ }^{6}$ |  |
| 7075-164 | w | 123 | 5,640 | - | 40 | 0 | EX-P | NADC (7) |
| 7075.773 | w | 123 | 5,640 | - | 20 | 0 | SC-P | NADC (7) |
| 7075-773 | w | 403 | 6,780 | - | SII | 0 | SII-P: SL-XF (E) | NADC (7) |
| 7075-173 | s | 403 | 6,780 | - | SII | 0 | SII-P: SL-XF (E) | NADC (7) |
| 7075-173 | w | 751 | 5,640 | - | PR | 0 | P (PR) | NADC (7) |
| 7075-173 | w | 197 | 2,340 | - | 0 |  | C; E; SL-IG | NADC (7) |
| 7075-173 | W | 402 | 2,370 | - | 31 | 0 | B (E) $\mathrm{P}_{\text {; }} \mathrm{XF}$ (E) | NADC (7) |
| 7075-173 | s | 402 | 2,370 | - | 0 | 0 | B | NADC (7) |
| Alctad 7075-0 | w | 1,064 | 5,300 | - | - | - | U | CEL (4) |
| Aclad 7075-76 | w | 123 | 5,640 | - | - | - | B; 1-IG | NADC (7) |
| 7079.16 | w | 123 | 5,640 | 5.2 | 79 | 36 | $C_{i} \mathrm{E}_{\mathrm{i}} \mathrm{P}$ (PR) $; \mathrm{XF}$ | CFL (4) |
| 7079-T6 | w | 123 | 5,640 | 0.7 | 4 | 9 | $\mathrm{C}_{i} \mathrm{P}$ | MEL (5) |
| 7079-T6 | S | 123 | 5,640 | 3.0 | 77 | 32 | $\mathrm{C}_{;} \mathrm{E} ; \mathrm{P}$ ( PR$) ; \mathrm{XF}$ | CEL (4) |
| 7079-T6 | w | 403 | 6,780 | 5.1 | 77 | 36 | $C_{i} \mathrm{~S}-\mathrm{L}: \mathrm{P}$ (PR) | CEL (4) |
| 7079-T6 | S | 403 | 6.780 | 3.4 | 57 | 38 | C; S-E; $\mathrm{P}^{\text {P }}$ | CELL ( 4 ) |
| 7079-T6 | w | 751 | 5,640 | 2.8 | 79 | 79 | $\mathrm{C}(\mathrm{PR}) ; \mathrm{P}(\mathrm{PR}) ; \mathrm{XF}$ | CEL (4) |
| 7079-T6 | w | 751 | 5,640 | 2.2 | $+1$ | 80 | $\mathrm{S}_{-\mathrm{C}}^{\mathrm{i}} \mathrm{P}$ | MLEI, (5) |
| 7079-T6 | S | 751 | 5,640 | 2.7 | 77 | 77 | $\mathrm{C}(\mathrm{PR})$; P (PR) ; XF | CEL (4) |
| 7079-16 | w | 197 | 2,340 | 4.7 | 77 | 46 | C; E; P (PR); XF | CEL ( 4 ) |
| 7079-T6 | S | 197 | 2,340 | 3.0 | 49 | 55 | C; E; P; XF | CEL (4) |
| 7079-T6 | w | 402 | 2,370 | 1.6 | 16 | 35 | C; S-E; P | CEL (4) |
| 7079-T6 | s | 402 | 2,370 | 1.4 | 21 | 27 | C; S-E; P | CEL (4) |
| Alclad 7079-T6 | w | 197 | 2,340 | <0.1 | 3 | 0 | $p^{\prime \prime \prime}$ | REY (14) |
| 7178-T6 | w | 123 | 5,640 | 4.1 | 43 | 0 | $\mathrm{P}_{\mathrm{i}} \mathrm{XF}$ | CFLL (4) |
| 7178-T6 | w | 123 | 5,640 | - | 63 | 0 | P (PR) : S-IG | NADC (7) |
| 7178-T6 | w | 403 | 6,780 | 4.9 | 15 | 35 | C; S-E; P; XF | CELL (4) |
| 7178-T6 | w | 403 | 6,780 | - | PR | 0 | S-E; P (PR) | NADC (7) |
| 7178-T6 | s | 403 | 6,780 | 4.4 | 25 | 30 |  | CEL (4) |
| 7178-T6 | w | 751 | 5,640 | 1.5 | 36 | 30 | C: S-E; P; XF | CEL ( 4 ) |
| 7178-T6 | w | 751 | 5,640 | 1 | PR | 0 | S-E; P (PR) | NADC (7) |
| 7178-T6 | w | 1,064 | 5,300 | 1.4 | 45 | 39 | C; S E E P $;$ XF | CELL (t) |

Table 78. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  |  |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | $\begin{aligned} & \text { Maximum } \\ & \text { Pit Depth }{ }^{b} \\ & \text { (mils) } \end{aligned}$ | Crevice Depth $^{b}$ <br> (mils) | Type ${ }^{\text {b }}$ |  |
| 7178-T6 | W | 197 | 2,340 | 3.2 | 0 | - | C; XF | CEL (4) |
| 7178-T6 | W | 197 | 2,340 | 0.2 | 50 | 0 | P (PR) | REY (14) |
| 7178-T6 | W | 197 | 2,340 | - | 31 | - | $C ; I G ; P ; S-X F(E)$ | NADC (7) |
| 7178-T6 | S | 197 | 2,340 | - | SL | - | C; IG; SL-P; XF (E) | NADC (7) |
| 7178-T6 | S | 197 | 2,340 | 2.4 | 0 | - | C; XF | CEL (4) |
| 7178-T6 | W | 402 | 2,370 | 3.3 | 34 | 34 | C; E; P; XF | CEL (4) |
| 7178-T6 | W | 402 | 2,370 | - | 82 | 0 | S -E; P (PR) | NADC (7) |
| 7178-T6 | S | 402 | 2,370 | - | - | 0 | E; P | NADC (7) |
| 7178-T6 | S | 402 | 2,370 | 1.8 | 31 | 34 | C; E; P; XF | CEL (4) |
| Alclad 7178-T6 | W | 751 | 5,640 | - | PR | 0 | $\mathrm{P}(\mathrm{PR})$ | NADC (7) |


bottom sediments.
prationad $=\mathrm{yd}$
$=$ Severe
$=$ Scattered
= Shallow
$=$ Slight
$=$ Uniform
$=$ Weld Bead
$=$ Weld Bead
$=$ Exfoliated (delaminated)

S
SC
SH
SL
U
WB
${ }^{k}$ Welded by the TIG process and aged.
$\mathrm{N}=$ Numerous
$\mathrm{NC}=$ No visible corrosion
$\mathrm{P}=$ Pitting
$\underset{=}{\mathscr{U}}-\underset{\sim}{\sum} \mathrm{z} \underset{\sim}{\mathcal{Z}} \approx$
${ }^{d}$ Transverse butt weld, MRD 7-5 electrode, TIG process.
${ }^{c}$ Numbers refer to references at end of report.

$$
\mathrm{HC}=\text { Honeycomb }
$$

$=$ Incipient
$=$ Intergranular
Line
Moderate
$=$ Numerous

$$
\mathrm{P} \quad=\text { Pitting }
$$

${ }^{\prime}$ Transverse butt welded, 7039 wire, MilG process. ${ }^{m} 2.6$-mil-thick cladding. ${ }^{e}$ Reheat treated to T6 condition after welding. $f_{\text {Part of one specimen missing. }}$
$g_{80} \%$ of cladding gone. ${ }^{1} 0 \%$ of cladding gone. ${ }^{i} 3.1$-mil-thick cladding.
$j 15 \%$ of cladding gone.

Table 79. Stress Corrosion of 7000 Series Aluminum Alloys

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure <br> (day) | Depth <br> (ft) | Number of Specimens Exposed | Number Failed | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| 7002-T6 | 18 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| 7002-T6 | 45 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| 7002-T6 | 18 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 7002-T6 | 30 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 7002-T6 | 45 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 7002-T6 | 45 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Alclad 7002-T6 | 18 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| Alclad 7002-T6 | 45 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| Alclad 7002-T6 | 18 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| Alclad 7002-T6 | 29 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| Alclad 7002-T6 | 45 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| Alclad 7002-T6 | 43 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 7075-T6 | 22 | 30 | 403 | 6,780 | 3 | 1 | NADC (7) |
| 7075-T6 | 55 | 75 | 403 | 6,780 | 3 | $b$ | NADC (7) |
| 7075-T6 | 22 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 7075-T6 | 55 | 75 | 197 | 2,340 | 3 | 2 | NADC (7) |
| 7075-T6 | 22 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 7075-T6 | 55 | 75 | 402 | 2,370 | 3 | 2 | NADC (7) |
| 7075-T6A ${ }^{\text {c }}$ | 12 | 30 | 403 | 6.780 | 3 | $b$ | NADC (7) |
| 7075-T6A ${ }^{c}$ | 30 | 75 | 403 | 6,780 | 3 | $b$ | NADC (7) |
| $7075-\mathrm{T} 6 \mathrm{~A}^{c}$ | 12 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 7075-T6A ${ }^{\text {c }}$ | 30 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 7075-T6A ${ }^{\text {c }}$ | 12 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 7075-T6A ${ }^{\text {c }}$ | 30 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 7075-T73 | 24 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| 7075-T73 | 51 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| 7075-T73 | 24 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 7075-T73 | 51 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 7075-T73 | 24 | 30 | 402 | 2.370 | 3 | 0 | NADC (7) |
| 7075-T73 | 51 | 75 | +02 | 2,370 | 3 | 0 | NADC (7) |
| 7079-T6 | 34 | 50 | 402 | 2,370 | 3 | 2 | CEL (4) |
| 7079-T6 | 50 | 75 | 402 | 2,370 | 3 | 3 | CEL ( 4 ) |
| Alclad 7079-T6 | 35 | 50 | 402 | 2,370 | 3 | 1 | CEL (4) |
| Alclad 7079-T6 | 53 | 75 | 402 | 2,370 | 3 | 2 | CEL (4) |
| 7178-T6 | 19 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| 7178-T6 | 49 | 75 | 403 | 6,780 | 3 | $b$ | NADC (7) |
| 7178-T6 | 19 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 7178-T6 | 49 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| 7178-T6 | 19 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| 7178-T6 | 41 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 7178-T6 | 49 | 75 | 402 | 2,370 | 3 | 3 | NADC (7) |
| 7178-T6 | 61 | 75 | 402 | 2,370 | 3 | 3 | CLL (4) |
| XAPOO1-T6 | 6 | 30 | 403 | 6,780 | 3 | $b$ | NADC (7) |
| XAP001-T6 | 15 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |

[^24]Table 80. Changes in Mechanical Properties of 7000 Series Aluminum Alloys Due to Corrosion

| Alloy | Exposure (day) | Depth <br> (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (ksi) | \% Change | Original (ksi) | \% Change | Original (\%) | \% Change |  |
| 7002-T6 ${ }_{6}$ | 123 | 5,640 | 76 | -10 | 68 | -12 | 12 | -8 | NADC (7) |
| 7002-T6 ${ }^{\text {b }}$ | 123 | 5,640 | 33 | -32 | - | - | 2 | -100 | NADC (7) |
| 7002-T6 $6^{\text {b,c,d }}$ | 123 | 5,640 | - | -100 | - | -100 | - | -100 | NADC (7) |
| 7002-T6 | 403 | 6,780 | 70 | -62 | 58 | -71 | 14 | -91 | CEL (4) |
| 7002-T6 | 403 | 6,780 | 76 | -12 | 68 | -15 | 12 | -8 | NADC (7) |
| 7002-T6 ${ }_{\text {b }}{ }^{\text {d }}$ | 751 | 5,640 | 76 | -8 | 68 | -12 | 12 | +17 | NADC (7) |
| 7002-T $6^{\text {b,d }}$ | 197 | 2,340 | - | -100 | - | -100 | - | $-100$ | NADC (7) |
| 7002-T6 | 402 | 2,370 | 70 | -1 | 58 | 0 | 14 | 0 | CEL (4) |
| 7002-16 | 402 | 2,370 | 76 | -10 | 68 | -16 | 12 | -14 | NADC (7) |
| Alclad 7002-T6 | 403 | 6.780 | 65 | -1 | 54 | +3 | 15 | -14 | CEL (4) |
| Alclad 7002-T6 | 402 | 2,370 | 65 | -3 | 54 | 0 | 15 | -10 | CEL (4) |
| 7039-T6 | 123 | 5,640 | 69 | -4 | 60 | -3 | $1+$ | -16 | CEL (4) |
| $7039-\mathrm{T}^{\text {d }}{ }_{d}$ | 403 | 6,780 | 69 | $-100$ | 60 | $-100$ | $1+$ | $-100$ | CEL ( 4 ) |
| 7039-T6 ${ }^{\text {d }}$ | 751 | 5,640 | 69 | -100 | 60 | -100 | 14 | -100 | CEL (4) |
| 7039-T6 | 197 | 2,340 | 69 | -2 | 60 | -2 | 14 | -2 | CEL (4) |
| 7039-T6 ${ }^{\text {d }}$ | 402 | 2,370 | 69 | -100 | 60 | -100 | 14 | -100 | CEL (4) |
| 7039-T64 | 181 | 5 | 63 | +2 | 53 | +6 | 16 | +13 | CEL (4) |
| 7039-T64 ${ }^{\text {e }}$ | 181 | 5 | 63 | $+17$ | 53 | +4 | 16 | +67 | CEL (4) |
| 7075-T6 | 123 | 5,640 | 83 | 0 | 76 | +1 | 11 | -5 | NADC (7) |
| 7075-T6 | 403 | 6,780 | 83 | -37 | 76 | -28 | 11 | -94 | NADC (7) |
| 7075-T6 | 751 | 5,640 | 83 | -13 | 76 | -9 | 11 | -84 | NADC (7) |
| 7075-T6 | 1,064 | 5.300 | 83 | -13 | 76 | -10 | 11 | -52 | NADC (7) |
| 7075-T6 | 197 | 2,340 | 83 | -8 | 76 | -4 | 11 | -55 | NADC (7) |
| 7075-T6 | 402 | 2,370 | 83 | -20 | 76 | - | 11 | -91 | NADC (7) |
| $7075-\mathrm{T} 6 \mathrm{~A}^{f}$ | 197 | 2,340 | 71 | -22 | 61 | -18 | 10 | -10 | NADC (7) |
| 7075-T6A | 402 | 2,370 | 71 | +7 | 61 | -10 | 10 | -20 | NADC (7) |
| 7075-T64 | 123 | 5.640 | 71 | $-41$ | 61 | -31 | 10 | -100 | NADC (7) |
| 7075-773 | 123 | 5,6+0 | 72 | 0 | 62 | 0 | 10 | +10 | NADC (7) |
| 7075-T73 | 403 | 6,780 | 72 | -11 | 62 | -3 | 10 | -75 | NADC (7) |
| 7075-T73 | 751 | 5,6+0 | 72 | -8 | 62 | -8 | 10 | 0 | NADC (7) |
| 7075-T73 | 197 | 2,340 | 72 | -3 | 62 | 0 | 10 | +10 | NADC (7) |
| 7075-T73 | 402 | 2,370 | 72 | 0 | 62 | +9 | 10 | -70 | NADC (7) |
| Alclad 7075-T6 | 123 | 5,640 | 76 | 0 | 68 | 0 | 10 | -10 | NADC (7) |
| 7079-T6 | 123 | 5,640 | 76 | -66 | 67 | -63 | 11 | -74 | CEL (4) |
| 7079-T6 | 403 | 6,780 | 76 | -43 | 67 | -52 | 11 | -70 | CEL (4) |
| 7079-T6 ${ }^{\text {d }}$ | 751 | 5,640 | 76 | -100 | 67 | -100 | 11 | -100 | CEL (4) |
| 7079-T6 | 197 | 2,3+0 | 76 | -17 | 67 | -4 | 11 | -23 | CEL (4) |
| 7079-T6 | 402 | 2.370 | 76 | -1 | 67 | 0 | 11 | -22 | CEL (4) |
| Alclad 7079-T6 | 402 | 2,370 | 78 | -2 | 69 | -1 | 12 | +4 | CEL (4) |
| 7178-76 | 123 | 5,640 | 88 | -19 | 80 | -16 | 10 | -16 | CEL (4) |
| 7178-T6 ${ }^{\text {g }}$ | 123 | 5,640 | 89 | -49 | 65 | - | 11 | -91 | NADC (7) |
| 7178-T6 | 403 | 6.780 | 88 | -41 | 80 | -41 | 10 | -82 | CEL (4) |
| 7178-T6 | 403 | 6,780 | 89 | -56 | 65 | - | 11 | -91 | NADC (7) |

Table 80. Continued.

| Alloy | Exposure (day) | Depth (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | $\begin{gathered} \text { Original } \\ (\text { ksi }) \end{gathered}$ | \% Change | Original (ksi) | \% Change | Original ( ${ }^{\circ} \mathrm{o}$ ) | \% Change |  |
| 7178-T6 | 751 | 5.640 | 88 | -48 | 80 | -64 | 10 | -58 | CEL (4) |
| 7178-T6 | 751 | $5.6+0$ | 89 | -90 | 65 | - | 11 | -91 | NADC (7) |
| 7178-T6 | 1,064 | 5,300 | 88 | -4 | 80 | -33 | 10 | +12 | CEL (4) |
| 7178-T6 | 197 | 2,340 | 88 | -19 | 80 | -37 | 10 | -52 | CEL (4) |
| 7178-T6 | 197 | 2,340 | 89 | -17 | 65 | 0 | 11 | -10 | NADC (7) |
| 7178-T6 | 402 | 2,370 | 88 | -24 | 80 | -21 | 10 | $-80$ | CEL (4) |
| 7178-T6 | 402 | 2,370 | 89 | -58 | 65 | - | 11 | $-100$ | NADC (7) |

${ }^{4}$ Numbers refer to references at end of report.
$b_{\text {Welded, MDR } 7-5 \text { electrode, TIG process. }}$
${ }^{c}$ Reheat treated to T6 condition after welding.
${ }^{d}$ Specimens too corroded (exfoliated) to machine into tensile specimens.
e'Transverse butt weld, 7039 wire, MIG process.

$g_{\text {Properties transverse to the direction of rolling. }}$

## SECTION 7

## TITANIUM ALLOYS

Titanium and titanium alloys owe their corrosion resistance to a protective oxide film. This film resists attack by oxidizing solutions, in particular those containing chloride ions. It has outstanding resistance to corrosion and pitting in marine environments and other chloride salt solutions.

The chemical compositions of the titanium alloys are given in Table 81, their corrosion rates and types of corrosion in Table 82, their susceptibility to stress corrosion in Table 83, and the effects of exposure on their mechanical properties in Table 84.

### 7.1. CORROSION

The corrosion rates and type of corrosion of the titanium alloys are given in Table 82 .

Except for two alloys, there was no corrosion of any of the titanium alloys during exposures in surface seawater or at depths of 2,500 and 6,000 feet. Reference 15 reported a corrosion rate of 0.19 mpy for unalloyed titanium and of 0.18 mpy for $6 \mathrm{Al}-4 \mathrm{~V}$ after 123 days of exposure at the 6,000 -foot depth, but no corrosion of these same alloys after 751 days of exposure at the 6,000 -foot depth. Also, no visible corrosion was reported. For practical purposes these values are considered to be inconsequential. DeLuccia, Reference 17, reported cracking in the heat-affected zone parallel to the weld bead in alloy $6 \mathrm{Al}-4 \mathrm{~V}$ after 197 days of exposure at the 2,500 -foot depth. Investigation of the weldments showed that the welds had been made under improper conditions and were contaminated with oxygen which made them brittle.

Alloys $75 \mathrm{~A}, 0.15 \mathrm{Pd}, 5 \mathrm{Al}-2.5 \mathrm{Sn}, 6 \mathrm{Al}-4 \mathrm{~V}$, $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}, 6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}$, and $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}$ were both unwelded and welded. They were fusionwelded by the inert-gas shielded arc, nonconsumable tungsten electrode process (TIG). There were transverse butt welds across the 6 -inch dimension of the specimens and 3 -inch-diameter ring welds in the centers of $6 \times 12$-inch specimens. The welded specimens were intentionally not stress relieved in order to
simulate the conditions present in a welded structure, i.e., to retain the maximum residual internal welding stresses. The process of placing a circular weld in a specimen imposes very high residual stresses in the specimen. Such circular welds simulate multiaxial stresses imposed in structures or parts fabricated by welding. There was no visible corrosion of these welded alloys except for stress corrosion cracking of alloy $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}$. This will be discussed under 7.2.

Alloy $6 \mathrm{Al}-4 \mathrm{~V}$ was also exposed as:
(1) Wire, $0.020-0.045^{-}$, and 0.063 -inch diameter.
(2) Cables, $1 / 16$-inch ( $1 \times 19$ ), $1 / 4$-inch ( $6 \times 19$ ), 1/4-inch ( $6 \times 19$ ) with Type 304 stainless steei swaged ends, and $1 / 4$-inch ( $6 \times 19$ ) with ends tied with mild steel wire.
(3) Flash-welded tube.
(4) Flash-welded sphere.
(5) Piece from broken sphere.
(6) Welded rings 9.625 -inch OD x 1.125 -inch wide $\times 8.75$-inch ID. One ring was unstressed and the others were stressed up to a maximum of $60,000 \mathrm{psi}$.

There was no visible corrosion on any of the above specimens except for the AISI Type 304 swaged fittings and the mild steel wire. The faying surfaces of the Type 304 stainless steel fittings were severely attacked by crevice corrosion. The rate of this crevice corrosion was probably increased by the galvanic couple formed by the two dissimilar metals, with the stainless steel being the anode of the couple. The mild steel wire used to tie the end of one titanium cable was corroded almost through by galvanic corrosion; the mild steel wire was anodic to the titanium cable.

### 7.2. STRESS CORROSION

Specimens of the alloys were stressed in various ways and to values equivalent to $30,35,50$, and $75 \%$
of their respective yield strengths at the surface and at depths of 2,500 and 6,000 feet for different periods of time.

The majority of the specimens were deformed by bowing to obtain the desired tensile stress in the central 2 -inch length of the outer surface of the specimen. Many of these specimens, butt-welded by the TIG process, were positioned such that the transverse weld bead was at the apex of the bow in the 2 -inch length. Other specimens, $6 \times 12$-inch, had a 3 -inch-diameter circular weld bead placed in the center. The stresses induced by the welding operation were not relieved in order to retain the maximum residual stresses in the specimens. Still other specimens were in the shape of welded rings, 9-5/8 inches outside diameter, which were deformed different amounts in order to induce tensile stresses in the periphery at the ends of the restraining rods.

The results of the stress corrosion tests are given in Table 83. There were no stress corrosion cracking failures of any of the alloys, both unwelded and butt-welded, stressed at values equivalent to as high as $75 \%$ of their respective yield strengths for 180 days of exposure at the surface, 402 days at the 2,500 -foot depth, and 751 days at the 6,000 -foot depth, except for the butt-welded $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}$ alloy. The unrelieved butt-welded $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{~A} 1$ alloy failed by stress corrosion cracking when stressed at values equivalent to $75 \%$ ( $94,500 \mathrm{psi}$ ) of its yield strength after 35,77 , and 105 days of exposure at the surface in the Pacific Ocean. The stress corrosion cracks were in the heat-affected zones at the edges of and parallel to the weld beads.

The butt-welded $6 \times 12$-inch specimens of $13-\mathrm{V}-11 \mathrm{Cr}-3 \mathrm{Al}$ alloy failed by stress corrosion during 398,540 , and 588 days of exposure at the surface due to the unrelieved residual welding stresses. The stress corrosion cracks were perpendicular to and extended across the weld beads from side to side.

The $6 \mathrm{Al}-4 \mathrm{~V}$ alloy rings stressed as high as 60,000 psi (approximately $50 \%$ of its yield strength) did not fail by stress corrosion cracking during 402 days of exposure at the 2,500 -foot depth.

Alloys $75 \mathrm{~A}, 0.15 \mathrm{Pd}, 5 \mathrm{Al}-2.5 \mathrm{Sn}, 7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}$, $6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}, 6 \mathrm{Al}-4 \mathrm{~V}$, and $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}$ were exposed with an unrelieved 3 -inch-diameter circular weld bead in the center of $6 \times 12$-inch specimens. Only the $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}$ alloy failed by stress corrosion cracking because of the residual welding stresses. Failure by stress corrosion cracking occurred first after 181 days of exposure at the surface. Thereafter, failures first occurred during 189 days of exposure when partially embedded in the bottom sediments and during 751 days of exposure in the seawater at the 6,000 -foot depth. At the 2,500 -foot depth the first failure occurred during 402 days of exposure in the seawater. The cracks in all cases extended radially across the weld beads. In some cases, the cracks changed direction by $90 \%$ and propagated circumferentially around the outside of the weld bead. In general, the $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}$ alloy was more susceptible to stress corrosion cracking in seawater at the surface than at depth in the Pacific Ocean.

### 7.3. MECHANICAL PROPERTIES

The effects of exposure in seawater on the mechanical properties of the titanium alloys are given in Table 84. The mechanical properties of the titanium alloys were not adversely affected.
Table 81. Chemical Composition of Titanium Alloys

| Alloy | C | Fe | N | H | O | Al | V | Cr | Other | Ti ${ }^{\text {a }}$ | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Titanium | $<0.1$ | - | 0.02 | - | - | - | - | - | - | R | INCO (3) |
| 75A | 0.027 | 0.20 | 0.026 | 0.004 | - | - | - | - | - | R | CEL (4) |
| 75 A | 0.025 | 0.13 | 0.016 | 0.003 | 0.28 | - | - | - | - | R | CEL (4) |
| 75A | 0.025 | 0.14 | 0.017 | 0.003 | 0.32 | -- | - | - | - | R | CEL (4) |
| 0.15 Pd | 0.022 | 0.06 | 0.009 | 0.003 | 0.15 | - | - | - | 0.15 Pd | R | CEL (4) |
| 0.15 Pd | 0.022 | 0.05 | 0.012 | 0.004 | - | - | - | - | 0.15 Pd | R | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}$ | 0.025 | 0.34 | 0.014 | 0.011 | - | 5.1 | - | - | 2.2 Sn | R | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}$ | 0.022 | 0.27 | 0.013 | 0.006 | 0.18 | 5.1 | - | - | 2.4 Sn | R | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}$ | 0.025 | 0.35 | 0.013 | 0.008 | 0.17 | 5.1 | - | - | 2.5 Sn | R | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}$ | 0.023 | 0.06 | 0.006 | 0.002 | 0.07 | 7.0 | - | - | 1.0 Ta | R | CEL (4) |
| $4 \mathrm{Al}-3 \mathrm{Mo}-1 \mathrm{~V}$ | - | -- | - | - | - | 4.30 | 0.95 | - | 2.91 Mo | R | NADC (7) |
| $4 \mathrm{Al}-3 \mathrm{Mo}-1 \mathrm{~V}^{c}$ | 0.08 | 0.25 | 0.05 | 0.15 | - | 4.25 | 1.0 | - | 3.0 Mo | R | CEL (4) |
| 4Al-3Mo-1V | - | 0.1 | - | - | - | 4.5 | 0.9 | 0.2 | 3.7 Mo | R | CEL (4) |
| $8 \mathrm{Mn}^{\text {c }}$ | 0.20 | - | 0.07 | 0.15 | - | - | - | - | 8.0 Mn | R | CEL (4) |
| 140A | - | 1.9 | - | - | - | $<0.1$ | $<0.1$ | 2.1 | 1.9 Mo | R | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | - | - | - | - | - | 5.61 | 4.13 | - | -- | R | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 0.023 | 0.15 | 0.015 | 0.007 | - | 6.0 | 3.9 | - | - | R | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 0.025 | 0.13 | 0.015 | 0.007 | - | 5.9 | 4.0 | - | - | R | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 0.022 | 0.08 | 0.013 | 0.007 | 0.11 | 5.8 | 4.0 | -- | - | R | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | - | 0.1 | - | - | - | 7.2 | 5.2 | $<0.1$ | - | R | CEL (4) |
| $7 \mathrm{Al}-12 \mathrm{Zr}$ | - | - | - | - | - | 5.93 | -- | - | 11.27 Zr | R | NADC (7) |
| $6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}$ | 0.02 | 0.06 | 0.006 | 0.002 | 0.077 | 6.1 | - | - | $\begin{aligned} & 2.2 \mathrm{Cb} \\ & 1.1 \mathrm{Ta} \\ & 0.74 \mathrm{Mo} \end{aligned}$ | R | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}$ | 0.027 | 0.17 | 0.027 | 0.008 | - | 3.1 | 13.4 | 11.4 | - | R | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}$ | 0.021 | 0.14 | 0.027 | 0.010 | 0.12 | 3.0 | 13.6 | 10.9 | - | R | CEL (4) |

[^25]Table 82. Corrosion Rates and Types of Corrosion of Titanium Alloys

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Type ${ }^{\text {b }}$ |  |
| Titanium | W | 123 | 5,640 | $<0.1$ | NC | INCO (3) |
| Titanium | W | 123 | 5,640 | 0.19 | NC | MEL (5) |
| Titanium | S | 123 | 5,640 | $<0.1$ | NC | INCO (3) |
| Titanium | W | 403 | 6,780 | $<0.1$ | NC | INCO (3) |
| Titanium | S | 403 | 6,780 | $<0.1$ | NC | INCO (3) |
| Titanium | W | 751 | 5,640 | $<0.1$ | NC | INCO (3) |
| Titanium | W | 751 | 5,640 | $<0.1$ | NC | MEL (5) |
| Titanium | S | 751 | 5,640 | $<0.1$ | NC | INCO (3) |
| Titanium | W | 1,064 | 5,300 | $<0.1$ | NC | INCO (3) |
| Titanium | S | 1,064 | 5,300 | $<0.1$ | NC | INCO (3) |
| Titanium | W | 197 | 2,340 | $<0.1$ | NC | INCO (3) |
| Titanium | W | 197 | 2,340 | $<0.1$ | NC | Boeing (6) |
| Titanium | S | 197 | 2,340 | $<0.1$ | NC | INCO (3) |
| Titanium | W | 402 | 2,370 | $<0.1$ | NC | INCO (3) |
| Titanium | S | 402 | 2,370 | $<0.1$ | NC | INCO (3) |
| Titanium | W | 181 | 5 | $<0.1$ | NC | INCO (3) |
| Titanium | W | 366 | 5 | $<0.1$ | NC | INCO (3) |
| 75A | W | 123 | 5,640 | 0.0 | NC | CEL (4) |
| 75A | S | 123 | 5,640 | 0.0 | NC | CEL (4) |
| $75 \mathrm{~A}^{d}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $75 \mathrm{~A}^{e}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $75 A^{d}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| 75A ${ }^{e}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| 75A | W | 403 | 6,780 | 0.0 | NC | CEL (4) |
| 75A | S | 403 | 6,780 | 0.0 | NC | CEL (4) |
| 75A | W | 751 | 5,640 | 0.0 | NC | CEL (4) |
| 75A | W | 197 | 2,340 | 0.0 | NC | CEL (4) |
| 75A | S | 197 | 2,340 | 0.0 | NC | CEL (4) |
| 75A | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| 75 A | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| 75A | W | 181 | 5 | 0.0 | NC; FS | CEL (4) |
| $75 \mathrm{~A}^{d}$ | W | 181 | 5 | 0.0 | NC; FS | CEL (4) |
| $75 \mathrm{~A}^{e}$ | W | 181 | 5 | 0.0 | NC; FS | CEL (4) |
| 75A | W | 398 | 5 | 0.0 | NC | CEL (4) |
| $75 A^{d}$ | W | 398 | 5 | 0.0 | NC | CEL (4) |
| $75 \mathrm{~A}^{e}$ | W | 398 | 5 | 0.0 | NC | CEL (4) |
| 75 A | W | 540 | 5 | 0.0 | NC | CEL (4) |
| $75 A^{d}$ | W | 540 | 5 | 0.0 | NC | CEL (4) |
| $75 \mathrm{~A}^{e}$ | W | 540 | 5 | 0.0 | NC | CEL (4) |

Continued

Table 82. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Type ${ }^{\text {b }}$ |  |
| 75A | W | 588 | 5 | 0.0 | NC | CEL (4) |
| $75 \mathrm{~A}^{\text {d }}$ | W | 588 | 5 | 0.0 | NC | CEL (4) |
| $75 \mathrm{~A}^{e}$ | W | 588 | 5 | 0.0 | NC | CEL (4) |
| $0.15 \mathrm{Pd}^{d}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $0.15 \mathrm{Pd}^{e}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $0.15 \mathrm{Pd}^{d}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| $0.15 \mathrm{Pd}^{e}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| $0.15 \mathrm{Pd}^{d}$ | W | 181 | 5 | 0.0 | NC; FS | CEL (4) |
| $0.15 \mathrm{Pd}^{e}$ | W | 181 | 5 | 0.0 | NC; FS | CEL (4) |
| $0.15 \mathrm{Pd}^{d}$ | W | 398 | 5 | 0.0 | NC | CEL (4) |
| $0.15 \mathrm{Pd}^{e}$ | W | 398 | 5 | 0.0 | NC | CEL (4) |
| $0.15 \mathrm{Pd}^{d}$ | W | 540 | 5 | 0.0 | NC | CEL (4) |
| $0.15 \mathrm{Pd}^{e}$ | W | 540 | 5 | 0.0 | NC | CEL (4) |
| $0.15 \mathrm{Pd}^{d}$ | W | 588 | 5 | 0.0 | NC | CEL (4) |
| $0.15 \mathrm{Pd}^{e}$ | W | 588 | 5 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{\text {d }}$ | W | 123 | 5,640 | $<0.1$ | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{e}$ | W | 123 | 5,640 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | S | 123 | 5,640 | $<0.1$ | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{e}$ | S | 123 | 5,640 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{e}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{e}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | W | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{e}$ | W | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | S | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{e}$ | S | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | W | 751 | 5,640 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{e}$ | W | 751 | 5,640 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | W | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{e}$ | W | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{\text {d }}$ | S | 197 | 2.340 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{e}$ | S | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{\text {d }}$ | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{\text {e }}$ | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
|  | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | W | 181 | 5 | 0.0 | NC; FS | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{e}$ | W | 181 | 5 | 0.0 | NC; FS | CEL (4) |

Table 82. Continued.

| Alloy | Environment ${ }^{a}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Type ${ }^{\text {b }}$ |  |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | W | 398 | 5 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{e}$ | W | 398 | 5 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | W | 540 | 5 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{\text {e }}$ | W | 540 | 5 | 0.0 | NC | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | W | 588 | 5 | 0.0 | NC | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{\text {d }}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{e}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{d}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{e}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{d}$ | W | 181 | 5 | 0.0 | NC; FS | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{e}$, | W | 181 | 5 | 0.0 | NC; FS | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{\text {d }}$ | W | 398 | 5 | 0.0 | NC | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{e}$ | W | 398 | 5 | 0.0 | NC | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{\text {d }}$ | W | 540 | 5 | 0.0 | NC | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{e}$ | W | 540 | 5 | 0.0 | NC | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{\text {d }}$ | W | 588 | 5 | 0.0 | NC | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{e}$ | W | 588 | 5 | 0.0 | NC | CEL (4) |
| 4Al-3Mo-1V | W | 123 | 5,640 | 0.0 | NC | NADC (7) |
| 4Al-3Mo-1V | W | 403 | 6,780 | 0.0 | NC | NADC (7) |
| 4Al-3Mo-1V | S | 403 | 6,780 | 0.0 | NC; BD | NADC (7) |
| 4Al-3Mo-1V | W | 751 | 5,640 | 0.0 | NC | NADC (7) |
| 4Al-3Mo-1V | S | 751 | 5,640 | 0.0 | NC; BD | NADC (7) |
| $4 \mathrm{Al}-3 \mathrm{Mo}-1 \mathrm{~V}$ | W | 1,064 | 5,300 | 0.0 | NC | CEL (4) |
| 4Al-3Mo-1V | W | 1,064 | 5,300 | 0.0 | NC | NADC (7) |
| $4 \mathrm{Al}-3 \mathrm{Mo}-1 \mathrm{~V}$ | S | 1,064 | 5,300 | 0.0 | NC; BD | NADC (7) |
| 4Al-3Mo-1V | W | 197 | 2,340 | 0.0 | NC | NADC (7) |
| 4Al-3Mo-1V | S | 197 | 2,340 | 0.0 | NC; BD | NADC (7) |
| 4Al-3Mo-1V | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| 4Al-3Mo-1V | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| 8 Mn | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| 8 Mn | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| 140A | W | 1,064 | 5,300 | 0.0 | NC | CEL (4) |
| $7 \mathrm{Al}-12 \mathrm{Zr}$ | W | 123 | 5,300 | $<0.1$ | NC | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 123 | 5,640 | $<0.1$ | NC | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 123 | 5,640 | 0.18 | NC | MEL (5) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 123 | 5,640 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{d}$ | W | 123 | 5,640 | 0.0 | NC | CEL (4) |

Table 82. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Type ${ }^{\text {b }}$ |  |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | W | 123 | 5,640 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | S | 123 | 5,640 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {d }}$ | S | 123 | 5,640 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | S | 123 | 5,640 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{d}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{d}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{2}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 403 | 6,780 | 0.0 | NC | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{f}$ | W | 403 | 6,780 | 0.0 | NC | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{d}$ | W | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | W | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | S | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{d}$ | S | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | S | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 751 | 5,640 | 0.0 | NC | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 751 | 5,640 | $<0.1$ | NC | MEL (5) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 751 | 5,640 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{d}$ | W | 751 | 5,640 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | W | 751 | 5,640 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 1,064 | 5,300 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 197 | 2,340 | 0.0 | NC | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{f}$ | W | 197 | 2,340 | 0.0 | HAZ (CR) ${ }^{g}$ | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{d}$ | W | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | W | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | S | 197 | 2,340 | 0.0 | NC; BD | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | S | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {d }}$ | S | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | S | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 402 | 2,370 | 0.0 | NC | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{d}$ | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {d }}$ | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | $181$ | 5 | $0.0$ | NC: FS | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {d }}$ | W | 181 | 5 | 0.0 | NC; FS | CEL (4) |

Table 82. Continued.

| Alloy | Environment ${ }^{*}$ | Exposure (day) | Depth (ft) | Corrosion |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Type ${ }^{\text {b }}$ |  |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | W | 181 | 5 | 0.0 | NC; FS | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 398 | 5 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {d }}$ | W | 398 | 5 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | W | 398 | 5 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | W | 540 | 5 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{d}$ | W | 540 | 5 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | W | 540 | 5 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {d }}$ | W | 588 | 5 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{e}$ | W | 588 | 5 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}^{\text {d }}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}^{e}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| $6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}^{d}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| $6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}^{e}$ | S | 189 | 5,900 | 0.0 | NC; BD | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}$ | W | 123 | 5,640 | $<0.1$ | NC | MEL (5) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | W | 123 | 5,640 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | W | 123 | 5,640 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | S | 123 | 5,640 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | S | 123 | 5,640 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{\text {d }}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | W | 189 | 5,900 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | S | 189 | 5,900 | 0.0 | SCC ${ }^{\text {b }}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | S | 189 | 5,900 | 0.0 | NC, BD | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d, i}$ | W | 360 | 4,200 | 0.0 | $\mathrm{SCC}^{j}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d, i}$ | S | 360 | 4,200 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{\text {d }}$ | W | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | W | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | S | 403 | 6,780 | 0.0 | SCC ${ }^{j}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | S | 403 | 6,780 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}$ | W | 751 | 5,640 | $<0.1$ | NC | MEL (5) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{\text {d }}$ | W | 751 | 5,640 | 0.0 | $\mathrm{SCC}^{j}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}{ }^{\text {d }}$ | W | 751 | 5,640 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{\text {d }}$ | W | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | W | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | S | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | S | 197 | 2,340 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | S | 402 | 2,370 | 0.0 | SCC ${ }^{j}$ | CEL (4) |

Continued

Table 82. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure <br> (day) | Depth <br> (ft) | Corrosion |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | Rate (mpy) | Type ${ }^{b}$ |  |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | W | 181 | 5 | -0.0 | SCC ${ }^{j}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | W | 181 | 5 | 0.0 | NC; FS | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | W | 398 | 5 | 0.0 | $\operatorname{SCC}(6)^{j}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | W | 398 | 5 | 0.0 | $\mathrm{SCC}(2)^{k}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | W | 540 | 5 | 0.0 | $\operatorname{SCC}(12)^{\prime}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | W | 540 | 5 | 0.0 | $\operatorname{SCC}(1)^{k}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{\text {d }}$ | W | 588 | 5 | 0.0 | $\operatorname{SCC}(19){ }^{\prime}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{e}$ | W | 588 | 5 | 0.0 | $\operatorname{SCC}(1)^{k}$ | CEL (4) |

${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimens were embedded in the bottom sediments.
${ }^{b}$ Symbols for types of corrosion:
$\mathrm{BD}=$ Bluish discoloration on portion in bottom sediment.
$\mathrm{CR}=$ Cracked
$\mathrm{FS}=$ Fouling stains
HAZ $=$ Heat-affected zone
$\mathrm{NC}=$ No visible corrosion
$\mathrm{SCC}=$ Stress corrosion cracking
Numbers in parentheses are number of stress corrosion cracks
${ }^{c}$ Numbers refer to references at end of report.
${ }^{d}$ Unrelieved 3 -in.-diam weld bead in center of specimen, TIG process, nonconsumable tungsten electrode.
${ }^{e}$ Unrelieved butt weld across width of specimen, TIG process, nonconsumable tungsten electrode.
$f_{\text {TIG welded, commercially pure titanium filler metal used, unrelieved. }}$
${ }^{g}$ Cracked in heat-affected zone of unrelieved transverse butt weld.
${ }^{b}$ Two cracks in each specimen perpendicular to and across circular weld beads; some branching; penetrated through 0.125 -in.-thick plate. Bluish discoloration on portions of specimens in sediment.
${ }^{i}$ Exposed in Tongue-of-the-Ocean, Atlantic, $0.3-\mathrm{kt}$ current, $4.5^{\circ} \mathrm{C}, 5.18-\mathrm{ml} /$ i oxygen.
${ }^{j}$ Cracks radially across circular weld bead into base metal outside heat-affected zone.
${ }^{k}$ Cracks across weld bead normal to direction of welding.

Table 83. Stress Corrosion of Titanium Alloys

| Alloy | Stress <br> (ksi) | Percent <br> Yield Strength | Exposure <br> (day) | Depth <br> (ft) | Number of Specimens Exposed | Number Failed | Source ${ }^{a}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $75 A^{b}$ | 41 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $75 \mathrm{~A}^{\text {b,c }}$ | 41 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $75 \mathrm{~A}^{b}$ | 62 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $75 A^{b, c}$ | 62 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| 75A | 25 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| 75A | 35 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| 75A | 53 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| 75A | 25 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| 75A | 35 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| 75A | 53 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| 75A | 35 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| 75A | 53 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $75 \mathrm{~A}^{6}$ | 29 | 35 | 180 | 5 | 3 | 0 | CEL (4) |
| $75 A^{b}$ | 41 | 50 | 180 | 5 | 3 | 0 | CEL (4) |
| $75 A^{b}$ | 62 | 75 | 180 | 5 | 3 | 0 | CEL (4) |
| $75 \mathrm{~A}^{\text {d }}$ | $e$ | - | 181 | 5 | 4 | 0 | CEL (4) |
| $0.15 \mathrm{Pd}^{b}$ | 25 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $0.15 \mathrm{Pd}^{b, c}$ | 25 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $0.15 \mathrm{Pd}^{b}$ | 37 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $0.15 \mathrm{Pd}^{b, c}$ | 37 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $0.15 \mathrm{Pd}{ }^{b}$ | 24 | 35 | 180 | 5 | 3 | 0 | CEL (4) |
| $0.15 \mathrm{Pd}^{b}$ | 34 | 50 | 180 | 5 | 3 | 0 | CEL (4) |
| $0.15 \mathrm{Pd}^{b}$ | 51 | 75 | 180 | 5 | 3 | 0 | CEL (4) |
| $0.15 \mathrm{Pd}^{d}$ | $e$ | - | 181 | 5 | 4 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b}$ | 43 | 35 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b}$ | 61 | 50 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b}$ | 92 | 75 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{\text {d }}$ | $e$ | - | . 123 | 5,640 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{6}$ | 62 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b, c}$ | 62 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b}$ | 93 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b, c}$ | 93 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b}$ | 43 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{6}$ | 61 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{6}$ | 92 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | $e$ | -- | 403 | 6,780 | 2 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{6}$ | 43 | 35 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{6}$ | 61 | 50 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{6}$ | 92 | 75 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | $e$ | - | 751 | 5,640 | 3 | 0 | CEL (4) |

Continued

Table 83. Continued.

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure <br> (day) | Depth <br> (ft) | Number of Specimens Exposed | Number Failed | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{\text {b }}$ | 43 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b}$ | 61 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{6}$ | 92 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | $e$ | - | 197 | 2,340 | 2 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b}$ | $e$ | - | 402 | 2,370 | 2 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{6}$ | 43 | 35 | 180 | 5 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{\text {b }}$ | 62 | 50 | 180 | 5 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b}$ | 93 | 75 | 180 | 5 | 3 | 0 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{d}$ | $e$ | - | 181 | 5 | 4 | 0 | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}{ }^{\text {b }}$ | 50 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{\text {b,c }}$ | 50 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}{ }^{\text {b }}$ | 75 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{\text {b, }}$ c | 75 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}{ }^{\text {b }}$ | 35 | 35 | 180 | 5 | 3 | 0 | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{b}$ | 50 | 50 | 180 | 5 | 3 | 0 | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{\text {b }}$ | 75 | 75 | 180 | 5 | 3 | 0 | CEL (4) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{\text {d }}$ | $e$ | - | 181 | 5 | 4 | 0 | CEL (4) |
| $6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}^{b}$ | 60 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}^{\text {b }}$, ${ }^{\text {a }}$ | 60 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}^{\text {b }}$ | 89 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}-1 \mathrm{Mo}^{\text {b,c }}$ | 89 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 48 | 35 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{6}$ | 49 | 35 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 68 | 50 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {b }}$ | 70 | 50 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 102 | 75 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {b }}$ | 105 | 75 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {d }}$ | e | - | 123 | 5,640 | 2 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {b }}$ | 66 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {b, }}$ | 66 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {b }}$ | 99 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{b, c}$ | 99 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 43 | 30 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 48 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {b }}$ | 49 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 71 | 50 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 68 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{6}$ | 70 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 107 | 75 | 403 | 6,780 | 3 | 0 | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 102 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |

Continued

Table 83. Continued.

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure (day) | Depth <br> (ft) | Number of Specimens Exposed | Number Failed | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $6 \mathrm{Al}-4 \mathrm{~V}^{6}$ | 105 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{d}$ | $e$ | - | 403 | 6,780 | 2 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 48 | 35 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {b }}$ | 49 | 35 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 68 | 50 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {b }}$ | 70 | 50 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 102 | 75 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {b }}$ | 105 | 75 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{d}$ | $e$ | - | 751 | 5,640 | 2 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 43 | 30 | 197 | 2,340 | 3 | 0 | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 48 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {b }}$ | 49 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 68 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{6}$ | 70 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 107 | 75 | 197 | 2,340 | 3 | 0 | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 102 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{6}$ | 105 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {d }}$ | $e$ | - | 197 | 2,340 | 2 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 43 | 30 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 68 | 50 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 107 | 75 | 402 | 2,370 | 3 | 0 | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 102 | 75 | 402 | 2,370 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {d }}$ | $e$ | - | 402 | 2,370 | 2 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {b }}$ | 46 | 35 | 180 | 5 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{b}$ | 66 | 50 | 180 | 5 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{\text {b }}$ | 99 | 75 | 180 | 5 | 3 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{d}$ | $e$ | - | 181 | 5 | 4 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{\text {b }}$ | 49 | 35 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 70 | 50 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{\text {b }}$ | 105 | 75 | 123 | 5,640 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | $e$ | - | 123 | 5,640 | 2 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 63 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b, c}$ | 63 | 50 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 95 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b, c}$ | 95 | 75 | 189 | 5,900 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | $e$ | - | 189 | 5,900 | 2 | $1{ }^{f}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 49 | 35 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 70 | 50 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 105 | 75 | 403 | 6,780 | 2 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | $e$ |  | 403 | 6,780 | 2 | ${ }_{1} f$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{\text {b }}$ | 49 | 35 | 751 | 5,640 | 3 | 0 | CEL (4) |

Continued

Table 83. Continued.

| Alloy | Stress <br> (ksi) | Percent <br> Yield <br> Strength | Exposure <br> (day) | Depth <br> (ft) | Number of <br> Specimens <br> Exposed | Number <br> Failed | Source $^{a}$ |
| :--- | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 70 | 50 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 105 | 75 | 751 | 5,640 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al} \mathrm{d}^{d}$ | $e$ | - | 751 | 5,640 | 2 | $1^{g}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 49 | 35 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 70 | 50 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 105 | 75 | 197 | 2,340 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | $e$ | - | 197 | 2,340 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | $e$ | - | 402 | 2,370 | 2 | $1^{f}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 44 | 35 | 180 | 5 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 63 | 50 | 180 | 5 | 3 | 0 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 95 | 75 | 180 | 5 | 3 | $3^{b}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | $e$ | - | 181 | 5 | 4 | $1^{g}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | $e$ | - | 398 | 5 | 2 | $2^{i}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | $e$ | - | 398 | 5 | 2 | $2^{j}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | $e$ | - | 540 | 5 | 2 | $1^{i}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | $e$ | - | 540 | 5 | 2 | $2^{k}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | $e$ | - | 588 | 5 | 2 | $1^{i}$ | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{d}$ | $e$ | - | 588 | 5 | 2 | $2^{l}$ | CEL (4) |

${ }^{\text {a }}$ Numbers refer to references at end of report.
${ }^{b}$ Unrelieved butt weld across width of specimen, TIG process, nonconsumable tungsten electrode, weld at apex of bow.
${ }^{c}$ Partially embedded in bottom sediment.
${ }^{d}$ Unrelieved 3-in.-diam weld bead in center of specimen, TIG process, nonconsumable tungsten electrode.
${ }^{e}$ Residual welding stresses.
$f_{\text {Specimen partially embedded in bottom sediment; cracked radially across the weld bead. }}$
${ }^{g}$ Specimen in seawater; cracked radially across the weld bead.
${ }^{b}$ Specimens failed in heat-affected zones after 35,77 , and 105 days of exposure.
${ }^{i}$ Cracks across weld beads.
${ }^{j}$ Six cracks radially across weld beads.
${ }^{k}$ Twelve cracks radially across weld beads.
${ }^{\prime}$ Nineteen cracks radially across weld beads.

Table 84. Changes in Mechanical Properties of Titanium Alloys Due to Corrosion

| Alloy | $\begin{gathered} \text { Exposure } \\ \text { (day) } \end{gathered}$ | Depth (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (ksi) | \% Change | Original <br> (ksi) | \% Change | Original (\%) | \% Change |  |
| 75A | 123 | 5,640 | 87 | +4 | 70 | +5 | 30 | -5 | CEL (4) |
| 75A | 403 | 6,780 | 87 | +4 | 70 | +4 | 30 | -16 | CEL (4) |
| 75A | 751 | 5,640 | 87 | +4 | 70 | +11 | 30 | -15 | CEL (4) |
| 75A | 197 | 2,340 | 87 | +5 | 70 | +8 | 30 | -11 | CEL (4) |
| 75A | 402 | 2,370 | 87 | +6 | 70 | +7 | 30 | -13 | CEL (4) |
| 75 A | 181 | 5 | 87 | +8 | 70 | +9 | 30 | -18 | CEL (4) |
| $75 \mathrm{~A}^{\text {b }}$ | 181 | 5 | 101 | +2 | 84 | $-1$ | 25 | -24 | CEL (4) |
| $0.15 \mathrm{Pd}^{b}$ | 181 | 5 | 66 | +3 | 47 | +4 | 28 | -33 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b}$ | 403 | 6,780 | 130 | $+10$ | 123 | $+10$ | 14 | -13 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b}$ | 751 | 5,640 | 130 | +4 | 123 | +5 | 14 | -11 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b}$ | 402 | 2,370 | 130 | +5 | 123 | +4 | 14 | -15 | CEL (4) |
| $5 \mathrm{Al}-2.5 \mathrm{Sn}^{b}$ | 181 | 5 | 138 | $+4$ | 124 | +1 | 17 | -20 | CEL (4) |
| 8 Mn | 402 | 2,370 | 132 | $+2$ | 116 | +3 | 12 | $+33$ | CEL (4) |
| 4Al-3Mo-1V | 123 | 5,640 | 155 | 0 | 115 | -3 | 15 | 0 | NADC (7) |
| 4Al-3Mo-1V | 403 | 6,780 | 155 | +2 | 115 | +2 | 15 | 0 | NADC (7) |
| 4 Al -3Mo-1V | 751 | 5,640 | 155 | -17 | 115 | -29 | 15 | 0 | NADC (7) |
| 4Al-3Mo-1V | 1,064 | 5,300 | 155 | -8 | 115 | -13 | 15 | +7 | NADC (7) |
| 4Al-3Mo-1V | 402 | 2,370 | 201 | +1 | 180 | -2 | 4 | $+19$ | CEL (4) |
| $7 \mathrm{Al}-12 \mathrm{Zr}$ | 123 | 5,640 | 132 | +1 | 126 | +1 | 17 | -24 | NADC (7) |
| $7 \mathrm{Al}-2 \mathrm{Cb}-1 \mathrm{Ta}^{b}$ | 181 | 5 | 111 | +4 | 98 | +1 | 18 | -16 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 123 | 5,640 | 172 | -3 | 143 | +7 | 5 | 0 | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 403 | 6,780 | 140 | +16 | 136 | +16 | 14 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{6}$ | 403 | 6,780 | 148 | +4 | 139 | -1 | 13 | -13 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 751 | 5,640 | 172 | 0 | 143 | +8 | 5 | +50 | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 751 | 5,640 | 140 | 0 | 136 | +4 | 14 | 0 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{6}$ | 751 | 5,640 | 148 | +3 | 139 | +4 | 13 | -13 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 197 | 2,340 | 172 | -27 | 143 | -22 | 5 | -40 | NADC (7) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 197 | 2,340 | 140 | +2 | 136 | +3 | 14 | +2 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 402 | 2,370 | 140 | +7 | 136 | +6 | 14 | -1. | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{6}$ | 402 | 2,370 | 148 | +2 | 139 | +1 | 13 | -8 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}$ | 181 | 5 | 140 | +3 | 136 | +2 | 14 | -9 | CEL (4) |
| $6 \mathrm{Al}-4 \mathrm{~V}^{6}$ | 181 | 5 | 138 | +3 | 132 | -6 | 14 | -29 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 403 | 6,780 | 144 | +18 | 140 | +18 | 9 | -2 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 751 | 5,640 | 144 | +18 | 140 | $+20$ | 9 | +6 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{b}$ | 402 | 2,370 | 144 | $+6$ | 140 | +5 | 9 | -22 | CEL (4) |
| $13 \mathrm{~V}-11 \mathrm{Cr}-3 \mathrm{Al}^{\text {b }}$ | 181 | 5 | 134 | +5 | 126 | +8 | 20 | -36 | CEL (4) |

[^26]
## SECTION 8

## MISCELLANEOUS ALLOYS

The miscellaneous alloys are those alloys or metals which are "one of a kind" or do not belong to any of the previous classes of alloys discussed. These alloys would not be considered constructional because of their price, mechanical properties, scarcity, and, in some cases, poor corrosion resistance. Many of them could be used advantageously in specialty or unique applications.

The chemical compositions of the miscellaneous alloys are given in Table 85, and their corrosion rates and types of corrosion in Table 86.

Alloys columbium, gold, platinum, $90 \%$ platinum- $10 \%$ copper, $75 \%$ platinum- $25 \%$ copper, tantalum, and tantalum-tungsten (Ta60) were uncorroded during exposure both at depth and at the surface.

Alloy MP35N neither corroded nor was susceptible to stress corrosion during 189 days of exposure at the 6,000 -foot depth. The MP35N bolts and nuts were in a block of $6 \mathrm{Al}-4 \mathrm{~V}$ titanium and were torqued to $50 \mathrm{ft}-\mathrm{lb}$. There was also no galvanic corrosion of either member of the couple.

The corrosion of the three magnesium alloys (M1A, AZ31B, and HK31A) and beryllium was so rapid that their use in seawater would be impractical.

Platinum alloys containing 50 and $75 \%$ copper were etched and pitted after 402 days of exposure at the 2,500 -foot depth. Such alloys are usually used for contacts in electrical applications. These two alloys would not be satisfactory for use in seawater.

Silver was attacked by the uniform thinning type of corrosion in seawater. The thin tarnish-like film is an excellent insulator; hence, silver could not be used as electrical contacts in seawater.

### 8.1. DURATION OF EXPOSURE

The effects of duration of exposure on some of the miscellaneous alloys are shown in Figures 18, 19, and 20 . The corrosion rates of arsenical, chemical, and tellurium lead, lead-tin solder, tin, and zinc decreased with duration of exposure (Figures 18 and
19), except for lead-tin solder and zinc at the 2,500 -foot depth. The corrosion rates of these two alloys increased with increasing time of exposure. At the 6,000 -foot depth the corrosion rate of tin increased initially, and, thereafter, decreased with increasing time of exposure. The extremely high corrosion rate for tin after 751 days of exposure (Table 86), obviously, is an error and must be disregarded.

Only surface seawater data were available for molybdenum and tungsten, and the effects of duration of exposure are shown in Figure 20. The corrosion rate of molybdenum decreased, becoming asymptotic with increasing time of exposure. The corrosion rate of tungsten increased linearly with time, at least during the first 760 days of exposure.

### 8.2. EFFECT OF DEPTH

The effects of depth on the corrosion of some of the miscellaneous alloys after 1 year of exposure are shown in Figure 21. The corrosion of lead, lead-tin solder, molybdenum, tungsten, and tin was greater at the surface than at depth in the Pacific Ocean. Only the corrosion of zinc was greater at depth than at the surface.

### 8.3. EFFECT OF CONCENTRATION OF OXYGEN

The effects of changes in the concentration of oxygen in seawater are shown in Figure 22. The corrosion rates of arsenical, chemical, and tellurium lead, lead-tin solder, molybdenum, tungsten, and tin were higher at the highest concentration of oxygen than at the lowest, but the increases were not necessarily proportional or linear. Only the corrosion of zinc was not uniformly influenced by changes in the concentration of oxygen in seawater between the limits of 0.4 to $5.75 \mathrm{ml} / \mathrm{l}$.

### 8.4. MECHANICAL PROPERTIES

The effects of exposure on the mechanical properties of columbium, molybdenum, and tantalum are given in Table 87. The mechanical properties of these three alloys were not impaired by exposure in seawater.


Figure 18. Effect of duration of exposure in seawater on the corrosion of lead and lead-tin solder at the surface and at depth.


Figure 19. Effect of duration of exposure in seawater on the corrosion of tin and zinc at the surface and at depth.


Figure 20. Effect of duration of exposure in surface seawater on the corrosion of molybdenum and tungsten.


Figure 21. Effect of depth on the corrosion of miscellaneous alloys after 1 year of exposure.


Figure 22. Effect of concentration of oxygen in seawater on the corrosion of miscellaneous alloys after 1 year of exposure.

Table 85. Chemical Composition of Miscellaneous Metals and Alloys, Percent by Weight

| Alloy | Chemical Composition | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: |
| Beryllium | 99.0 Be | Boeing (6) |
| Columbium | 99.75 Cb | CEL (4) |
| Columbium | 99.8 Cb | CEL (4) |
| Gold | 99.999 Au | CEL (4) |
| Lead, antimonial | $99 \mathrm{~Pb}, 6 \mathrm{Sb}$ | INCO (3) |
| Lead, chemical | 99.9 Pb | INCO (3) |
| Lead, tellurium | $99+\mathrm{Pb}, 0.04 \mathrm{Te}$ | INCO (3) |
| Lead-tin solder | $67 \mathrm{~Pb}, 33 \mathrm{Sn}$ | INCO (3) |
| Magnesium, MIA | $99 \mathrm{Mg}, 1 \mathrm{Mn}$ | INCO (3) |
| Magnesium, AZ31B | $96 \mathrm{Mg}, 2.6 \mathrm{Al}, 1.1 \mathrm{Zn}, 0.4 \mathrm{Mn}$ | INCO (3) |
| Magnesium, AZ31B | 95.3 Mg, 3.5 Al, 0.94 $\mathrm{Zn}, 0.25 \mathrm{Mn}$ | NADC (7) |
| Magnesium, HK31A | 96.2 Mg, 2.67 Th, 0.67 $\mathrm{Zr}, 0.45 \mathrm{Mn}$ | NADC (7) |
| Molybdenum | 99.9 Mo | CEL (4) |
| Platinum | 99.99 Pt | CEL (4) |
| Platinum-copper, 90-10 | $90 \mathrm{Pt}, 10 \mathrm{Cu}$ | CEL (4) |
| Platinum-copper, 75-25 | $75 \mathrm{Pt}, 25 \mathrm{Cu}$ | CEL (4) |
| Platinum-copper, 50-50 | $50 \mathrm{Pt}, 50 \mathrm{Cu}$ | CEL (4) |
| Platinum-copper, 25-75 | $25 \mathrm{Pt}, 75 \mathrm{Cu}$ | CEL (4) |
| Silver | 99.999 Ag | CEL (4) |
| Tantalum | 99.9 Ta | CEL (4) |
| Tantalum | $99.9 \mathrm{Ta}, 0.010 \mathrm{C}, 0.010 \mathrm{O}, 0.005 \mathrm{~N}, 0.002 \mathrm{H}$ | CEL (4) |
| Tantalum-tungsten, Ta 60 | $88.8-91.3 \mathrm{Ta}, 8.5-11 \mathrm{~W}$ | CEL (4) |
| Tin | 99.95 Sn | INCO (3) |
| Tungsten | 99.95 W | CEL (4) |
| Zinc | $99.9 \mathrm{Zn}, 0.09 \mathrm{~Pb}, 0.01 \mathrm{Fe}$ | INCO (3) |
| MP35N | $35 \mathrm{Co}, 35 \mathrm{Ni}, 20 \mathrm{Cr}, 10 \mathrm{Mo}$ | CEL (4) |

[^27]Table 86. Corrosion Rates and Types of Corrosion of Miscellaneous Alloys

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth (ft) | Corrosion |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & (m p y) \end{aligned}$ | Type ${ }^{b}$ |  |
| Beryllium | W | 197 | 2,340 | 2.0 | $\mathrm{P}(\mathrm{PR})(27)$ | Boeing (6) |
| Columbium | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| Columbium | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| Columbium | W | 182 | 5 | 0.0 | NC | CEL (4) |
| Columbium | W | 364 | 5 | 0.0 | NC | CEL (4) |
| Columbium | W | 723 | 5 | 0.0 | NC | CEL (4) |
| Columbium | W | 763 | 5 | 0.0 | NC | CEL (4) |
| Gold | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| Gold | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| Lead, antimonial | W | 123 | 5,640 | 0.8 | U | INCO (3) |
| Lead, antimonial | S | 123 | 5,640 | 1.3 | U | INCO (3) |
| Lead, antimonial | W | 403 | 6,780 | 0.3 | U | INCO (3) |
| Lead, antimonial | S | 403 | 6,780 | 0.5 | U | INCO (3) |
| Lead, antimonial | W | 751 | 5,640 | 0.2 | U | INCO (3) |
| Lead, antimonial | S | 751 | 5,640 | 0.3 | U | INCO (3) |
| Lead, antimonial | W | 1,064 | 5,300 | 0.2 | U | INCO (3) |
| Lead, antimonial | S | 1,064 | 5,300 | 0.3 | U | INCO (3) |
| Lead, antimonial | W | 197 | 2,340 | 0.3 | U | INCO (3) |
| Lead, antimonial | S | 197 | 2,340 | 0.2 | U | INCO (3) |
| Lead, antimonial | W | 402 | 2,370 | 0.3 | U | INCO (3) |
| Lead, antimonial | S | 402 | 2,370 | 0.6 | U | INCO (3) |
| Lead, antimonial | W | 182 | 5 | 1.2 | U | INCO (3) |
| Lead, antimonial | W | 366 | 5 | 0.5 | U | INCO (3) |
| Lead, chemical | W | 123 | 5,640 | 0.8 | U | INCO (3) |
| Lead, chemical | S | 123 | 5,640 | 0.6 | U | INCO (3) |
| Lead, chemical | W | 403 | 6,780 | 0.2 | U | INCO (3) |
| Lead, chemical | S | 403 | 6,780 | 0.1 | U | INCO (3) |
| Lead, chemical | W | 751 | 5,640 | 0.1 | U | INCO (3) |
| Lead, chemical | S | 751 | 5,640 | 0.1 | U | INCO (3) |
| Lead, chemical | W | 1,064 | 5,300 | 0.1 | U | INCO (3) |
| Lead, chemical | S | 1,064 | 5,300 | 0.3 | U | INCO (3) |
| Lead, chemical | W | 197 | 2,340 | 0.3 | U | INCO (3) |
| Lead, chemical | S | 197 | 2,340 | 0.2 | U | INCO (3) |
| Lead, chemical | W | 402 | 2,370 | 0.2 | U | INCO (3) |
| Lead, chemical | S | 402 | 2,370 | 0.3 | U | INCO (3) |
| Lead, chemical | W | 182 | 5 | 0.8 | U | INCO (3) |
| Lead, chemical | W | 366 | 5 | 0.5 | U | INCO (3) |
| Lead, tellurium | W | 123 | 5,640 | 1.1 | U | INCO (3) |
| Lead, tellurium | S | 123 | 5,640 | 1.3 | U | INCO (3) |
| Lead, tellurium | W | 403 | 6,780 | 0.3 | U | INCO (3) |
| Lead, tellurium | S | 403 | 6,780 | 0.3 | U | INCO (3) |
| Lead, tellurium | W | 751 | 5,640 | 0.1 | U | INCO (3) |
| Lead, tellurium | S | 751 | 5,640 | 0.2 | U | INCO (3) |
| Lead, tellurium | W | 1,064 | 5,300 | 0.1 | U | INCO (3) |
| Lead, tellurium | S | 1,064 | 5.300 | 0.3 | U | INCO (3) |
| Lead, tellurium | W | 197 | 2,340 | 0.3 | U | INCO (3) |

Table 86. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Type ${ }^{\text {b }}$ |  |
| Lead, tellurium | S | 197 | 2,340 | 0.3 | U | $1 \mathrm{NCO}(3)$ |
| Lead, tellurium | W | 402 | 2.370 | 0.2 | U | INCO (3) |
| Lead, tellurium | S | 402 | 2,370 | 0.3 | U | INCO (3) |
| Lead, tellurium | w | 182 | 5 | 1.0 | U | $1 \mathrm{NCO}(3)$ |
| Lead, tellurium | w | 366 | 5 | 0.5 | U | INCO (3) |
| Lead-tin solder | w | 123 | 5,640 | 1.2 | U | INCO (3) |
| Lead-tin solder | S | 123 | 5,640 | 0.7 | U | INCO (3) |
| Lead-tin solder | W | 403 | 6,780 | 1.1 | U | INCO (3) |
| Lead-tin solder | S | 403 | 6,780 | 0.6 | U | INCO (3) |
| Lead-tin solder | W | 751 | 5,640 | 0.3 | U | INCO (3) |
| Lead-tin solder | S | 751 | 5,640 | 0.4 | U | [ $\mathrm{NCO}(3)$ |
| Lead-tin solder | W | 1,064 | 5,300 | 0.3 | U | INCO (3) |
| Lead-tin solder | S | 1,064 | 5,300 | 0.5 | G | INCO (3) |
| Lead-tin solder | w | 197 | 2,340 | 0.5 | U | INCO (3) |
| Lead-tin solder | S | 197 | 2,340 | 0.3 | U | INCO (3) |
| Lead-tin solder | W | 402 | 2,370 | 0.6 | U | INCO (3) |
| Lead-tin solder | S | 402 | 2,370 | 0.7 | U | INCO (3) |
| Lead-tin solder | W | 182 | 5 | 3.7 | U | INCO (3) |
| Lead-tin solder | w | 366 | 5 | 1.5 | G | $1 \mathrm{NCO}(3)$ |
| Magnesium, M1A | W | 123 | 5,640 | $+2.0$ | S | INCO (3) |
| Magnesium, M1A | S | 123 | 5,6+0 | 43.0 | S | inco (3) |
| Magnesium, M1A | W | 403 | 6.780 | >20.0 | 100 C | [NCO (3) |
| Magnesium, M1A | s | 403 | 6.780 | 7.3 | S-G | INCO (3) |
| Magnesium, M1A | W | 751 | 5,670 | $\geqslant 10.0$ | 100 C | INCO (3) |
| Magnesium, M1A | S | 751 | 5,640 | $>12.0$ | 100 C | INCO (3) |
| Magnesium, M1A | W | 1,064 | 5.300 | - | 100 C | INCO (3) |
| Magnesium, M1A | S | 1,064 | 5,300 | 2.9 | S | INCO (3) |
| Magnesium, M11A | W | 197 | 2,340 | - | 100 C | INCO (3) |
| Magnesium, M1A | S | 197 | 2,340 | 9.0 | S | $1 \mathrm{NCO}(3)$ |
| Magnesium, AZ31B | w | 123 | 5.640 | $>59.0$ | 100 C | INCO (3) |
| Magnesium, AZ31B | w | 123 | 5,640 | - | P (PR) | NADC (7) |
| Magnesium, AZ31B | S | 123 | 5,6+0 | 27.0 | S | INCO (3) |
| Magnesium, AZ31B | w | 403 | 6,780 | $>20.0$ | 100 C | INCO (3) |
| Magnesium, AZ31B | W | $+03$ | 6.780 | - | $\mathrm{N}-\mathrm{P}$ (PR) | NADC (7) |
| Magnesium, AZ318 | S | 403 | 6.780 | 6.6 | S-G | $1 \mathrm{NCO}(3)$ |
| Magnesium, AZ31B | w | 751 | 5,640 | $>10.0$ | 100 C | inco (3) |
| Magnesium, AZ31B | S | 751 | 5,640 | $>10.0$ | 100 C | $1 \mathrm{NCO}(3)$ |
| Magnesium, AZ31B | w | 1,064 | 5,300 | - | 100 C | INCO (3) |
| Magnesium, AZ31B | S | 1,064 | 5,300 | $>7.0$ | 100 C | INCO (3) |
| Magnesium, AZ31B | W | 197 | 2,340 | - | 100 C | INCO (3) |
| Magnesium, AZ31B | w | 197 | 2,340 | - | EX-G | NADC (7) |
| Magnesium, AZ31B | S | 197 | 2,340 | >15.0 | 50 C | INCO (3) |
| Magnesium, AZ31B | W | 402 | 2,370 | - | 100 C | INCO (3) |
| Magnesium, AZ31B | S | 402 | 2.370 | 11.0 | S | INCO (3) |
| Magnesium, AZ31B | W | 182 | 5 | $>40.0$ | 95C | 1 NCO (3) |
| Magnesium, AZ31B | w | 366 | 5 | $>20.0$ | 100 C | INCO (3) |

Table 86. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth <br> (ft) | Corrosion |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & \text { (mpy) } \end{aligned}$ | Type ${ }^{\text {b }}$ |  |
| Magnesium, HK31A | w | 403 | 6,780 | - | S-Gt | NADC (7) |
| Magnesium, HK31A | w | 197 | 2,340 |  | EX G ${ }^{\text {d }}$ | NADC (7) |
| Molybdenum | w | 402 | 2,370 | 0.8 | C (9); U | CEL (4) |
| Molybdenum | S | 402 | 2,370 | 0.8 | C (6); U | CEL (4) |
| Molybdenum | W | 182 | 5 | 1.4 | U | CEL (4) |
| Molybdenum | w | 364 | 5 | 1.1 | U-ET | CEL (4) |
| Molybdenum | w | 723 | 5 | 1.1 | G | CEL (4) |
| Molybdenum | w | 763 | 5 | 1.0 | C (6); G | CEL (4) |
| Platinum | w | 402 | 2,370 | 0.0 | NC | CEL (4) |
| Platinum | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $90 \mathrm{Pt}-10 \mathrm{Cu}$ | w | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $90 \mathrm{Pt}-10 \mathrm{Cu}$ | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $75 \mathrm{Pt}-25 \mathrm{Cu}$ | w | 402 | 2,370 | 0.0 | NC | CEL ( + ) |
| $75 \mathrm{Pt}-25 \mathrm{Cu}$ | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| $50 \mathrm{Pt}-50 \mathrm{Cu}$ | W | 402 | 2,370 | - | ET | CEL. (4) |
| $50 \mathrm{Pt}-50 \mathrm{Cu}$ | S | 402 | 2,370 | - | ET; P | CEL (4) |
| $25 \mathrm{Pt}-75 \mathrm{Cu}$ | W | 402 | 2,370 | - | ET; P | CEL (4) |
| $25 \mathrm{Pt}-75 \mathrm{Cu}$ | S | 402 | 2,370 | - | ET; P | CEL (4) |
| Silver | w | 402 | 2,370 | 0.6 | U | CEL (4) |
| Silver | S | 402 | 2,370 | 0.5 | U | CEL (4) |
| Tantalum | W | 402 | 2,370 | 0.0 | NC | CEL (4) |
| Tantalum | S | 402 | 2,370 | 0.0 | NC | CEL (4) |
| Tantalum | w | 182 | 5 | 0.0 | NC | CEL (4) |
| Tantalum | w | 364 | 5 | 0.0 | NC | CEL (4) |
| Tantalum | w | 723 | 5 | 0.0 | NC | CEL (4) |
| Tantalum | W | 763 | 5 | 0.0 | NC | CEL (4) |
| Ta-60 | w | 182 | 5 | 0.0 | NC | CEL (4) |
| Ta-60 | w | 364 | 5 | 0.0 | NC | CEL (4) |
| Ta-60 | W | 723 | 5 | 0.0 | NC | CEL (4) |
| Ta-60 | W | 763 | 5 | 0.0 | NC | CEL (4) |
| Tin | W | 123 | 5,640 | 0.5 | G | INCO (3) |
| Tin | S | 123 | 5,640 | 0.6 | G | INCO (3) |
| Tin | W | 403 | 6,780 | 1.4 | P (17) | INCO (3) |
| Tin | S | 403 | 6,780 | 1.7 | SH-CR | INCO (3) |
| Tin | W | 751 | 5,640 | 20.4 | P (PR)(30) | INCO (3) |
| Tin | S | 751 | 5,640 | 0.1 | G | INCO (3) |
| Tin | W | 1,064 | 5,300 | $<0.1$ | SC-ET | INCO (3) |
| Tin | S | 1,064 | 5,300 | 0.3 | G | INCO (3) |
| Tin | W | 197 | 2,340 | 1.8 | C (2); CR (2) | INCO (3) |
| Tin | S | 197 | 2,340 | $<0.1$ | P (2) | INCO (3) |
| Tin | W | 402 | 2,370 | 1.6 | SC-P (9) | INCO (3) |
| Tin | S | 402 | 2,370 | 0.1 | C (1) | INCO (3) |
| Tin | w | 182 | 5 | 8.3 | $\mathrm{P}(\mathrm{PR})(30)$ | INCO (3) |
| Tin | w | 366 | 5 | 2.8 | $\mathrm{C}(\mathrm{PR})(30) ; \mathrm{P}(\mathrm{PR})(30)$ | INCO (3) |

Continued

Table 86. Continued.

| Alloy | Environment ${ }^{\text {a }}$ | Exposure (day) | Depth (ft) | Corrosion |  | Source ${ }^{\text {c }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  | $\begin{aligned} & \text { Rate } \\ & (m p y) \end{aligned}$ | Type ${ }^{b}$ |  |
| Tungsten | W | 402 | 2,370 | 0.6 | U | CEL (4) |
| Tungsten | S | 402 | 2,370 | 0.5 | U | CEL (4) |
| Tungsten | W | 182 | 5 | 2.8 | U | CEL (4) |
| Tungsten | W | 364 | 5 | 3.2 | U | CEL (4) |
| Tungsten | W | 723 | 5 | 3.7 | U | CEL (4) |
| Tungsten | W | 763 | 5 | 4.0 | U | CEL (4) |
| Zinc | W | 123 | 5,640 | 6.7 | P (13) | INCO (3) |
| 7inc | S | 123 | 5,640 | 5.0 | $P(25)$ | INCO (3) |
| Zinc | W | 403 | 6,780 | 5.9 | C (PR)(30) | INCO (3) |
| Zinc | S | 403 | 6.780 | 0.2 | G | INCO (3) |
| Zinc | W | 751 | 5,640 | 3.6 | G | INCO (3) |
| Zinc | S | 751 | 5,640 | 2.8 | G | INCO (3) |
| Zinc | W | 1.064 | 5,300 | 2.4 | CR; L | INCO (3) |
| Zinc | S | 1,064 | 5,300 | 0.7 | G | INCO (3) |
| Zinc | W | 197 | 2,340 | 2.3 | P (2) | $1 \mathrm{NCO}(3)$ |
| Zinc | S | 197 | 2,340 | 0.3 | G | INCO (3) |
| Zinc | W | 402 | 2,370 | 2.8 | G | INCO (3) |
| Zinc | S | 402 | 2,370 | 2.4 | GASL | INCO (3) |
| Zinc | W | 182 | 5 | 4.5 | P (5) | INCO (3) |
| Zinc | W | 366 | 5 | 2.8 | P (10) | INCO (3) |

${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimens were embedded in the bottom sediments.
${ }^{b}$ Symbols for types of corrosion:

| C | $=$ Crevice |
| :--- | :--- |
| CR | $=$ Cratering |
| ET | $=$ Etched |
| EX | $=$ Extensive |
| G | $=$ General |
| N | $=$ Numerous |
| NC | $=$ No visible corrosion |
| P | $=$ Pitting |


| PR | $=$ Perforated |
| ---: | :--- |
| S | $=$ Severe |
| SC | $=$ Scattered |
| SH | $=$ Shallow |
| U | $=$ Uniform |
| 50 C | $=50 \%$ corroded |
| 95 C | $=95 \%$ corroded |
| 100 C | $=100 \%$ corroded |

GASI = General above sediment line.

Numbers in parentheses indicate maximum depth in mils.
${ }^{c}$ Numbers refer to references at end of report.
${ }^{\prime}$ Thick, black, brittle crust of corrosion products.
Table 87. Changes in Mechanical Properties of Miscellaneous Alloys Due to Corrosion

| Alloy | Exposure (day) | Depth (ft) | Tensile Strength |  | Yield Strength |  | Elongation |  | Source ${ }^{\text {a }}$ |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  | Original (ksi) | \% Change | Original <br> (ksi) | \% Change | Original <br> (\%) | \% Change |  |
| Molybdenum | 402 | 2,370 | 14 | +3 | 12 | +9 | 17 | -20 | CEL (4) |
| Molybdenum ${ }^{\text {b }}$ | 402 | 2,370 | 14 | +3 | 12 | +4 | 17 | -29 | CEL (4) |
| Molybdenum | 181 | 5 | 14 | -2 | 12 | +1 | 17 | -14 | CEL (4) |
| Molybdenum | 364 | 5 | 14 | -5 | 12 | +2 | 17 | -29 | CEL (4) |
| Tantalum | 402 | 2,370 | 49 | -3 | 37 | -4 | 49 | -6 | CEL (4) |
| Tantalum ${ }^{\text {b }}$ | 402 | 2,370 | 49 | -3 | 37 | +1 | 49 | -5 | CEL (4) |
| Tantalum | 181 | 5 | 49 | -4 | 37 | -11 | 49 | -5 | CEL (4) |
| Tantalum | 364 | 5 | 49 | -3 | 37 | -18 | 49 | +66 | CEL (4) |
| Columbium | 402 | 2,370 | 114 | +6 | 102 | -3 | 12 | -17 | CEL (4) |
| Columbium ${ }^{b}$ | 402 | 2,370 | 114 | -3 | 102 | 0 | 12 | 0 | CEL (4) |
| Columbium | 181 | 5 | 114 | -3 | 102 | -2 | 12 | +2 | CEL (4) |
| Columbium | 364 | 5 | 114 | -5 | 102 | -20 | 12 | +33 | CEL (4) |

[^28]
## SECTION 9

## WIRE ROPES

Wire ropes of many different chemical compositions, with different types of coatings, of different sizes, and of different types of construction were exposed in seawater at depth to determine their corrosion behavior. Some were stressed in tension to determine their susceptibility to stress corrosion or whether stress increased their rates of corrosion.

The chemical compositions of the ropes are given in Table 88, and their corrosion behavior in Table 89.

There was no visible corrosion on rope numbers $15,18,19,20,21,22,41$ ( 751 days, 6,000 feet), 48, $49,50,51,52$, and 53 . Rope number 15 was Type 316 stainless steel modified by adding silicon and nitrogen and was exposed for 189 days at the 6,000 -foot depth. Wire rope number 41 , conventional Type 316, was uncorroded after 751 days of exposure at the 6,000 -foot depth, but was rusted with some internal wires broken and crevice corrosion after 1,064 days of exposure; its breaking strength was decreased by $41 \%$ after 1,064 days of exposure at the 6,000 -foot depth. Because the conventional Type 316 stainless steel was not corroded until between 751 and 1,064 days of exposure, it cannot be stated that the addition of silicon and nitrogen to the Type 316 stainless steel improved its corrosion resistance.

Rope numbers $18,19,20$, and 21 were nickel base alloys. Rope numbers 20 and 21 were also uncorroded when lying on or in the bottom sediment.

Rope number 22 was a cobalt base alloy which was also uncorroded when lying on or in the bottom sediment. It, also, was not susceptible to stress corrosion in either the seawater or the bottom sediment when stressed to $40 \%$ of its breaking strength.

Rope numbers 48, 49, 50, 51, 52, and 53 were $6 \mathrm{Al}-4 \mathrm{~V}-\mathrm{Ti}$ ropes and wires. The ropes and wires themselves were not corroded, but the Type 304 stainless steel fittings and steel tie wires were severely corroded galvanically.

All the other wire ropes, coated or uncoated, were corroded to varying degrees of severity, the most severe being the parting of the wires.

Bare steel wires $1,2,3,35$, and 36 , as expected, were completely covered with rust. There was no loss of strength after periods of exporsure of as long as

1,064 days. These rupes had been lubricated during fabrication; the lubricant on the vuter surfaces of the ropes had disappeared, but not on the internal surfaces during exposure. One rope, No. 2, was degreased pror to exposure as a result, it was more severely corroded than the others on the outside surfaces, and there was light rust on many of the internal wires: One rope, No. 3, was degreased, then wrapped with 10 -mil-thick polyethylene tape prior to exposure. There was heavy rust undemeath the tape for a distance of about 3 feet from each end and there was light rust on about $75 \%$ of the internal wires. Wires 35 and 36 had been stressed in tension to $20 \%$ of their respective breaking strengths prior to exposure. These two ropes were covered with rust with no rust on the internal wires, did not fail by stress corrosion, and had no decrease in their breaking strengths.

The galvanized (zinc-coated) ropes were numbers $4,5,6,23,24,25,26,27,28,37$, and 38 . The zinc coatings protected the steel wires, but there was no good correlation between the weight or thickness of coating and the duration of protection. In general, except for the electrogalvanized coating, the heavier the coating the longer the period of time before rust appeared on the ropes. The breaking strengths of the ropes were not impaired by exposures for as long as 1,064 days of exposure. Also, rope numbers 37 and 38 were not susceptible to stress corrosion when stressed to $20 \%$ of their respective breaking strengths.

Rope numbers $7,8,9$, and 40 , in addition to being galvanized, were also jacketed with plastic coatings. In all cases, seawater penetrated along the interfaces between the ropes and the jackets. There was some light rust on the strands of rope number 40 underneath the poly(vinyl chloride) jacket after 751 days of exposure. The polyurethane (rope number 7) and polyethylene (rope numbers 8 and 9) jackets protected the galvanized ropes to a considerable extent. The jackets were not punctured or broken, but seawater had penctrated to the metal ropes through the end terminations. That water had penetrated to the interface between the jackets and the ropes was proven by puncturing the jackets, at which time seawater spurted out under considerable
pressure. When a terminal was removed from one end of each specimen, the zinc coatings were gone from the portions of the ropes which had been inside the terminals and the wires of the strands were rusted. The polyethylene jacket on one rope (number 9) had been punctured in many places prior to exposure. After exposure these holes were filled with white corrosion products; there was pressure underneath the jacket; and there was no rust on the wires except inside the terminals.

Rope numbers $39,43,44$, and 45 were aluminized (coated with a layer of aluminum). Aluminum coatings afforded considerable protection to steel ropes in the same manner as did zinc coatings. A $0.38-\mathrm{oz} / \mathrm{sq} \mathrm{ft}$ of aluminum coating ( 1.4 mils thick) afforded protection to steel rope for about the same period of time as did an $0.83-\mathrm{oz} / \mathrm{sq} \mathrm{ft}$ of zinc coating (1.4 mils thick). In dcep-ocean environments, equal thicknesses of coatings of zinc and aluminum protected steel ropes for about the same periods of time, but on a weight basis zinc was about twice as heavy as aluminum. There was no decrease in breaking strength caused by corrosion. Also, rope number 39 was not susceptible to stress corrosion when stressed at $20 \%$ of its breaking strength.

Rope numbers $10,11,12,13,14,15,16,17,29$, $30,31,32,33,34,41$, and 72 were stainless steels of different chemical compositions. The 0.1875 -inchdiameter Type 304 stainless steel ropes (10, 11, 12, $13,29,30$, and 31 ), stress relieved and not stress relieved, were corroded by crevice, pitting, and tunnel corrosion; and many of the wires had parted because of corrosion, particularly internal wires. There were only rust spots on the larger diameter, 0.250 -through- 0.375 -inch, Type 304 ropes $(32,33$, and 34) for equivalent periods of exposure. The addition of vanadium and nitrogen (rope number 16) to the Type 304 composition did not improve the corrosion resistance of the Type 304 stainless steel. The addition of copper (rope number 14) to the Type 316 stainless steel composition impaired its corrosion resistance, while the addition of silicon and nitrogen (rope number 15) did not appear to have any influence. The conventional Type 316 stainless steel (rope number 41) was uncorroded after 751 days of exposure, but after 1,064 days there were many internal wires broken as a result of attack by crevice corrosion. The breaking strengths of most of the
stainless steel ropes were impaired by exposure in seawater at depth. Rope numbers 41 and 42 were not susceptible to stress corrosion when stressed at $20 \%$ of their respective breaking strengths.

Two Type 304 stainless steel ropes (numbers 46 and 47) were clad with $90 \%$ copper- $10 \%$ nickel alloy. Rope number 46 , which had a clad layer 0.7 inch thick, had a green color after 402 days of exposure, indicating that the $\mathrm{Cu}-\mathrm{Ni}$ clad layer had not been completely sacrificed. However, rope number 47 , which had a clad layer 0.3 mil thick, was covered with a light film of rust, indicating that it had been completely sacrificed during the same period of time. In both cases the internal wires of the ropes were uncorroded.
Table 88. Chemical Composition of Wire Ropes

| Rope | C | Mn | P | S | Si | Ni | Cr | Mo | Cu | Co | $\mathrm{Fc}^{\text {a }}$ | Other |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{Fe}-\mathrm{Cr}-\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cu}$ | 0.065 | 1.55 | 0.010 | 0.013 | 1.39 | 13.90 | 18.64 | 2.44 | 1.95 | - | 12 | $0.060 \mathrm{~N},<0.01 \mathrm{Zr},<0.02 \mathrm{~V}$ |
| Fe-Cr-Ni-Mto-Si-N | 0.072 | 1.60 | 0.013 | 0.015 | 2.28 | 13.80 | 18.70 | 2.47 | $<0.02$ | - | R | $0.17 \mathrm{~N},<0.01 \mathrm{Zr},<0.02 \mathrm{~V}$ |
| Fe-Cr-Ni-V-N | 0.070 | 1.35 | 0.012 | 0.014 | 0.98 | 13.70 | 19.56 | $<0.01$ | $<0.02$ | - | R | $0.15 \mathrm{~N},<0.01 \mathrm{Zr}, 3.50 \mathrm{~V}$ |
| $\mathrm{Fe}-\mathrm{Cr}-\mathrm{Ni}-\mathrm{Si}$ | 0.063 | 1.51 | 0.005 | 0.008 | 1.92 | 17.82 | 17.82 | 0.02 | 0.03 | - | R | - |
| $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr}-\mathrm{C}$ | 0.01+ | 0.45 | 0.010 | 0.008 | 0.53 | 60.2 | 15.9 | 15.0 | - | - | 3.56 | 0.080 V, 3.14 W |
| $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr}-\mathrm{Mo}$ | - | - | - | - | - | 35.0 | 20.0 | 10.0 | - | 35.0 | - | - |
| Ni-Cr-M10 103 | 0.02 | - | - | - | - | 67.0 | 18.0 | 14.0 | - | - | - | 0.50 Cb |
| Ni -Cr-M10 625 | 0.05 | - | - | 0.007 | 0.30 | 61.0 | 22.0 | 9.0 | 0.10 | - | 3.0 | $\mathrm{Cb}+4.0 \mathrm{Ta}$ |
| $\mathrm{Co}-\mathrm{Cr}-\mathrm{Ni}-\mathrm{Fe}-\mathrm{Mo}$ | 0.050 | 1.96 | - | - | 0.74 | 14.96 | 19.84 | 7.14 | - | 40.46 | 14.60 | $0.058 \mathrm{Al}, 0.07 \mathrm{Be}, 0.058 \mathrm{Zr}, 0.011 \mathrm{Ca}$ |
| Cr-Mn-N | 0.07 | 14.3 | 0.021 | 0.003 | 0.67 | 0.27 | 18.4 | - | - | - | R | 0.48 N |
| AISI type $30 \psi^{b}$ | 0.08 | - | - | - | - | 10.0 | 19.0 | - | - | - | R | - |
| AISI type $316^{b}$ | 0.08 | - | - | - | - | 12.0 | 17.0 | 2.5 | - | - | R | - |
| $\begin{aligned} & 90-10 \text { Clad } 304^{b} \\ & 30+\text { Core } \\ & 90-10 \text { Clad } \end{aligned}$ | $0.08$ | - | - | - | - | $10.0$ | 19.0 - | - | $\stackrel{-}{90.0}$ | $\begin{aligned} & - \\ & - \end{aligned}$ | $\begin{aligned} & \mathrm{R} \\ & - \end{aligned}$ | - |
| Aircraft cable ${ }^{c}$ <br> Plow steel ${ }^{d}$ <br> Improved plow stecl ${ }^{l^{\prime}}$ <br> Monitor stect $f$ <br> Monitor $\triangle \mathrm{A}$ steel $^{g}$ |  |  |  |  |  |  |  |  |  |  |  |  |

[^29]${ }^{i}$ No chemical composition requirements; but strength requirements very high, especially for processed carbon steel wire. $d_{\text {No chemical composition requirements; only mechanical strength requirements. }}$
'No chemical composition requirements; mechanical strength requirements $15 \%$ higher than those for plow stecl.
$f_{\text {Same as plow stecl. }}$
${ }^{s}$ Same as improved plow steel.

Table 89. Corrosion of Wire Ropes

| Rope No. | Alloy ${ }^{\text {a }}$ | Coating | Diameter (in.) | Construction | Stress on Rope (lb) | Exposure (day) | Depth (ft) | Breaking Load |  |  | Remarks |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |  |  |  | Original <br> (lb) | Final <br> ( lb ) | $\therefore$ Change |  |
| 1 | Plow steel | lubricated | 0.875 | $7 \times 19$ | 0 | 123 | 6.000 | 48,200 | 48,200 | 0 | Rust on crowns of outside wires: lubricant still in grooves: inside: wires bright; tensile fracture. |
| 1 | Plow steel | lubricated | 0875 | $7 \times 19$ | 0 | 751 | 6,000 | 48.200 | 45,800 | -5 | Outside: $100 \%$ rust; inside: wires bright; tensile fracture. |
| 2 | Plow steel | degreased | 0.875 | $7 \times 19$ | 0 | 123 | 6,000 | 48,200 | 48,200 | 0 | Outside: $\mathbf{1 0 0 \%}$ rust i inside: wires light rust, few bright spots; tensile and torsion fractures. |
| 2 | Plow steel | degreased | 0.875 | $7 \times 19$ | 0 | 751 | 6.000 | +8,200 | 49,300 | +2 | Outside: $100 \%$ rust; inside: wires light rust, few bright spots: tensile fracture. |
| 3 | Plow steel | degreased; covered with 10 mil polyethylene tape | 0.875 | $7 \times 19$ | 0 | 123 | 6.000 | 48,200 | 48.900 | +2 | Rust underneath tape for about 3 feer from ends: inside: $\mathbf{5 0 \%}$ light rust, $50 \%$ bright; tensile and torsion fractures. |
| 3 | Plow steel | degreased; covered with $10-\mathrm{mil}$ polyethylene tape | 0.875 | $7 \times 19$ | 0 | 751 | 6.000 | 48,200 | 48,200 | 0 | Heavy rust at edges and underneath tape for about 3 feet from ends, inside: $\mathbf{7 5} \%$ light rust, $\mathbf{2 5 \%}$ bright: tensile fracture. |
| 4 | Improved plow steel | $\mathrm{zn} 0.50 \mathrm{oz} / \mathrm{ft}^{2}$ | 0250 | $3 \times 19$ | 0 | 189 | 6,000 |  | - | - | Outside: light, uniform rust, heavy in some grooves. |
| 5 | Improved plow steel | $\mathrm{zr} 0.70 \mathrm{oz} / \mathrm{ft}^{2}$ | 0.500 | $3 \times 19$ | 0 | 189 | 6,000 | - |  |  | Outside: yellow with few areas of heavy rust in grooves. |
| $\bigcirc$ | Improved plow steel | $\mathrm{Zn} 0.90 \mathrm{oz} / \mathrm{ft}^{2}$ | 0500 | $3 \times 7$ | 0 | 189 | 6.000 | - | - | - | Outside: grey-yellow, few areas of white corrosion products. few areas of heavy yellow corrosion products in grooves. |
| 7 | Improved plow stee | $\begin{gathered} \mathrm{Zn} 0.70 \mathrm{oz} / \mathrm{ft}^{2} \text {. } \\ \text { polyurethane jacket } \end{gathered}$ | 0.500 | $3 \times 19$ | 0 | 189 | 6,000 |  | - | - | No breaks in coating; white corrosion products on ends of terminals; terminations not watertight. |
| 8 | Improved plow stee! | $\begin{gathered} \mathrm{Zn} 0.70 \mathrm{oz} / \mathrm{ft}^{2}, \\ \text { polyethylene jacket } \end{gathered}$ | 0.500 | $3 \times 19$ | 0 | 189 | 6.000 |  | - | - | No breaks in coating; white corrosion products on ends of terminals; terminations not watertight. |
| 9 | Improved plow steel | $\mathrm{Zn} 0.70 \mathrm{oz} / \mathrm{ft}^{2}$, punctured polyethylene jacket | 0.500 | $3 \times 19$ | 0 | 189 | 6.000 | - | - | - | No rust at punctures in jacket; no breaks in coating; white corrosion products at ends of terminals; terminations not watertight. |
| 10 | AISI Type 304 | not stress relieved | 0.1875 | $3 \times 19$ | 0 | 189 | 6,000 | - | - | - | Dull grey with light rust stains; $\mathbf{1 2 - i n}$. length near center of wire covered with heavy red rust: some broken wires; when cleaned, many broken wires, tunnel, pitting, and crevice corrosion. |
| 11 | AISI Type 304 | stress relieved | 0.1875 | $3 \times 19$ | 0 | 189 | 6,000 | - | - | - | Dull grey; some light rust stains; heavy rust at edges of silicone porting compound; broken wires at edge of silicone compound under heavy rust at one end (crevice corrosion): when cleaned, numerous broken wires, internal tunnel, pitting, and crevice corrosion. |
| 12 | AISI Type 304 | not stress relieved | 0.1875 | $3 \times 7$ | 0 | 189 | 6,000 | - | - | - | Dull grey; light rust and some heavy rust in some areas; crevice corrosion; when cleaned, broken wires, tunnel, pitting, and crevice corrosion. |

Continued

Table 89. Continued.

| Rope No. | Alloy ${ }^{\text {a }}$ | Coating | Diameter (in.) | Construction | Stresson Rope (lb) | Exposure (day) | Depth <br> (ft) | Breaking Load |  |  | Remarks |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |  |  |  | Originai <br> (1b) | Final (b) | \% Change |  |
| 13 | AISI Type 304 | stress relieved | 0.1875 | $3 \times 7$ | 0 | 189 | 6,000 | - | - | - | Dull grey : light rust stains; few pits on crowns of outside wires: when cleaned, crevice, pitting, and tunnel corrosion. |
| 14 | $\mathrm{Fe}-\mathrm{Cr}-\mathrm{Ni}-\mathrm{Ma}-\mathrm{Cu}$ | bare | 0.125 | $3 \times 7$ | 0 | 189 | 6,000 | - | - | - | Dull grey with mottled light yellow stans; when cleaned, incipient crevice corrosion. |
| 15 | $\mathrm{Fe}-\mathrm{Cr}-\mathrm{Ni}-\mathrm{Mo}-\mathrm{Si}-\mathrm{N}$ | bare | 0.125 | $1 \times 7$ | 0 | 189 | 6,000 | - | - | - | No visible corrosion; metallic sheen sell present. |
| 16 | $\mathrm{Fe}-\mathrm{Cr}_{\mathrm{r}} \cdot \mathrm{Ni}-\mathrm{V}-\mathrm{N}$ | bare | 0.125 | $1 \times 7$ | 0 | 189 | 6,000 | - | - | - | Falled by tunnel and crevice corrosion underneath silicone potting compound: only end loops recovered. |
| 17 | $\mathrm{Fe}-\mathrm{Cr}-\mathrm{Ni}-\mathrm{Sr}$ | bare | 0.125 | $1 \times 7$ | 0 | 189 | 6,000 | - | - | - | Grey, no visible corrosion; when cleaned, many areas of shight crevice corrosion and shallow pitting. |
| 18 | $\mathrm{Ni}-\mathrm{Co}-\mathrm{Cr}_{\mathrm{T}} \mathrm{Mo}$ | bare | 00625 | $1 \times 7$ | 0 | 189 | 6,000 | - | - | - | No visible corrosion; original metallic sheen intact. |
| 19 | $\mathrm{Ni}-\mathrm{Mo}-\mathrm{Cr} \mathrm{C}$ | bate | 00625 | $1 \times 7$ | 0 | 189 | 6,000 | - | - | - | No visible corrosion; original metallic sheen intact. |
| 20 | Ni-Cr-Mo 625 | bare | 0250 | $7 \times 19$ | 0 | 189 | 6,000 | 7,400 | - | - | No visible corrosion; orginal metallic sheen intact. |
| - 20 | $\mathrm{Ni}-\mathrm{Cr}$-Mo $625^{\text {b }}$ | bare | 0250 | $7 \times 19$ | 0 | 189 | 6,000 | 7.400 | - | - | No visible corrosion; orginal metallic sheen intact. |
| 21 | Ni-Cr-Mo 103 | bare | 0250 | $7 \times 19$ | 0 | 189 | 6.000 | 7.000 | - | - | No visible corrosion: original metallic sheen intact. |
| 21 | $\mathrm{Ni}-\mathrm{Cr}$-Mo 103 ${ }^{\text {b }}$ | bare | 0.250 | $7 \times 19$ | 0 | 189 | 6,000 | 7,000 | - | - | No visible corrosion: orginal metalic sheen intact. |
| 22 | $\mathrm{Co}-\mathrm{Cr}-\mathrm{Ni}-\mathrm{Fe}-\mathrm{Mo}$ | bare | 0.1875 | $3 \times 19$ | 0 | 189 | 6,000 | 4,000 | - | - | Original blue film gone; no visible corrosion. |
| 22 | $\mathrm{CO}-\mathrm{Cr}-\mathrm{N} \cdot \mathrm{Fe}-\mathrm{Mo}^{\circ}$ | bare | 0.1875 | $3 \times 19$ | 1,600 | 189 | 6.000 | 4,000 | - | - | No wre failures, original blue film gone; no visible corrosion. |
| 22 | $\mathrm{Co}-\mathrm{Cr}-\mathrm{Ni}-\mathrm{Fe}-\mathrm{Mo}^{\text {b }}$ | bare | 01875 | $3 \times 19$ | 0 | 189 | 6,000 | 4,000 | - | - | Onginal blue film gone: no visible corrosion. |
| 22 | $\mathrm{Co}-\mathrm{Cr}-\mathrm{Ni}-\mathrm{Fe}-\mathrm{Mo}^{\text {b }}$ | bare | 01875 | $3 \times 19$ | 1.600 | 189 | 6.000 | 4,(000 | - | $\sim$ | No wire failures; original blue film gone: no visible corrosion. |
| 23 | Alrcraft cord | Zn 0.40 oz/ft ${ }^{2}$ | 00938 | $7 \times 7$ | 0 | 403 | 6.000 | 1,100 | 1,100 | 0 | Dark grey to black: tensile fracture. |
| 23 | Aircraft cord | $\mathrm{zn} 0.40 \mathrm{oz/ff}^{2}$ | 0.0938 | $7 \times 7$ | 0 | 197 | 2,500 | 1,100 | 1,000 | -9 | Outside. 100\% rust: inside: grey, tensile fracrure. |
| 24 | Aircraft cable | $\mathrm{zn} 0.40 \mathrm{oz} / \mathrm{fr}^{2}$ | 0.125 | $7 \times 19$ | 0 | 403 | 6,000 | 2,000 | 1,000 | -50 | Dark grey to black; inside. grey: tensile fracture. |
| 24 | Alrcraft cable | $\mathrm{znO}_{0} 00 \mathrm{z} / \mathrm{ft}^{2}$ | 0125 | $7 \times 19$ | 0 | 197 | 2,500 | 2,000 | 1.800 | -10 | Outside - 100\% rust; insude: grey; tensile and torsion fractures. |
| 25 | Aircraft cable | $\mathrm{Zn} 0.50 \mathrm{oz} / \mathrm{ft}^{2}$ | 01875 | $7 \times 19$ | 0 | 403 | 6,000 | 3,500 | 4,000 | +14 | Outside. dark grey to black; inside: grey. tensile and torsion fractures. |
| 25 | Arctaft cable | $\mathrm{zn} 0.500 \mathrm{z} / \mathrm{ft}^{2}$ | 01875 | $7 \times 19$ | 0 | 197 | 2,500 | 3,500 | 3,700 | +6 | Outwide dark grey: inside: grey; tensile and torston fractures. |
| 26 | Wire rope | $2 \mathrm{n} 0.50 \mathrm{oz} / \mathrm{ft}^{2}$ | 0.1875 | $1 \times 7$ | 0 | 403 | 6,000 | 2,600 | 2,600 | 0 | Outside $90{ }^{\circ}$ rust, inside: grey, tensile fracture. |
| 26 | Wire rope | $\mathrm{zn} 0.50 \mathrm{oz/ft}{ }^{2}$ | 01875 | $1 \times 7$ | 0 | 197 | 2,500 | 2.600 | 2,500 | -4 | Outside: medrum grey, few rust spors: inside: grey; tensile fracture. |

Continued

Table 89. Continued

| Rope No. | Alloy ${ }^{\text {a }}$ | Coating | Diameter (in.) | Construction | Stress on Rope (b) | Exposure (day) | Depth <br> (ft) | Breaking Load |  |  | Remarks |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |  |  |  | Original <br> (Ib) | Final <br> (b) | \% Change |  |
| 27 | Wire rope | zn 0.85 oz/ft ${ }^{2}$ | 0.250 | $1 \times 7$ | 0 | 403 | 6,000 | 5,900 | 4,600 | -22 | Outside: 100\% yellow; inside: grey; tensile fracture. |
| 27 | Wire rope | $\mathrm{zn} 0.85 \mathrm{oz} / \mathrm{ft}^{2}$ | 0.250 | $1 \times 7$ | 0 | 197 | 2,500 | 5,900 | 5,300 | -10 | Outside: medium grey, few rust spots: inside: grey; tensile fracture. |
| 28 | Aircraft cable | $\mathrm{Zn} 0.60 \mathrm{oz} / \mathrm{ft}^{2}$ | 0.250 | $7 \times 19$ | 0 | 403 | 6,000 | 6,100 | 5.900 | -2 | Outside: dark grey to black; inside: grey; tensile and torsion fractures. |
| 28 | Aircraft cable | $\mathrm{Zn} 0.60 \mathrm{oz} / \mathrm{ft}^{2}$ | 0.250 | $7 \times 19$ | 0 | 197 | 2,500 | 6,100 | 6,200 | +2 | Outside: dark grey to black; inside: grey; tensile fracture. |
| 29 | Aircraft cord | bare, Type 304 | 0.0938 | $7 \times 7$ | 0 | 403 | 6,000 | 800 | 100 | -88 | Internal strands corroded; tunnel, crevice, and pitting: many wires parted by corrosion. |
| 29 | Aircraft cord | bare, Type 304 | 0.0938 | $7 \times 7$ | 0 | 197 | 2,500 | 800 | 800 | 0 | Ourside: few rust spots; inside: bright; tensile fracture. |
| 30 | Aircraft cable | bare, Type 304 | 0.125 | $7 \times 19$ | 0 | 403 | 6,000 | 1.600 | 200 | -88 | Outside: few rust stains; inside: broken wires, crevice and pitting corrosion. |
| 30 | Aircraft cable | bare, Type 304 | 0.125 | $7 \times 19$ | 0 | 197 | 2,500 | 1,600 | 1,800 | +13 | Outside: crignal metallic lustre; inside: bright; tenssle fracture. |
| 31 | Aircraft cable | bare. Type 304 | 0.1875 | $7 \times 19$ | 0 | 403 | 6,000 | 2,700 | 100 | -96 | Outsise: many rust stains and broken wires, pitting, tunnel and crevice corrosion: inside: many broken wires, pitting, tunnel, and crevice corrosion. |
| 31 | Aitcraft cable | bare, Type 304 | 0.1875 | $7 \times 19$ | 0 | 197 | 2.500 | 2,700 | 2,800 | +4 | Outside: few rust spots; inside: bright; tensile fracture. |
| 32 | Aircraft cable | bare, Type 304 | 0.250 | $7 \times 19$ | 0 | 403 | 6,000 | 5,100 | 5,000 | -2 | Outside. few yellow stains; inside: bright; tensile fracture. |
| 32 | Aircraft cable | bare, Type 304 | 0.250 | $7 \times 19$ | 0 | 197 | 2,500 | 5,100 | 5,100 | 0 | Ourside. orignal metallic sheen; inside: bright: tensile and torsion fractures. |
| 33 | Aircraft cable | bare. Type 304 | 03125 | $7 \times 19$ | 0 | 403 | 6,000 | 7.100 | 7,700 | +8 | Outside: few rust stains, inside: bright; tensile and torsion fractures. |
| 33 | Aircraft cable | bare, Type 304 | 0.3125 | $7 \times 19$ | 0 | 197 | 2.500 | 7,100 | 7,000 | $-1$ | Outside: orignal metallic lustre; inside: bright; tensile fracture. |
| 34 | Aircraft cable | bare, Type 304 | 0.375 | $7 \times 19$ | 0 | 403 | 6,000 | 11,900 | 12,700 | -2 | Outside: few rust stains, inside: bright; tensile and torsion fractures. |
| 34 | Aircraft cable | bare. Type 304 | 0.375 | $7 \times 19$ | 0 | 197 | 2.500 | 11.900 | 11,600 | -3 | Outside: few rust stains: inslde: bright; tenstle fracrure. |
| 35 | Plow steel | lubricated | 0.325 | $1 \times 19$ | 2,100 | 751 | 6,000 | 10,700 | 10,700 | 0 | No stress failure, outside: $100 \%$ rust; inside: bright; tensile fracture. |
| 35 | Plow steel | luinivated | 0.325 | $1 \times 19$ | 2,100 | 1,064 | 6,000 | 10,700 | 11,500 | +7 | No stress failure, outside: $100 \%$ rust; inside: bright; tensile fracture. |
| 36 | Improved plow steel | lubricatcu | 0.326 | $1 \times 19$ | 2.900 | 751 | 6,000 | 14,300 | 14,900 | +4 | No stress failure, outside: $100 \%$ rust: inside: bright; tensile fracture. |

Table 89. Continued.

|  | Alloy ${ }^{\text {a }}$ | Coating | Diameter (in.) | Construction | Stress on Rope (lb) | Exposure (day) | Depth (ft) | Breaking Load |  |  | Remarks |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\begin{aligned} & \text { Rope } \\ & \text { No. } \end{aligned}$ |  |  |  |  |  |  |  | Original (b) | Final ( lb ) | \% Change |  |
| 36 | Improved plow steel | lubricated | 0.326 | $1 \times 19$ | 2.900 | 1,064 | 6.000 | 14,300 | 15,300 | +7 | No stress failure; outside: $100 \%$ rust; inside: bright; tensile failure. |
| 37 | Plow steel | $\mathrm{Zn} 0.83 \mathrm{oz} / \mathrm{ft}^{2}$ | 0.340 | $1 \times 19$ | 2.100 | 751 | 6.000 | 10,400 | 10.900 | +5 | No stress failure; outside: grey with $5 \%$ scattered rust: unside. grey; tensile fracture. |
| 37 | Plow steel | $\mathrm{zn} 0.83 \mathrm{oz} / \mathrm{ft}^{2}$ | 0.340 | $1 \times 19$ | 2,100 | 1,064 | 6,000 | 10,400 | 8.600 | -17 | No stress failure, ourside- $20 \%$ rust, $80 \%$ yellow: inside: grey. tensile fracture. |
| 38 | PJow steet | Zn 1.50 oz $/ \mathrm{ft}^{2}$, electrogalvanized | 0.335 | $1 \times 19$ | 2.200 | 751 | 6.000 | 10,901 | 11,100 | +2 | No stress fallure: outside: $\mathbf{5 0 \%}$ rust, $\mathbf{5 0 \%}$ grey; inside grey; tensile fracture. |
| 38 | Plow steel | $\begin{aligned} & \mathrm{Zn} 1.50 \text { oz/ft } \mathrm{f}^{2}, \\ & \text { etectrogalvanzed } \end{aligned}$ | 0.335 | $1 \times 19$ | 2.200 | 1,064 | 6,000 | 10.900 | 11.600 | +6 | No stress failure: outside: $95 \%$ rust: inside: grey; rensile fracture. |
| 39 | Plow steel | Al $0.38 \mathrm{oz} / \mathrm{ft}^{2}$ | 0.335 | $1 \times 7$ | 1,400 | 751 | 6.000 | 6,900 | 7,000 | +1 | No stress fallure: outside white corrosion products with $10^{\circ}$. rust stans; inside: grey; tensile fracture. |
| 39 | Plow steel | Al $0.38 \mathrm{o} / / \mathrm{ft}^{2}$ | 1) 335 | $1 \times 7$ | 1.400 | 1,064 | 6,000 | 6.900 | 6.500 | -6 | No stress fallure: outside. white corrosion products plus $50^{\circ}$ rust: inside: grey and light rust: tensile fracture. |
| 40 | Plow steel | $\begin{aligned} & \mathrm{Zn} 0.17 \mathrm{oz} / \mathrm{ft}^{2} \text {. } \\ & \text { polyvinyl chloride ja ket } \end{aligned}$ | 0.195 | $7 \times 7$ | 250 | 751 | 6,000 | 1.300 | 1,210 | -8 | Nu stress failure: PVC, dull; some light rust on strands underneath PVC; tensile fracture. |
| 40 | Plow stes 1 | $\begin{gathered} \mathrm{Zn}_{\mathrm{n}} 0.17 \mathrm{oz} / \mathrm{ft}^{2} \\ \text { polyvinyl chloride jacket } \end{gathered}$ | 0.195 | $7 \times 7$ | 250 | 1,064 | 6.000 | 1,300 | 1,100 | -15 | No stress failure, PVC, dull; some light rust on strands underneath PVC: tenstie and torston fractures. |
| 41 | Aıretaft cable | bare, Type 316, luhriated | 0.135 | $7 \times 7$ | 350 | 751 | 6,000 | 1,700 | 1,400 | -18 | No stress fallure: outside original metallic lustre. inside: bright; tensile fracture. |
| 41 | Aircraft cable | bare, Type 316, Iutricated | 0.135 | $7 \times 7$ | 350 | 1,064 | 6,000 | 1,700 | 1,000 | -41 | No stress failure; outside: $\mathbf{5 0 \%}$ rust, crevice corrosion, insiderusted wires, some broken, crevice corrosion, tensile and brittle fractures |
| 42 | Aırcraft cable | bare, $18 \mathrm{Cr} \cdot 14 \mathrm{Mm}-0.5 \mathrm{~N}$ | 0.395 | $7 \times 19$ | 2,500 | 751 | 6,000 | 12,400 | 11,400 | -8 | No stress failures, outside. few rust spors, insıde: bright $_{i}$ torsion fracture. |
| 42 | Arcraft cable | bare, $18 \mathrm{Cr}-14 \mathrm{Mn}-0.5 \mathrm{~N}$ | 0.395 | $7 \times 19$ | 2,500 | 1,064 | 6,000 | 12,400 | 12.500 | +1 | No stress failure, outside considerable rust and broken wires, inside some broken wires in all strands: torsion fracture. |
| 43 | Improved pluw stee! | Al $0.11 \mathrm{oz} / \mathrm{ft}^{2}$ | 0.1875 | $7 \times 7$ | 0 | 402 | 2,500 | 3,900 | 3,500 | -10 | Outside: white corrosion products with light rust stains. |
| 4. | Improved plow steel | Al $0.19 \mathrm{oz/ft}{ }^{2}$ | 0.250 | $1 \times 19$ | 0 | 402 | 2,500 | 8,800 | 7,800 | -11 | Outside: motted white and grey |
| 45 | 1 mproved plow steel | Al $0.19 \mathrm{oz} / \mathrm{ft}^{2}$ | 0.3125 | $1 \times 19$ | 0 | 402 | 2,500 | 14,000 | 13,000 | -7 | Outside grey with some white corrosion products. |
| 46 | Type 304 | $\begin{gathered} 90 \mathrm{Cu}-10 \mathrm{~N}, 0.0007 \mathrm{in} . \\ 0.50 \mathrm{oz} / \mathrm{ft}^{2} \end{gathered}$ | 0.1875 | $7 \times 7$ | 0 | 402 | 2,500 | 3,900 |  |  | Outside. green. |

Table 89. Continued.

| Rope No. | Alloy ${ }^{\text {a }}$ | Coating | Diameter (in.) | Construction | Stress on Rope (b) | Exposure (day) | Depth (ft) | Breaking Load |  |  | Remarks |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  |  |  |  |  |  |  |  | Orignal (lb) | Final <br> (lb) | \% Change |  |
| 47 | Type 304 | $\begin{gathered} 90 \mathrm{Cu}-10 \mathrm{Ni}, 0.0003 \mathrm{in} ., \\ 0.25 \mathrm{oz} / \mathrm{ft}^{2} \end{gathered}$ | 0.280 | $37 \times 7$ | 0 | 402 | 2.500 | - | - |  | Outside: light film rust; inside: bright, uncorroded. |
| 48 | $6 \mathrm{Al}-4 \mathrm{~V}-\mathrm{T}_{1}$ | bare, Type 304 firtings | 0250 | $6 \times 19$ | 0 | +02 | 2,500 | - | - |  | Fittungs rusted, crevice and galvanic corrosion, cable, no corrosion |
| 49 | 6Al-4- $\mathrm{T}_{1}$ | bare | 0.0625 | 1×19 | - | 402 | 2.500 |  | - |  | No corrosion except one steel wire which had become introduced during fabrication. |
| so | 6Al-4V-Ti | bare, steel fitting. steel tie wire | 0.0625 | $6 \times 19$ | 0 | 402 | 2,500 |  | - |  | Steel fitting and steel tee wire corroded galvanically, cable, no corrosion. |
| 51 | $6 \mathrm{Al}-4 \mathrm{~V}-\mathrm{Ti}$ | bare | 0063 | - | 0 | 402 | 2.500 | - |  |  | No corrosion |
| 52 | $6 \mathrm{Al}-4 \mathrm{~V}-\mathrm{Ti}$ | bare | 0045 | - | 0 | 402 | 2.500 | - |  |  | No corrosion. |
| 53 | $6 \mathrm{Al}-4 \mathrm{~V}-\mathrm{T}_{1}$ | bare | 0020 |  | $\bigcirc$ | 402 | 2.500 | - |  | - | No corroston. |

${ }^{2}$ Immersed in seawater unless otherwise specified.
$b_{1}$ mmersed in bottom sedment.

## SECTION 10

## REFERENCES

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## 13.

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[^0]:    ${ }^{a}$ Formerly Marine Engineering Laboratory (MEL), Annapolis, Maryland.

[^1]:    ${ }^{a}$ Numbers refer to references at end of report.
    ${ }^{b}$ No composition limits.
    ${ }^{c}$ High-strength, low-alloy steel.
    ${ }^{d}$ With mill scale.
    ${ }^{e}$ High-strength steel.
    

[^2]:    ${ }^{\text {a }}$ Copper Development Association alloy number.
    ${ }^{b}$ Numbers refer to references at end of report.

[^3]:    ${ }^{a}$ Copper Development Association alloy number.
    ${ }^{b} \mathrm{~W}=$ Totally exposed in seawater on sides of structure $; \mathrm{S}=$ Exposed in base of structure
    so that a portion of each specimen was embedded in the bottom sediments.
    ${ }^{c}$ Symbols for types of corrosion:
    GBSL = General below sediment line
    = Incipient
    $=$ Nonuniform
    $=$ Pitting
    $=$ Perforation
    $=$ Severe
    $=$ Slight
    $=$ Uniform
    ${ }_{\mathrm{I}}^{\mathrm{I}}$
    NU
    P
    PR
    PR
    SL
    = Crevice
    $=$ Coppering, a selective attack where copper
    appears on surface
    $=$ Cratering
    $=$ Edge
    $=$ Etching
    EBSL $=$ Etched below sediment line $\mathrm{G}=$ General

    Numbers indicate maximum depth in mils.
    ${ }^{d}$ Numbers refer to references at end of report.
    ${ }^{e}$ Much less below sediment line.
    $f_{\text {Exposed at Francis L. LaQue Corrosion Laboratory, INCO, Wrightsville Beach, N.C. }}$ $g_{\text {Cast alloy. }}$
    ${ }^{b}$ No visible corrosion below sediment line.

[^4]:    ${ }^{a}$ Copper Development Association alloy number.
    ${ }^{b}$ Numbers refer to references at end of report.

[^5]:    ${ }^{a}$ Numbers refer to references at end of report.

[^6]:    ${ }^{a}$ Numbers refer to references at end of report.
    ${ }^{b}$ Broke in weld.
    ${ }^{c}$ Partially embedded in bottom sediment.

[^7]:    Numbers indicate maximum depth in mils.

[^8]:    ${ }^{a}$ Numbers refer to references at end of report.
    ${ }^{b}$ Age hardened.
    ${ }^{c}$ Age hardened and cold worked.

[^9]:    ${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimen were embedded in the bottom sediments.
    ${ }^{b}$ Numbers refer to references at end of report.

[^10]:    ${ }^{a}$ Numbers refer to references at end of report.

[^11]:    ${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimen were embedded in the bottom sediments.
    $=$ Pitting
    $=$ Perforate
    $\mathrm{P}=$ Perforated
    
    $\mathrm{T}=$ Tunnel
    $\mathrm{NC}=$ No visible
    ${ }^{c}$ Numbers refer to references at end of report.

[^12]:    ${ }^{a}$ Numbers refer to references at end of report.

[^13]:    ${ }^{a} \mathrm{R}=$ remainder.

[^14]:    ${ }^{a} W=$ Totally exposed in seawater on sides of structure; $S=$ Exposed in base of structure so that the lower portions of the
    specimens were embedded in the bottom sediments.
    $=$ Severe
    $=$ Scattered
    $=$ Stress corrosion cracked
    $=$ Slight
    $=$ Scattered pinpoint pitting
    $=$ Tunnel
    $=$ Uniform
    $\mathrm{WB}=$ Weld bead
    $\mathrm{X}=$ Portion of
    S
    SC
    SPP
    T
    U
    WB
    $\mathrm{X}=$ Portion of specimens missing due to corrosion

[^15]:    ${ }^{c}$ Numbers refer to references at end of report.
    $d_{\text {Three-inch-diameter circular weld in center of specimen. }}$.
    ${ }^{e}$ Transverse butt weld.

[^16]:    ${ }^{a} \mathrm{~W}=$ Totally exposed in seawater on sides of structure; $\mathrm{S}=$ Exposed in base of structure so that the lower portions of the specimen
    were embedded in the bottom sediments.
    ${ }^{b}$ Numbers refer to references at end of report.
    ${ }^{c}$ Transverse butt weld at center of 12 -inch length.
    ${ }^{d}$ Too badly corroded to obtain tensile specimens.

[^17]:    ${ }^{c}$ Numbers refer to references at end of report.

[^18]:    ${ }^{a} W=$ Totally exposed in seawater on sides of structure; $S=$ Exposed in base of structure so that the lower portions of the specimens were embedded in
    the bottom sediments.
    ${ }^{b}$ Symbols for types of corrosion:
    D
    D $\quad$ Deep
    E
    $\mathrm{EX}=$ Edge
    $\begin{aligned} & =\text { Blisters } \\ & =\text { Crevice }\end{aligned}$

    IG = Intergranular
    Line

    IG
    L
    MD

    $$
    \mathrm{MD}=\text { Moderate }
    $$

[^19]:    ${ }^{b}$ Footnote $b$ continued:
    aron
    $\mathrm{NC}=$ No
    $\mathrm{NPF}=$ No paint failure
    $=$ Pitting
    ${ }^{c}$ Numbers refer to references at end of report.
    ${ }^{d}$ Welded with 2319 rod by the TIG process.
    ${ }^{e}$ Welded.

[^20]:    ${ }^{a}$ Numbers refer to references at end of report.
    ${ }^{b}$ Specimens lost at sea during exposure.

[^21]:    ${ }^{a} \mathrm{R}=$ remainder.
    ${ }^{b} \mathrm{Si}+\mathrm{Fe}=0.70$.

[^22]:    ${ }^{a}$ Numbers refer to references at end of report.

[^23]:    ${ }^{a} \mathrm{R}=$ remainder
    ${ }^{b_{S i}}+\mathrm{Fe}=0.23$
    ${ }^{c} \mathrm{Si}+\mathrm{Fe}=0.40$

[^24]:    "Numbers refer to references at end of report.
    ${ }^{b}$ Missing when STU was recovered.
    ${ }^{c}$ Alloy $7076-\mathrm{T} 6$ heated 8 hours at $350^{\circ} \mathrm{F}$, air cooled.

[^25]:    ${ }^{a} \mathrm{R}=$ remainder.
    ${ }^{b}$ Numbers refer to references at end of report.
    ${ }^{c}$ Nominal compositions.

[^26]:    "Numbers refer to references at end of report.
    $b_{\text {Unrelieved butt weld across width of specimens, TIG process, nonconsumable tungsten electrode. }}$.

[^27]:    ${ }^{a}$ Numbers refer to references at end of report.

[^28]:    ${ }^{a}$ Numbers refer to references at end of report.
    ${ }^{b}$ Exposed in bottom sediments.

[^29]:    ${ }^{b}$ Nominal chemical composition

